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In-plane chemical pressure essential for superconductivity in BiCh₂-based (Ch: S, Se) layered structure

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BiCh₂-based compounds (Ch: S, Se) are a new series of layered superconductors, and the mechanisms for the emergence of superconductivity in these materials have not yet been elucidated. In this study, we investigate the relationship between crystal structure and superconducting properties of the BiCh₂-based superconductor family, specifically, optimally doped Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS₂ and LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)₂. We use powder synchrotron X-ray diffraction to determine the crystal structures. We show that the structure parameter essential for the emergence of bulk superconductivity in both systems is the in-plane chemical pressure, rather than Bi-Ch bond lengths or in-plane Ch-Bi-Ch bond angle. Furthermore, we show that the superconducting transition temperature for all REO_{0.5}F_{0.5}BiCh₂ superconductors can be determined from the in-plane chemical pressure.

Most superconductors with a high transition temperature (T_c) possess a layered crystal structure. Typical examples of layered high- T_c superconductors are Cu-oxide¹ and Fe-based superconductors²; the crystal structures of these superconductors are composed of alternating superconducting layers (CuO₂ or FeAs layers, respectively) and electrically-insulating spacer layers. The spacer layers are essential for the emergence of low-dimensional electronic states within the superconducting layers, which sometimes result in unconventional pairing mechanisms. One of the attractive features of the layered superconductors is the wide variety of crystal structures. New superconductors can be designed by stacking superconducting layers and various types of spacer layers; the superconducting properties depend on the type of the spacer layer. In fact, many Fe-based superconductors containing various types of spacer layers have been discovered, several of which attained high T_c ²⁻⁸.

With this wide selection of related layered superconductors having various structures and properties, one can determine which crystal structure parameters are essential for the emergence of superconductivity in a given family of layered superconductors. It was found that T_c for the Fe-based family can be estimated from a crystal structure parameter, such as the As-Fe-As bond angle⁹ or the anion height from the Fe square lattice (anion = As, P, Se, and Te)¹⁰. Changes in these parameters strongly affect Fermi surface configurations and pairing symmetry, subsequently affecting the superconducting properties of Fe-based superconductors¹¹. In fact, clarification of which crystal structure parameter influences superconducting properties is one of the most important challenges for understanding the mechanisms

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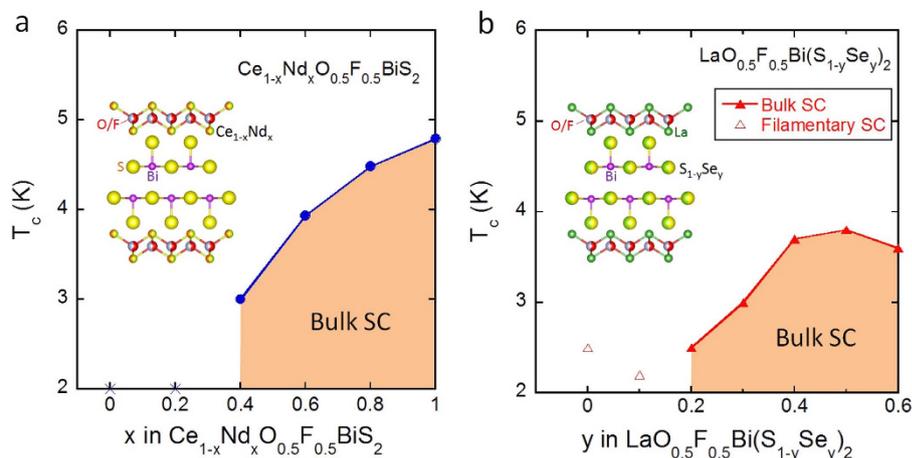


Figure 1. Superconductivity phase diagrams of $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$.

(a) Superconductivity phase diagrams of $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$. For $x = 0$ and 0.2 , superconducting transition is not observed at $T > 2$ K. For $0.4 \leq x \leq 1$, superconducting transition with a large shielding signal, with which we could regard the samples as a bulk superconductor (Bulk SC), is observed. T_c increases with increasing x . The inset figure shows a schematic of crystal structure of $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$. (b) Superconductivity phase diagrams of $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$. For $y = 0$ and 0.1 , superconducting transition is observed but their shielding signals are very small as a bulk superconductor (Filamentary SC). For $y \geq 0.2$, superconducting transition with a large shielding signal is observed (Bulk SC). T_c increases with increasing y up to $y = 0.5$. The inset figure shows a schematic image of crystal structure of $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$.

of superconductivity in a new series of layered superconductors; this also provides a direct strategy to design new layered superconductors with high T_c .

In 2012, we reported layered superconductors composed of alternate stacks of BiS_2 superconducting layers and various spacer layers^{12,13}. Band calculations suggested that the parent compound of BiS_2 -based superconductors is an insulator containing Bi^{3+} . Superconductivity is induced when electron carriers are doped into the BiS_2 layers (in Bi -6p orbitals) by element substitution in the spacer layers^{12,14}. For example, a parent compound REOBiS_2 becomes a superconductor when O^{2-} in the spacer layers is partially substituted by F^- (namely, $\text{REO}_{1-x}\text{F}_x\text{BiS}_2$)¹³. To date, 12 types of parent compounds of BiS_2 -based superconductors have been discovered: REOBiS_2 ($\text{RE} = \text{La}^{13}$, Ce^{15} , Pr^{16} , Nd^{17} , Sm^{18} , Yb^{19} , and $\text{Bi}^{20,21}$), AFBiS_2 ($\text{A} = \text{Sr}^{22,23}$, Eu^{24}), $\text{Bi}_6\text{O}_8\text{S}_5^{12}$, $\text{Bi}_3\text{O}_2\text{S}_3^{25}$, and $\text{Eu}_3\text{F}_4\text{Bi}_2\text{S}_4^{26}$. Superconductivity was also observed in a BiSe_2 -based compound, $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiSe}_2^{27}$. Exploring new these compounds is a growing field of research in condensed matter physics.

As described above, electron-carrier doping is necessary for the emergence of superconductivity in the BiCh_2 family. Some high-pressure (HP) studies, however, suggest that the crystal structure is also important for the emergence of superconductivity in these materials^{28–32}. One such study investigated the effect of HP on T_c , in which $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ showed remarkable changes in superconductivity under HP. $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ did not exhibit bulk superconductivity (bulk SC), though it showed a filamentary (weak) superconductivity signal with a T_c of 2.5 K¹³. With the application of HP, $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ become a bulk superconductor and T_c drastically increases from 2.5 K to over 10 K^{13,28–32}. T_c enhancement under HP has also been observed in other $\text{REO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ superconductors³². These facts strongly suggest that superconducting properties of BiCh_2 -based superconductors are correlated with changes in crystal structure, which is analogous to findings for the Fe-based family^{9,10}.

Furthermore, our recent studies concerning the effect of isovalent-substitution on superconductivity suggest that optimization of crystal structure is important for the emergence of bulk SC and the ability to attain a high T_c in optimally doped $\text{REO}_{0.5}\text{F}_{0.5}\text{BiCh}_2$. We use the example of $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ ³³ for following discussion. Since in this crystal structure, the valence of Ce and Nd are both 3+, electron carriers in these compounds are essentially the same: the formal valence of Bi is 2.5+. However, bulk SC is induced by the systematic substitution of Ce by Nd, and T_c increases with increasing Nd concentration (x) as shown in Fig. 1a. The emergence of superconductivity in this material was explained by uniaxial lattice shrinkage along the a -axis and optimization of the lattice shrinkage ratio, c/a ³³. Another example of isovalent-substitution systems is $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ ³⁴. In this material, the S^{2-} site within the superconducting layers is systematically substituted by Se^{2-} . Therefore, the formal valence of Bi (2.5+) should not change with Se substitution. Bulk SC is induced by Se substitution, and T_c increases with increasing Se concentration (y) as shown in Fig. 1b. Se substitution enhances the metallic conductivity of this system and induces bulk SC through lattice volume expansion³⁴.

These experimental results obtained from HP studies on $\text{REO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ ^{13,28–32} and isovalent-substitution studies on $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ ^{33,34}, confirm that superconducting properties

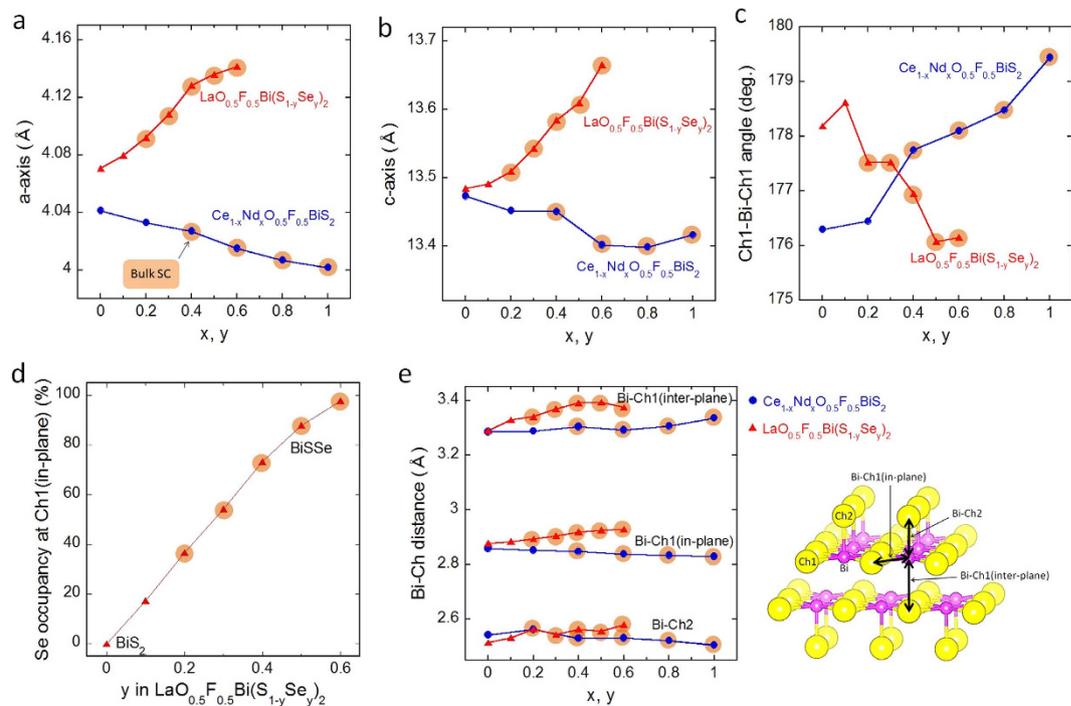


Figure 2. Crystal structure parameters for $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$. (a) Lattice constant of a for $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ is plotted as a function of x (or y). Data points for the samples showing Bulk SC in Fig. 1 are highlighted with orange filled circles in (a–e). (b) Lattice constant of c for $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ is plotted as a function of x (or y). (c) Evolution of Ch1–Bi–Ch1 angle for $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ as a function of x (or y). (d) Se occupancy at the in-plane Ch1 site in $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ as a function y . (e) Evolution of three different Bi–Ch distances in $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ as a function of x (or y). The right image describes the Bi, Ch1 and Ch2 sites, and three Bi–Ch distances: Bi–Ch1 (in-plane), Bi–Ch1 (inter-plane), and Bi–Ch2.

of the BiCh_2 family are influenced by not only electron carrier concentration, but also crystal structure. Although previous studies suggest the importance of in-plane Bi–S distance¹⁸ and/or in-plane S–Bi–S angle³⁵ on superconductivity, the universal relationship between crystal structure and superconductivity (and its T_c) in the BiCh_2 family has not been clarified. In this study, we aim to determine a crystal structure parameter which universally explains the emergence of superconductivity and evolution of T_c in the BiCh_2 family. We studied two isovalent-substitution systems, $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$, which exhibit similar superconductivity phase diagrams as shown in Fig. 1. To determine changes in crystal structure parameters and their relationship to superconductivity, we performed powder synchrotron X-ray diffraction (XRD) and Rietveld refinement for these two systems.

Here, we show that the structure parameter essential for the emergence of bulk SC in the BiCh_2 -based family is the in-plane chemical pressure, but not the Bi–Ch bond lengths or in-plane Ch–Bi–Ch angle. Furthermore, we show that T_c for $\text{REO}_{0.5}\text{F}_{0.5}\text{BiCh}_2$ superconductors can be universally determined from the in-plane chemical pressure. We believe that the important role of in-plane chemical pressure in the evolution of superconductivity in $\text{REO}_{0.5}\text{F}_{0.5}\text{BiCh}_2$ demonstrated here will be useful for designing new Bi–Ch-based layered superconductors with high T_c .

Results

Evolution of crystal structure parameters. We performed powder synchrotron XRD and Rietveld refinement analysis for $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$. Typical Rietveld refinement profiles for $x=0.6$ and $y=0.5$ are displayed in the supplementary information (Figure S1). Although minute impurity phases of the RE fluorides were detected, the structures of the main phase was refined using a tetragonal $P4/nmm$ space group. The crystal structure parameters are plotted as a function of x (or y) in Fig. 2; the crystal structure data are listed in the supplementary information (Table S1). In Fig. 2, the data points for the samples showing bulk SC are highlighted with orange-filled circles. From Fig. 1, we note that $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ exhibit similar superconductivity phase diagrams as a function of x or y . Therefore, if a crystal structure parameter for these two series changes similarly, the parameter should be considered essential for the emergence of superconductivity in these series.

Figure 2a,b show the x (or y) dependences of the lattice constants a and c for $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ and $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$, respectively. In $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$, both lattice constants decrease with increasing x due to increasing concentration of Nd^{3+} (with an ionic radius of 112 pm, assuming a coordination number of 8), which is smaller than Ce^{3+} (with an ionic radius of 114 pm). Conversely, in $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$, both lattice constants increase with increasing y due to the increase of Se^{2-} (with an ionic radius of 198 pm, assuming a coordination number of 6) concentration, which is larger than S^{2-} (with an ionic radius of 184 pm). These contrasting changes in lattice constants suggest that the evolution of superconductivity in these two series cannot be explained by simple lattice contraction or expansion.

Figure 2c shows the x (or y) dependences of Ch1-Bi-Ch1 angle for $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ and $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$. The Ch1-Bi-Ch1 angle is an indicator of the flatness of the Bi-Ch1 plane. Since electrons in the hybridized Bi-6p/Ch-p orbitals (S-3p or Se-4p) are essential for the emergence of superconductivity in $BiCh_2$ -based superconductors, a flatter Bi-Ch1 plane should facilitate the emergence of bulk SC with high T_c . A study of the crystal structure of $CeO_{1-x}F_xBiS_2$ with different F concentrations demonstrated that a flatter Bi-S1 plane resulted in higher superconducting properties³⁶. In $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$, the Ch1-Bi-Ch1 angle approaches 180° with increasing x , which indicates that the Bi-S1 plane becomes flatter with Nd substitution, and superconductivity is induced. In contrast, in $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$, the dependence of the Ch1-Bi-Ch1 angle on Se concentration decreases with increasing y , leading to distortion of the Bi-Ch1 plane. Therefore, the Ch1-Bi-Ch1 angle (i.e. the flatness of the Bi-Ch1 plane) cannot explain the evolution of superconductivity in these $BiCh_2$ -based superconductors. The contrasting changes in the in-plane structure (flatness) may be due to the difference in Ch^{2-} ions at the Ch1 site. Figure 2d shows the y dependence of Se occupancy at the Ch1 site, indicating that Se ions selectively occupy the in-plane Ch1 site. At $x=0.5$, approximately 90% of Ch1 sites are occupied with Se. Recently, a similar site selectivity of Se in $LaO_{1-x}F_xBiS_2$ single crystals was reported³⁷. The preferential occupation of Se^{2-} , which has a larger ionic radius, in the two-dimensional Bi-Ch1 plane may cause distortion.

Figure 2e shows the x (or y) dependences of the three different Bi-Ch distances. Figure 2e (right) demonstrates the $BiCh_2$ layer structure, where one Bi ion is coordinated by six Ch ions. The shortest Bi-Ch distance is the Bi-Ch2 distance along the c -axis. The Bi-Ch2 distance in $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ decreases slightly with increasing x . In contrast, the Bi-Ch2 distance in $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$ increases with increasing y . Therefore, this indicates that Bi-Ch2 distance cannot be correlated to the evolution of superconductivity within these two series. The Bi-Ch1 (in-plane) distance in $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ decreases with increasing x . In contrast, that in $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$ increases with increasing y . Thus, the in-plane Bi-Ch1 distance cannot explain the evolution of superconductivity. The longest Bi-Ch distance is Bi-Ch1 (inter-plane), which roughly corresponds to the inter-layer distance of two $BiCh_2$ layers. The Bi-Ch1 (inter-plane) distance in $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ does not change much upon Nd substitution, while that in $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$ shows a noticeable increase with increasing y . Therefore, the Bi-Ch1 (inter-plane) distance cannot typically explain the evolution of superconductivity.

Although we expected the Ch1-Bi-Ch1 angle or one of the Bi-Ch distances to exhibit similar behaviour in $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ and $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$, we could not identify any clear correlation between them. However, the assumption that the in-plane Bi-Ch1 distance should, to a certain extent, correlate with the evolution of superconductivity stems from the facts that the superconductivity is induced in the Bi-Ch1 plane, and superconducting properties change remarkably with change in the crystal structure without altering the electron carrier concentration (F concentration). Therefore, we introduce the concept of *in-plane chemical pressure* to discuss the relationship between in-plane structure and the evolution of superconductivity in $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$, $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$, and other $REO_{0.5}F_{0.5}BiCh_2$ superconductors.

Influence of in-plane chemical pressure on superconductivity. Figure 3a shows schematics of compression or expansion of the Bi-Ch plane caused by Nd or Se substitution. In $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$, Bi-Ch1 planes are compressed owing to a decrease in the volume of spacer layers with increasing Nd concentration. The compression of the Bi-Ch1 plane results in an enhancement of the packing density of $Bi^{2.5+}$ and S^{2-} ions within the superconducting plane: this is the so-called *in-plane chemical pressure*. In $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$, the in-plane Bi-Ch1 distance increases with increasing occupancy of Se at the Ch1 site. However, the increase of the in-plane Bi-Ch1 distance is smaller than that expected from the difference in the ionic radii of S^{2-} and Se^{2-} because the composition of the spacer layer (LaO) remains constant in $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$. Therefore, the packing density of $Bi^{2.5+}$ and Ch^{2-} ions in the superconducting plane is enhanced. This situation is similar to the enhancement of in-plane chemical pressure in $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$. In order to compare the magnitude of in-plane chemical pressure in the two series, we define in-plane chemical pressure using equation (1).

$$\text{In-plane chemical pressure} = (R_{Bi} + R_{Ch1})/Bi-Ch1 \text{ (in-plane)} \quad (1)$$

R_{Bi} is the ionic radius of $Bi^{2.5+}$. Here, we assume that the ionic radius of $Bi^{2.5+}$ is 104.19 pm, which is obtained from the average of the six Bi-S bonds (four in-plane Bi-S1 bonds, one inter-plane Bi-S1 bond, and one Bi-S2 bond) determined from the structure analysis of a single crystal of $LaO_{0.54}F_{0.46}BiS_2$ ³⁵. R_{Ch1} is the ionic radius of the chalcogen at the Ch1 site: 184 and 198 pm for S^{2-} and Se^{2-} , respectively. In the case of $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$, we calculated an average value for R_{Ch1} using the occupancy of Se at the

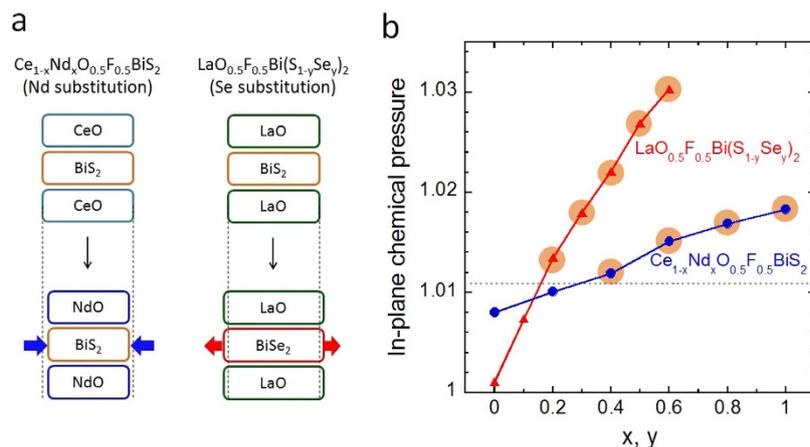


Figure 3. Influence of in-plane chemical pressure to crystal structure and superconductivity in $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ and $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$. (a) Schematics of changes in crystal structure with increasing in-plane chemical pressure in $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ and $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$. In $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$, the volume of spacer layer decreases with increasing Nd concentration (x), and Bi-S plane is compressed; hence, in-plane chemical pressure is enhanced. In $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$, the volume of superconducting Bi-Ch1 layer increases with increasing Se concentration (y). However, the expansion of Bi-Ch1 plane is smaller than that expected from the ionic radii of S^{2-} and Se^{2-} because the composition of the spacer layer (LaO layer) remains constant; hence, in-plane chemical pressure is enhanced as well as in $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$. (b) In-plane chemical pressure of $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ and $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$, calculated using equation (1), are plotted as a function of x (or y). In both systems, bulk SC is induced with increasing chemical pressure. The dashed line at an in-plane chemical pressure of ~ 1.011 is an estimated boundary of Bulk-SC and non-SC regions.

Ch1 site. The Bi-Ch1 (in-plane) distances were obtained from the Rietveld refinement (Fig. 2e). We note that the chemical pressure derived from ionic radii is a simplified estimation, and it cannot describe the exact orbital overlap between Bi and Ch. Nevertheless, the estimation appears very useful to discuss the relationship between crystal structure and superconductive properties of these systems.

The calculated in-plane chemical pressure is plotted as a function of x (or y) in Fig. 3b. For both systems, the in-plane chemical pressure increases with increasing x (or y). Surprisingly, the chemical pressure at which bulk SC is induced is similar (above ~ 1.011) in both $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ and $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$. Based on these experimental facts, we suggest that the emergence of bulk SC in these systems can be explained by the increase of in-plane chemical pressure. Enhancement of the in-plane chemical pressure would enhance the overlap of Bi-6p and Ch-p orbitals, which would result in an increase of the metallic conductivity and induce bulk SC in the $REO_{0.5}F_{0.5}BiCh_2$ family.

Discussion

Figure 4 shows the plot of T_c for $Ce_{1-x}Nd_xO_{0.5}F_{0.5}BiS_2$ and $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$ as a function of in-plane chemical pressure. To clarify a general tendency of superconductivity in the $REO_{0.5}F_{0.5}BiCh_2$ family, we added data points of $Nd_{0.8}Sm_{0.2}O_{0.5}F_{0.5}BiS_2$ and $Nd_{0.6}Sm_{0.4}O_{0.5}F_{0.5}BiS_2$ (from this study), $PrO_{0.5}F_{0.5}BiS_2$ ¹⁶, and single crystals of $NdO_{0.5}F_{0.5}BiS_2$ ³⁸, $SmO_{0.5}F_{0.5}BiS_2$ ¹⁸, and $LaO_{0.5}F_{0.5}BiS_2$ ($y = 1$)³⁹. Interestingly, data points for all the $REO_{0.5}F_{0.5}BiS_2$ -type compounds are located on a single curve bounding the blue region. Notably, the data points for $CeO_{0.5}F_{0.5}BiS_2$ and $SmO_{0.5}F_{0.5}BiS_2$, which do not exhibit superconductivity, are located to the left of the boundary (in-plane chemical pressure < 1.011). These facts suggest that the emergence of superconductivity and the T_c of $REO_{0.5}F_{0.5}BiS_2$ -type materials simply depend on the magnitude of in-plane chemical pressure. According to the Bardeen-Cooper-Schrieffer (BCS) theory involving electron-phonon mechanisms⁴⁰, the enhancement of T_c can be explained by an increase of phonon frequency and/or an enhancement of the density of states at the Fermi level. Usually, in metals, an increase in the orbital overlap should decrease the density of states because of an increase of bandwidth. Therefore, the increase in the density of state cannot simply explain the enhancement of T_c in this series. Although phonon frequency may be enhanced with increasing in-plane chemical pressure, further experimental and theoretical investigations are needed to elucidate the mechanisms of the enhancement of T_c with increasing in-plane chemical pressure.

A further inspection of Fig. 4 reveals that T_c for $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$ and $LaOBiSe_2$, shown by the red region, are clearly lower than those for the $REO_{0.5}F_{0.5}BiS_2$ -type series. An important observation is that T_c vs. in-plane chemical pressure plots for $REO_{0.5}F_{0.5}BiS_2$ and $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$ lie in different regions. For the $LaO_{0.5}F_{0.5}Bi(S_{1-y}Se_y)_2$ compounds, in-plane chemical pressure is relatively higher than that in $REO_{0.5}F_{0.5}BiS_2$, but their T_c 's are noticeably lower than those of $REO_{0.5}F_{0.5}BiS_2$. As described above, Se

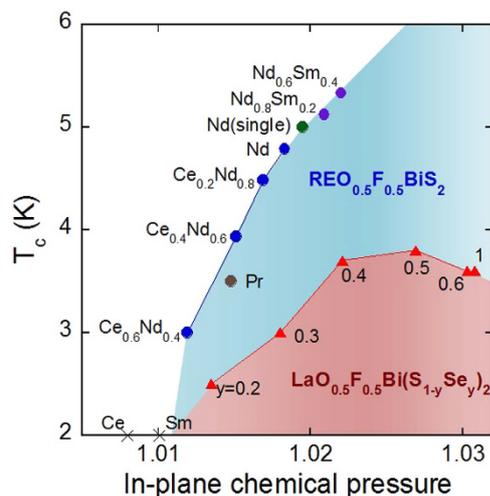


Figure 4. Relationship between T_c and degree of in-plane chemical pressure in $\text{REO}_{0.5}\text{F}_{0.5}\text{BiCh}_2$. The data of T_c and the in-plane chemical pressure of $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ are plotted with those of $\text{Nd}_{0.8}\text{Sm}_{0.2}\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{Nd}_{0.6}\text{Sm}_{0.4}\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ (from this study), $\text{PrO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ ¹⁶, and single crystals of $\text{NdO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ ³⁸, $\text{SmO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ ¹⁸, and $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiSe}_2$ ($y=1$)³⁹. The curves for $\text{REO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ (blue) and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ (red) lie on different regions.

preferably occupies the in-plane Ch1 site. Therefore, it can be considered that superconductivity in $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ is induced in the Bi-(S,Se) or Bi-Se plane. From the perspective of the BCS theory, the lower T_c in the Bi-Se plane than in the Bi-S plane can be explained by a lower phonon frequency in the former owing to the larger atomic number of Se. To confirm the above assumption and the mechanisms of superconductivity in the BiCh_2 -based superconductor family, further studies of other $\text{REO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ superconductors, such as $\text{CeO}_{0.5}\text{F}_{0.5}\text{BiSe}_2$ or $\text{NdO}_{0.5}\text{F}_{0.5}\text{BiSe}_2$, are needed for comparison. From Fig. 4, we conclude that a higher T_c should be obtained for the Bi-S plane with a higher in-plane chemical pressure.

Finally, we briefly discuss the HP phase of $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiS}_2$, which shows the highest T_c among BiCh_2 -based superconductors^{13,28–32}. T. Tomita *et al.* reported that the crystal structure of $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ above 0.7 GPa was monoclinic³¹. In the monoclinic structure, the Bi-S1 plane is distorted into a zigzag chain with a Bi-S1 distance of 2.72 Å; the zigzag chains are connected with a Bi-S1 distance of 3.03 Å. In addition, we have previously reported the in-plane anisotropy of the upper critical field within the Bi-S1 plane in an HP phase of $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiS}_2$, implying the emergence of quasi-one-dimensional superconducting states in the distorted plane⁴¹. Calculating the value of chemical pressure by substituting a Bi-S1 distance of 2.72 Å in equation (1) yields chemical pressure of 1.06. In this case, the chemical pressure is not the in-plane chemical pressure but instead is the quasi-one-dimensional chemical pressure. A chemical pressure of 1.06 and T_c of 10 K roughly lie on the extrapolated T_c -chemical pressure curve for the other $\text{REO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ superconductors shown in Fig. 4. This suggests that the enhancement of quasi-one-dimensional chemical pressure, but not in-plane chemical pressure, is responsible for the evolution of superconductivity in $\text{REO}_{0.5}\text{F}_{0.5}\text{BiCh}_2$ compounds. To further discuss and understand the relationship between chemical pressure and superconductivity in the BiCh_2 family, investigations of crystal structure and superconducting properties of new BiS_2 -based, BiSe_2 -based and/or BiTe_2 -based compounds with various spacer layers are needed.

In summary, we analysed the crystal structure of the optimally doped BiCh_2 -based superconductors, $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ using powder synchrotron XRD. We investigated the relationship between the crystal structure and the superconducting properties for these compounds. We found that an enhancement of the in-plane chemical pressure could induce bulk SC in both systems. Furthermore, we revealed that T_c for the $\text{REO}_{0.5}\text{F}_{0.5}\text{BiCh}_2$ superconductors could be determined by the magnitude of in-plane chemical pressure and type of Ch element that composes the superconducting Bi-Ch planes. In addition, we extended our results to include a monoclinic phase (HP phase) of $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiS}_2$. We suggested that the enhancement of the quasi-one-dimensional chemical pressure within a Bi-Ch chain, but not of the in-plane (two-dimensional) chemical pressure, is required for high T_c in BiCh_2 -based superconductors.

Methods

Polycrystalline samples of $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ used in this study were prepared using solid state reaction. The detailed synthesis procedures were described in previous reports^{33,34}. Powder synchrotron X-ray powder diffraction measurements were performed at room temperature at the BL02B2 experimental station of SPring-8 (JASRI; Proposal No. 2014B1003, 2014B1071, and 2015A1441). The wavelength of the radiation beam was 0.49542(4) Å. We have performed the Rietveld refinement

(RIETAN-FP⁴²) for $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-y}\text{Se}_y)_2$ using a typical structure model of REOBiCh₂-based superconductor with a tetragonal space group of $P4/nmm$ ^{35,37}. Contributions from impurity phases of rare-earth fluorides (REF₃) and/or Bi₂S₃ are included in Rietveld refinement. In addition, we have analysed a crystal structure for two related compounds of $\text{Nd}_{0.8}\text{Sm}_{0.2}\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and $\text{Nd}_{0.6}\text{Sm}_{0.4}\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$ to enrich the discussion part³³. The obtained crystal structure parameters are summarized in Table S1. The schematic images of crystal structure were drawn using VESTA⁴³.

References

1. Bednorz, J. G. & Müller, K. A. Possible high T_c superconductivity in the Ba–La–Cu–O system. *Z. Physik B Condensed Matter* **64**, 189–193 (1986).
2. Kamihara, Y. *et al.* Iron-Based Layered Superconductor $\text{La}[\text{O}_{1-x}\text{F}_x]\text{FeAs}$ ($x = 0.05\text{--}0.12$) with $T_c = 26$ K. *J. Am. Chem. Soc.* **130**, 3296–3297 (2008).
3. Chen, X. H. *et al.* Superconductivity at 43 K in $\text{SmFeAsO}_{1-x}\text{F}_x$. *Nature* **453**, 761–762 (2008).
4. Ren, Z. A. *et al.* Superconductivity at 55 K in Iron-Based F-Doped Layered Quaternary Compound $\text{Sm}[\text{O}_{1-x}\text{F}_x]\text{FeAs}$. *Chinese Phys. Lett.* **25**, 2215 (2008).
5. Rotter, M., Tegel, M. & Johrendt, D. Superconductivity at 38 K in the Iron Arsenide $(\text{Ba}_{1-x}\text{K}_x)\text{Fe}_2\text{As}_2$. *Phys. Rev. Lett.* **101**, 107006 (1–4) (2008).
6. Wang, X. C. *et al.* The superconductivity at 18 K in LiFeAs system. *Solid State Commun.* **148**, 538–540 (2008).
7. Ogino, H. *et al.* Superconductivity at 17 K in $(\text{Fe}_2\text{P}_2)(\text{Sr}_4\text{Sc}_2\text{O}_6)$: a new superconducting layered pnictide oxide with a thick perovskite oxide layer. *Supercond. Sci. Technol.* **22**, 075008 (1–4) (2009).
8. Zhu, X. *et al.* Transition of stoichiometric $\text{Sr}_2\text{VO}_3\text{FeAs}$ to a superconducting state at 37.2 K. *Phys. Rev. B* **79**, 220512 (1–4) (2009).
9. Lee, C. H. *et al.* Effect of Structural Parameters on Superconductivity in Fluorine-Free LnFeAsO_{1-y} (Ln = La, Nd). *J. Phys. Soc. Jpn.* **77**, 083704 (1–4) (2008).
10. Mizuguchi, Y. *et al.* Anion height dependence of T_c for the Fe-based superconductor. *Supercond. Sci. Technol.* **23**, 054013 (1–5) (2010).
11. Kuroki, K. *et al.* Pnictogen height as a possible switch between high- T_c nodeless and low- T_c nodal pairings in the iron-based superconductors. *Phys. Rev. B* **79**, 224511 (1–16) (2009).
12. Mizuguchi, Y. *et al.* BiS₂-based layered superconductor $\text{Bi}_4\text{O}_4\text{S}_3$. *Phys. Rev. B* **86**, 220510 (1–5) (2012).
13. Mizuguchi, Y. *et al.* Superconductivity in Novel BiS₂-Based Layered Superconductor $\text{LaO}_{1-x}\text{F}_x\text{BiS}_2$. *J. Phys. Soc. Jpn.* **81**, 114725 (1–5) (2012).
14. Usui, H., Suzuki, K. & Kuroki, K. Minimal electronic models for superconducting BiS₂ layers. *Phys. Rev. B* **86**, 220501 (1–5) (2012).
15. Xing, J. *et al.* Superconductivity Appears in the Vicinity of an Insulating-Like Behavior in $\text{CeO}_{1-x}\text{F}_x\text{BiS}_2$. *Phys. Rev. B* **86**, 214518 (1–5) (2012).
16. Jha, R. *et al.* Synthesis and superconductivity of new BiS₂ based superconductor $\text{PrO}_{0.5}\text{F}_{0.5}\text{BiS}_2$. *J. Supercond. Nov. Magn.* **26**, 499–502 (2013).
17. Demura, S. *et al.* BiS₂-based superconductivity in F-substituted NdOBiS₂. *J. Phys. Soc. Jpn.* **82**, 033708 (1–3) (2013).
18. Thakur, G. S. *et al.* Synthesis and properties of $\text{SmO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ and enhancement in T_c in $\text{La}_{1-y}\text{Sm}_y\text{O}_{0.5}\text{F}_{0.5}\text{BiS}_2$. *Inorg. Chem.* **54**, 1076–1081 (2015).
19. Yazici, D. *et al.* Superconductivity of F-substituted LnOBiS₂ (Ln = La, Ce, Pr, Nd, Yb) compounds. *Philos. Mag.* **93**, 673 (1–8) (2012).
20. Okada, T. *et al.* Topotactic Synthesis of a new BiS₂-based superconductor $\text{Bi}_2(\text{O},\text{F})\text{S}_2$. *Appl. Phys. Express* **8**, 023102 (1–4) (2015).
21. Shao, J. *et al.* Superconductivity in $\text{BiO}_{1-x}\text{F}_x\text{BiS}_2$ and possible parent phase of $\text{Bi}_4\text{O}_4\text{S}_3$ superconductor. *Supercond. Sci. Technol.* **28**, 015008 (1–6) (2015).
22. Lin, X. *et al.* Superconductivity induced by La doping in $\text{Sr}_{1-x}\text{La}_x\text{FBiS}_2$. *Phys. Rev. B* **87**, 020504 (1–4) (2013).
23. Jha, R., Tiwari, B. & Awana, V. P. S. Appearance of bulk superconductivity under hydrostatic pressure in $\text{Sr}_{0.5}\text{RE}_{0.5}\text{FBiS}_2$ (RE = Ce, Nd, Pr, and Sm) compounds. *J. Appl. Phys.* **117**, 013901 (1–7) (2015).
24. Zhai, H. F. *et al.* Possible Charge-density wave, superconductivity and f-electron valence instability in EuBiS_2F . *Phys. Rev. B* **90**, 064518 (1–9) (2014).
25. Phelan, W. A. *et al.* Stacking Variants and Superconductivity in the Bi–O–S System. *J. Am. Chem. Soc.* **135**, 5372–5374 (2013).
26. Zhai, H. F. *et al.* Anomalous Eu Valence State and Superconductivity in Undoped $\text{Eu}_3\text{Bi}_2\text{S}_4\text{F}_4$. *J. Am. Chem. Soc.* **136**, 15386–15393 (2014).
27. Maziopa, A. K. *et al.* Superconductivity in a new layered bismuth oxyselenide: $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiSe}_2$. *J. Phys.: Condens. Matter* **26**, 215702 (1–5) (2014).
28. Deguchi, K. *et al.* Evolution of superconductivity in $\text{LaO}_{1-x}\text{F}_x\text{BiS}_2$ prepared by high pressure technique. *EPL* **101**, 17004 (p1–p5) (2013).
29. Mizuguchi, Y. *et al.* Stabilization of High- T_c Phase of BiS₂-Based Superconductor $\text{LaO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ Using High-Pressure Synthesis. *J. Phys. Soc. Jpn.* **83**, 053704 (1–4) (2014).
30. Kotegawa, H. *et al.* Pressure Study of BiS₂-Based Superconductors $\text{Bi}_4\text{O}_4\text{S}_3$ and $\text{La}(\text{O},\text{F})\text{BiS}_2$. *J. Phys. Soc. Jpn.* **81**, 103702 (1–4) (2012).
31. Tomita, T. *et al.* Pressure-Induced Enhancement of Superconductivity and Structural Transition in BiS₂-Layered $\text{LaO}_{1-x}\text{F}_x\text{BiS}_2$. *J. Phys. Soc. Jpn.* **83**, 063704 (1–4) (2014).
32. Wolowiec, C. T. *et al.* Enhancement of superconductivity near the pressure-induced semiconductor–metal transition in the BiS₂-based superconductors $\text{LnO}_{0.5}\text{F}_{0.5}\text{BiS}_2$ (Ln = La, Ce, Pr, Nd). *J. Phys.: Condens. Matter* **25**, 42220 (1–6) (2013).
33. Kajitani, J. *et al.* Chemical pressure effect on superconductivity of BiS₂-based $\text{Ce}_{1-x}\text{Nd}_x\text{O}_{1-y}\text{F}_y\text{BiS}_2$ and $\text{Nd}_{1-z}\text{Sm}_z\text{O}_{1-y}\text{F}_y\text{BiS}_2$. *J. Phys. Soc. Jpn.* **84**, 044712 (1–6) (2015).
34. Hiroi, T. *et al.* Evolution of superconductivity in BiS₂-based superconductor $\text{LaO}_{0.5}\text{F}_{0.5}\text{Bi}(\text{S}_{1-x}\text{Se}_x)_2$. *J. Phys. Soc. Jpn.* **84**, 024723 (1–4) (2015).
35. Miura, A. *et al.* Crystal structures of $\text{LaO}_{1-x}\text{F}_x\text{BiS}_2$ ($x \sim 0.23, 0.46$): effect of F doping on distortion of Bi–S plane. *J. Solid State Chem.* **212**, 213–217 (2014).
36. Miura, A. *et al.* Structure, Superconductivity, and Magnetism of $\text{Ce}(\text{O},\text{F})\text{BiS}_2$ Single Crystals. *Cryst. Growth Des.* **15**, 39–44 (2015).
37. Tanaka, M. *et al.* Site Selectivity on Chalcogen Atoms in Superconducting $\text{La}(\text{O},\text{F})\text{BiSe}$. *Appl. Phys. Lett.* **106**, 112601 (1–5) (2015).
38. Nagao, M. *et al.* Structural Analysis and Superconducting Properties of F-Substituted NdOBiS₂ Single Crystals. *J. Phys. Soc. Jpn.* **82**, 113701 (1–4) (2013).
39. Tanaka, M. *et al.* First single crystal growth and structural analysis of superconducting layered bismuth oxyselenide; $\text{La}(\text{O},\text{F})\text{BiSe}_2$. *J. Solid State Chem.* **219**, 168–172 (2014).

40. Bardeen, J., Cooper, L. N. & Schrieffer, J. R. Theory of Superconductivity. *Phys. Rev.* **108**, 1175–1204 (1957).
41. Mizuguchi, Y. *et al.* Anisotropic upper critical field of BiS₂-based superconductor LaO_{0.5}F_{0.5}BiS₂. *Phys. Rev. B* **89**, 174515 (1–7) (2014).
42. Izumi, F. & Momma Three-Dimensional Visualization in Powder Diffraction. *Solid State Phenom.* **130**, 15–20 (2007).
43. Momma, K. & Izumi, F. VESTA: a three-dimensional visualization system for electronic and structural analysis. *J. Appl. Crystallogr.* **41**, 653–658 (2008).

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Author Contributions

Y.M. and A.M. planned the research. Y.M., J.K., T.H. and O.M. prepared polycrystalline samples and studied superconducting properties. Y.M., A.M., K.T., N.K., E.M., C.M. and Y.K. performed synchrotron XRD and crystal structure analysis. Y.M. wrote the manuscript.

Additional Information

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