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OPEN First-principles calculated decomposition pathways for LiBH₄ nanoclusters

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We analyze thermodynamic stability and decomposition pathways of LiBH₄ nanoclusters using grandcanonical free-energy minimization based on total energies and vibrational frequencies obtained from density-functional theory (DFT) calculations. We consider (LiBH₄)_n nanoclusters with n = 2 to 12 as reactants, while the possible products include (Li)_n, (B)_n, (LiB)_n, (LiH)_n, and Li₂B_nH_n; off-stoichiometric $Li_n B_n H_m$ ($m \le 4n$) clusters were considered for n = 2, 3, and 6. Cluster ground-state configurations have been predicted using prototype electrostatic ground-state (PEGS) and genetic algorithm (GA) based structural optimizations. Free-energy calculations show hydrogen release pathways markedly differ from those in bulk LiBH₄. While experiments have found that the bulk material decomposes into LiH and B, with $L_{12}B_{12}H_{12}$ as a kinetically inhibited intermediate phase, (LiBH₄)_n nanoclusters with n < 12are predicted to decompose into mixed $Li_n B_n$ clusters via a series of intermediate clusters of $Li_n B_n H_m$ $(m \le 4n)$. The calculated pressure-composition isotherms and temperature-pressure isobars exhibit sloping plateaus due to finite size effects on reaction thermodynamics. Generally, decomposition temperatures of free-standing clusters are found to increase with decreasing cluster size due to thermodynamic destabilization of reaction products.

Hydrogen is a promising next-generation energy carrier due to its high energy density, high energy conversion efficiency, and absence of harmful emissions. However, on-board hydrogen storage is still a serious challenge for developing economically viable hydrogen-powered passenger vehicles¹. To achieve widespread commercialization, hydrogen storage systems should simultaneously possess several characteristics such as safety, high gravimetric and volumetric densities, fast reversible hydrogen release and uptake under moderate pressures and temperatures matched to the operating conditions of proton exchange membrane (PEM) fuel cells.

Complex metal hydrides emerged as feasible high-density hydrogen storage materials after Bogdanović and Schwickardi demonstrated reversible (de)hydrogenation reactions in transition metal doped sodium alanate $(NaAlH_4)^2$. Spurred by this discovery, many other complex hydrides have been explored as candidate hydrogen storage materials, including alanates, amides, and borohydrides. Lithium borohydride, LiBH₄, has received particular attention owing to its high gravimetric ($18.5 \text{ wt.}\% \text{ H}_2$) and volumetric ($121 \text{ g H}_2/\text{L}$) densities. However, due to its high thermodynamic stability and slow kinetics, hydrogen release from bulk LiBH₄ requires very high temperatures that are incompatible with proton-exchange membrane (PEM) fuel cells³⁻⁵. In addition, the tendency to release diborane during dehydrogenation leads to irreversibility and causes poisoning of the fuel cell⁶⁻¹⁰.

Destabilization has been extensively explored as a means of improving the thermodynamics of hydrogen storage reactions¹¹⁻¹⁴. The main idea of this approach is to add a second reactant, which upon decomposition forms a stable low-energy product phase and lowers the overall reaction enthalpy. The canonical example of a destabilized reaction is $LiBH_4 + MgH_2 \rightarrow LiH + MgB_2 + (5/2)H_2$, where the addition of MgH₂ leads to the formation of MgB₂ as a low-energy product phase. This significantly lowers the reaction enthalpy relative to the decomposition reaction of the pure compound, LiBH4 \rightarrow LiH + B + (3/2)H₂. Unfortunately, destabilization is not very effective in lowering the kinetic barriers to hydrogen release, leading to only modest improvement in the hydrogen release temperatures.

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Another approach to improving the thermodynamics and kinetics of hydrogen release consists of using LiBH₄ nanoparticles incorporated in support materials with nanoscale pores, such as nanoporous carbon or metal-organic frameworks (MOFs)^{9,10,15-24}. Besides improving the kinetics and lowering the temperature of hydrogen release, this strategy has also been shown to suppress the formation of diborane^{19,25,26}. First-principles density-functional theory (DFT) based studies^{27,28} have investigated the thermodynamic properties of nano-LiBH₄, suggesting that bonding with the nanoporous carbon support plays a key role in improving the thermodynamic properties of nano-LiBH₄. However, due to the large configuration space, these studies did not perform comprehensive examination of the most stable cluster structures and compositions as functions of cluster size. Hence, a complete picture of the size-dependent thermodynamics properties of LiBH₄ nanoparticles and their decomposition products is not yet available.

Decomposition pathways and pressure-composition isotherms are determined using free energy minimization in the grand canonical ensemble²⁹. In addition to clusters with the chemical composition of bulk LiBH₄ and its predicted bulk decomposition products Li₂B₁₂H₁₂, LiH and B³⁰, we considered (Li)_n, (LiB)_n, and Li₂B_nH_n clusters as possible products. To assess the role of other cluster compositions, further detailed studies were performed for the decomposition reactions of $(\text{LiBH}_4)_2$, $(\text{LiBH}_4)_3$ and $(\text{LiBH}_4)_6$ including $\text{Li}_2\text{B}_2\text{H}_n$ (n = 1 to 7), $\text{Li}_3\text{B}_3\text{H}_n$ (n = 1to 11) and $Li_6B_6H_n$ (n = 1 to 23) clusters as possible intermediate products. Our analysis shows that the reaction end products are changed from $(\text{LiH})_n + B_n + (3/2)H_2$ to $(\text{LiB})_n + 2nH_2$ when the size of reactants is reduced to the nanoscale. The calculated reaction enthalpies decrease from 238 (n = 2) to 133 kJ/mol H₂ (n = 12) with increasing cluster size, remaining significantly above the bulk decomposition enthalpy of LiBH₄. The hydrogen release temperatures are predicted to behave in a non-trivial way, with multiple intermediates appearing in the simulated dehydrogenation processes. The finite size effects and presence of intermediates manifest in multiple plateaus in the calculated release curves as functions of temperature, while the calculated pressure-composition isotherms and temperature-composition isobars exhibit sloping plateaus. Our study provides an in-depth understanding of the thermodynamics of hydrogen release from nanoscale particles and shows the importance of engineering appropriate support materials that can bind the reaction products and lower the reaction enthalpies to achieve reversible hydrogen storage in a practically viable range of temperatures and pressures.

Results

Structure of reactants and products. We start by briefly describing our findings for the atomic geometries of the reactants and products, which are summarized in Fig. 1. The reactants, LiBH₄ nanoclusters, are composed of the cation, Li⁺, and the anionic group, BH₄⁻. We find that the lowest energy conformations of small (LiBH₄)_n clusters are chainlike, forming closed rings. The tetrahedral BH₄⁻ anion is a near perfect tetrahedron in all cases with some elongation in the B-H bond length from 1.22 to 1.24 Å. Note that the length of the B-H bond in bulk LiBH₄ is 1.22 Å. The diameter of LiBH₄ clusters increases gradually from 6.0 to 10.4 Å upon increasing the cluster size from n=2 to 12. (LiH)_n clusters form rocksalt-derived structures for sizes $n \ge 4$.

For boron clusters, our GA method reproduced the lowest energy structures found in earlier studies^{31–33}. Small boron clusters are found to form 2-dimensional planar structures except for the B₉ cluster which has two boron atoms on two sides of the circular substructure with seven boron atoms forming a ring. Changes in the cluster geometries with increasing number of boron atoms indicate that boron nanoclusters prefer to have a coordination number of six.

For pure lithium clusters, the GA again finds the same structures as those in previous studies^{34,35}. Unlike boron clusters, lithium clusters agglomerate in three-dimensional motifs when the the cluster sizes exceed four atoms.

In the mixed lithium-boron clusters (LiB)_n, the boron atoms generally tend to gather in the center of the cluster with lithium atoms surrounding them on the periphery. In addition, for $n \le 5$ the arrangement of boron atoms is the same as in the pure boron clusters. When n > 5, boron atoms in the center are distorted due to interactions with the surrounding lithium atoms. We also found that the B-B distances in the center of the (LiB)_n clusters are gradually increased from 1.53 Å for n = 2 to 1.85 Å for n = 12, while the Li-B bond lengths stay approximately constant. It appears that interactions with the outer Li ions and charge transfer from Li to B combine to push the B atoms outward.

Cluster energies. To better understand the thermodynamic stability, reaction pathways and reaction enthalpies, we first discuss the cluster-size dependent energetics of reactants and reaction products. Figure 2 shows the relative energies of clusters normalized by their bulk energies without the zero-point energy (ZPE) corrections. This quantity is defined as the ratio of the energy of the cluster per formula unit versus the energy of the bulk phase. Bulk lithium is calculated in the $Im\overline{3}m$ body-centered cubic lattice, bulk boron is in the α -B phase, and bulk LiH is in the rocksalt structure. Bulk LiBH₄ has several near-degenerate ground-state structures³⁶, and the Pnma structure is used in our calculations³⁷. As for bulk LiB, we used a P6₃/mmc structure reported in a previous experimental study³⁸. The calculated reaction enthalpy for the decomposition of bulk LiBH₄ at T = 0 K is $\Delta H = 60 \text{ kJ/(mol H_3)}$ with zero-point energy (ZPE) corrections, while excluding ZPE we obtain 82 kJ/(mol H_2). The predicted decomposition temperature is $T_{dec} = 340$ °C at p = 1 bar hydrogen pressure, which is about 100 °C below experimental results. We find that the stabilities of the reaction products are dramatically reduced with decreasing cluster size, with LiH and LiB clusters being more stable than pure Li and B clusters. However, the behavior of the reactant LiBH₄ clusters is very different: the energy per formula unit of $(LiBH_4)_n$ is nearly flat from n=2 to n=12, which signals that their thermodynamic stability is reduced only slightly upon decreasing cluster size. This suggests that most of the binding energy is stored in the polar covalent B-H bonds, while the electrostatic interactions between lithium and BH_4^- complexes are relatively weak. Examination of the size-dependent stabilities of each cluster type shows that clusters with even-numbered formula units are generally more stable than the odd ones. For pure clusters, Li_n and B_n, we find that Li₃, Li₅, Li₉, Li₁₁, B₇, B₉, and B₁₁ possess certain geometric symmetries, but they are unstable with respect to decomposition into even-numbered n+1 and n-1

B _n	Lin	(LiH) _n	(LiBH ₄) _n	(LiB) _n	Li ₂ (BH) _n
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11					
12				X	

Figure 1. First-principles relaxed cluster geometries found in this work. B, Li, and H atoms are the blue, green and yellow spheres, respectively.

clusters due to unpaired electron filling in the highest occupied molecular orbital. As for the LiH and LiB clusters, even though the highest orbitals in $(LiH)_5$, $(LiH)_7$, $(LiH)_{11}$, and $(LiB)_9$ are fully occupied, these clusters are unstable because of broken geometric symmetry. Moreover, $(LiH)_{10}$ and $(LiB)_5$ are also unstable because they are at the structural transition point between distinct bonding topologies.

Decomposition pathways of LiBH₄ **clusters.** Previous computational work³⁰ has shown that, thermodynamically, decomposition of bulk LiBH₄ should occur via the formation of an intermediate closoborane $\text{Li}_2\text{B}_{12}\text{H}_{12}$ compound. However, the formation of closoboranes is kinetically inhibited in experiments and decomposition proceeds directly to a mixture of LiH and B. We begin our discussion of the decomposition pathways of LiBH₄ nanoclusters by considering simplified decomposition reactions with closoborane $\text{Li}_2\text{B}_n\text{H}_n$, elemental (Li_n and (B)_n, and binary (LiH)_n and (LiB)_n clusters as the end products; the full treatment using the ensemble grand canonical formalism is given below. First, we consider the following reaction pathways:

$$\left(\text{LiBH}_{4}\right)_{n} \rightarrow \left(\text{LiH}\right)_{n} + B_{n} + \frac{3n}{2}H_{2},\tag{1}$$

$$\left(\mathrm{LiBH}_{4}\right)_{n} \to \mathrm{Li}_{n} + \mathrm{B}_{n} + 2n\mathrm{H}_{2},\tag{2}$$

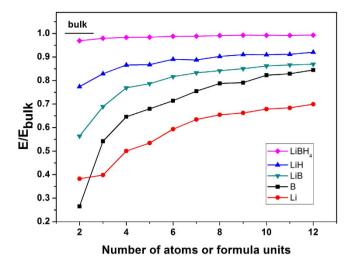


Figure 2. Calculated DFT total energies (without zero-point energy corrections) of $(\text{LiBH}_4)_n$, $(\text{LiH})_n$, $(\text{LiB})_n$, B_n and Li_n clusters, normalized to the corresponding bulk values.

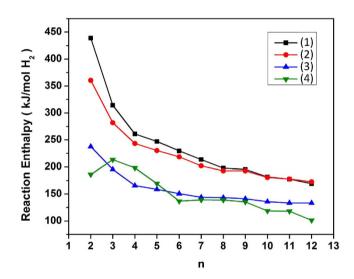


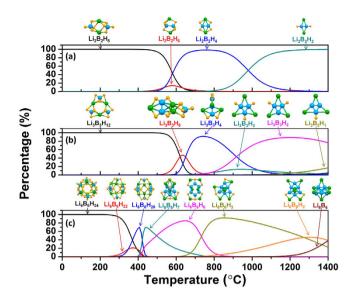
Figure 3. Calculated DFT reaction enthalpies (without ZPE correction) of the four decomposition pathways of nano-LiBH₄ given by Eqs 1–4, shown as functions of the cluster size n.

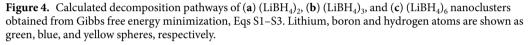
$$(\text{LiBH}_4)_n \to (\text{LiB})_n + 2n\text{H}_2, \tag{3}$$

$$(\text{LiBH}_4)_n \to \text{Li}_2\text{B}_n\text{H}_n + (\text{LiH})_{n-2} + (n+1)\text{H}_2. \tag{4}$$

Equations 1 and 2 are the nanocluster analogues of the bulk decomposition reactions occurring at T = 400 and 900 °C, respectively³. Reaction 4 is the nanocluster analogue of the bulk decomposition reaction into Li₂B₁₂H₁₂, accounting for the formation of various closoborane $(B_nH_n)^{2-}$ species with $n \le 12$, while the product of Eq. 1 can be obtained by further decomposition of a mixture of Li₂B_nH_n and $(LiH)_{n-2}$. Finally, Eq. 3, is similar to the favored decomposition reaction predicted for nano-NaAlH₄²⁹.

The calculated DFT reaction enthalpies of Eqs 1–4 without vibrational free energy contributions are shown in Fig. 3. In all cases, the calculated DFT reaction enthalpies are larger than the decomposition enthalpy of bulk LiBH₄ [82 kJ/(mol H₂)]. This does not mean that LiBH₄ nanoclusters are generally more stable with respect to decomposition than the bulk compound because these data are based on a limited number of possible reactions and reaction products; we will show below that the Gibbs free energy minimization approach and inclusion of product clusters with intermediate hydrogen content can lower the reaction enthalpies significantly. The trend towards higher reaction enthalpies for the hypothesized reactions in Eqs 1–4 can be easily explained from the size-dependence of the cluster binding energies in Fig. 2: since the (LiBH₄)_n clusters show weak size dependence, while all the product clusters show much stronger finite size energy penalty, the reaction enthalpies are significantly higher than the bulk limit. It is interesting to note that of all four reaction pathways, the bulk-like Eq. 1 is





the least favorable one in nano-LiBH₄, which shows that the reaction path can change dramatically upon reducing the particle size to the nanometer regime; similar effects have been predicted for sodium alanate^{29,39}. Both pathways in Eqs 2 and 3 liberate all the hydrogen, and our results show that Eq. 3 is more favorable than Eq. 2 because the binary (LiB)_n clusters are more stable than the mixture of (Li)_n and (B)_n clusters. We note that the formation of Li₂B_nH_n clusters via Eq. 4 is the preferred path for n = 6 to 12 when closoborane species form as the end product, while the pathway in Eq. 3 is preferred for n = 3 to 5. In general, we find that the calculated reaction enthalpies decrease with increasing cluster size *n*, and are significantly higher than the calculated DFT enthalpies for decomposition of bulk-LiBH₄ into Li₂B₁₂H₁₂³⁰. Due to the high stability of B₁₂H₁₂²⁻ complex anions, the reaction enthalpy of Eq. 4 drops significantly from n = 11 to n = 12, and one may expect that these species will persist as the dominant end product for n > 12.

To investigate the possibility that intermediate products other than those in Eqs 1–4 may form during decomposition, we have carried out calculations for clusters with the general formula $\text{Li}_n B_n H_m$, where 0 < m < 4n and n = 2, 3, and 6. For this series, we adopted the random search scheme rather than the genetic algorithm. The random search scheme is implemented as follows. New initial structures were randomly generated by removing one or more H atoms from nano-LiBH₄ and by gradually adding H atoms to LiB nanoclusters, breaking symmetries by slightly distorting all atoms away from the original positions. These randomly generated structures were then relaxed using DFT forces to a local minimum configuration, and the energies of the most favorable structures at each composition were used in the Gibbs ensemble formalism given by Eqs S1–S3 (See Supplementary Information). The zero point energy, temperature-dependent vibrational free energy, and the free energy of the diatomic H₂ gas are included in these calculations. Our results for the cluster type fractions p_f are summarized in Fig. 4 assuming a hydrogen pressure of p = 0.01 bar, which is representative of experimental measurements.

Figure 4(a) shows the calculated dehydrogenation pathway of $(\text{LiBH}_4)_2$ and the structures of the predicted intermediate products as functions of temperature. $(\text{LiBH}_4)_2$ nanoclusters start to release hydrogen at approximately 450 °C by first forming $\text{Li}_2\text{B}_2\text{H}_6$. The structure of this cluster resembles diborane, with a significant difference that the donation of two electrons from Li atoms reconfigures the three-center two-electron bonds of diborane into a direct B-B bond. At 500 °C, $\text{Li}_2\text{B}_2\text{H}_4$ clusters form and exist as the dominant specie from 580 to 980 °C. Finally, $\text{Li}_2\text{B}_2\text{H}_2$ clusters appear at 800 °C and dominate until full dehydrogenation into (LiB)₂ as the end product above 1400 °C (not shown). We note that the intermediate products in the dehydrogenation pathway of (LiBH_4)₂ all contain an even number of hydrogen atoms, and each intermediate reaction step releases one H₂ molecule per cluster.

Figure 4(b) shows the decomposition pathway of $(\text{LiBH}_4)_3$. It is seen that hydrogen release starts at 530 °C where two H₂ molecules per cluster are released to form Li₃B₃H₈, which exists in a narrow temperature interval. The latter cluster is essentially a superposition of Li₂B₂H₄ and $(\text{LiBH}_4)_1$ clusters, which explains its marginal stability. In the next step, two more H₂ molecules are released to form Li₃B₃H₄, bypassing Li₃B₃H₆ entirely. At temperatures of 920 °C and higher, a Li₃B₃H₂ cluster is formed by releasing another H₂ molecule, before full dehydrogenation to the end product Li₃B₃. Small fraction of Li₃B₃H₃ and Li₃B₃H₁ clusters with odd-numbered hydrogen atoms is predicted to appear intermittently.

Finally, Fig. 4(c) shows the dehydrogenation pathway of $(LiBH_4)_6$. Even-numbered $Li_6B_6H_{22}$ and $Li_6B_6H_{20}$ clusters stage a brief appearance at 250 and 300 °C, respectively. The structures of these clusters can be described as derived from the parent borohydride cluster by releasing a hydrogen molecule and forming a $B_2H_6^{2-}$ complex anion. It is tempting to hypothesize that $2BH_4^{2-} \rightarrow B_2H_4^{2-}$ also represents a *kinetically* viable hydrogen release

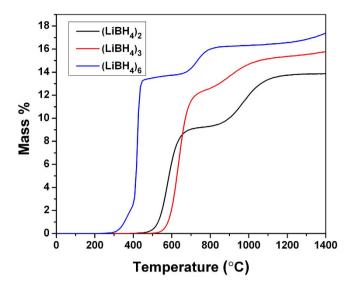


Figure 5. Mass fraction of released hydrogen as a function of temperature at constant hydrogen pressure p = 0.01 bar.

pathway, but this has to be verified by separate studies of the activation energies. In any case, the negative charge carried by the BH_4^- anions should present a significant barrier to bringing these complexes together for the reaction. Most of the hydrogen is released around 420 °C with the appearance of $Li_6B_6H_7$ and $Li_6B_6H_6$ clusters. The latter cluster exhibits a high degree of symmetry and appears to be stable over a wide range of temperatures up to approximately 700 °C. In the final reaction steps, $Li_6B_6H_3$, $Li_6B_6H_2$ and Li_6B_6 show up one by one as the majority cluster phases. Again, it is seen that most of the stable clusters in the decomposition sequence contain an even number of hydrogen atoms, with the brief appearance of $Li_6B_6H_7$ as the only exception.

Based on the data in Fig. 4, we have calculated the cumulative mass fraction of released H_2 as a function of temperature at constant pressure (i.e., temperature-composition isobar). Figure 5 shows the results of these calculations. It is seen that the calculated hydrogen release curves exhibit nontrivial behavior as functions of temperature and cluster size.

For instance, the initial onset of hydrogen release occurs at a lower temperature for the n = 2 cluster than for the n = 3 cluster. This is due to the fact that the n = 2 cluster can release one H₂ molecule and form Li₂B₂H₆ (which consists of two Li⁺ ions and a B₂H₆²⁻ anion), while the corresponding Li₃B₃H₁₀ compound is unfavorable because it involves a mixture of B₂H₆²⁻ and BH₄⁻ anions. Furthermore, all three clusters exhibit two sloping plateaus in their dehydrogenation curves. Focusing on the n = 6 cluster, most of the hydrogen comes off in a narrow temperature interval before the first plateau, corresponding to successive removal of hydrogen molecules and eventual formation of Li₂B₆H₆ with an overall formula Eq. 4. In the second stage, a smaller amount of hydrogen is released by decomposing Li₂B₆H₆ and forming Li₂B₆H₃, etc. With increasing cluster size of nano-LiBH₄, we expected that the amount of hydrogen released in the first-stage will increase and that the main hydrogen release step will shift to lower temperatures. According to the reaction enthalpies in Fig. 3, the overall reaction for $n \le 12$ is expected to proceed according to Eq. 4, even though it seems likely that the full pathway will consist of multiple intermediate steps, similar to those discussed above. For n > 12, we hypothesize that the formation of clusters with Li-compensated B_mH_m²⁻ complex anions ($6 \le m \le 12$) will be favorable, but the existence of intermediate steps is also likely, in a manner similar to the n = 6 case. Unfortunately, reliable structure prediction for such large systems is beyond the capabilities of current computational methods.

Due to the increased enthalpy difference between the reactants and products, to achieve dehydrogenation at the same pressure as bulk LiBH_{4} , $(\text{LiBH}_{4})_6$ nanoclusters need temperatures that are approximately 200 °C higher. Since experimental measurements on nanoconfined borohydrides^{18–20} often find lowering of the desorption temperature, this indicates that the supporting substrate plays an important role by binding the supported nanoclusters and lowering the enthalpy of the reaction products relative to $(\text{LiBH}_4)_n$.

While dehydrogenation is usually carried out by increasing temperature at a constant pressure, as shown in Figs 4 and 5, hydrogen absorption measurements are commonly done by increasing pressure at constant temperature. Hence, pressure-composition isotherms (PCT), showing the equilibrium pressure of the H₂ gas coexisting with a material with a given the hydrogen content, are of considerable interest. For bulk materials, these curves show flat regions corresponding to equilibria between the hydrogen gas and two solid phases on the opposite sides of the miscibility gap. In nanomaterials, clusters with different hydrogen content play the role of the coexisting solid phases. However, in contrast to bulk systems where the plateaus are perfectly flat in the thermodynamic limit with no kinetic limitations, in finite size clusters the plateaus acquire slope. This is illustrated in Fig. 6, which shows the calculated DFT pressure-composition isotherms for (LiBH₄)₆ clusters at three different temperatures. The left-hand side is the dehydrogenated phase, Li₆B₆H₆, corresponding 4.6 wt.% H₂ on the horizontal axis. The fully hydrogenated phase, (LiBH₄)₆ is on the right-hand side, at 18.5 wt.% H₂. Going right-to-left, the sloping plateaus describe the series of increasingly hydrogen-poor clusters in the decomposition sequence of (LiBH₄)₆.

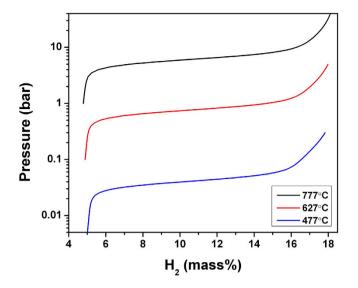


Figure 6. Calculated pressure-composition isotherms for (LiBH₄)₆ clusters.

with the slope arising from the intermediate steps and from the fact that the cluster probabilities p_f in Eq. S2 vary continuously with pressure and temperature. We note that sloping plateaus are seen in experimental study of pressure-composition isotherms of bulk-LiBH₄⁵, but this is likely due to kinetic limitations, not thermodynamic effects. Additionally, the apparent width of the plateau increases slightly with increasing temperature, which we attribute to the narrowing of the thermodynamic stability windows for the intermediate cluster compositions between (LiBH₄)₆ and Li₂B₆H₆.

Conclusions

We present a detailed computational study of the decomposition thermodynamics of small LiBH₄ nanoclusters using DFT-based ground state structure prediction algorithms and total DFT free energies, including vibrational contributions. With a few exceptions at very small sizes, nanoclusters favor the same decomposition sequence into closoborane $Li_2B_{\mu}H_{\mu}$ species as bulk LiBH₄. The calculated reaction enthalpies are higher than those for bulk materials because the total formation energies of the reaction product clusters approach their bulk values significantly slower than $(\text{LiBH}_4)_n$. Hence, reduction of cluster size is not expected to improve the thermodynamic properties of LiBH₄ for use in reversible hydrogen storage. Since experimental studies have demonstrated significant reduction of hydrogen release temperatures in supported nanoclusters, this confirms the importance of cluster-substrate binding energies for the thermodynamics of hydrogen release reactions. It also suggests that technologically favorable thermodynamics of LiBH₄ nanocluster decomposition could be engineered by varying the microscopic structure and composition of the porous substrate to lower the enthalpies of the reactants and achieve hydrogen release at temperatures compatible with the operating conditions of PEM fuel cells. Furthermore, we find that in all cases the decomposition pathways of (LiBH₄)₂, (LiBH₄)₃, and (LiBH₄)₆ consist of multiple intermediate stages, most of them separated by release of an integer number of H₂ molecules. While in bulk systems each reaction occurs along a sharp temperature-pressure curve given by the van't Hoff relation, in nanoclusters the transitions between the cluster phases of different compositions are smooth and characterized by broad coexistence regions on the temperature and pressure; within these regions, clusters of different compositions coexist and their equilibrium fractions vary continuously. As a result, the calculated pressure-composition isotherms exhibit sloping plateaus and smooth variation near the regions of stable bulk compositions.

Computational Methods. The structural and thermodynamic properties of small (LiBH₄)_n ($n \le 12$) clusters are studied using first-principles DFT^{40,41} calculations in conjunction with state-of-the-art genetic algorithm (GA)^{42–44} and prototype electrostatic ground state search (PEGS)⁴⁵ for structure predictions. The detailed descriptions of our computational methods can be found in the Supplementary Information. Decomposition pathways and pressure-composition isotherms are determined using free energy minimization in the grand canonical ensemble (see Supplementary Information).

References

- 1. Schlapbach, L. & Züttel, A. Hydrogen-storage materials for mobile applications. Nature 414, 353 (2001).
- Bogdanović, B. & Schwickardi, M. Ti-doped alkali metal aluminium hydrides as potential novel reversible hydrogen storage materials. J. Alloys Compd. 253–254, 1 (1997).
- 3. Züttel, A. et al. LiBH₄ a new hydrogen storage material. J. Power Sources 118, 1 (2003).
- 4. Orimo, S. et al. Dehydriding and rehydriding reactions of LiBH₄. J. Alloys Compd. 404-406, 427 (2005).
- 5. Mauron, P. et al. Stability and Reversibility of LiBH₄. J. Phys. Chem. B 112, 906 (2008).
- Kostka, J., Lohstroh, W., Fichtner, M. & Hahn, H. Diborane Release from LiBH₄/Silica-Gel Mixtures and the Effect of Additives. J. Phys. Chem. C 111, 14026 (2007).
- Au, M. et al. Stability and Reversibility of Lithium Borohydrides Doped by Metal Halides and Hydrides. J. Phys. Chem. C 112(47), 18661 (2008).

- Cai, W. T., Wang, H., Jiao, L. F., Wang, Y. J. & Zhu, M. Remarkable irreversible and reversible dehydrogenation of LiBH₄ by doping with nanosized cobalt metalloid compounds. *Int. J. Hydrogen Energy* 38, 3304 (2013).
- 9. Sun, W. W. *et al.* Nanoconfinement of lithium borohydride in Cu-MOFs towards low temperature dehydrogenation. *Dalton Trans.* **40**, 5673 (2011).
- 10. Si, X. L. et al. Metals (Ni, Fe)-Incorporated Titanate Nanotubes Induced Destabilization of LiBH₄. J. Phys. Chem. C 115, 9780 (2011).

11. Reilly, J. J. & Wiswall, R. H. Reaction of hydrogen with alloys of magnesium and copper. *Inorg. Chem.* **6(12)**, 2220–2223 (1967).

- 12. Reilly, J. J. & Wiswall, R. H. Reaction of hydrogen with alloys of magnesium and nickel and the formation of Mg2NiH4. *Inorg. Chem.* 7(11), 2254–2256 (1968).
- Siegel, D. J., Wolverton, C. & Ozolinš, V. Reaction energetics and crystal structure of Li₄BN₃H₁₀ from first principles. *Phys. Rev. B* 75, 014101 (2007).
- 14. Vajo, J. J. & Skeith, S. L. Reversible Storage of Hydrogen in Destabilized LiBH₄. J. Phys. Chem. B 109(9), 3719–3722 (2005).
- 15. Sun, T. et al. Confined LiBH₄: Enabling Fast Hydrogen Release at ~100 °C. Int. J. Hydrogen Energy **37**, 18920 (2012).
- 16. Guo, L. J. *et al.* Enhanced desorption properties of LiBH₄ incorporated into mesoporous TiO₂. *Int. J. Hydrogen Energy* **38**, 162 (2013).
- Zhang, Y. *et al.* LiBH₄ nanoparticles supported by disordered mesoporous carbon: Hydrogen storage performances and destabilization mechanisms. *Int. J. Hydrogen Energy* **32**, 3976 (2007).
- Gross, A. F., Vajo, J. J., Van Atta, S. L. & Olson, G. L. Enhanced Hydrogen Storage Kinetics of LiBH₄ in Nanoporous Carbon Scaffolds. J. Phys. Chem. C 112, 5651 (2008).
- 19. Fang, Z. Z. et al. et al. Kinetic- and thermodynamic-based improvements of lithium borohydride incorporated into activated carbon. Acta Materialia 56, 6257 (2008).
- Cahen, S., Eymery, J.-B., Janot, R. & Tarascon, J.-M. Improvement of the LiBH₄ hydrogen desorption by inclusion into mesoporous carbons. J. Power Sources 189, 902 (2009).
- Gosalawit-Utke, R. et al. Destabilization of LiBH₄ by nanoconfinement in PMMAcoBM polymer matrix for reversible hydrogen storage. Int. J. Hydrogen Energy 39, 5019 (2014).
- Zhang, L. T. et al. Superior dehydrogenation performance of nanoscale lithium borohydride modified with fluorographite. Int. J. Hydrogen Energy 39, 896 (2014).
- Ngene, P., van Zwienen, M. (Rien) & de Jongh, P. E. Reversibility of the hydrogen desorption from LiBH₄: a synergetic effect of nanoconfinement and Ni addition. *Chem. Commun.* 46, 8201 (2010).
- Li, C., Peng, P., Zhou, D. W. & Wan, L. Research progress in LiBH4 for hydrogen storage: A review. *Int. J. Hydrogen Energy* 36, 14512 (2011).
 Liu, X. F., Peaslee, D., Jost, C. Z. & Majzoub, E. H. Controlling the Decomposition Pathway of LiBH₄ via Confinement in Highly Ordered Nanoporous Carbon. *J. Phys. Chem. C* 114, 14036 (2010).
- Liu, X. F., Peaslee, D., Jost, C. Z., Baumann, T. F. & Majzoub, E. H. Systematic Pore-Size Effects of Nanoconfinement of LiBH₄: Elimination of Diborane Release and Tunable Behavior for Hydrogen Storage Applications. *Chem. Mater.* 23, 1331 (2011).
- Hazrati, E., Brocks, G. & de Wijs, G. A. First-Principles Study of LiBH₄ Nanoclusters and Their Hydrogen Storage Properties. J. Phys. Chem. C 116, 18038 (2012).
- Mason, T. & Majzoub, E.H. Effects of a Carbon Surface Environment on the Decomposition Properties of Nanoparticle LiBH₄: A First-Principles Study. J. Phys. Chem. C 118, 8852–8858 (2014).
- Majzoub, E. H., Zhou, F. & Ozolinš, V. First-Principles Calculated Phase Diagram for Nanoclusters in the Na-Al-H System: A Single-Step Decomposition Pathway for NaAlH₄. J. Phys. Chem. C 115, 2636 (2011).
- Ozolinš, V., Majzoub, E. H. & Wolverton, C. First-Principles Prediction of Thermodynamically Reversible Hydrogen Storage Reactions in the Li-Mg-Ca-B-H System. J. Am. Chem. Soc. 131, 230 (2009).
- 31. Boustani, I. Systematic *ab initio* investigation of bare boron clusters:mDetermination of the geometry and electronic structures of B_n (n = 2-14). *Phys. Rev. B* 55, 16426 (1997).
- 32. Nguyen, M. T., Matus, M. H., Ngan, V. T., Grant, D. J. & Dixon, D. A. Thermochemistry and Electronic Structure of Small Boron and Boron Oxide Clusters and Their Anions. J. Phys. Chem. A 113, 4895 (2009).
- 33. Tai, T. B., Grant, D. J., Nguyen, M. T. & Dixon, D. A. Thermochemistry and Electronic Structure of Small Boron Clusters (B_n, n=5-13) and Their Anions. J. Phys Chem A 114, 994 (2010).
- 34. Gardet, G., Rogemond, F. & Chermette, H. Density functional theory study of some structural and energetic properties of small lithium clusters. *J. Chem. Phys.* **105**, 9933 (1996).
- 35. Fournier, R., Cheng, J. B. Y. & Wong, A. Theoretical study of the structure of lithium clusters. J. Chem. Phys. 119, 9444 (2003).
- Zhang, Y., Wang, Y., Michel, K. & Wolverton, C. First-principles insight into the degeneracy of ground-state LiBH₄ structures. *Phys. Rev. B* 86, 094111 (2012).
- Soulié, J-Ph, Renaudin, G., Cerný, R. & Yvon, K. Lithium boro-hydride LiBH₄: I. Crystal structure. J. Alloys Compd. 346, 200 (2002).
 Liu, Z. J., Qu, X. H., Huang, B. Y. & Li, Z. Y. Crystal structure and morphology of a new compound, LiB. J. Alloys Compd. 311, 256 (2000).
- 39. Mueller, T. & Ceder, G. Effect of Particle Size on Hydrogen Release from Sodium Alanate Nanoparticles. ACS Nano 4, 5647 (2010).
- 40. Kresse, G. & Hafner, J. Ab initio molecular dynamics for liquid metals. *Phys. Rev. B* 47, 558 (1993).
- Kresse, G. & Furthmüller, J. Efficient iterative schemes for *an initio* total-energy calculations using plane-wave basis set. *Phys. Rev. B* 54, 11169 (1996).
- 42. Deaven, D. M. & Ho, K. M. Molecular Geometry Optimization with a Genetic Algorithm. Phys. Rev. Lett. 75, 288 (1995).
- 43. Ho, K. M. et al. Structures of medium-sized silicon clusters. Nature 392, 582 (1998).
- 14. Chuang, F. C., Wang, C. Z. & Ho, K. M. Structure of neutral aluminum clusters Al_n ($2 \le n \le 23$): Genetic algorithm tight-binding calculations. *Phys. Rev. B* 73, 125431 (2006).
- 45. Majzoub, E. H. & Ozolinš, V. Prototype electrostatic ground state approach to predicting crystal structures of ionic compounds: Application to hydrogen storage materials. *Phys. Rev. B* 77, 104115 (2008).

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Author Contributions

F.C.C. and V.O. conceived and initiated the study. Z.Q.H. performed the GA calculations, W.C.C. performed the DFT calculations, and E.H.M. performed the PEGS calculations. W.C.C., F.C.C. and V.O. performed the analysis and contributed the discussions. F.C.C., W.C.C. and V.O. wrote the manuscript. All authors reviewed the manuscript.

Additional Information

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