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## Multicolor light-emitting devices with $Tb_2O_3$ on silicon

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Great efforts have been devoted to achieving efficient Si-based light-emitting devices. Here we report new light-emitting devices fabricated with  $Tb_2O_3$  on Si substrates. Intense green electroluminescence was observed, with a turn-on voltage of about 8 V. The green emission is attributed to the characteristic transitions of  $Tb^{3+}$  ions in  $Tb_2O_3$ . The electroluminescence mechanisms of the  $Tb_2O_3$  light-emitting devices are discussed. In addition, visible and near infrared electroluminescence was observed in rare-earth ( $Eu^{3+}$ ,  $Sm^{3+}$  and  $Yb^{3+}$ ) doped  $Tb_2O_3$  light-emitting devices.

Received: 27 July 2016  
Accepted: 10 January 2017  
Published: 21 February 2017

Si-based photonics have been regarded as an effective way to improve data transfer rates of information processing systems<sup>1</sup>. In the past decades, great efforts have been devoted to achieving efficient Si-based light sources<sup>2–6</sup>. Rare-earth (RE) doped  $SiO_2$  has attracted a lot of interest due to their high luminescence efficiency and wide spectral range extending from ultraviolet (UV) to infrared (IR) ranges<sup>7–9</sup>. Previously, efficient visible light-emitting devices (LEDs) based on RE-doped metal-oxide-semiconductor (MOS) structures have been demonstrated<sup>10–12</sup>. However, both the light emission yield and the reliability of the RE-doped oxides devices are limited by charge trapping and RE clustering effects<sup>13</sup>.

$Tb_2O_3$  is a very attractive member of RE oxides since it is a direct and wide-band-gap semiconductor ( $\sim 3.8$  eV), and its lattice constant ( $\sim 10.73$  Å) matches Si ( $\sim 5.431$  Å)<sup>14</sup>. Haugrud *et al.* reported that Ca-doped  $Tb_2O_3$  is a p-type semiconductor<sup>15</sup>. In particular,  $Tb^{3+}$  ions have been widely used as ideal activator ions for green display devices, detectors, and lasers<sup>16</sup>. In these applications, different emission wavelengths have been achieved owing to the energy transfer between  $Tb^{3+}$  and other RE ions (such as  $Eu^{3+}$ ,  $Sm^{3+}$ ,  $Nd^{3+}$ ,  $Yb^{3+}$ )<sup>17–20</sup>. However, so far, there has been no reports on electroluminescence (EL) of  $Tb_2O_3$ .

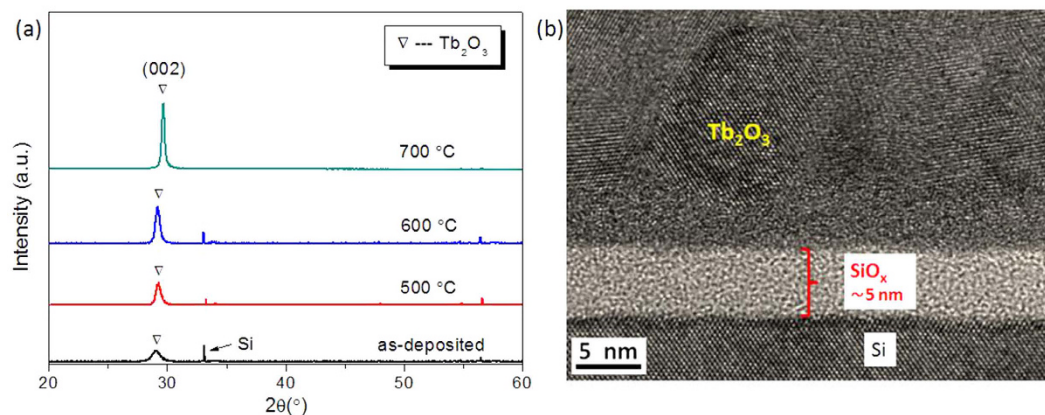
Here we demonstrate, for the first time, new green LEDs using  $Tb_2O_3$  on Si substrates.  $Tb_2O_3$  thin films were prepared by magnetron sputtering and annealed in an  $O_2$  ambient. Strong green emission was observed, with a turn-on voltage of 8 V. EL mechanisms of the  $Tb_2O_3$  LEDs are also discussed. In addition, intense red, orange, and near IR emissions are obtained from RE<sup>3+</sup> ( $Eu^{3+}$ ,  $Sm^{3+}$  and  $Yb^{3+}$ ) doped  $Tb_2O_3$  LEDs on Si substrates, respectively. Our results show that  $Tb_2O_3$  LEDs can be potentially used in display, optical communication, and other Si-based optoelectronics.

### Results and Discussion

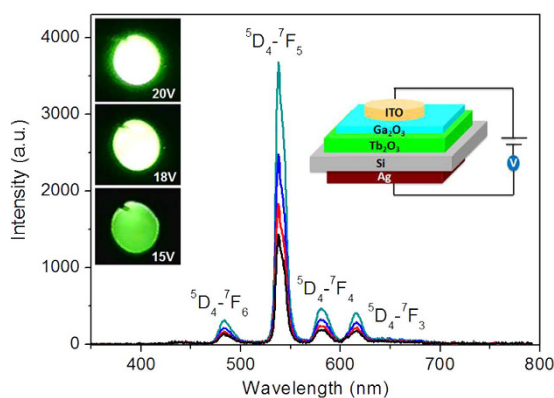
The crystalline phases of as-deposited and annealed  $Tb_2O_3$  films were investigated by X-ray diffraction (XRD). As shown in Fig. 1(a), the as-deposited film shows a weak diffraction peak at  $2\theta \approx 29.0^\circ$ , which corresponds to (002) plane of hexagonal  $Tb_2O_3$ . This peak increases and narrows with increasing the annealing temperature. Strong peak of (002) plane of  $Tb_2O_3$  is observed when the annealing temperature is further raised to above 700 °C. The lattice structure of the films is investigated by using a transmission electron microscope (TEM). The high-magnification TEM image shows the presence of crystalline areas in the  $Tb_2O_3$  film. As shown in Fig. 1(b), there is an amorphous  $SiO_x$  layer at the  $Tb_2O_3$ /Si interface, with a thickness of about 5 nm.

A  $Tb_2O_3$  LED was fabricated with the structure diagram shown in the inset of Fig. 2. Intense green EL is observed when a positive voltage is applied on the indium tin oxide (ITO) layer, while no EL is detectable under reverse biases. The turn-on voltage of the device is as low as 8 V. The emission is bright enough to be observed by naked eyes under normal room light. The EL spectrum shows peaks at 484, 540, 582, and 616 nm, which correspond to  $^5D_4-^7F_6$ ,  $^5D_4-^7F_5$ ,  $^5D_4-^7F_4$ , and  $^5D_4-^7F_3$  transitions of  $Tb^{3+}$ , respectively<sup>16</sup>. The  $^5D_4-^7F_5$  transition is the most intense one and features a double-peak structure, which can be attributed to the crystal field splitting of the ground state. When increasing the forward bias, the EL spectral shape remains unchanged, with the intensity increasing with the applied voltage from 8 V up to 20 V. The forward current of the device reaches 2.8 mA when

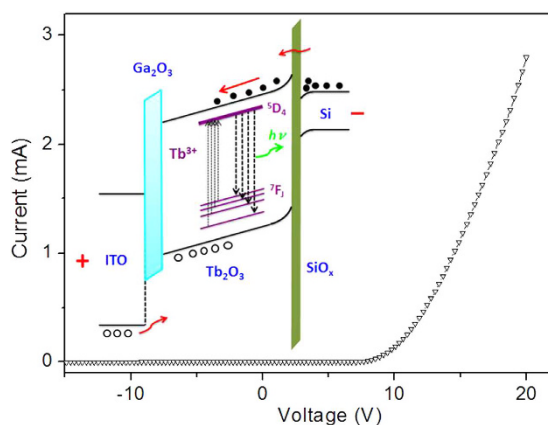
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**Figure 1. Lattice structure of the films.** (a) XRD of as-deposited and annealed  $\text{Tb}_2\text{O}_3$  films ( $\sim 200$  nm) at 500, 600, and 700 °C for 1 hour, respectively. (b) High-magnification TEM image of a  $\text{Tb}_2\text{O}_3$  film.



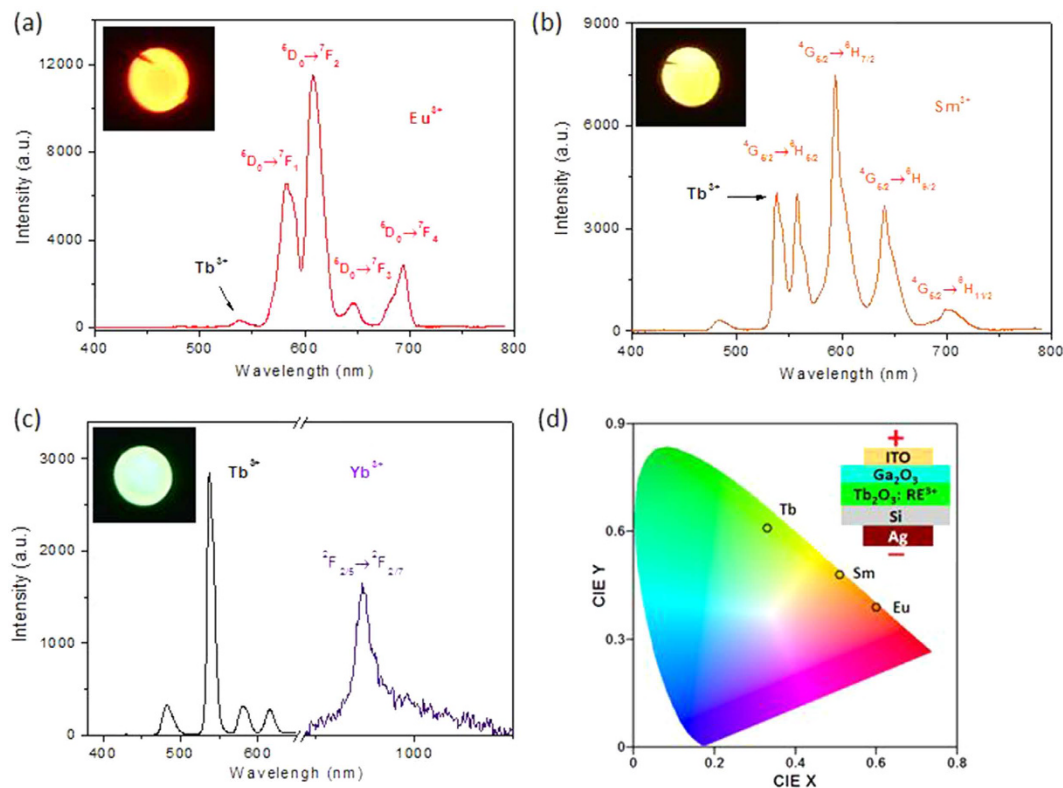
**Figure 2. EL performance of the  $\text{Tb}_2\text{O}_3$  LED.** EL spectra of the LED at the voltage of 12–20 V, the insets show the structure diagram of the LED and EL photos of the device at different voltages.



**Figure 3. Electrical characterization of  $\text{Tb}_2\text{O}_3$  LED.** Current-voltage characteristic of the device. The inset shows the energy band diagram of the devices and the charge transfer process.

the forward bias is 20 V, while the reverse leakage current is minimal. These results show that the  $\text{Tb}_2\text{O}_3$  LED has excellent rectification performance.

The EL mechanism is schematically illustrated in the inset of Fig. 3. When a sufficiently high forward bias is applied, the energy bands of both  $\text{Tb}_2\text{O}_3$  and  $\text{SiO}_x$  bend upward along the electric field direction. According to Zhu *et al.*<sup>21–23</sup>, a trap-assisted tunneling (TAT) mechanism dominates the conduction mechanism at the EL-enabling voltages. When a sufficiently high forward bias voltage is applied between the two electrodes, a large number of electrons in Si accumulate in Si/SiO<sub>x</sub> interface and then reach the conduction band of  $\text{Tb}_2\text{O}_3$  by



**Figure 4.** EL performance of the LEDs. (a–c) EL spectra of the red-, orange- and near IR-emitting devices from  $\text{Eu}^{3+}$ ,  $\text{Sm}^{3+}$ ,  $\text{Yb}^{3+}$  doped LED at forward biases of 20 V, respectively. (d) CIE coordinates of the green-, red- and orange-emitting devices.

tunneling through the  $\text{SiO}_x$  barrier. Meanwhile, holes are injected from ITO electrode into the  $\text{Ga}_2\text{O}_3$  layer and then enter the  $\text{Tb}_2\text{O}_3$ . The holes accumulate at  $\text{Tb}_2\text{O}_3/\text{SiO}_x$  interface due to the  $\text{SiO}_x$  barrier. Thus, direct impact excitation of  $\text{Tb}^{3+}$  ions are exerted when the kinetic energies exceeds the threshold energy.

We have demonstrated a bright green EL device based on  $\text{Tb}_2\text{O}_3$ . To further explore the possibility of using  $\text{Tb}_2\text{O}_3$  as a host material to RE ions to achieve devices of other colors, we further fabricated EL devices with RE-doped  $\text{Tb}_2\text{O}_3$ . Intense red EL is observed from  $\text{Eu}^{3+}$  doped  $\text{Tb}_2\text{O}_3$  LED. As shown in Fig. 4(a), the emission peaks are at about 582, 619, 646, and 694 nm, corresponding to  ${}^5\text{D}_0-{}^7\text{F}_j$  ( $J = 1, 2, 3,$  and  $4$ ) transitions of  $\text{Eu}^{3+}$ . The highest peak at 619 nm corresponds to the  $\text{Eu}^{3+}$  electric dipole transitions of  ${}^5\text{D}_0-{}^7\text{F}_2$ <sup>17</sup>. In Fig. 4(b), strong orange EL is observed from  $\text{Sm}^{3+}$  doped  $\text{Tb}_2\text{O}_3$  LED. The characteristic of the  ${}^4\text{G}_{5/2}-{}^6\text{H}_j$  ( $J = 5/2, 7/2, 9/2,$  and  $11/2$ ) transitions of  $\text{Sm}^{3+}$  ions are appeared. The  $\text{Sm}^{3+}$  emission peaks are from transitions of  ${}^4\text{G}_{5/2}-{}^6\text{H}_{5/2}$  (558 nm),  ${}^4\text{G}_{5/2}-{}^6\text{H}_{7/2}$  (593 nm),  ${}^4\text{G}_{5/2}-{}^6\text{H}_{9/2}$  (640 nm), and  ${}^4\text{G}_{5/2}-{}^6\text{H}_{11/2}$  (701 nm)<sup>18</sup>. In Fig. 4(c), both green emission of  $\text{Tb}^{3+}$  and near IR emission of  $\text{Yb}^{3+}$  are obtained. The characteristic peak of  $\text{Yb}^{3+}$  is attributed to the transition from  ${}^2\text{F}_{5/2}$  to  ${}^2\text{F}_{7/2}$ <sup>19</sup>. As shown in Fig. 4(d), The CIE coordinates of the green-, red- and orange-emitting devices are (0.33, 0.61), (0.60, 0.39) and (0.51, 0.48), respectively.

In summary, new LEDs from Si-based  $\text{Tb}_2\text{O}_3$  are fabricated. Intense green EL was observed, with a turn-on voltage of about 8 V. The green emission centered at 484, 540, 582, and 616 nm, corresponding to the  ${}^5\text{D}_4-{}^7\text{F}_j$  transitions of  $\text{Tb}^{3+}$  in  $\text{Tb}_2\text{O}_3$ , where  $J = 6, 5, 4,$  and  $3$ . The EL intensity increases with the applied voltage up to 20 V. In addition, red, orange, and near infrared EL were observed from  $\text{RE}^{3+}$  ( $\text{Eu}^{3+}$ ,  $\text{Sm}^{3+}$  and  $\text{Yb}^{3+}$ ) doped  $\text{Tb}_2\text{O}_3$  LEDs, respectively. Our results could provide a possible route for achieving stable and highly efficient Si-based LEDs.

## Methods

About 200 nm  $\text{Tb}_2\text{O}_3$  films and RE-doped  $\text{Tb}_2\text{O}_3$  films were deposited on n-type Si (100) substrates by magnetron co-sputtering technique. The Si substrates were cleaned by dipping in a dilute HF solution ( $\text{HF}:\text{H}_2\text{O} = 1:7$ ) for 60 s. Tb (99.95%) target was sputtered in  $\text{Ar}:\text{O}_2 = 15:5$  atmosphere, at a substrate temperature of 150 °C. The deposition rate was 0.4 Å/s. RE ions (RE = Sm, Eu, and Yb) were doped in  $\text{Tb}_2\text{O}_3$  films by sputtering with Sm (99.95%), Eu (99.95%) and Yb (99.95%) targets, respectively.  $\text{Ga}_2\text{O}_3$  layer (~20 nm) was deposited by sputtering with  $\text{Ga}_2\text{O}_3$  target. The as-deposited samples were annealed in  $\text{O}_2$  ambient at 500, 600, or 700 °C for 1 hour, respectively. We fabricated the LEDs as schematically illustrated in the inset of Fig. 2. ITO and Ag electrodes were deposited on the surface of the film and the back side of the Si substrate, respectively, both by magnetron sputtering.

The crystal structure characterization was carried out by using Bruker D8 ADVANCE XRD with Cu-K $\alpha$  radiation, and the morphology of the samples was determined by TEM (Hitachi, H8100 200 kV). The EL spectra of the devices and I–V characteristics were measured by a system of an ACTON 150 CCD spectrometer and a Keithley 2410 source meter, respectively.

## References

- Hirschman, K. D., Tsybeskov, L., Duttagupta, S. P. & Fauchet, M. Silicon-based visible light-emitting devices integrated into microelectronic circuits. *Nature* **384**, 338–341 (1996).
- Pavesi, L., Negro, L. D., Mazzoleni, C., Franzò, G. & Priolo, F. Optical gain in silicon nanocrystals. *Nature* **408**, 440–444 (2000).
- Yuan, C. K., Anthony, R., Kortshagen, U. R. & Holmes, R. J. High-efficiency silicon nanocrystal light-emitting devices. *Nano Lett.* **10**, 1952–1956 (2011).
- Nguyen, H. P. T. *et al.* p-Type modulation doped InGaN/GaN dot-in-a-Wire white-light-emitting diodes monolithically grown on Si(111). *Nano Lett.* **11**, 1919–1924 (2011).
- Hsieh, Y. P. *et al.* Electroluminescence from ZnO/Si-nanotips light-emitting diodes. *Nano Lett.* **9**, 1839–1843 (2009).
- Triviño, N. V. *et al.* Integrated photonics on silicon with wide bandgap GaN semiconductor. *Appl. Phys. Lett.* **102**, 081120 (2013).
- Izeddin, I., Moskalenko, A. S., Yassievich, I. N., Fujii, M. & Gregorkiewicz, T. Nanosecond Dynamics of the Near-Infrared Photoluminescence of Er-Doped SiO<sub>2</sub> Sensitized with Si Nanocrystals. *Phys. Rev. Lett.* **97**, 207401 (2006).
- Li, L. *et al.* Investigation on white light emissions from CeO<sub>2</sub>/Dy<sub>2</sub>O<sub>3</sub> multilayer films based on silicon substrates. *Vacuum* **112**, 38–41 (2015).
- Li, L. *et al.* Investigation on photoluminescence properties of CeO<sub>2</sub>/Sm<sub>2</sub>O<sub>3</sub> multilayer films based on Si substrates. *Phys. Status Solidi B*. **251**, 737–740 (2014).
- Sun, J. M. *et al.* Efficient ultraviolet electroluminescence from a Gd-implanted silicon metal-oxide-semiconductor device. *Appl. Phys. Lett.* **85**, 3387–3389 (2004).
- Rebohle, L. *et al.* Strong electroluminescence from SiO<sub>2</sub>-Tb<sub>2</sub>O<sub>3</sub>-Al<sub>2</sub>O<sub>3</sub> mixed layers fabricated by atomic layer deposition. *Appl. Phys. Lett.* **104**, 251113 (2014).
- Li, L. *et al.* A novel violet/blue light-emitting device based on Ce<sub>2</sub>Si<sub>2</sub>O<sub>7</sub>. *Sci. Rep.* **5**, 16659 (2015).
- Polman, A. *et al.* Erbium in crystal silicon limits. *J. Appl. Phys.* **77**, 1256–1262 (1995).
- Svetlana, V. *et al.* MOCVD synthesis of terbium oxide films and their optical properties. *Chem. Vap. Deposition*, **21**, 150–155 (2015).
- Haugsrud, R., Larring, Y. & Norby, T. Proton conductivity of Ca-doped Tb<sub>2</sub>O<sub>3</sub>. *Solid State Ionics*, **176**, 2957–2961 (2005).
- Kulakcia, M. & Turan, R. Improvement of light emission from Tb-doped Si-based MOS-LED using excess Si in the oxide layer. *J. Lumin.* **137**, 37–42 (2013).
- Gupta, A., Brahme, N. & Bisen, D. P. Electroluminescence and photoluminescence of rare earth (Eu, Tb) doped Y<sub>2</sub>O<sub>3</sub> nanophosphor. *J. Lumin.* **155**, 112–118 (2014).
- Mani, K. P. *et al.* Spectroscopic investigation on tunable luminescence by energy transfer in Tb<sub>2-x</sub>Sm<sub>x</sub>(MoO<sub>4</sub>)<sub>3</sub> nanophosphors. *Opt. Mater.* **42**, 237–244 (2015).
- Terra, I. A. A. *et al.* Analysis of energy transfer processes in Yb<sup>3+</sup>-Tb<sup>3+</sup> co-doped, low-silica calcium aluminosilicate glasses. *J. Appl. Phys.* **110**, 083108 (2011).
- Charu, C. D., Joshi, B. C., Upreti, D. K. & Bhawana, K. Non-radiative energy transfer from Tb<sup>3+</sup> to Ho<sup>3+</sup> ions in zinc phosphate glass. *Indian J. Pure Appl. Phys.* **49**, 398–400 (2011).
- Zhu, C. *et al.* Multicolor and near-infrared electroluminescence from the light-emitting devices with rare-earth doped TiO<sub>2</sub> films. *Appl. Phys. Lett.* **107**, 131103 (2015).
- Ly, C. *et al.* Electroluminescence from metal-oxide-semiconductor devices with erbium-doped CeO<sub>2</sub> films on silicon. *Appl. Phys. Lett.* **106**, 141102 (2015).
- Rebohle, L. *et al.* Strong electroluminescence from SiO<sub>2</sub>-Tb<sub>2</sub>O<sub>3</sub>-Al<sub>2</sub>O<sub>3</sub> mixed layers fabricated by atomic layer deposition. *Appl. Phys. Lett.* **104**, 251113 (2014).

## Acknowledgements

This work was financially supported by the National Science Foundation of China (Grant no. 60977017, 61275058) and the Fundamental Research Funds for the Central Universities (2013JBM101, 2015JBC028).

## Author Contributions

L. X. Y. conceived the idea and designed the study. L. L. carried out the experiment and prepared the manuscript. L. L., S. W. W., G. Y. M., X. Y. and L. X. Y. discussed the results and commented the manuscript.

## Additional Information

**Competing financial interests:** The authors declare no competing financial interests.

**How to cite this article:** Li, L. *et al.* Multicolor light-emitting devices with Tb<sub>2</sub>O<sub>3</sub> on silicon. *Sci. Rep.* **7**, 42479; doi: 10.1038/srep42479 (2017).

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