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Assembly of luminescent ordered multilayer thin-films based on oppositely-charged MMT and magnetic NiFe-LDHs nanosheets with ultra-long lifetimes

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In this present report, luminescent ordered multilayer thin films (OMFs) based on oppositely-charged inorganic nanosheets and the different oppositely-charged chromophores were fabricated via layer-by-layer assembly method. Exfoliated layered double hydroxides (LDHs) and montmorillonite (MMT) nanosheets with opposite charges can be expected to provide a pseudo electronic microenvironment (PEM) which has not been declared in previous literatures, and transition metal-bearing LDHs nanosheets can offer an additional ferromagnetic effect (FME) for the chromophores at the same time. Surprisingly, the luminescent lifetimes of those OMFs with PEM and FME are significantly prolonged compared with that of the pristine chromophores, even much longer than those of OMFs without oppositely-charged and ferromagnetic architecture. Therefore, it is highly expected that the PEM and FME formed by oppositely-charged and transition metal-bearing inorganic nanosheets have remarkable influence on obtaining better optical property, which suggests a new potential way to manipulate, control and develop the novel light-emitting materials and optical devices.

Layered materials have attracted attentions to their rich fundamental physics and potential applications in future multifunctional devices, and the nanosheets of those have been used as building units for making new designed organic-inorganic or inorganic-inorganic nanomaterials due to their intrinsic unique two-dimensional structure^{1–7}. LDHs and MMT, the inorganic layered minerals containing ionic nanosheets, have partially been widely applied in sensing, catalysis and energy storage *etc*^{8–13}. As the successful liquid exfoliation of layered materials and the fruitful assembly of the inorganic nanosheets with polyelectrolyte, other different inorganic nanosheets, even quantum dots and so on^{14–21}, layer-by-layer assembly method has been masterly used to build layered and ordered functional materials. Transition metal-bearing LDHs are known to have broader technological applications attributing to their special catalytic, electronic, optical, and magnetic properties^{22–28}. Specially, Sasaki *et al.* confirmed that transition metal-bearing LDHs' nanosheets can act as nanoscale ferromagnetic layers at room temperature, and their multilayer assemblies exhibited significant magneto-optical response²⁹. On the other hand, Liu *et al.*³⁰ utilized PVA as intermediate linkers to obtain uniform multilayer films (MMT/PVA/CoAl-LDHs/PVA)_n, and other scientists successfully fabricated the nano-dimensional hybrid materials with homogeneously precipitation of ionic nanosheets on the oppositely-charged nanosheets' surface via a simple hydrothermal process^{31–34}. But there is no relational reports that oppositely-charged nanosheets can be expected to form a PEM between the interlayer, and how the PEM affects the functional molecules in it.

Organic luminescent materials, imputed to their chemical stability and excellent luminescence properties, give rise to many possible applications in chemical sensors, optical devices, photovoltaic cells *etc*^{35–37}. But attempts to implement this idea have been hampered by their poor optical stabilities and relatively short service lifetime. Many scientists tried changing the external factors such as thermal and mechanical stimuli, to modify chemical structure of molecules for luminescent materials, thus improve the optical properties^{38–41}. Recently, organo-erbium systems, a potential organic optical amplifier material, combined a relatively longer fluorescence lifetime



to produce a medium with exceptional sensitization⁴². Especially, Duan's group successfully utilized non-magnetic LDHs nanosheets assemble well-oriented photoemissive quantum well structures in order to suppress the chromophores' π - π stacking^{43–48}. Herein, what we are most interested in are whether organic luminescent materials can be intercalated into the oppositely-charged MMT and magnetic NiFe-LDHs nanosheets, and how the PEM and FME affect the optical properties.

In this work, in order to confirm the universality of the PEM and FME's contribution on the different series chromophores, two kinds of chromophores with opposite charge, a photoactive divalent cationic bis(*N*-methylacridinium) (BNMA) and a luminescent π -conjugated anionic polymer poly [5-methoxy-2-(3-sulfopropoxy)-1,4-phenylenevinylene] potassium (APPV), were assembled with the NiFe-LDHs nanosheets and MMT nanosheets to fabricate series luminescent ordered multilayer thin films, respectively. In the nano-architecture, as the assembled PEM can provide the electronic field in nano-level system and the NiFe-LDHs can offer a constant FME to the chromophores (Figure 1), the OMFs exhibit remarkable optical properties with reasonably longer luminescent lifetimes. Herein, in this work, not only does we successfully fabricate the OMFs with PEM and FME' architecture, which can promote the hybrid systems with various optical functions, but also puts forward a new concept about the PEM and FME' effects on chromophores firstly.

Results

Characterization of luminescent OMFs based on ferromagnetic NiFe-LDHs containing photoactive divalent cations. The growths of $(\text{NiFe-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs by UV-vis absorption spectra were measured after different deposition cycles (Figure 2a), and the intensities of the characteristic absorption bands of BNMA at 264.0 nm and 371.5 nm correlate linearly with the number of deposition cycles (Figure 2a, inset), demonstrating a stepwise and

regular growth procedure. Furthermore, the fluorescence emission intensity at 508 nm displays a consistent increase with the number of deposition cycles, as shown in Figure 2b, and no obvious red or blue shifts occur compared with the pristine BNMA powders, suggesting there is no formation of BNMA aggregation throughout the whole assembly processing. This can also be visualized by irradiating the thin films under daylight and UV light (365 nm), respectively (Figure 2c), where these films exhibit uniform enhanced luminescence with increasing the number of deposition cycles. The $(\text{NiFe-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs surface is microscopically smooth and uniform, according to the top view of the SEM image (Figure 2d). Moreover, the side view of the SEM image (Figure 2d, inset) show that OMFs are continuous and homogeneous, with the obvious layer structure. Furthermore, small angle XRD peaks intensities of the as-prepared OMFs increasing with the layer number indicate that the OMFs are assembled by a stepwise and regular growth procedure with a period of *ca.* 4.7 nm (Figure 2e). Above all, the OMFs exhibit an ordered and regular film growth procedure and uniform enhanced luminescence as the number of deposition cycles increasing.

In order to verify the effect of NiFe-LDHs nanosheets' ferro-magnetism, the OMFs were studied by detecting luminescent lifetimes (Figure 2f). Surprisingly, it is amazing that the luminescent lifetimes of $(\text{NiFe-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs (12.75–13.85 ns) are prolonged nearly 3-fold compared with the $(\text{MgAl-LDHs}/\text{BNMA}@\text{PVS})_n$ OMFs (4.61–4.88 ns), and are also amazingly prolonged about 37-fold compared with the pristine powder BNMA (0.37 ns)⁴⁷. The profound increasing is partially ascribed to the isolation effect imposed by the rigid NiFe-LDHs nanosheets, preventing the formation of BNMA aggregation. But most importantly, NiFe-LDHs can offer a constant FME, thus definitely prolonging their luminescent lifetimes. Above all, FME is beneficial to obtain better photoemission behavior in the OMFs system.

Characterization of luminescent OMFs with PEM and FME containing photoactive divalent cations. The UV-visible absorption spectra of $(\text{MMT}/\text{BNMA}@\text{PVA}/\text{NiFe-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs with varying numbers of assembly steps are shown in Figure 3a, and a linear increase in absorbance at 264.0 nm or 371.5 nm was observed upon increasing the number of bilayers (Figure 3a, inset). Furthermore, the fluorescence emission intensity at 508 nm and photos under daylight and UV light (365 nm) of these films display uniform enhanced luminescence increasing with the number of deposition cycles, as shown in Figure 3b and 3c. SEM images (Figure 3d) shows that the films are so smooth, continuous and homogeneous and XRD patterns show that the OMFs are significantly ordered and grow with a period of *ca.* 7.4 nm (Figure 3e). Hence, the OMFs, with an ordered and regular structure and uniform enhanced yellow-green luminescence as the number of deposition cycles increasing, can be obtained.

In order to univocally attribute PEM's contribution, the OMFs were studied by detecting luminescent lifetimes, as shown in Figure 3f. Meaningfully, it is amazing that the luminescent lifetimes of $(\text{MMT}/\text{BNMA}@\text{PVA}/\text{NiFe-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs (15.38–16.52 ns) are extended more than 3-fold compared with the $(\text{MgAl-LDHs}/\text{BNMA}@\text{PVS})_n$ OMFs (4.61–4.88 ns), are amazingly prolonged nearly 45-fold compared with the pristine BNMA powders (0.37 ns)⁴⁷, and also significantly present a gradient growth compared with the $(\text{NiFe-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs. The profound increasing is partially due to the negative-charged MMT nanosheets' electrostatic attraction to BNMA and the isolation effect imposed by the rigid LDHs and MMT nanosheets, preventing the formation of BNMA aggregation. What's more, positive charged NiFe-LDHs and negative charged MMT nanosheets can form PEM and NiFe-LDHs is able to offer a constant FME, which will constrain the electron vibration in the conjugate structure of chromophores definitely, thus prolonging their luminescent lifetimes.

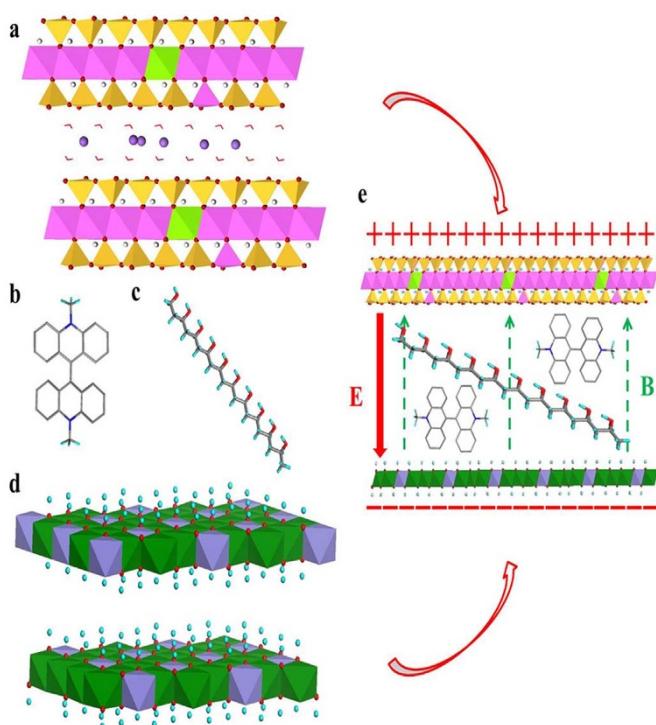


Figure 1 | (a) A representation of MMT, (b) Structure of APPV (c) Structure of PVA Chemical formula of APPV, (d) A representation of NiFe-LDHs, (e) The assembly of OMFs with the PEM and FME. pink: aluminum element, green: magnesium element, yellow: silicon element, dark green: nickel element, purple: iron element.

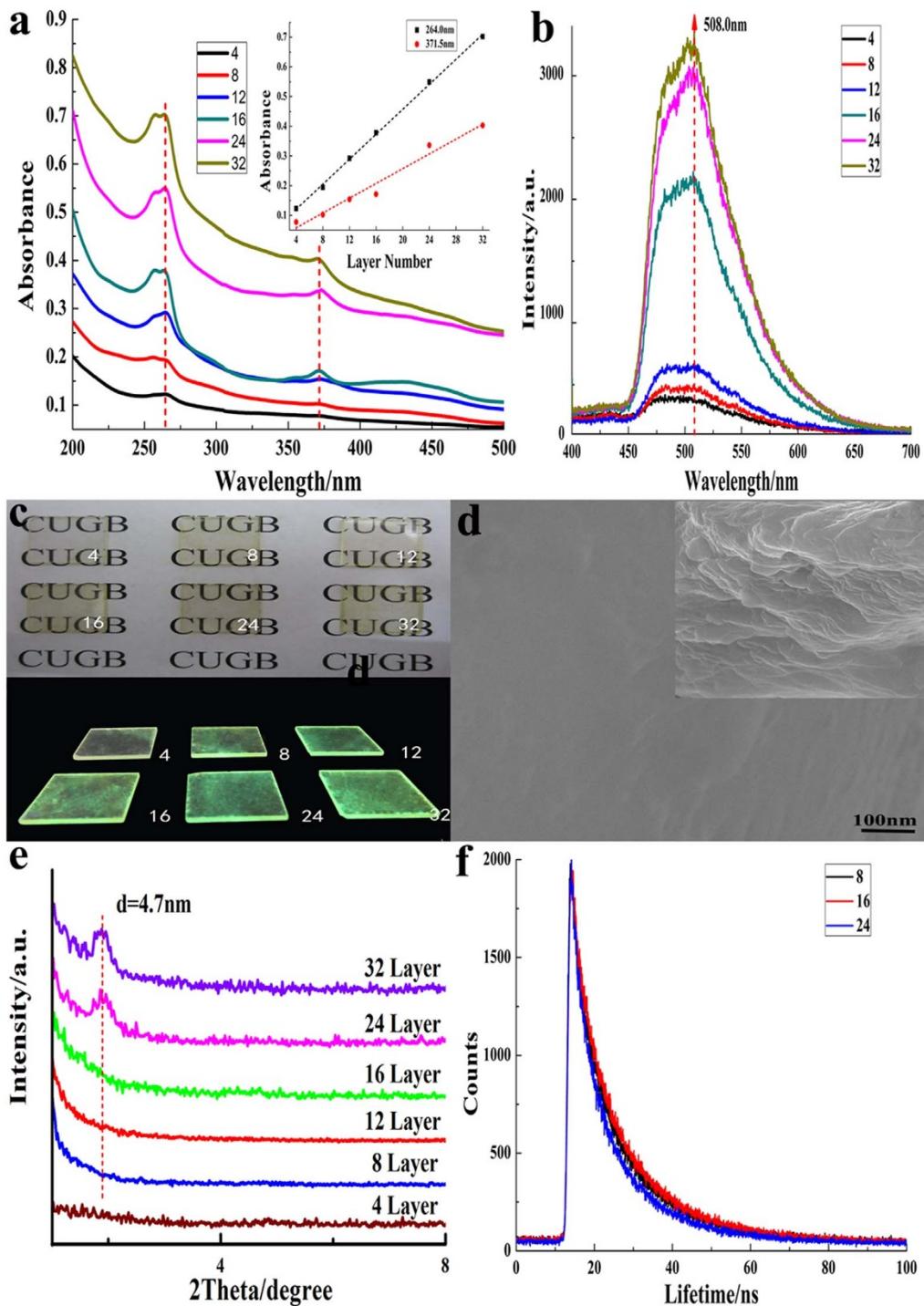


Figure 2 | Characterization of $(\text{NiFe-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs (a) UV-visible absorption spectra, the inset shows the absorbance increasing linear relationship in 264.0 nm and 371.5 nm. (b) Photoluminescence spectra. (c) Photographs of OMFs with different bilayer numbers under daylight and UV light (365 nm), respectively. (d) The top view of the SEM image, the inset shows the side view of the SEM image. (e) Small angle XRD patterns. (f) Fluorescence decay profiles.

Above all, the PEM and FME can significantly prolong the luminescent lifetimes, simultaneously.

Characterization of luminescent OMFs based on ferromagnetic NiFe-LDHs containing luminescent π -conjugated anionic polymer. In order to confirm the universality of the novel concept for more series chromophores, we selected a luminescent π -conjugated anionic polymer APPV to assemble the luminescent OMFs. The UV-visible absorption spectra of $(\text{NiFe-LDHs}/\text{APPV}@\text{PVA})_n$

OMFs with varying numbers of assembly steps are shown in Figure 4a, and the intensities of absorption peaks exhibit a linear correlation with the number of deposition cycles (Figure 4a, inset). Furthermore, the fluorescence emission intensity also displays a consistent increase with deposition cycles, as shown in Figure 4b, and photos of these films under daylight and UV light (365 nm) (Figure 4c) exhibit uniform enhanced luminescence with increasing n . SEM image (Figure 4d) and XRD patterns (Figure 4e) show that the film surface is smooth, continuous and homogeneous,

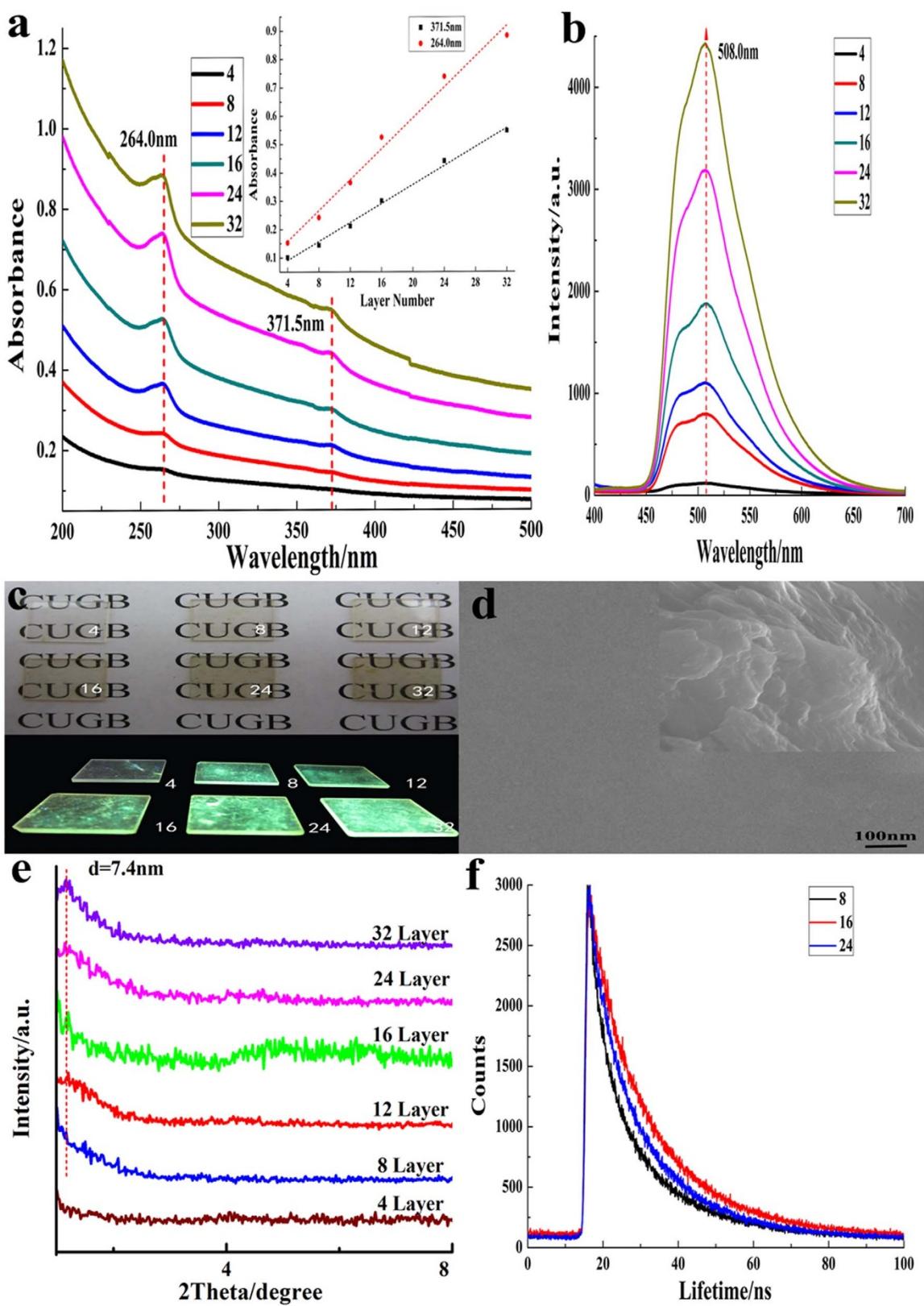


Figure 3 | Characterization of $(\text{MMT}/\text{BNMA}@\text{PVA}/\text{NiFe-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs (a) UV-visible absorption spectra, the inset shows the absorbance increasing linear relationship in 264.0 nm and 371.5 nm. (b) Photoluminescence spectra. (c) Photographs of OMFs with different bilayer numbers under daylight and UV light (365 nm), respectively. (d) The top view of the SEM image, the inset shows the side view of the SEM image. (e) Small angle XRD patterns. (f) Fluorescence decay profiles.

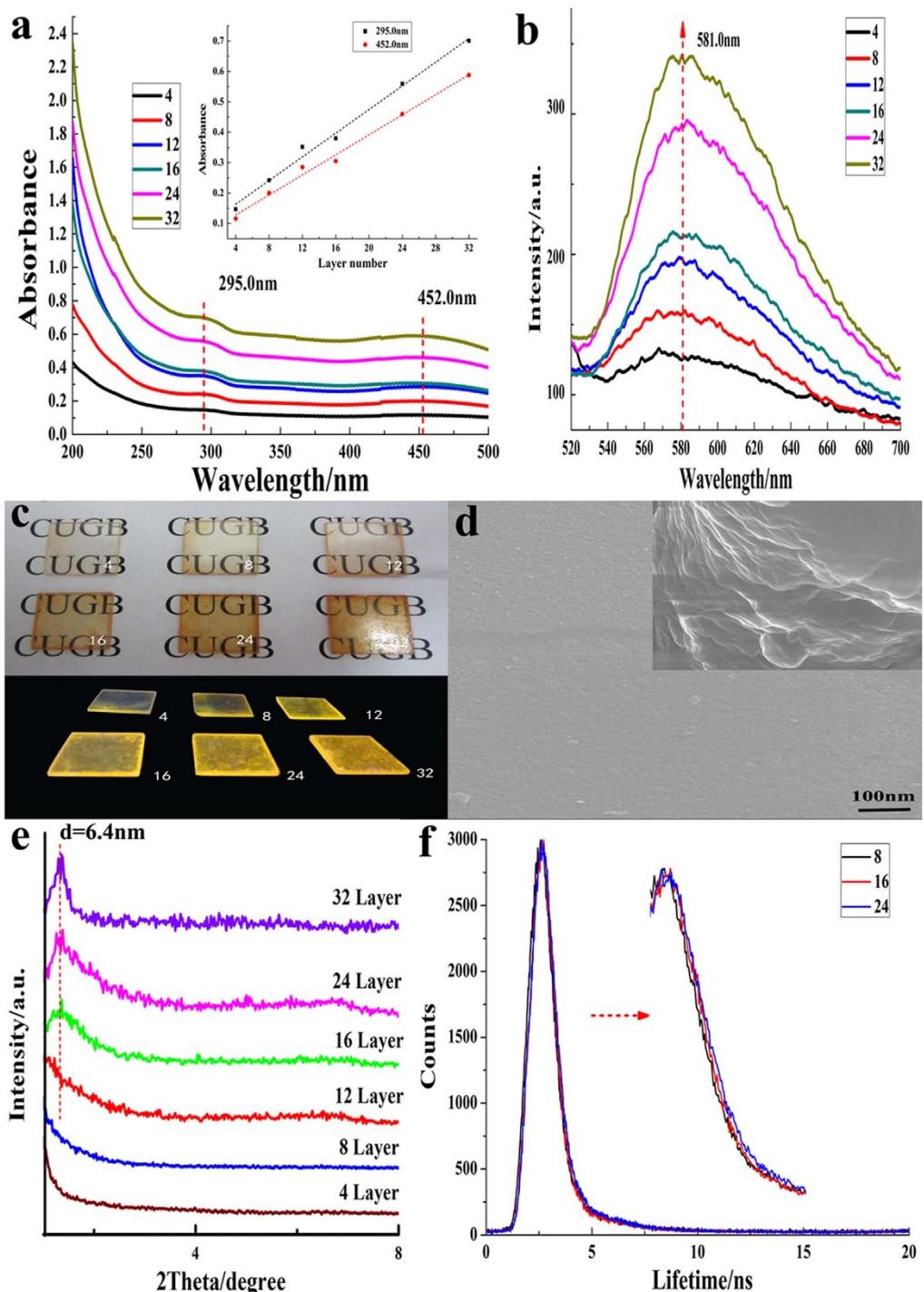


Figure 4 | Characterization of $(\text{NiFe-LDHs}/\text{APPV}@\text{PVA})_n$ OMFs (a) UV-visible absorption spectra, the inset shows the absorbance increasing linear relationship in 295.0 nm and 452.0 nm. (b) Photoluminescence spectra. (c) Photographs of OMFs with different bilayer numbers under daylight and UV light (365 nm), respectively. (d) The top view of the SEM image, the inset shows the side view of the SEM image. (e) Small angle XRD patterns. (f) Fluorescence decay profiles.

growing with a period of ca. 6.4 nm. Above all, the OMFs exhibit an ordered and regular film growth procedure and uniform enhanced yellow luminescence as the deposition cycles increasing.

It is amazing that the luminescent lifetimes of $(\text{NiFe-LDHs}/\text{APPV}@\text{PVA})_n$ OMFs (1.26–1.38 ns) (Figure 4f) are about 1.5 times as long as $(\text{MgAl-LDHs}/\text{APPV})_n$ OMFs' (0.66–0.81 ns), and are also amazingly 2.3 times the size of the pristine APPV solution's (0.60 ns)⁴⁸. This remarkable increase is partially related to the uniform dispersion of the APPV anion between NiFe-LDHs nanosheets, with the existence of PVA polyanions. But the critical and important

reason is that NiFe-LDHs can offer a constant FME, which affect the conjugate structure's electron cloud of chromophores, thus definitely prolonging their luminescent lifetimes. Above all, FME is beneficial to obtain better photoemission behavior in the OMFs system containing the luminescent π -conjugated anionic polymer.

Characterization of luminescent OMFs with PEM and FME containing luminescent π -conjugated anionic polymer. In order to univocally attribute PEM's contribution to the π -conjugated anionic polymer, we also assemble the OMFs containing APPV

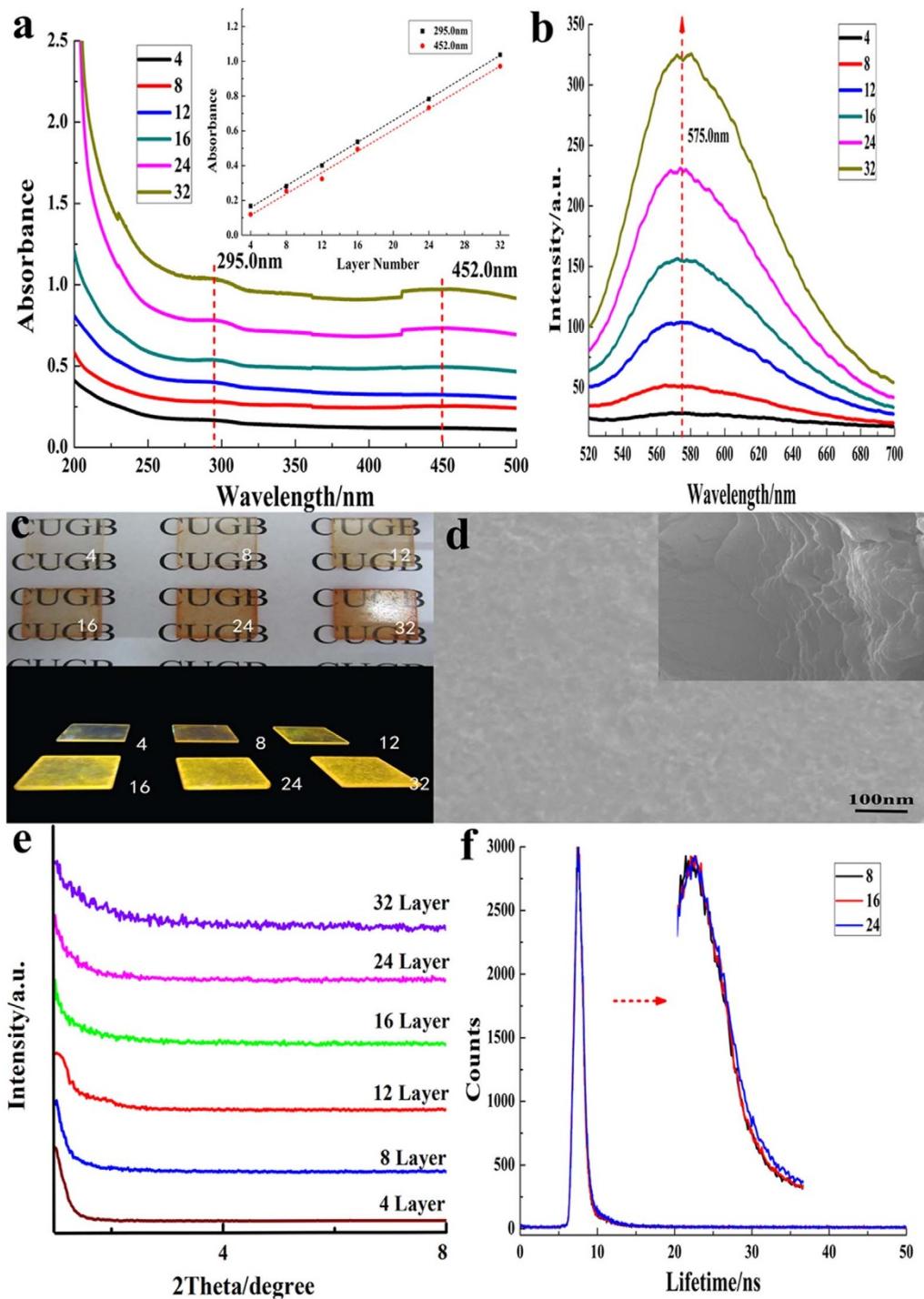


Figure 5 | Characterization of $(\text{MMT}/\text{APPV}@\text{PVA}/\text{NiFe-LDHs}/\text{APPV}@\text{PVA})_n$ OMFs (a) UV-visible absorption spectra, the inset shows the absorbance increasing linear relationship in 295.0 nm and 452.0 nm. (b) Photoluminescence spectra. (c) Photographs of OMFs with different bilayer numbers under daylight and UV light (365 nm), respectively. (d) The top view of the SEM image, the inset shows the side view of the SEM image. (e) Small angle XRD patterns. (f) Fluorescence decay profiles.

based on NiFe-LDHs and MMT nanosheets. The UV-visible absorption spectra of $(\text{MMT}/\text{APPV}@\text{PVA}/\text{NiFe-LDHs}/\text{APPV}@\text{PVA})_n$ OMFs with varying numbers of assembly steps are shown in Figure 5a, and linear increases in absorbance at 295.0 nm and 452.0 nm were observed upon increasing the number of deposition cycles (Figure 5a, inset), the fluorescence emission intensity also

displays a consistent increase with the number of deposition cycles, as shown in Figure 5b, and photos of these films exhibit uniform enhanced luminescence with increasing n under daylight and UV light (365 nm) (Figure 5b). SEM image (Figure 4d) show that the film surface is smooth, uniform, continuous and homogeneous, but small angle XRD have no peak, which verify the

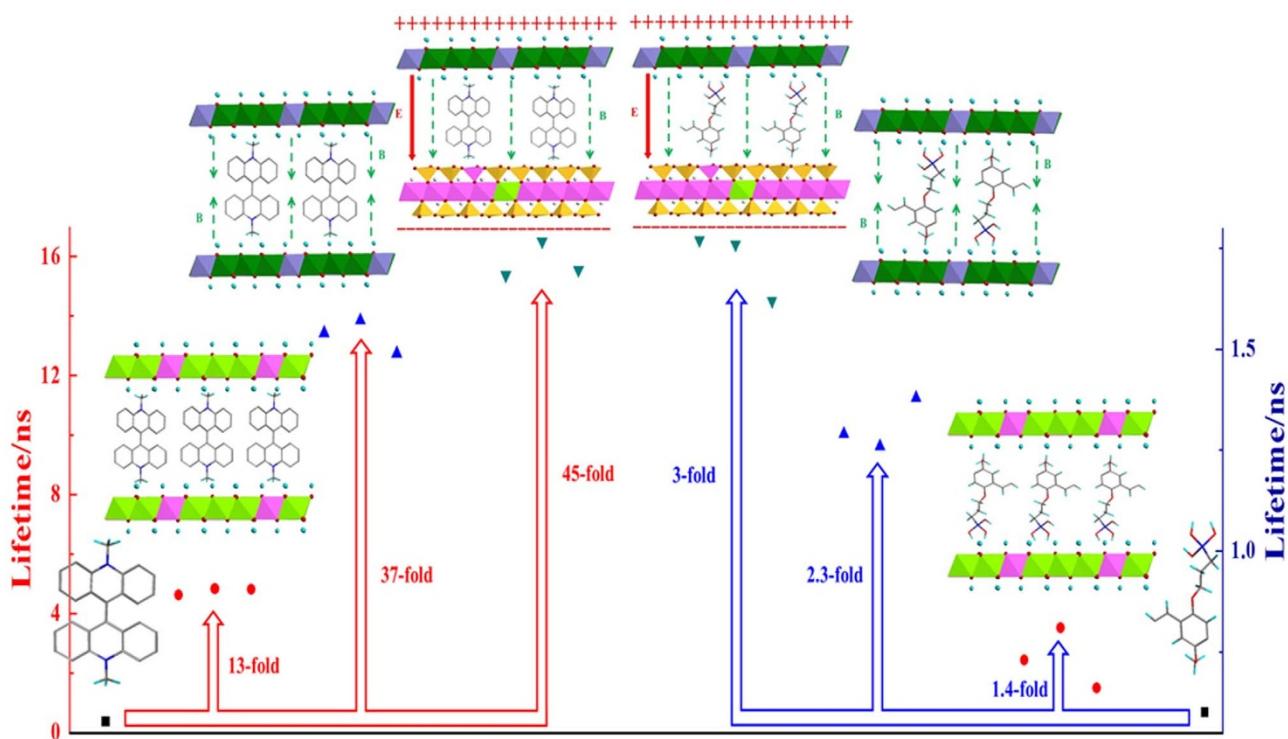


Figure 6 | The comparison of BNMA and APPV' lifetimes under the different environments. Black dots represent the luminescent lifetimes of pristine BNMA or APPV, the red stand for the lifetimes of $(\text{MgAl-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs or $(\text{MgAl-LDHs}/\text{APPV})_n$ OMFs, and the blue show the luminescent lifetimes of $(\text{NiFe-LDHs}/\text{BNMA}@\text{PVA})_n$ OMFs or $(\text{NiFe-LDHs}/\text{APPV}@\text{PVA})_n$ OMFs. The dark green stand for the luminescent lifetimes of $(\text{MMT}/\text{BNMA}@\text{PVA}/\text{NiFe-LDHs}/\text{BNMA}@\text{VA})_n$ OMFs or $(\text{MMT}/\text{APPV}@\text{PVA}/\text{NiFe-LDHs}/\text{APPV}@\text{PVA})_n$ OMFs.

interlayer spacing are enough large. Hence, the OMFs, with an ordered and regular structure and uniform enhanced yellow luminescence as the number of deposition cycles increasing, can be obtained.

Wondrously, it is amazing that the luminescent lifetimes of $(\text{MMT}/\text{APPV}@\text{PVA}/\text{NiFe-LDHs}/\text{APPV}@\text{PVA})_n$ OMFs (1.62–1.77 ns) are extended approximate 2.3-fold compared with the $(\text{MgAl-LDHs}/\text{APPV})_n$ OMFs (0.66–0.81 ns), are amazingly increased more than 3-fold compared with the pristine APPV solution (0.60 ns)⁴⁸, and are also significantly multiplied by 1.3 times of the $(\text{NiFe-LDHs}/\text{APPV}@\text{PVA})_n$ OMFs, as shown in Figure 5f. The prominent increasing is partially referred to the negative-charged LDHs nanosheets' electrostatic attraction to APPV and the isolation effect imposed by the rigid LDHs and MMT nanosheets, preventing the formation of APPV aggregation. But most dominantly, oppositely-charged NiFe-LDHs and MMT nanosheets can form PEM and NiFe-LDHs is able to offer a FME, which affect the conjugate structure's electron cloud of chromophores, thus definitely prolonging their luminescent lifetimes. According to this remarkable increase, the theory, as mentioned above, that PEM and FME can definitely prolong chromophores' luminescent lifetimes is verified.

Discussion

To sum up, this work assembled series of novel luminescent OMFs containing cationic BNMA or anionic APPV via layer-by-layer assembly method. Importantly, the chromophores were successfully confined in the oppositely-charged and ferromagnetic inorganic layer hosts, and it is demonstrated that the PEM and FME are fairly beneficial to enhancing the lifetimes of OMFs through apparently constraining electron vibration in the conjugate structure of chromophores, finally prolonging their luminescent lifetimes. As shown in Figure 6, we compared the lifetimes of BNMA and APPV under the different environments (the pristine, the nonmagnetic LDHs' isolation, FME plus the LDHs' isolation, and PEM, FME plus the

LDHs' and MMT' isolation). As introducing the different effect from the inorganic nanosheets, the significant growth presents a regular three-step. The first step is because that the rigid LDHs isolation can eliminate the interlayer π - π stacking interaction, as confirmed by Duan *et al*^{47,48}. The next conspicuous step is caused by transition metal-bearing LDHs nanosheets can offer FME which can affect the spinning of electron, thus extending the lifetimes. The last noteworthy step is partially due to the initiate MMT nanosheets isolation, but the crucial reason is that the PEM formed by oppositely-charged LDHs and MMT nanosheets, which can definitely impact electron vibration in the conjugate structure of chromophores. In this nano-system, the PEM and FME can influence the variations in the electronic structure, such as increasing in electronic $\pi^*-\pi$ transition energy and oscillator strength, thus impact the relaxation times of the electron spins, finally prolonging the lifetimes^{49–54}. Above all, PEM and FME are fairly beneficial to obtain better photoemission behavior in the OMFs system.

Therefore, the as-fabricated OMFs are expected to have much flexibility and be potential for manipulating, controlling and developing novel optoelectrical, optomagnetic, even photomagnetoelectric devices. Further work is ongoing with the research on lifetime of various luminescent guests imbedded in series of different layered inorganic nanosheets with PEM and FME.

Methods

Reagents and materials. All the chemicals are analytical grade and used as received without further purification. Bis(N-methylacridinium) (BNMA MW = 510.50) and poly[5-methoxy-2-(3-sulfopropoxy)-1,4-phenylenevinylene] potassium salt solution(APPV, 0.25 wt% in H₂O) were purchased from Sigma Chemical. Co. Ltd, and polyvinyl alcohol (PVA, DP = 1750 ± 50) was purchased from Tianjin Fuchen Chemical Reagent Plant. Na-montmorillonite (MMT) was purchased from Zhejiang Sanding Co. Ltd. Ni(NO₃)₂·6H₂O, Fe(NO₃)₃·9H₂O were all supplied by the Xilong Chemical Plant. NaOH, NH₃·H₂O, H₂O₂ (30%), H₂SO₄ (95%–98%) was supplied by Beijing Chemical Reagent Company.



Characterization. UV-visible absorption spectra were measured in the range from 200 to 500 nm on a TU-1901 Double beam UV-vis spectrophotometer with slit width of 5.0 nm. The fluorescence spectra were performed on F-4600

Fluorospectrophotometer. The fluorescence decays measurements of OMFs were recorded by using an Edinburgh Instruments' Steady and transient time-resolved fluorescence spectrometer. X-ray diffraction patterns were recorded using a Rigaku 2500 VB2+PC diffractometer under the conditions: 40 kV, 50 mA. The morphology of thin films was investigated by using a scanning electron microscope (SEMHitachi S-3500), and the accelerating voltage applied was 20 kV. The morphology of layered double hydroxides was investigated by using a transmission electron microscope (Hitachi H9000).

Synthesis of NiFe-LDHs. The NiFe-LDHs was prepared via hydrothermal coprecipitation method. Typically, $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ were dissolved in 50 mL boiled deionized water to form a mixed salts solution with the final concentrations of 0.8 and 0.4 mol/L, respectively. The solution of sodium hydroxide (50 mL, 2.4 mol/L) is mixed with the salts solution under vigorous stirring. Then the suspension was stirred at 80°C for 24 h under nitrogen gas flow. The precipitate was centrifuged, washed with hot deionized water and dried in a vacuum at 60°C for 12 h.

Exfoliated of NiFe-LDHs and MMT. The unilamellar, positively charged LDHs nanosheets were obtained by vigorously agitating 0.1 g NiFe-LDHs in 100 mL formamide at room temperature under a N_2 gas flow for 2 days. The MMT powder (1 g) was dissolved in 1000 mL distilled water. After vigorous stirring for 4 weeks, the upper suspensions with exfoliated MMT nanosheets were obtained after centrifugation at 10000 rpm for 10 min.

Fabrication of (NiFe-LDHs/BNMA@PVA)_n OMFs and NiFe-LDHs/APPV@PVA)_n OMFs. The quartz slide ($1.5 \times 1.5 \text{ cm}^2$) was first cleaned in the solution ($\text{NH}_3 \cdot \text{H}_2\text{O} : \text{H}_2\text{O}_2 = 7 : 3$), then cleaned with H_2SO_4 for 30 min, and finally rinsed and washed thoroughly with deionized water. PVA was dissolved in distilled water to obtain 1wt% aqueous solution, the BNMA (0.1 g) was dissolved in 100 mL deionized water, and then the solution of BNMA and PVA were mixed (1:1 in volume) to form BNMA@PVA solution in which the concentration of BNMA is 0.5 g/L. (NiFe-LDHs/BNMA@PVA)_n OMFs were fabricated by depositing alternatively through LDHs nanosheets solution and BNMA@PVA solution for n cycles. The resulting films were dried with a nitrogen gas flow for 2 min. Fabrication of (NiFe-LDHs/APPV@PVA)_n OMFs was similar with that of (NiFe-LDHs/BNMA@PVA)_n OMFs.

Fabrication of OMFs (MMT/BNMA@PVA/NiFe-LDHs/BNMA@PVA) and (MMT/APPV@PVA/NiFe-LDHs/APPV@PVA)_n OMFs. Quartz slides ($1.5 \times 1.5 \text{ cm}^2$) were cleaned in a "piranha" solution ($\text{H}_2\text{SO}_4 : \text{H}_2\text{O}_2 = 3 : 1$ in volume), and then thoroughly rinsed with distilled water and dried under nitrogen flow. PVA was dissolved in distilled water to obtain 1wt% aqueous solution, the BNMA (0.1 g) was dissolved in 100 mL deionized water, and then the solution of BNMA and PVA were mixed (1:1 in volume) to form BNMA@PVA solution in which the concentration of BNMA is 0.5 g/L. The (MMT/BNMA@PVA/NiFe-LDHs/BNMA@PVA)_n OMFs were fabricated by applying a cyclic repetition of the following steps: a) dipping the quartz slide into MMT solution for 5 min, then thoroughly rinsing it with distilled water and drying it; b) dipping it into BNMA@PVA solution for 5 min, then thoroughly rinsing and drying it; c) dipping it into exfoliated NiFe-LDHs suspension for 5 min, then rinsing and drying it; d) dipping it into BNMA@PVA solution for 5 min, then followed by distilled water washing and drying it. All these procedures were repeated n times to produce OMFs of (MMT/BNMA@PVA/NiFe-LDHs/BNMA@PVA)_n OMFs. Fabrication of (MMT/APPV@PVA/NiFe-LDHs/APPV@PVA)_n OMFs was similar with that of (MMT/BNMA@PVA/NiFe-LDHs/BNMA@PVA)_n OMFs.

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Author contributions

M.T.L. designed the whole experiment and analyzed the data of the work. He also revised the manuscript critically and approved the final version for submission. T.L.W. did the most of the experiment and drafted the paper. Moreover, he revised the paper with Liu seriously and submitted the final version of paper as well. H. W. M. helped M.T.L. design the research and gave the interpretation of the whole data. Y.F., K.R.H. and C.G. helped to perform the fluorescence decays characterization and analysis of the data. All the authors have read the paper and approved the final version of the paper for submission.

Additional information

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