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## Ti<sup>3+</sup>-self doped brookite TiO<sub>2</sub> single-crystalline nanosheets with high solar absorption and excellent photocatalytic CO<sub>2</sub> reduction

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**Black brookite TiO<sub>2</sub> single-crystalline nanosheets with outstanding photocatalytic activity toward CO<sub>2</sub> reduction is prepared by a facile oxidation-based hydrothermal reaction method combined with post-annealing treatment. Large amount of Ti<sup>3+</sup> defects are introduced into the bulk of brookite nanoparticles, which increases the solar energy absorption and enhances the photocatalytic activity.**

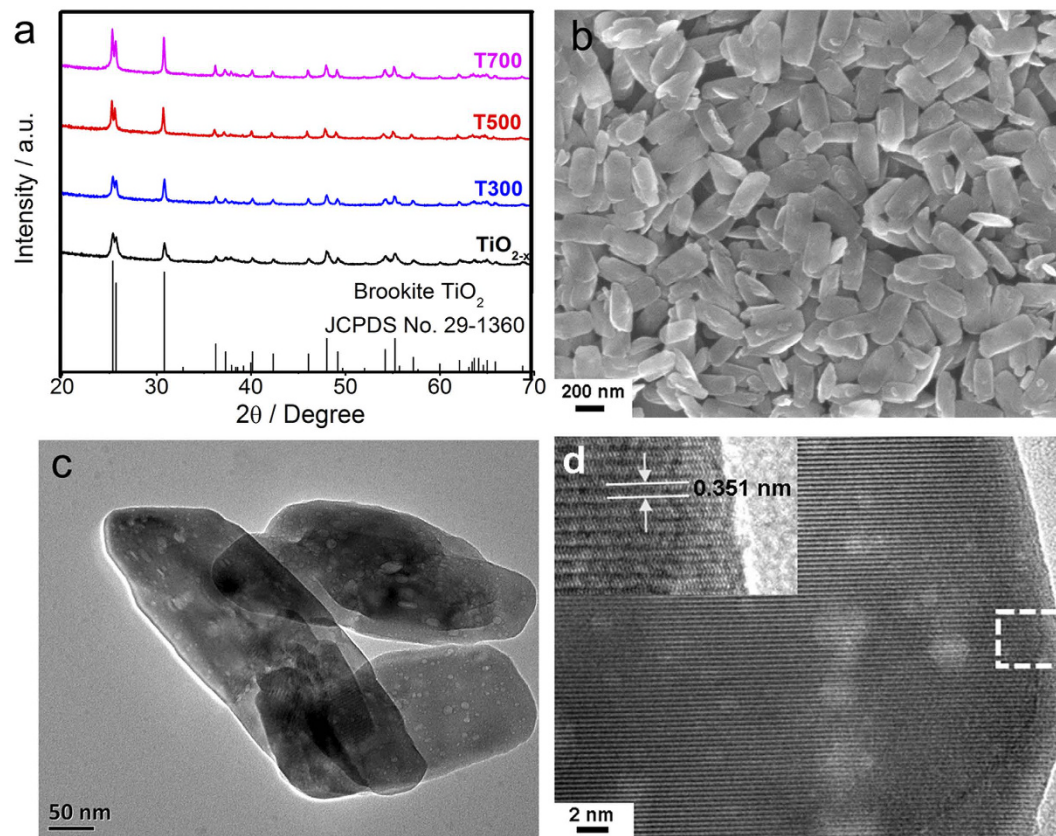
The emission of greenhouse gases, particularly carbon dioxide (CO<sub>2</sub>), could result in the global climate change and unhealthy regional air quality<sup>1,2</sup>. To reduce the emission of CO<sub>2</sub> and to achieve a sustainable energy future, novel materials and new promising “green chemistry” technologies are desired to convert CO<sub>2</sub> into useful chemical compounds and fuels using sunlight. CO<sub>2</sub> photoreduction in engineered systems, however, still faces severe challenges like low conversion yields and quantum efficiencies due to fast electron-hole recombination, narrow sunlight absorption range, and backward reactions<sup>3–6</sup>. Strategies to resolve those challenges have been a research hotspot in the area<sup>7–10</sup>. Among the studied semiconductor photocatalysts, titanium dioxide (TiO<sub>2</sub>) is by far the most popular catalyst because of its low cost, suitable bandgap and nature abundance<sup>11–13</sup>. From this prospect, TiO<sub>2</sub> is still a promising photocatalyst for CO<sub>2</sub> photoreduction due to its favorable band edges because TiO<sub>2</sub> has a higher CB edge, which will promote the reaction of CO<sub>2</sub> reduction with H<sub>2</sub>O that has a high reduction potential<sup>14</sup>.

TiO<sub>2</sub> naturally exists in three polymorphs: anatase, brookite and rutile. Among them, only TiO<sub>2</sub> anatase, rutile, and anatase-rutile mixed phase (e.g., Degussa P25) have been studied for CO<sub>2</sub> photoreduction<sup>15</sup>. In other words, naturally occurring metastable brookite is the least investigated one because of the difficulties to obtain its pure form<sup>16</sup>. Based on the previous model for the structure-property relationship of photo-catalytic materials, the photocatalytic activity of brookite would be inferior to anatase but superior to rutile<sup>17–19</sup>. For brookite, lower packing factor, in accordance with higher structural openness degree, is associated with stronger ability for electron-hole separation and transfer, and usually results in better photocatalytic activity<sup>20,21</sup>. Considering that brookite has a relative open structure, its photocatalytic performance is worth investigating such as in CO<sub>2</sub> photoreduction.

Many efforts have been made in decades to improve the photocatalytic activity of brookite, including preparation of nanosized brookite with various morphologies and studies of the involved shape-dependent photocatalytic performance<sup>22–26</sup>. However, the application of brookite TiO<sub>2</sub> has been limited to the ultraviolet (UV) range, which makes it an inefficient photocatalyst for fully exploiting solar light. Narrowing down the TiO<sub>2</sub> bandgap to turn it into an efficient material for solar energy conversion has been an ongoing challenge. Recently, the chemistry of structurally defective TiO<sub>2</sub> with Ti<sup>3+</sup> self-doping has been developed to solve the challenges in broad spectral range of photon absorption. Reduction of Ti<sup>4+</sup> into Ti<sup>3+</sup> is often achieved by harsh and costly physical methods, such as high temperature and pressurized hydrogenation<sup>27–30</sup>, plasma treatment<sup>31,32</sup>, vacuum activation and electron beams irradiation<sup>33</sup> to improved light absorption and photocatalytic activities. Thus, a facile method is greatly desired to get defective brookite TiO<sub>2</sub> with Ti<sup>3+</sup> self-doping.

In this work, we developed an innovative facial oxidation-based hydrothermal method to synthesize bulk Ti<sup>3+</sup> self-doped brookite TiO<sub>2</sub> single-crystalline nanosheets (denoted as TiO<sub>2-x</sub> if no further heat treatment). By post-annealing at different temperatures, Ti<sup>3+</sup> self-doped brookite TiO<sub>2</sub> with controllable amounts of Ti<sup>3+</sup> defects

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**Figure 1.** (a) XRD patterns of as-prepared brookite  $\text{TiO}_{2-x}$  and the blue  $\text{TiO}_{2-x}$  after postannealing treatment at different temperatures for 3 h. (b) SEM image of the typical T500 sample. (c) TEM image of the T500 sample. (d) HRTEM image taken from the edge of the single crystal.

is achieved. The black brookite  $\text{TiO}_{2-x}$  (T500) shows drastically enhanced visible-light absorption and exhibits excellent photocatalytic activity toward  $\text{CO}_2$  reduction.

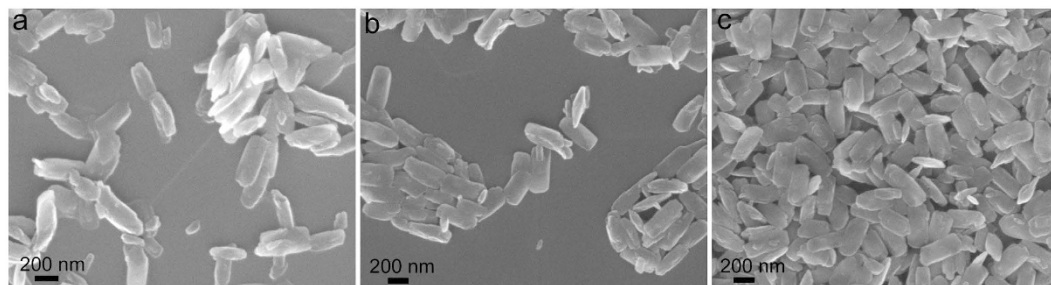
## Results

The X-ray diffraction (XRD) patterns of the high quality brookite  $\text{Ti}^{3+}$ -self doped  $\text{TiO}_2$  is displayed in Fig. 1a. The patterns show that all the samples are pure brookite phase without any other impurity phase (rutile or anatase phase). All the diffraction positions and intensity distributions can be indexed to the brookite corresponding to the JCPDS card No. 29–1360. By increasing the post-annealing temperature, the characteristic diffraction peaks of defective brookite  $\text{TiO}_2$  becomes stronger and sharper due to the increase in the crystallinity. When the temperature reaches 700 °C, no rutile or anatase characteristic diffraction peaks were detected, indicating that the brookite phase can be stabilized at higher temperatures.

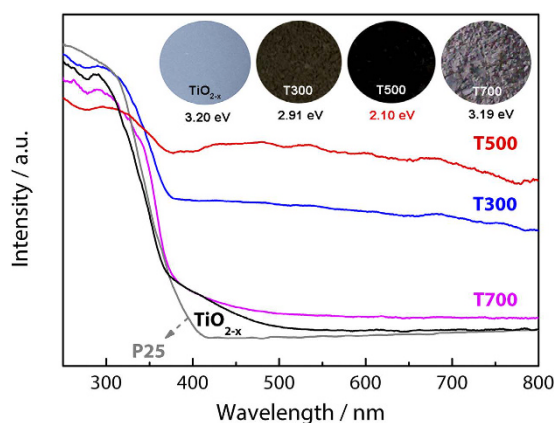
The morphology of the synthesized brookite powders is investigated by scanning electron microscopy (SEM) and transmission electron microscopy (TEM) as shown in Fig. 1b,c, respectively. Clearly, the  $\text{Ti}^{3+}$ -self doped brookite shows platelike features with uniform sizes of length up to 400 nm and thickness up to 10 nm. The obtained  $\text{TiO}_2$  particles are uniformly dispersed without obvious aggregation. There is no further particle growth with an increase of the post-annealing temperature. Distinct crystal planes and smooth fringes can be observed clearly from the magnified image of a single particle as shown in Fig. 1c.

In order to unveil the microstructure transformation of the brookite after post-annealing treatment, the  $\text{TiO}_{2-x}$  sample treated at 500 °C were examined by high resolution transmission electron microscopy (HRTEM) as shown in Fig. 1d. The uniform lattices indicate that the  $\text{Ti}^{3+}$  self-doped brookite  $\text{TiO}_2$  nanoparticles are highly crystallized. In addition, the spacing of the fringes paralleling to the top and bottom of the vertical nanoplate is 0.351 nm (Fig. 1d, inset), which can be attributed to the (210) facet of brookite. There is no obvious difference in lattice fringes width (e.g., those for rutile phase or  $\text{TiH}_2$ ) indicating that the as-prepared sample is pure brookite phase. This result is consistent with XRD analysis. Additionally, as shown in Fig. 2, there is no obvious difference among the other samples, indicating that the temperature change has very limited influence on the morphology and crystallite size of these nanocrystals.

The top insert of Fig. 3 is photographic images of  $\text{Ti}^{3+}$  self-doped brookite  $\text{TiO}_2$  samples prepared with postannealing treatment at different temperatures for 3 h in a  $\text{N}_2$  gas flow and their optical band gap estimated by Kubelka-Munk function. The results show that the color of brookite  $\text{TiO}_2$  samples is remarkably different from white perfect  $\text{TiO}_2$  nanocrystal particles. From previous research, this phenomenon indicates the existence of  $\text{Ti}^{3+}$  defects<sup>30</sup>. The color of the brookite turns from blue to brown (T300) and further to black (T500). The color



**Figure 2.** SEM images of (a)  $\text{TiO}_{2-x}$ , (b) T300, and (c) T700.



**Figure 3.** UV-vis diffuse reflectance spectra of the as-prepared brookite  $\text{TiO}_{2-x}$ , T300, T500 and T700.

variation indicates the enhanced light absorption of brookite after post-annealing treatment up to 500 °C. As the post-annealing treatment temperature further increases, the dark black color is not maintained but gradually faded to a light grey blue at 700 °C.

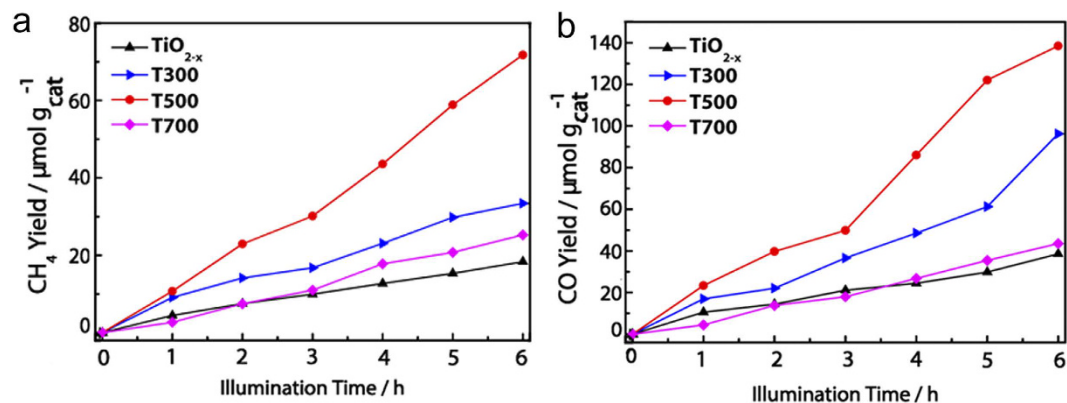
Figure 3 displays the UV-vis diffuse reflectance spectra (UV-vis DRS) of as-prepared brookite  $\text{TiO}_{2-x}$ , T300, T500 and T700 nanoparticles, and Degussa P25 as reference was also examined. The absorption peak at 390 nm is attributed to the intrinsic bandgap absorption of crystalline brookite  $\text{TiO}_{2-x}$ . The as-prepared  $\text{TiO}_{2-x}$  sample (blue in color) exhibits an obvious stronger absorption between 400 and 800 nm compared to P25. This strong absorption is attributed to the existence of  $\text{Ti}^{3+}$  defects which induce a continuous vacancy band of electronic states just below the conduction band edge of  $\text{TiO}_{2-x}$ . The existence of  $\text{Ti}^{3+}$  defects is supported by ESR measurements that will be discussed in later section. Compared with  $\text{TiO}_{2-x}$ , the absorption in the visible light region is greatly enhanced after post-annealing treatment. As the annealing temperature increases from 300 °C to 500 °C, the light absorption increases gradually, in good agreement with the color transformation of brookite. However, the light absorption of T700 presents a dramatically decrease in comparison with the T500 sample, which is coincident with the observed color change trend of the samples (as depicted in the insert of Fig. 3). In the hydrothermally treated brookite  $\text{TiO}_{2-x}$ , the enhanced visible-light absorption is attributed to the fact that hydrothermally treated process introduces the disorder in the  $\text{TiO}_2$  and the resulted bandgap narrowing<sup>34</sup>. For the black brookite  $\text{TiO}_2$ , postannealing treatment not only increases the structural openness, but also introduces  $\text{Ti}^{3+}$  ( $3d^1$ ) in the  $\text{TiO}_{2-x}$  bulk<sup>35</sup>. Using the Kubelka–Munk function as the vertical axis to plot it against the photon energy, the optical band gaps of all samples can be derived, and the results are summarized in the top insert of Fig. 3. Among all the samples, the band gap of T500 (2.10 eV) is the narrowest one. These aspects may work together leading to the black colouration, enhanced light absorption and promoted photocatalytic activity of defective brookite  $\text{TiO}_2$ .

In this work, photoactivity of as prepared defective brookite  $\text{TiO}_{2-x}$  was explored by the heterogeneous photo-reduction of  $\text{CO}_2$  under visible light illumination (Fig. 4).  $\text{CH}_4$  and CO were detected as the major hydrocarbon products from the photo-reduction of  $\text{CO}_2$  over the as-synthesized  $\text{Ti}^{3+}$ -self doped brookite  $\text{TiO}_2$  photocatalysts. The yields of  $\text{CH}_4$  and CO ( $\mu\text{mol}/\text{g}_{\text{cat}}$ ) attained after 6 h of visible-light irradiation are shown in Fig. 4a,b (detailed calculation can refer to Eq. 1).

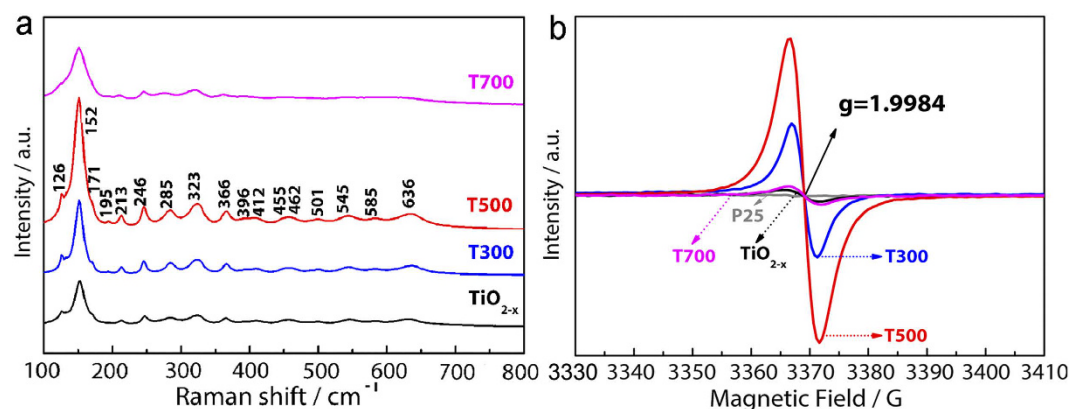
$$\text{Total } \text{CH}_4/\text{CO yield} = \frac{\text{total amount of } \text{CH}_4/\text{CO produced } (\mu\text{mol})}{\text{amount of photocatalyst used } (\text{g-catalyst})} \quad (1)$$

## Discussion

The low symmetry and large unit cell of brookite result in a diagnostically complex Raman spectrum compared to the other two polymorphs (anatase and rutile), thus Raman spectroscopy has been widely used to detect the



**Figure 4.** (a) CH<sub>4</sub> and (b) CO evolution over the brookite TiO<sub>2-x</sub>, T300, T500 and T700 samples for a period of 6 h visible-light illumination ( $\lambda \geq 420$  nm).



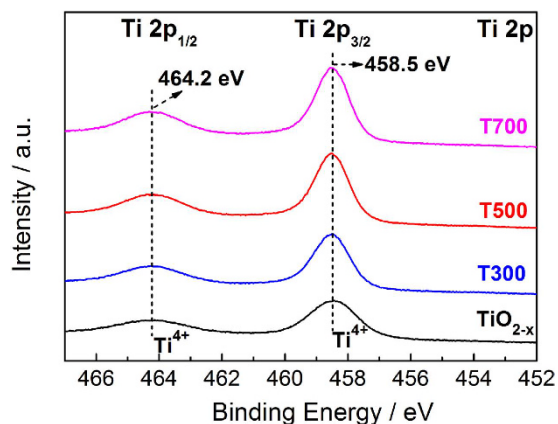
**Figure 5.** (a) Raman spectra and (b) low-temperature EPR spectra of the brookite TiO<sub>2-x</sub>, T300, T500 and T700 samples.

presence of the brookite phase in TiO<sub>2</sub> containing materials<sup>25</sup>. Therefore, Raman spectra of the before and after post-annealing treatment can further confirm the phase composition. As shown in Fig. 5a, the TiO<sub>2-x</sub> and the annealed defective TiO<sub>2</sub> at different temperatures display the typical Raman spectra of brookite, and 16 bands are identified in total, including seven A<sub>g</sub> (124, 151, 192, 244, 410, 544, 636 cm<sup>-1</sup>), four B<sub>1g</sub> (211, 318, 415, 500 cm<sup>-1</sup>), four B<sub>2g</sub> (365, 393, 460, 581 cm<sup>-1</sup>), and one B<sub>3g</sub> (287 cm<sup>-1</sup>). The characteristic Raman signals at 399 cm<sup>-1</sup> and 519 cm<sup>-1</sup> for anatase or at 447 cm<sup>-1</sup> and 612 cm<sup>-1</sup> for rutile are not observed. This further proves the high phase purity of brookite for the TiO<sub>2-x</sub> and the annealed defective TiO<sub>2</sub> samples, in good agreement with the XRD results.

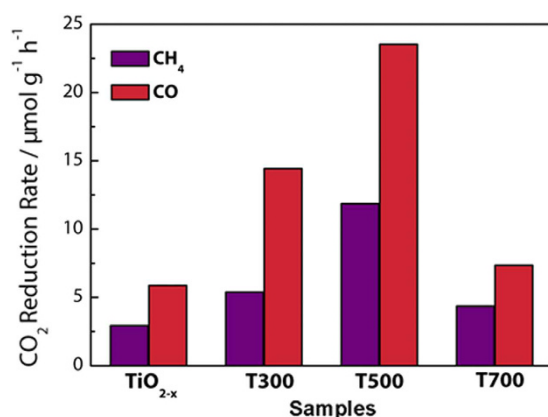
In addition, the intensity of high-resolution Eg Raman vibration modes increases as the post-annealing treatment temperature increases up to 500 °C. This is attributed to the effect of enhanced crystallinity upon the increase in annealing temperature, because higher crystallinity contains more Raman active facets, which is proportional to the Eg intensity fluctuations. When the temperature increases to 700 °C, the intensity of Raman vibration modes is decreased and the peak width is broadened, which may be due to the change in the distribution and concentration of Ti<sup>3+</sup>. The fact that the Ti<sup>3+</sup>-doped TiO<sub>2</sub> has quite different high-resolution Eg modes in Raman spectra with different defects distribution and concentration suggests that Ti<sup>3+</sup> should mainly localize within the bulk of the samples.

Electron paramagnetic resonance (EPR), which is highly sensitive to paramagnetic species containing unpaired electrons, has been widely used to characterize the existence of Ti<sup>3+</sup> defects. As indicated in Fig. 5b, the pristine TiO<sub>2</sub> contains almost completely Ti<sup>4+</sup> (3d<sup>0</sup>) and shows a negligible paramagnetic signal peak under the present EPR measurement condition at 110 K. In general, the surface Ti<sup>3+</sup> tends to adsorb O<sub>2</sub>, which could be reduced to superoxide radical anions (O<sub>2</sub><sup>-</sup>) with an EPR signal feature at  $g = 2.02$ <sup>36</sup>. However, no such signal appears for all samples, further demonstrating the absence of surface Ti<sup>3+</sup>.

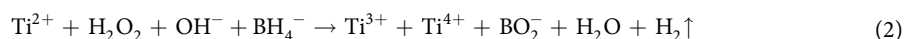
On the other hand, the EPR spectra of brookite defective TiO<sub>2</sub> nanoparticles show intense axially symmetry signals centered at the  $g$  value of 1.9984, which have been reported for Ti<sup>3+</sup> ions in brookite crystallite<sup>37</sup>, indicating that the free electrons occupy interior Ti position thereby generating Ti<sup>3+</sup> defects in bulk. Accordingly, the chemical equation in our work can be expressed as below:



**Figure 6.** Ti 2p XPS spectra of the brookite  $\text{TiO}_{2-x}$ , T300, T500 and T700 samples.



**Figure 7.**  $\text{CO}_2$  reduction rate of the brookite  $\text{TiO}_{2-x}$ , T300, T500 and T700 samples.



Furthermore, the EPR intensity increases along with the post-annealing temperature. The strongest value is obtained for T500, but the signal intensity declines sharply when further increasing the treatment temperature to 700 °C. By numerical double integration of the EPR spectra with an aqueous solution of  $\text{Cu}^{2+}$  as reference, the amount of  $\text{Ti}^{3+}$  centers for  $\text{TiO}_{2-x}$  sample is calculated to be  $0.7 \times 10^{19}$  spins/mol, equivalent to one  $\text{Ti}^{3+}$  out of every  $2.9 \times 10^4$   $\text{Ti}^{4+}$ . Whereas, the amount of  $\text{Ti}^{3+}$  centers for T500 sample is almost 10 times larger than that for  $\text{TiO}_{2-x}$ , namely, one  $\text{Ti}^{3+}$  in every  $2.9 \times 10^3$   $\text{Ti}^{4+}$ . This suggests that the concentration of bulk  $\text{Ti}^{3+}$  defect is greatly enhanced by the high temperature post-annealing treatment. From our previous study, the decreased  $\text{Ti}^{3+}$  defects upon treatment at 700 °C is mainly due to the difficulty of further diffusion of  $\text{Ti}^{3+}$  in enhanced crystallinity, leading to the dilution of the “colour center”<sup>38</sup>.

In order to confirm the existence of  $\text{Ti}^{3+}$ , additional characterizations are performed. X-ray photoelectron spectroscopy (XPS) is performed to further investigate the transformation of surface chemical bonding and to detect the electronic valence band position of the brookite samples. As shown in Fig. 6, the Ti 2p XPS spectra of brookite  $\text{TiO}_{2-x}$ , T300, T500 and T700 samples show the typical pattern of  $\text{Ti}^{4+}$  –O bonds in  $\text{TiO}_2$  with Ti  $2p_{3/2}$  and  $2p_{1/2}$  peaks centered at binding energies of 458.5 eV and 464.2 eV<sup>39</sup>. For all the samples, no peaks shift apparently to lower energy, indicating that  $\text{Ti}^{3+}$  species are separately located at bulk of the samples, and no different chemical states and disorders on the surface of samples. This agrees with the HRTEM observation, i.e., no obvious disordered layer in the edge of the samples.

The  $\text{CO}_2$  reduction rate is presented in Fig. 7. The  $\text{CO}_2$  reduction rate of annealed brookite  $\text{TiO}_2$  also improves dramatically; it shows drastically enhanced  $\text{CO}_2$  reduction rate when brookite  $\text{TiO}_{2-x}$  is annealed at 500 °C ( $11.9 \mu\text{mol} \cdot \text{g}_{\text{cat}}^{-1} \text{h}^{-1}$  for  $\text{CH}_4$  and  $23.5 \mu\text{mol} \cdot \text{g}_{\text{cat}}^{-1} \text{h}^{-1}$  for  $\text{CO}$ ). However, defective brookite prepared at 700 °C shows lower  $\text{CO}_2$  reduction rate ( $4.4 \mu\text{mol} \cdot \text{g}_{\text{cat}}^{-1} \text{h}^{-1}$  for  $\text{CH}_4$  and  $7.3 \mu\text{mol} \cdot \text{g}_{\text{cat}}^{-1} \text{h}^{-1}$  for  $\text{CO}$ ), but is still higher than untreated  $\text{TiO}_{2-x}$  sample. It can be deduced that the photocatalytic reduction performance agrees well with the light absorption and  $\text{Ti}^{3+}$  defects (numbers and distribution); the more the light absorption and the  $\text{Ti}^{3+}$  defects, the higher the photocatalytic reduction activity is.

Clearly, the engineered  $\text{Ti}^{3+}$ -self doped brookite catalysts developed in this work demonstrate a superior activity. It is the first time to present a facile approach to controllably synthesize  $\text{Ti}^{3+}$ -self doped brookite  $\text{TiO}_2$  as an

outstanding candidate for CO<sub>2</sub> photoreduction to produce CO and CH<sub>4</sub>. Further study on the selectivity of CO<sub>2</sub> reduction is going on.

In summary, this work illustrates that the oxidation-based hydrothermal synthesis of Ti<sup>3+</sup> self-doped brookite TiO<sub>2</sub> is an effective strategy to prepare uniform Ti<sup>3+</sup> self-doped brookite TiO<sub>2</sub> nanosheets. The introduction of Ti<sup>3+</sup> defects in the bulk of brookite enhances the visible light absorption and narrows the bandgap. Our study demonstrates that brookite can be successfully tuned to be highly active toward photocatalytic performance for CO<sub>2</sub> reduction through post-annealing at different temperatures, which illuminates the future research of brookite; the oxidation-based hydrothermal synthesis combined with post-annealing treatment may provide a novel path for tuning the photocatalysts from relatively inert to highly active and shine light on the greenhouse gases conversion into sustainable energy.

## Methods

**Materials.** Titanium hydride (TiH<sub>2</sub>, 98.0%) powder was purchased from Sigma-Aldrich Co., LLC. Hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>, 30.0%), sodium borohydride (NaBH<sub>4</sub>, 98.0%), hydrochloric acid (HCl, 36–38%, A.R.) and sodium hydroxide (NaOH, 98%, A.R.) were obtained from Sinopharm Chemical Reagent Co., Ltd. and used as received without any further purification. Double distilled water was used throughout the experiments.

**Synthesis of Ti<sup>3+</sup> self-doped brookite TiO<sub>2</sub> nanoparticles.** In a typical synthetic process, TiH<sub>2</sub> (0.256 g) and H<sub>2</sub>O (2 mL) were mixed in a 50 mL round-bottomed flask magnetically stirred for 5 min. Then H<sub>2</sub>O<sub>2</sub> (30 mL 30.0 wt%) as the oxidation was added dropwise to this dark grey suspensions, and this mixture was vigorously stirred for 12 h till it changed to yellowish gel-like state. After that, double distilled water (40 mL) was added under continuous magnetic stirring. A certain amount of NaOH (1.0 M) solution as the pH regulator was added gradually until the pH of the mixture solution was tuned to 9.0. NaBH<sub>4</sub> (0.4 g) as reducing agent was added to this light yellow transparency mixture and then transferred to the Teflon-lined stainless-steel autoclave immediately and hydrothermally treated at 180 °C for 24 h. The sample was then collected and added into HCl (50.0 mL, 1.0 M) solution to eliminate the sodium boron compounds. After stirring for 10 h, the powders were washed by distilled water and ethanol repetitively to remove the impurities (e.g., Na<sup>+</sup>, Cl<sup>-</sup>, BO<sub>3</sub><sup>2-</sup>). The obtained precipitate was dried under vacuum for 12 h to yield a grey blue TiO<sub>2</sub> nanocrystals powder, denoted as TiO<sub>2-x</sub>. Postannealing treatment of the TiO<sub>2-x</sub> sample was conducted under a N<sub>2</sub> gas flow (150 sccm) in a tube furnace at an elevated temperature of 300 °C, 500 °C and 700 °C for 3 h. Therefore, the samples obtained at the specific temperatures are designated as T300, T500 and T700, respectively.

**Characterization.** The crystal structures of the samples were identified on a Bruker D8 X-ray diffractometer with Cu K $\alpha$  radiation ( $\lambda = 0.15418$  nm). The morphology photographs of the samples were recorded by field emission scanning electron microscopy (FESEM; ZEISS SUPRA55VP) and transmission electron microscopy (TEM, JEOL-JEM 2100). Ultraviolet–visible (UV–vis) diffusion reflectance spectra of the samples were obtained on a SolidSpec-3700DUV spectrophotometer (Shimadzu) using BaSO<sub>4</sub> as reference to obtain absorption spectra for determining the band gap. Raman spectra were obtained on a laser Raman spectrometer (LabRAM HR Evolution RAMAN SPECTROMETER, HORIBA Scientific Ltd.) with a back scattering configuration using an Ar<sup>+</sup> laser (20 mW, 532 nm) as excitation source. The surface electronic state analysis was studied by X-ray photoelectron spectra (XPS), and the measurements were carried out on an X-ray photoelectron spectrometer (ESCALAB MK II) using Mg Ka (1253.6 eV) X-rays as the excitation source, with C 1s (284.6 eV) for calibration. Electron paramagnetic resonance (EPR) spectra were recorded on a Bruker Elexsys E500 spectrometer by applying an X-band (9.43 GHz, 1.5 mW) microwave with sweeping magnetic field at 110 K in cells that can be connected to a conventional high-vacuum apparatus (residual pressure < 10<sup>-4</sup> mbar). The concentration of Ti<sup>3+</sup> was determined by a numerical double integration of the EPR spectra in comparison with an aqueous solution of Cu<sup>2+</sup>.

**Photocatalytic reduction of CO<sub>2</sub> under visible-light.** The photocatalytic activities of the Ti<sup>3+</sup>-self doped brookite TiO<sub>2</sub> single-crystalline nanosheets were studied using a CO<sub>2</sub> photoreduction system at ambient condition in a continuous gas flow reactor. The CO<sub>2</sub> photoreduction process was performed under visible light irradiation with a 300 W Xe lamp (PLS-SXE300, Perfect Light Company, Beijing, China) equipped with an ultraviolet cut-off filter to provide visible light ( $\geq 420$  nm). The illumination intensity at the surface was 0.216 W/cm<sup>2</sup> as measured by a calibrated precision optical power meter (1916-C, Newport Corp.). The amount of photocatalyst used was held constant in all runs. Highly pure CO<sub>2</sub> (99.99%) was bubbled through water to produce a mixture of CO<sub>2</sub> and water vapor into the photoreactor at atmospheric pressure. Before switching on the light source, wet CO<sub>2</sub> was permitted to flow through the photoreactor at 30 mL/min for 30 min to eliminate any excess air and to ensure the complete adsorption of gas molecules. The photoreactor was operated in a continuous flow mode (2.0 mL/min flow rate), and the gaseous products in the reactor effluent were continuously analyzed for 6 h by a gas chromatograph (GC, Agilent 7890 A) equipped with both a thermal conductivity detector (TCD) and a flame ionization detector (FID).

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## Author Contributions

X.X. and L.W. performed the experiments and wrote the main manuscript text. C.W. guided the whole work. T.X. made contribution for discussions and critical revision of the manuscript. All authors reviewed the manuscript.

## Additional Information

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