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From nanocorals to nanorods to nanoflowers nanoarchitecture for efficient dye-sensitized solar cells at relatively low film thickness: All Hydrothermal Process

SUBJECT AREAS:

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Simple and low temperature hydrothermal process is employed to synthesize exotic nanostructures of TiO₂. The nanostructures are obtained merely by changing the nature of the precursors and processing parameters. The chloride and isopropoxide salts of titanium are used to grow high quality thin films comprising anatase nanocorals, rutile nanorods and rutile nanoflowers respectively. A novel route of addition of room temperature ionic liquid (RTIL) is used to synthesize hitherto unexplored nano-morphologies. The Bronsted Acidic Ionic Liquid [BAIL] 0.01 M, 1: 3-ethoxycarbonylethyl-1-methyl-imidazolium chloride [CMIM][HSO₄] RTIL directed growth of TiO₂ flowers with bunch of aligned nanorods are obtained. The structural, optical and morphological properties of hydrothermally grown TiO₂ samples are studied with the different characterization techniques. The influence of these exotic nano-morphologies on the performance of dye sensitized solar cells (DSSCs) is investigated in detail. It is found that [CMIM][HSO₄] can facilitate the formation of novel nanoflower morphology with uniform, dense, and collectively aligned in regular petal like oriented TiO₂ nanorods and hence improves the dye adsorption and the photovoltaic performance of DSSCs, typically in short-circuit photocurrent and power conversion efficiency. A best power conversion efficiency of 6.63% has been achieved on a DSSC based on nanoflowers (TNF) film obtained from a [CMIM][HSO₄] solution.

Dye Sensitized Solar Cells (DSSCs), a molecular approach to photovoltaic solar energy conversion, is one of the emerging solar technologies that offer the potential to reduce the cost of photovoltaic electricity production. During the past two decades, nanoporous polycrystalline titania has been extensively used in (DSSCs), which have been demonstrated to be a promising alternative of silicon based solar cells due to their relatively high solar-to-electric power conversion efficiency and low cost¹. In the DSSC, the mesoscopic structure of the titanium dioxide (TiO₂) electrode plays a significant role in increasing cell efficiency by providing the photosensitizer dye with much greater surface area for light harvesting². The mesoporous TiO₂ photoanodes that have the percolated links of the nanoparticles produce very large photocurrent due to their high surface area suitable for dye adsorption and electron energy level matching allowing for the injection of carriers from dye molecules to nanostructured TiO₂ films. Despite this advantage of the nanoparticle-based photoanodes, their numerous interparticle boundaries easily trap the charge carriers, which result in the decrease in the carrier mobility and the carrier lifetime. In the nanoparticle-based DSSCs, therefore, the carrier transport is a trap-limited diffusion process and the electron diffusion coefficient is several orders smaller than the expected value that is deduced from the physical properties of the single crystalline bulk TiO₂³. Nanowire/nanorods have demonstrated a great potential to achieve high diffusion coefficient of carriers in electric devices due to their unique one-dimensional (1D) structure⁴. Many attempts have been made for controlling the 1D nanostructured TiO₂ using different techniques which includes electrodeposition⁵, hydrothermal and Solvothermal^{6,7}, anodiza-



tion⁸, template based growth sol-gel electrophoresis⁹, and by using different surfactants¹⁰. Recently, single crystalline TiO₂ nanorods with a rutile phase have been successfully grown on the top of fluorine-doped tin oxide (FTO) films coated glass, by a hydrothermal method¹¹. Generally, spin-coated or screen printed Pt/FTO is used as counter electrodes for DSSC application. Recently, Y. H. Hu *et al.* had fabricated variety of counter electrodes like PEDOT¹², NiO¹³, Graphene^{14–16}, ZrN¹⁷, ZnO¹⁸.

In recent years, many inorganic nanostructures have been fabricated via various room temperature ionic liquid (RTIL)-involved processes, including electrodeposition, chemoreduction, sol-gel and solvothermal route. RTIL have unique properties such as extremely low volatility, wide liquid temperature range, good thermal stability, good dissolving ability, excellent microwave (MW) absorbing ability, designable structures, high ionic conductivity, and wide electrochemical window, etc¹². Moreover, RTIL is the excellent surfactant for the growth of nanostructured material. Very recently, RTILs have been used as solvents, reactants, or templates for the synthesis of inorganic nanomaterials with novel morphologies and improved properties^{13–15}. Further, recently IL based on imidazolium salts have been widely used as solvents for DSSCs¹⁶. Many inorganic nanostructures¹⁷, including titanium dioxide^{18–23}, have been fabricated via various ILs-involved processes. For titania nanomaterials, however, to the best of our knowledge, few works about the synthesis of rutile nanostructures have been reported in ILs solution^{24–32}. Recently, Kunlun Ding *et al.* successfully developed route for the synthesis of high quality TiO₂ nanocrystals in ionic liquid via a Microwave-Assisted Process³³.

In this article, we have studied effect of different TiO₂ morphology from nanocorals (TNC)^{34,35}, nanorods (TNR) to nanoflowers (TNF)³⁶ thin films on FTO substrate that were prepared by facile hydrothermal route. Herein, TNF film was prepared from 0.001 M of 1:3-ethoxy-carbonylethyl-1-methyl-imidazolium chloride [CMIM][HSO₄] RTIL mixed with Titanium tetraisopropoxide (TTIP):Hydrochloric Acid (HCl, 38%) (1:1 Volume) solution, and uniform, compressed and nanocrystalline TiO₂ nanoflowers were obtained. These nanostructures of TiO₂ films were further effectively loaded with N719-dye and studied its DSSCs properties. As compared to the TNC, TNR and P25, the TNF prepared from a solution containing the [CMIM][HSO₄] acids exhibited higher dye loading, resulting in higher short-circuit photocurrent and power conversion efficiency. Our results showed that the TNF sample exhibits higher short-circuit photocurrent density (13.15 mAcm⁻²) and power conversion efficiency (6.63%).

Results

The morphological features of the TiO₂ samples are investigated by Field Emission scanning electron microscopy (FESEM). Fig. 1 (a–d) shows the P25 TiO₂ nanoparticles that were deposited by doctor blade technique. FESEM images show the compactly arranged and uniformly deposited TiO₂ nanoparticles over the substrate. Fig. 1 (e–h) shows FESEM image of the TiO₂ nanocoral (TNC) sample that exhibits a novel coral-like morphology with nanopolyps. The diameter of the circular corals is about 400–500 nm, on top of which nanopolyps are nucleated and their average size is about 20 nm. This peculiar morphological feature is beneficial as it provides larger active surface area. Detailed growth mechanism is explained in our previous work^{34–36}. FESEM images of the TiO₂ nanorods (TNR) are shown in fig. 1 (i–l). Most of TNRs displayed microspheres containing nanorods of nearly 250 nm diameter. Fig. 1 (m–p) shows that the TiO₂ nanoflowers (TNF) bunch of aligned nanorods containing petals prepared with RTIL retain the rod like array geometry but the nanorod diameter drastically decreased to nearly 62 nm. It is also to be noted that the nanorods are tapered at the tip and separated to each other.

Fig. 2 (a–c) shows the FESEM, selected area electron diffraction (SAED), TEM and HRTEM pattern of TNC sample. The formation of the ring pattern within the dotted line indicates a highly crystalline sample. The morphology of the as-obtained anatase TiO₂ nanocorals were characterized by TEM, as shown in Fig. 2 (b). The size of each polyps is around 20–30 nm with high crystallinity. Fig. 2(c) shows an HRTEM image of the respective sample. The lattice planes are clearly indicates that the synthesized nanocorals are highly crystalline in nature. The regular periodic arrangement of the atoms to form atomic plane with an interplanar distance of 3.36 Å affirms the growth of anatase TiO₂ phase. Fig. 2 (d–f) shows FESEM, SAED and HRTEM micrographs of TNF sample. The spotted SAED indicates the single-crystalline nature of the rutile TiO₂ nanoflowers. The formation of the dark spot pattern indicates a single crystalline nature of the sample. It is believed that the synthesized films are composed of single-crystalline rutile TiO₂ nanorods that grow in the [001] direction. The clear lattice fringe of the single nanorod of TNF sample is single crystalline along their entire length (fig. 2 (e)). Interplanar spacings, $d_{110}=3.3\text{\AA}$ between the adjacent lattice fringes perpendicular to the rod axis can be assigned to the interplanar distance of rutile TiO₂ (110) and $d_{001}=2.9\text{\AA}$ is interplanar distance for the (001) planes along the rod axis that are obtained from the HRTEM lattice fringes of TNF. This indicates the formation of single crystalline rutile phase, in agreement with the XRD results (*Supporting information Figure S1 and S2*).

The Fourier transform Raman (FT-Raman) spectra of the films were recorded in the spectral range of 36–2000 cm⁻¹ using FT-Raman spectrometer (Bruker MultiRAM, Germany) that employs a Nd:YAG laser source with an excitation wavelength 1064 nm and resolution 4 cm⁻¹ at 336 mW laser power. The FT-Raman of the P25, TNC, TNR and TNF films are shown in Fig. 3. The ‘tetragonal anatase structured’ TiO₂ belongs to D194h (I41/amd) space group (SG) and following normal lattice $A_{1g}+B_{1g}+B_{2g}+E_g$. The first E_g peak at ~144 cm⁻¹, a characteristic of anatase TiO₂ was formed in the P25 as well as TNC samples. The peaks at 516 (corresponding to B_{1g}, A_{2g}) and 635 cm⁻¹ (correspond to E_g) modes of anatase TiO₂ are observed. The rutile phase of TiO₂ is tetragonal and exhibits symmetry characters of the space group with two TiO₂ molecules per unit cell. The TNR and TNF samples show four Raman active fundamental modes that reveals rutile TiO₂ at 143 cm⁻¹ (B_{1g}), 447 cm⁻¹ (E_g), 612 cm⁻¹ (A_{1g}), and 826 cm⁻¹ (B_{2g}) expressed as $A_{1g}+B_{1g}+B_{2g}+E_g$. For the rutile phase two prominent maxima at 445 cm⁻¹ (E_g) and 609 cm⁻¹ (A_{1g}), are comparable with that found in the rutile TiO₂ single crystal. In addition, there are second-order scattering features, the most prominent one at ~237 cm⁻¹ (E_g) peak due to the multiple-phonon scattering processes, which is also considered as a characteristic Raman peak of rutile type TiO₂²⁷. In the FT-Raman spectra of our samples, the E_g and A_{1g} modes, as well as the second-order effect at ~237 cm⁻¹, are the major features; the B_{1g} and B_{2g} modes are extremely weak or absent (fig 3). With conversion from TNRs to TNF, the wave number of all peaks decreased relevantly; in particular, the E_g (612 cm⁻¹) mode decreased by about 6 cm⁻¹, while the intensity of second-order E_g mode (~237 cm⁻¹) is drastically increased. The second order E_g at 237 cm⁻¹, a characteristic of rutile TiO₂, is more intense and slightly broader and shifted with respect to that of a TNRs sample. This nature is due to phonon confinement effect in which the decrease in the particle dimensions to the nanometer scale can cause a wave number shift and broadening of Raman peaks as a result of phonon confinement. The non-linear behavior of the E_g peak intensity suggests strong light scattering differences among samples, potentially related to changes on the morphological properties of the nanorods and nanoflowers. Figure S4 shows the N2 adsorption-desorption isotherm of the TiO₂ samples. The isotherm displays a typical type IV curve while the hysteresis loop is associated with narrow slit-like pores in the sample. The BET specific surface area of the P25, TNC, TNR and TNF sample

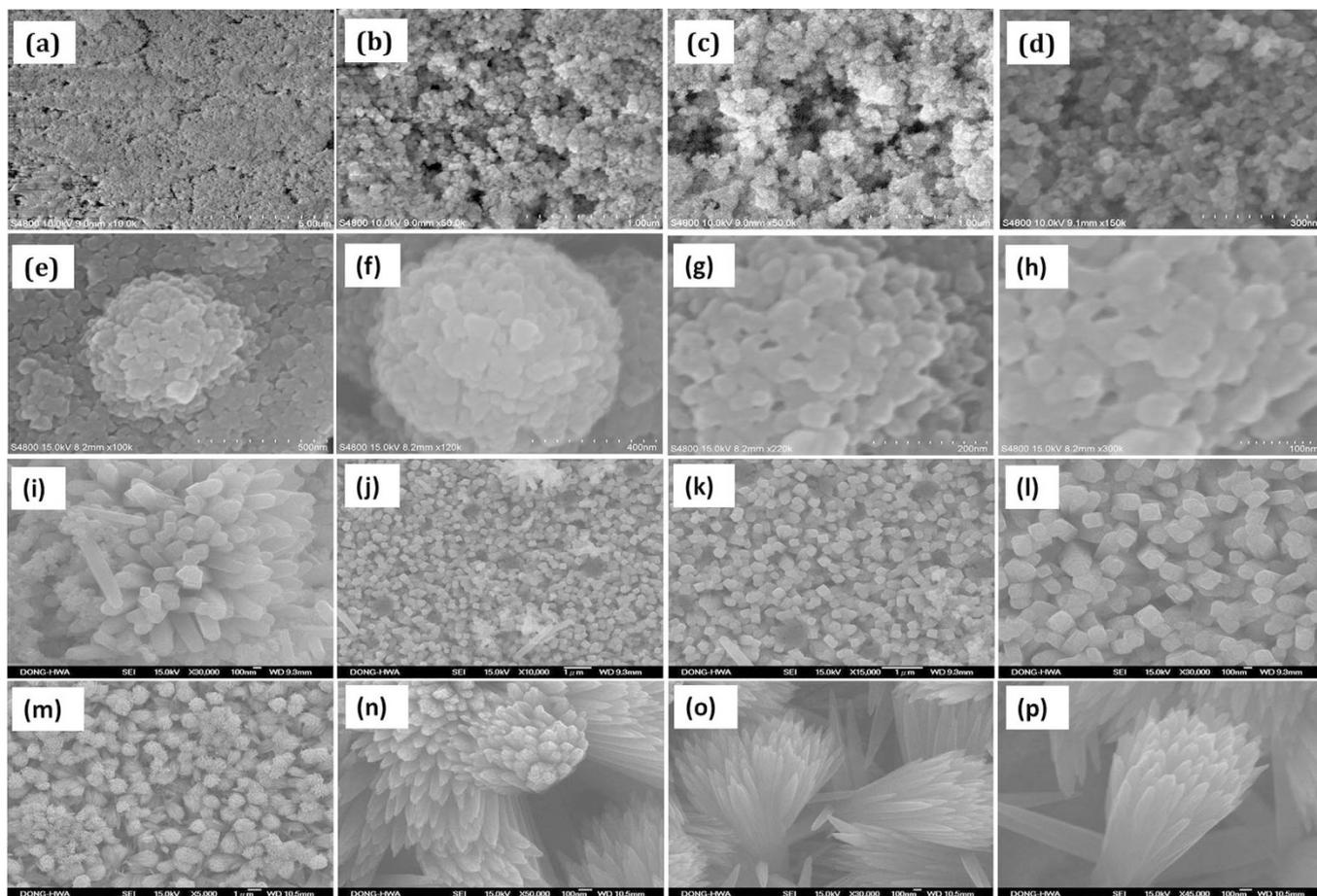


Figure 1 | FESEM images of hierarchical TiO₂ nanostructures: (a–d) Degussa P25 nanoparticles deposited by doctor blade, (sample-1). (e–h) TiO₂ nanocorals (TNC) deposited by hydrothermal process from saturated NaCl solvent having (TiCl₄) at 120°C, solutions (sample-2), (i–l) TiO₂ nanorods (TNR) deposited by hydrothermal process from TTIP in HCl:H₂O (1:1v) at 180°C (sample-3) and (m–p) TiO₂ nanofloweres (TNF) obtained by hydrothermal process in TTIP:HCl:H₂O:[CMIM][HSO₄] ionic liquid (sample-4).

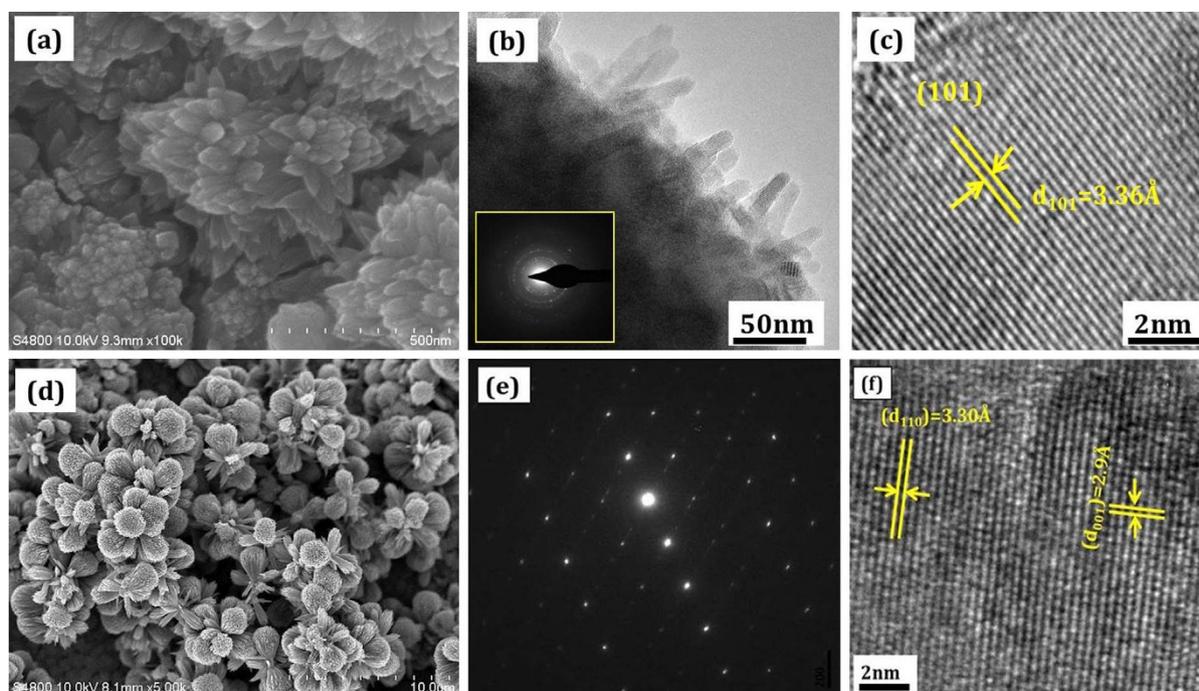


Figure 2 | High-magnification FESEM images, SAED and HRTEM characterizations of the of the different TiO₂ nanostructures in the form of (a–c) TNC, (d–f) TNF.

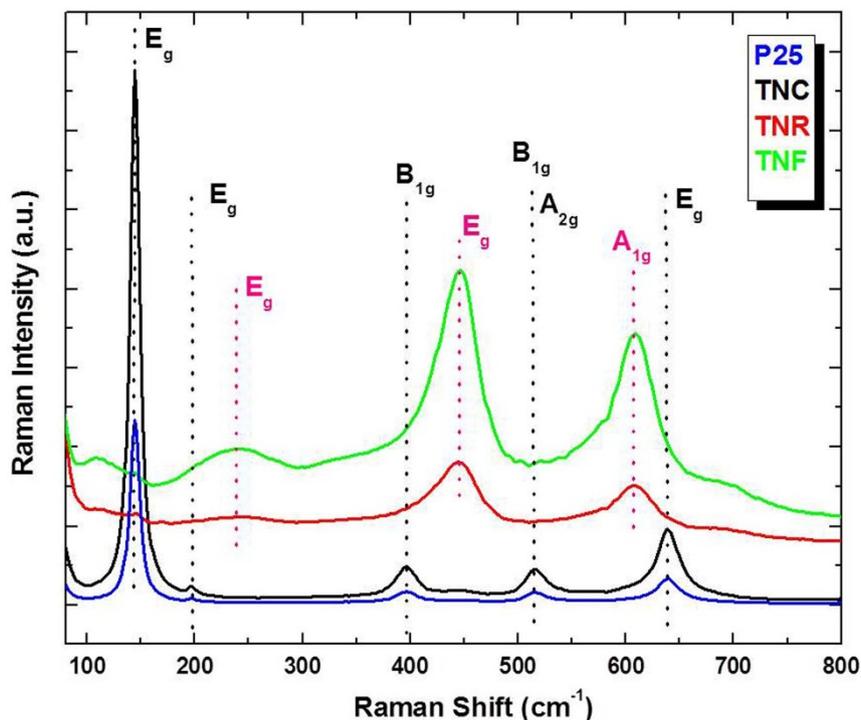


Figure 3 | FT-Raman spectra of synthesized TNC, TNR and TNF recorded in the wave number range of 35–800 cm^{-1} with 1064 nm Nd:YAG laser excitation wavelength.

are $54.7 \text{ m}^2\text{g}^{-1}$, $73.5 \text{ m}^2\text{g}^{-1}$, $89.4 \text{ m}^2\text{g}^{-1}$ and $106.1 \text{ m}^2\text{g}^{-1}$ respectively. From these data it is clear that the TNF sample has much higher surface area than other TiO_2 photoelectrodes. These results clearly reveal that TiO_2 nanoflowers facilitates larger amount of dye adsorption as well as effective light harvesting.

The DSSC devices assembled using the above films were tested under simulated AM1.5G solar illumination (100 mWcm^{-2}). The 0.3 mM ethanolic solution of N-719 (Di-tetrabutylammonium cis-bis(isothiocyanato)bis(2,2'-bipyridyl-4,4'-dicarboxylato)ruthenium (II), Sigma Aldrich) dye solution was used for dye loading. The standard iodide-based electrolyte (TEL-294H, EL-HPE, Dyesol) was used as the redox electrolyte. Figure 4 shows the J-V plots of different TiO_2 morphologies denoted as Dark (●) P25 (●), TNC (●), TNR (●) and TNF (●). The average photovoltaic data were summarized in Table 1. The DSSC based on P25 produced conversion efficiency (η) of 3.18% ($J_{\text{sc}} = 6.68 \text{ mAcm}^{-2}$, $V_{\text{oc}} = 0.692 \text{ V}$, $\text{FF} = 0.69$). The TNC sample shows η of 3.53% ($J_{\text{sc}} = 8.63 \text{ mAcm}^{-2}$, $V_{\text{oc}} = 0.706 \text{ V}$, $\text{FF} = 0.58$). In case of TiO_2 nanorods (TNR), it shows $J_{\text{sc}} = 11.24 \text{ mAcm}^{-2}$, $V_{\text{oc}} = 0.699 \text{ V}$, $\text{FF} = 0.62$. The conversion efficiency of TNR comes to be 4.87%. While, the nanoflowers grown from RTIL [CMIM][HSO_4] resulted in a big enhancement of J_{sc} by twice compared to P25. The TNF sample shows η of 6.63% ($J_{\text{sc}} = 13.15 \text{ mAcm}^{-2}$, $V_{\text{oc}} = 0.699 \text{ V}$, $\text{FF} = 0.67$). As a result, power conversion efficiency improved from 3.18% for P25 sample to 6.63% for TNF.

Using the same dye and redox electrolyte, a DSSC made of a film comprising 20 nm TiO_2 anatase nanoparticles produced 3.18% conversion efficiency ($J_{\text{sc}} = 6.68 \text{ mAcm}^{-2}$, $V_{\text{oc}} = 0.692 \text{ V}$, $\text{FF} = 0.69$). From Table 1 it is obvious that the nanoflower rutile film is superior to the P25 nanoparticulate anatase film in the generation of J_{sc} and $\eta\%$ even though anatase is known to show better efficiency. In order to verify this, the adsorbed amounts of dye were determined by measuring the eluted dye concentration from the porous TiO_2 structure. With UV-vis (UV-1800, Shimadzu) absorption spectroscopy P25 sample shows $89.1 \mu\text{mol}\cdot\text{cm}^{-2}$ which is less as compared to TNC, TNR and TNF samples. The amount of dye loaded by TNF

is higher ($129.9 \mu\text{mol}\cdot\text{cm}^{-2}$) compared to both TNC ($97.5 \mu\text{mol}\cdot\text{cm}^{-2}$) and TNR ($115.3 \mu\text{mol}\cdot\text{cm}^{-2}$) sample. From the above discussion it is clear that the increment in dye adsorption was simply proportional to the surface area of the TiO_2 morphology. Therefore TNF sample provides large surface area for dye adsorption which results in increased efficiency. From Table 1 it is clear that all samples shows FF less than 0.70, which may be due to high series resistance (R_s). In our experiment we used FTO with sheet resistance $15 \Omega\cdot\text{cm}^{-2}$ which will cause the increment in the series resistance.

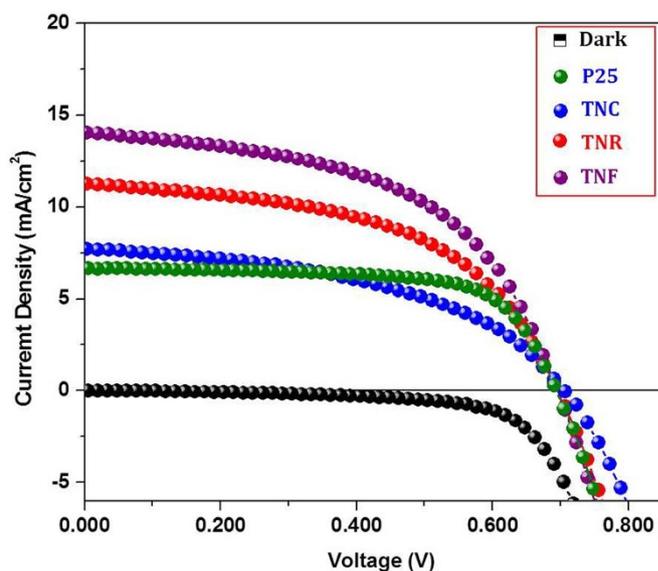


Figure 4 | J-V curves of the DSSCs assembled respectively from P25, TNC, TNR and TNF under simulated AM 1.5G solar light (100 mW cm^{-2}). P25 film obtained from commercial Degussa powder, TNC obtained from the TiCl_4 and sat. NaCl; TNR obtained from the $\text{TTIP}:\text{HCl}:\text{H}_2\text{O}$; TNF obtained from the $\text{TTIP}:\text{HCl}:\text{H}_2\text{O}:[\text{CMIM}][\text{HSO}_4]$.

Table 1 | Average photovoltaic performance parameters for DSSCs based on the different TiO₂ morphologies

| Sample | V _{oc} (V) | J _{sc} (mAcm ⁻²) ^[a] | R | R | FF (%) | (η) (%) | Adsorbed Dye (μmol cm ⁻²) | Surface Area m ² g ⁻¹ |
|------------|------------------------|---|--|--|-----------|------------|--|--|
| | | | at V _{oc} (Ω·cm ²) | at I _{sc} (Ω·cm ²) | | | | |
| P25 | 0.692 | 6.68 | 183 | 3152.9 | 69 | 3.18 | 89.1 | 54.7 |
| TNC | 0.706 | 8.63 | 78.1 | 2018.3 | 58 | 3.53 | 97.5 | 73.5 |
| TNR | 0.699 | 11.24 | 74.2 | 2129.5 | 62 | 4.87 | 115.3 | 89.4 |
| TNF | 0.699 | 13.15 | 45.81 | 809.7 | 67 | 6.63 | 129.9 | 106.1 |

[a] Device measured @ 0.5 × 0.5 cm active area illumination.

Open-circuit voltage-decay (OCVD) measurements were conducted to investigate the recombination kinetics of these nanostructures. Following the technique as reported by Zaban et al.³⁷ Open-circuit voltage-decay measurements were performed by monitoring the Voc transient during relaxation from an illuminated quasi equilibrium state to the dark equilibrium (Figure 5 (a)). When the visible light illumination on a DSSC at open circuit is interrupted, the excess electrons are removed due to recombination, with the photo-voltage decay rate directly related to the electron lifetime.

Appropriate use of OCVD equation, assumes that the recombination is linear with a first-order dependence on electron concentration and that electron recombination occurs only with the electrolyte. Figure 5 shows the OCVD curves measured for various DSSCs, and it appears that the TNF has the slower voltage decay rate as compared to other samples. Figure 5 (b) is the plot of the response time obtained by applying equation to the data in Figure 5 (a). Recombination in all samples is very slow and cells can sustain significant voltages many seconds after light has been turned off. In comparison to P25, TNC, TNR and TNF OCVD measurements, the TNF based DSSCs exhibit the much larger τ_n values compared to all other samples. The derived electron lifetimes from transient equation, also demonstrate that the TNF electrode has larger τ_n values compared with those for the other DSSCs electrodes. This result indicates that the recombination rate of photogenerated electrons with electrolyte-oxidized species is the slowest in the DSSCs fabricated based on TNF morphology. This effect is also responsible for the higher performance of the DSSCs electrode.

In order to understand the reasons behind such results, we decided to perform electrochemical impedance spectroscopy (EIS) on the assembled cells in the dark at -0.70 V applied potential. Analyzing EIS results using the appropriate transmission line model, gives insights into the interfacial charge transfer processes that take place at the working and counter electrodes/electrolyte interfaces. Typical EIS Nyquist plots of TiO₂ DSSCs devices (P25, TNC, TNR and TNF) and their corresponding equivalent circuit fitted plots are shown in Figure 6 (a–d). Inset shows its equivalent circuit used to analyze the impedance spectra. In general, under an applied potential of -0.70 V three phase angle peaks are present. The peak seen in the high frequency region (20 kHz–100 Hz) results from the charge transfer resistance (R_{CE}) and the double layer capacitance (C_{CE}) at the electrolyte/counter electrode interface, whereas the one observed at mid frequencies (1–100 Hz) is due to the charge transfer resistance (R_{CT}) of the electron recombination and the chemical capacitance (C_1) at the TiO₂/N719/electrolyte interface.

The third arc at low frequencies (0.1–1 Hz) is characteristic of the Warburg impedance (W_1) corresponding to the electrolyte's ions diffusion in a DSSC as seen in the case of TNC sample. The fitted impedance spectra for TNC sample consists of (R_3 , Q_2) which represents the electrode/electrolyte interface and R_1 is the shunt resistance in parallel with the TiO₂-Dye structure represented by W_1 , R_2 , C_1 . R_3 is the charge transfer resistance, R_{CE} and Q_2 is the double layer capacitance C_{CE} . While W_1 and R_2 (or R_{CT}) represent the diffusion resistance and series resistance of the TiO₂-Dye structure, C_1 repre-

sents the TiO₂-Dye junction capacitance. In case of nanorods, the W_1 gets replaced by the constant phase element Q_0 , while the remaining circuit elements are the same. For P25 and TNF the equivalent circuits are similar as shown in the inset. The product of the charge-transfer resistance of the electron recombination (R_{CT} or R_2 in the case of TNC and TNR while it is R_1 in the case of P25 and TNF) and the chemical capacitance (Q_1 or C_1) at the TiO₂/N719 interface corresponds to the electron lifetime in the TiO₂ film, $\tau_e = R_{CT} \times Q_1$.

The series and shunt resistances obtained after fitting the impedance data are shown in Table 2 along with the carrier lifetime estimates. From the Table-2 it is clear that, there is a significant increase in the carrier lifetime for all TiO₂ nanostructures prepared by hydrothermal method when compared to P25. The shunt resistance R_1 of device P25 and TNR is very high than that of $R_2=2531$ and $23 \Omega \cdot \text{cm}^{-2}$ for devices TNC and TNF measured at -0.70 V, respectively. Though the series resistance (R_2) of TNF is very less, the shunt resistance also is very small decreasing the efficiency. So better optimization of the TiO₂-Dye interface promises further increase in efficiency. Higher R_{CT} for TNC electrode results in lower fill factor, thereby lowering efficiency when compared to TNF based device.

For TNF sample increased J_{sc} may be due to typical morphology of TiO₂ sample which will facilitate for effective dye adsorption and the corresponding increase in current. Such morphology will also provide effective light harvesting accessibility. Further, J_{sc} increased significantly because the size and width of nanoflowers are thinner about (~61 nm) compared to TNR (~250 nm) sample, denser and more aligned, which yielded more dye loading and faster electron diffusion. Among the above-mentioned morphologies of nanoflowers (TNF), those grown from the TTIP:HCl:[CMIM][HSO₄] solution produced the highest short-circuit photocurrent (13.15 mAcm⁻²) and power conversion efficiency (6.63%). Therefore, we suggest that TNF are the best candidate for DSSCs application.

Discussion

In conclusion, we have presented first report on a synthetic strategy to deposit different nanostructured TiO₂ thin films by simple and cost effective hydrothermal route for dye sensitized solar cells (DSSCs) application. We have also demonstrated a facile way to synthesize novel nanoflower morphology of TiO₂ using [CMIM][HSO₄] room temperature ionic liquid (RTIL) which is non-toxic and are useful in developing green chemistry approach for efficient DSSCs applications. Furthermore, [CMIM][HSO₄] helped to achieve a high degree of crystallization even at mild annealing temperatures, such as 180°C. Among the above-mentioned morphologies TiO₂ nanoflowers (TNF), those grown from the TTIP:HCl:[CMIM][HSO₄] solution produced the highest short-circuit photocurrent (13.15 mAcm⁻²) and power conversion efficiency (6.63%). The TNFs are superior to those grown from the single step hydrothermal system in solar cell performance, typically with current density J_{sc} (13.15 mAcm⁻²). Even with a 1.8~μm long nanorod array film, a solar-to-electric power conversion efficiency of 6.63% has been achieved, which is more efficient than a DSSC based on a 1.8 μm thick nanoparticulate film. This method will open a new

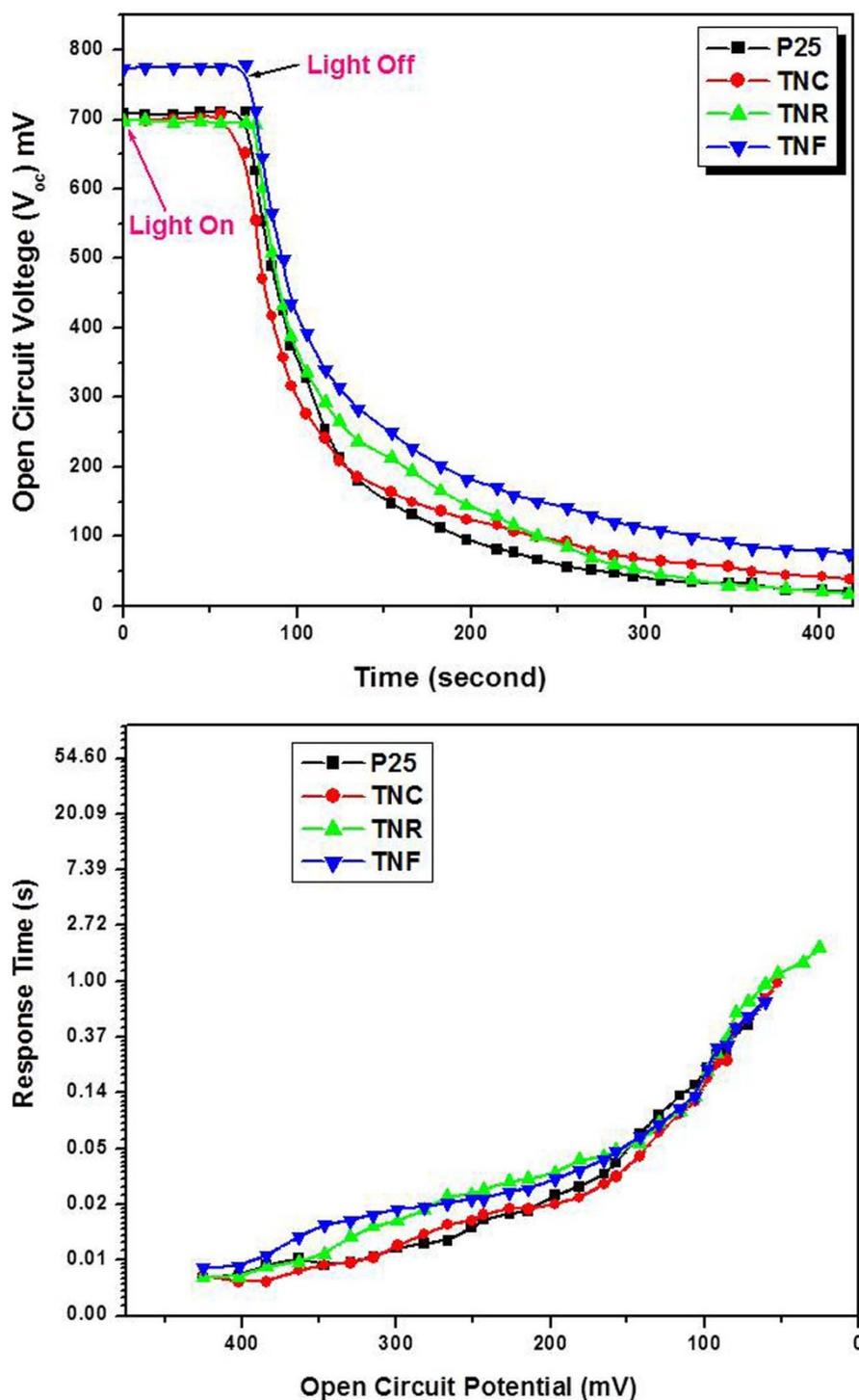


Figure 5 | (a) Photovoltage-decay measurement of TiO_2 DSSCs electrodes to illumination excitation (AM 1.5G solar light (100 mWcm^{-2})), (b) Response time determined by open circuit photovoltage decay for different DSSCs samples.

approach to develop hierarchical functional nanomaterials using ionic liquid and subsequent applications.

Methods

Materials. Titanium tetrachloride (TiCl_4) (99.98% Spectrochem, India), and Titanium tetraisopropoxide (TTIP) were purchased from Alfa Aesar. Hydrochloric acid (HCl, 35.46%), sulfuric acid (H_2SO_4 , 99.9%), nitric acid (HNO_3 , 38%), and sodium chloride (NaCl) were obtained from Thomas Baker. Absolute ethanol (99.9%) AR-grade was obtained from Changshu Yangyuan Chemicals China. Conducting FTO glass (fluorine-doped SnO_2 , resistance $15 \Omega \text{ cm}^{-2}$, transmittance 90%, Solaronix) was received from Solaronix, Switzerland.

Iodolyte AN-50 (Solaronix) redox electrolyte were used in the DSSCs. The sensitizer used in this work is an organic N719 dye (Di-Tetrabutylammonium cis-bis(isothiocyanato)bis(2,2-bipyridyl-4,4'-dicarboxylato)-ruthenium(II), 95% (NMR)) purchased from Dysol.

Hydrothermal Synthesis of TiO_2 Nanocorals (TNC). TiO_2 nanocorals (TNC) were synthesized by controlled hydrothermal process^{34,35}. In a typical experiment, hydrothermal inorganic precursor solution was prepared by mixing 0.05 M TiCl_4 carefully with absolute ethanol in ice-cold bath forming yellow colored titanium chloroalkoxide [$\text{TiCl}_2(\text{OEt})_2$] solution. A small amount of 0.001 M concentrated hydrochloric acid (HCl, 36%, Thomas Baker) was added as a catalyst and pH of solution was adjusted to 2.1 using nitric acid (HNO_3 , 38%, Thomas Beaker) as a

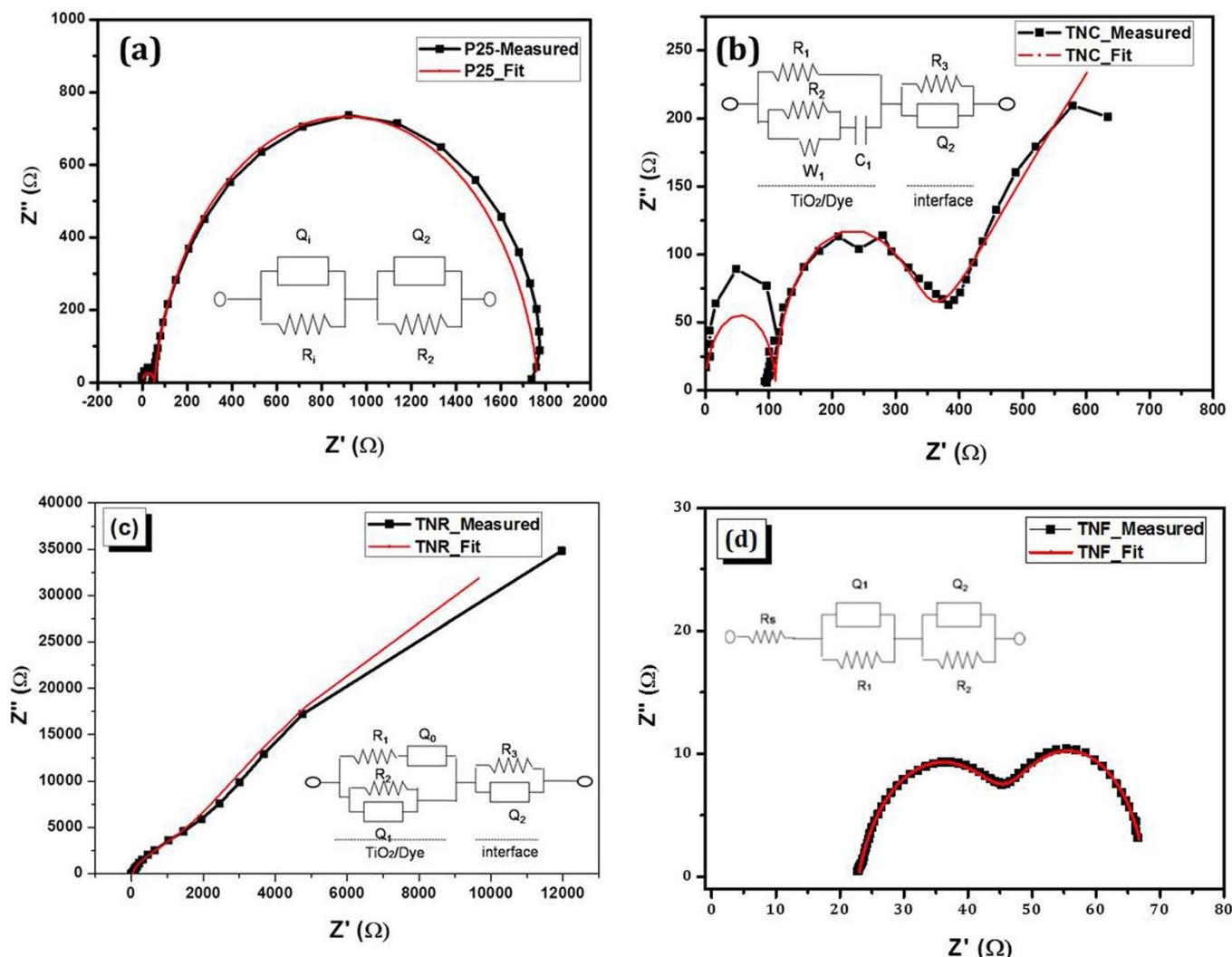


Figure 6 | (a–d) Impedance spectra of TiO₂ DSSCs electrodes measured in the dark under 20.70 V applied bias. Inset shows the respective equivalent circuits used to analyze the impedance data.

peptizer. The transparent solution thus prepared was further mixed with saturated solution of 6.8 M NaCl (Extra pure, Thomas Baker). The addition of water causes the hydrolysis of the inorganic moieties, producing in situ ethanol and HCl, the latter being the reason for the high stability of the solution. The final solutions are very acidic and stable for several months at ambient temperature (20–25 °C) or 1–2 weeks at 30 °C. The solution was stirred for 30 min using magnetic stirrer and transferred into a polytetrafluoroethylene (Teflon)-coated 25 ml vessel. The cleaned glass or TCO substrate was immersed into this solution vertically. It was then sealed and maintained at 120 °C for 3 h (at 1.5 kg/cm² pressure). Upon completion of the reaction, the autoclave was allowed to cool at room temperature and the deposited film was rinsed in double distilled water and finally dried at room temperature. The TiO₂ film was whitish, uniform, crack-free and adherent to the substrates, whose adherence was checked by a scotch tape test. The deposited film is designated as TNC.

Hydrothermal Synthesis of TiO₂ Nanorods (TNR). A TiO₂ TNR on conducting glass electrodes was prepared by the hydrothermal method. In a typical synthesis, titanium (IV) isopropoxide (0.5 mL) was added to equal volume of distilled water and concentrated HCl. The resulting mixture was stirred for 30 min. The clear transparent solution was then transferred to a Teflon-lined stainless steel autoclave (25 mL), and a piece of F-doped SnO₂ (FTO) glass (Solaronix, 15 Ω·cm⁻²) was immersed in the solution parallel along the wall. The autoclave was sealed and placed in an oven at 180 °C for 3 h. After cooling the autoclave to room temperature, the TiO₂ films were washed several times with distilled water and dried in oven at 70 °C for 1 h designated as TNR.

Hydrothermal Synthesis of TiO₂ Nanoflowers (TNF). TiO₂ nanoflowers (TNF) on conducting glass electrodes were prepared by the hydrothermal method as per our previous method³⁶. The TiO₂ nanoflowers (TNF) were synthesized by the addition of 0.001 M, 1: 3-ethoxycarbonylethyl-1-methyl-imidazolium chloride [CMIM][HSO₄] RTIL in above precursor solution and carried out the same reaction at same 180 °C temperature for 3 h and the sample was designated as TNF. For comparison we used P25 Degussa TiO₂ nanoparticles and 1.8 μm thick film was fabricated by using well known doctor blade technique. Briefly, 1 g TiO₂ nanoparticles paste was prepared using available literature and film was deposited on FTO coated glass substrate and annealed at 450 °C for 5 min.

Device fabrication and dye sensitized solar cell tests. The DSSCs properties were measured by Sol2A Oriel New Port Corporation USA, with Keithley-2420 source meter under 1.5 AM illumination. A compact and sealed dye sensitized solar cell (DSSCs) device was fabricated using a standard two-electrode configuration, comprising dye loaded Glass/FTO/P25 (with an active surface area of 0.5 cm²) as photoanode, platinum coated FTO, as a counter electrode which is sealed with the working electrode using a spacer (1 mm) of polyacralamyde. The platinum coated FTO counter electrode, was prepared by dropping a H₂PtCl₆ solution (0.7 mM) on an FTO glass followed by heating at 400 °C for 20 min in air. The dye-adsorbed

Table 2 | Fitted impedance parameters of the DSSC's for different devices

| Device | Q ₁ (F) | R ₂ Ω·cm ^{2(a)} | R ₁ Ω·cm ² | τ _e ± 17% (ms) |
|--------|------------------------------|--|-------------------------------------|------------------------------|
| P25 | (1.25e-8) ¹ | 53 | 150000 | 0.00066 |
| TNC | (1.033e-5) ¹ | 238 | 2246 | 0.245 |
| TNR | (1.495e-5) ^{0.6562} | 2531 | 228900 | 0.112 |
| TNF | (5.753e-5) ^{0.8007} | 23 | 23.8 | 0.116 |

[a] Measured under dark conditions with an applied potential = -0.70 V and a cell area = 0.5 × 0.51 cm².



nanostructured TiO₂ electrodes and the Pt counter electrode was sealed with 25 μm-thick Surlyn (SX1170–25, Solaronix) thermoplastic at a pressure of ~210 Kpacm⁻² and a temperature of about ~110°C. The Iodolyte AN-50 (Solaronix) electrolyte was inserted through predrilled hole from Pt/FTO counter electrode. Finally device was sealed using thermoplastic. All the measurements were carried out at room temperature in air.

Dye Adsorption Measurement. To measure the adsorbed dye concentration on the TiO₂ electrode surface, the dye loaded TiO₂ samples were desorbed by a 0.1 M NaOH solution in water and ethanol (50 : 50, v/v) for 10 min. The dimension of the TiO₂ electrode was 1 cm². The adsorbed dye was quantitatively determined from the absorbance at 510 nm measured by UV-1800, Shimadzu UV-vis spectrophotometer (Resolution 0.01 nm).

For more details of effective dye loading, characterization techniques used for TiO₂ samples, device fabrication and dye sensitized solar cell tests, XRD spectra of TiO₂ samples refer Fig. S1 and Fig S2, for FT-IR spectra of TiO₂ samples refer Fig. S3, and for Nitrogen adsorption-desorption isotherms plots of TiO₂ samples refer to Fig. S4.

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Author contributions

S.S.M., C.K.H. and P.S.P. contributed to the conception and design of the experiments, analysis of the data and writing the paper. S.S.M. carried out all experiments and wrote the paper. S.S.M. and C.A.B. performed EIS measurements, calculations and analyzed data. C.A.B., P.S.P. and P.N.B. participated in the scientific discussion and valuable suggestions during the course of this manuscript. All authors discussed the results and reviewed the manuscript.

Additional information

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