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Study on the lignin-derived sp^2 – sp^3 hybrid hard carbon materials and the feasibility for industrial production

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Hard carbon has been widely used in anode of lithium/sodium ion battery, electrode of supercapacitor, and carbon molecular sieve for CO₂ capture and hydrogen storage. In this study the lignin derived hard carbon products are investigated, and the conclusions are abstracted as follows. (1) The lignin derived hard carbon products consist of microcrystal units of sp^2 graphene fragments, jointed by sp^3 carbon atoms and forming sp^2 – sp^3 hybrid hard carbon family. (2) From the lignin precursors to the sp^2 – sp^3 hybrid hard carbon products, most carbon atoms retain their original electron configurations (sp^2 or sp^3) and keep their composition in lignin. (3) The architectures of lignin-derived hard carbon materials are closely dependent on the forms of their lignin precursors, and could be preformed by different pretreatment techniques. (4) The carbonization of lignin precursors follows the mechanism “carbonization in situ and recombination nearby”. (5) Due to the high carbon ratio and abundant active functional groups in lignin, new activation techniques could be developed for control of pore size and pore volume. In general lignin is an excellent raw material for sp^2 – sp^3 hybrid hard carbon products, a green and sustainable alternative resource for phenolic resin, and industrial production for lignin derived hard carbon products would be feasible.

Keywords Biomass, Lignin, Hard carbon, Sodium ion battery, Graphene

Hard carbon is a type of carbon materials that cannot be graphitized under very high temperature even over 2500 °C^{1,2}. On the other hand, at temperature around 1400 °C soft carbon could be converted to graphite crystal over 90%^{3,4}. In recent decades hard carbon has found important applications in anode materials of lithium ion battery^{5–7} and sodium ion battery^{8–12}, electrode materials of supercapacitor^{13–16}, and carbon molecular sieves for gas separations¹⁷, such as H₂/CH₄ and N₂/O₂ separations^{18–20}, and CO₂ capture²¹, all are urgent needed materials for the solutions of energy crisis, climate change, environment and ecology protection^{22–25}. The traditionally raw materials of hard carbon are several synthetic resins, such as epoxy resin, ion exchange resin, and phenolic resin^{26–28}, all are manufactured from coal and petroleum. Seeking for alternative and sustainable feedstocks for hard carbon production is an urgent task.

Traditionally biomass had long been the precursors of various carbon products for hundreds and thousands years. So far many researchers still keep searching different biomass species for various carbon materials, such as eucalyptus wood, corncobs, peanut shells, waste apples, and orange peel^{29–31}. On the other hand, more detailed study revealed that all biomass species consist of three components: cellulose, hemicellulose and lignin. The carbon products fabricated from the three components of biomass are quite different in structures and properties, also different from the carbon products directly produced from biomass species^{32–35}. Among the three components of biomass, lignin is the second-abundant component, very different from the other two carbohydrate components (cellulose and hemicellulose) in chemistry and physics properties. It has been noted that a variety of high-value carbon materials can be made from lignin, such as low-cost carbon fiber^{36,37} and carbon black to strengthen tires³⁸. Therefore, lignin is a green and sustainable resource with great potential to produce high-value carbon materials.

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The natural occurrent lignin has been estimated to be around 0.5–0.6 billion tons annually, as well as about 40–50 million tons of lignin from pulp and paper industry and 100,000–200,000 tons from cellulosic ethanol industry as a pollutant byproduct^{39,40}. Therefore the most abundant and available lignin resource is the pulp mills, where the cellulose are extracted from wood or grass feedstock for paper pulp, meanwhile lignin is discarded in black liquor. In black liquor lignin chemically combines with alkali (sodium hydroxide) through ester bonds and salt bonds, forming dissoluble alkali lignin. For high-quality hard carbon products, lignin has to be further extracted and purified from black liquor by means of acidification reaction or electro dialysis. This study is devoted to exploring technical and theoretical issues of lignin derived hard carbon, particularly the feasibility of its industrial production.

Methodology and lignin materials

Methodology

The lignin precursors and carbon products are measured and characterized using physical and chemical methods. The SEM (Scanning electron microscope) images are taken by using an instrument Hitachi S-3400. In order to obtain clearer SEM images, the samples are first coated by gold-beam. The crystal structures of lignin samples and derived carbon products are characterized by the HRTEM (High Resolution Transmission Electron Microscope). In this study the HRTEM images are taken by a commercial corporation (Tianhe (Shandong) Testing Technology Co, Ltd, <https://www.keysci.com/>) using Transmission Electron Microscope (Tecnai G2 F30). The Minimum Dose System and other helpful tools are integrated into this instrument. In order to avoid samples to be damaged, when the HRTEM images of lignin samples are taken, the irradiation intensity and imaging conditions are carefully adjusted and selected. The chemical structures of lignin samples are identified by using FTIR (Fourier transform infrared spectroscopy) instrument (Thermo fisher Nicolet IS10). The crystal structures of lignin derived hard carbon materials are characterized by using laser Raman spectroscopy (Renishaw inVia Raman spectrometer with a laser wavelength of 532 nm). The specific surface area (SSA) and pore structure of active carbon is measured and characterized by using Automatic Volumetric Sorption Analyzer (Autosorb-1MP, Quantachrome) with BET (Brunauer, Emmett and Teller)⁴¹ and DFT (density functional theory)⁴² methods. The atomic composition and electron states on the surface of lignin precursors and derived carbon products are analyzed by using XPS (PHI Quantera II)^{43,44}.

Lignin raw materials

In this study the black liquor was provided from Yuanfu biomass technology Co. Ltd in Sanming, Fujian, where bamboo and sodium hydroxide were used as the feedstock for paper pulp production.

Lignin structure and characters

Lignin is a random polymer of three phenol monomers without definite chemical formula and has no fixed crystal structure. An approximate empirical chemical formula of lignin ($C_{31}H_{34}O_{11}$)_n was proposed by Faix et al.⁴⁵. Based on this formula of lignin the element composition is around carbon 63.4%, oxygen 30%, and hydrogen 5.9%. Figure 1a shows a structural fragment of lignin, Fig. 1b is the three phenolic monomers (coniferyl alcohol, coumaryl alcohol, and syringyl alcohol), and Fig. 1c is a structural fragment of phenolic resin. In the molecular structure of lignin the sp^2 aromatic rings of phenol monomers are connected by sp^3 carbon short chains, forming a three dimensional network.

Phenolic resin is a widely used precursor for hard carbon materials^{26–28}. Comparing lignin with phenolic resin, there are three common grounds: (1) Both are the most carbon rich polymers among all organic polymers, the carbon ratio in lignin is 63.4% and in phenolic resin is 69%. (2) Both are made up of sp^2 carbon aromatic rings and jointed by sp^3 short carbon chains, forming 3D network. (3) Both are random polymers without fixed structures and molecular formulas. However, between lignin and phenolic resin there is a significant difference. Phenolic resin is a stable polymer, in which, except for hydroxyl groups, there are almost no other active functional groups. On the other hand in lignin there are many active functional groups, including alcohol hydroxyl, phenol hydroxyl, carbonyl, ester and ether groups, as indicated in the FTIR spectrum (Fig. 2). In the XPS spectrum of carbon in lignin (Fig. 3a) there are two overlapped peaks between 283 and 288 eV. The red one is the peak of C1s atoms centered at 284.7 eV, composed by the sub-peak of sp^2 carbon (284.5 eV) and the sub-peak of sp^3 carbon (285.1 eV); and the blue one is the peak of carbon atoms in oxygen groups, centered at 286.2 eV, composed by sub-peaks of three carbon–oxygen groups, as listed in Table 1. The XPS peak of oxygen in lignin is composed by four oxygen–carbon groups between 529 and 536 eV, as shown in Fig. 3b. The most favorable aspects of lignin are its very high carbon content (63.4%) and the sp^2 (72.7%) and sp^3 (27.3%) hybrid carbon structure that make it an excellent precursor for sp^2 – sp^3 hybrid hard carbon materials, also an excellent substitute for phenolic resin. A remarkable advantage of lignin precursors over phenolic resin is its active functional groups that give lignin new chemical characteristics for development of new carbonization and activation technologies, which will be studied more detailed in the next sections.

Recovery of lignin from black liquor

In this section lignin recovery from black liquor by means of acidification^{46,47} and electro dialysis^{48,49} were performed. In pulp black liquor the mass concentration of dissolved alkali lignin is around 10–18% and mixed by organic and inorganic impurities. The black liquor was first cleaned by using centrifugal filter, and the impurities were removed, including soil, sand, and insoluble organic matters. The filtered black liquor was concentrated to 20% mass percent, and was acidized by 30% H_2SO_4 to pH = 2.5, and lignin was precipitated in flocculent precipitate. The suspending lignin was filtered with centrifugal filter, then washed three times with deionized water to neutral. The recovered lignin from black liquor has a higher carbon content than that in biomass due

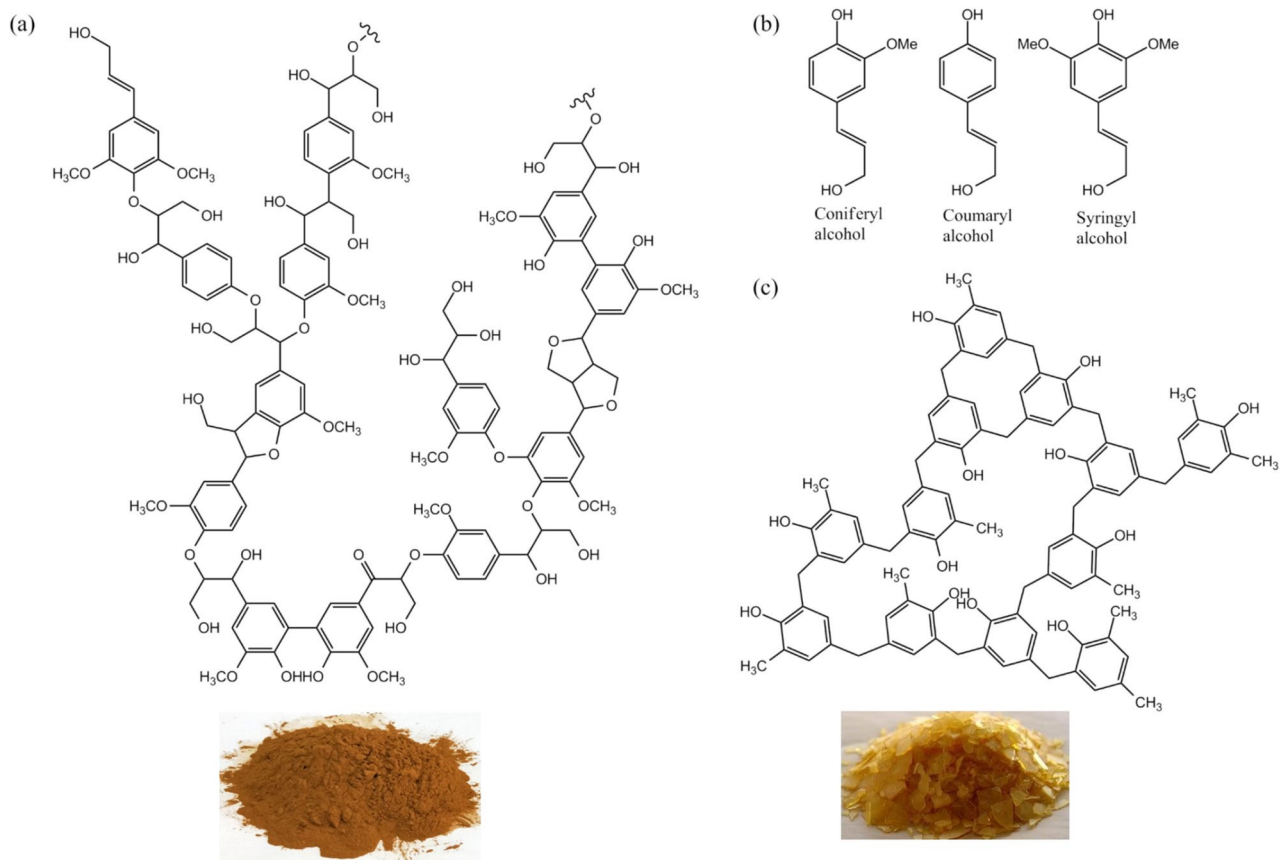


Figure 1. Molecular structures of lignin and contrast phenolic resin. **(a)** A molecular fragment of lignin. Lignin is the polymer of phenolic monomers, and has no definite molecular formula and no crystal structure. **(b)** Three phenolic monomers: coniferyl alcohol, coumaryl alcohol, and syringyl alcohol. **(c)** A molecular fragment of phenolic resin.

to dehydration reaction during the pulp cooking process⁵⁰. The acidification treatment of black liquor for lignin recovery is a lower cost procedure and could produce in large scale, however, large amount of acid is consumed, and pollutant waste water is produced.

Lignin recovery from black liquor by using electrodialysis (ED) technique^{48,49} was also performed in our research team. After black liquor was filtered using centrifugal filter removing impurities, lignin recovery was carried out in a six compartment ED cell with area 230 cm² and thickness 1.6 cm. The concentration of lignin in mother liquor was increased from 62 to 285 g L⁻¹. The suspending lignin precipitate was filtered with centrifugation filter, washed with pH = 2.5 diluted H₂SO₄ solution for removing the hemicellulose, then washed three times with deionized water to neutral. Based on our experience the problem of lignin depolymerization during electrodialysis reaction^{49,50} could be solved by aging reaction for a few days. Using electrodialysis approach for recovery of lignin from black liquor, not only can obtain pure lignin, but also can recycle the alkaline solution for pulp cooking, and obtain other useful fractions, such as xylan, in principle no wastes are produced⁵¹. Due to the remarkable merit, electrodialysis approach has received much attention for cleaner production and utilization of sustainable resource biomass.

Preparation of lignin precursors for hard carbon

Lignin has no definite molecular formula and no crystal structure. However, lignin could form various architectures under different pretreatment conditions that affect the structures of carbon products. The lignin recovered from black liquor is further pretreated by using following four methods in authors' team. (1) In the first method lignin, recovered from black liquor by acidification, was washed using DI water (deionized water) to neutral, then dried in oven at 120 °C for 6 h. (2) In the second method, after black liquor was acidized by 15% H₂SO₄ to pH = 2.5, suspending lignin precipitate was washed using DI water, then immediately transferred into a steel beaker, and was rapidly frozen and dried with liquid-nitrogen. (3) In the third method lignin, recovered by electrodialysis, was dissolved in 75% alcohol solution, then solidified using spray-drying technique. (4) In the fourth method lignin was solved in ethylene glycol solution, or *N,N*-dimethylformamide solution⁵². Then the lignin membrane was made by using electrospinning technique⁵³. Such electrospun lignin membrane, consisting of lignin fiber in micron or nanometer diameter, was heated to the melting point of lignin around 130 °C, and the lignin fiber membrane becomes lignin film in micron or nanometer thickness.

The SEM images of the lignin precursors, prepared using above four methods, are shown in Fig. 4a–d, labeled as lignin-a, lignin-b, lignin-c and lignin-d, respectively. As shown in Fig. 4a, the particles of lignin-a, prepared

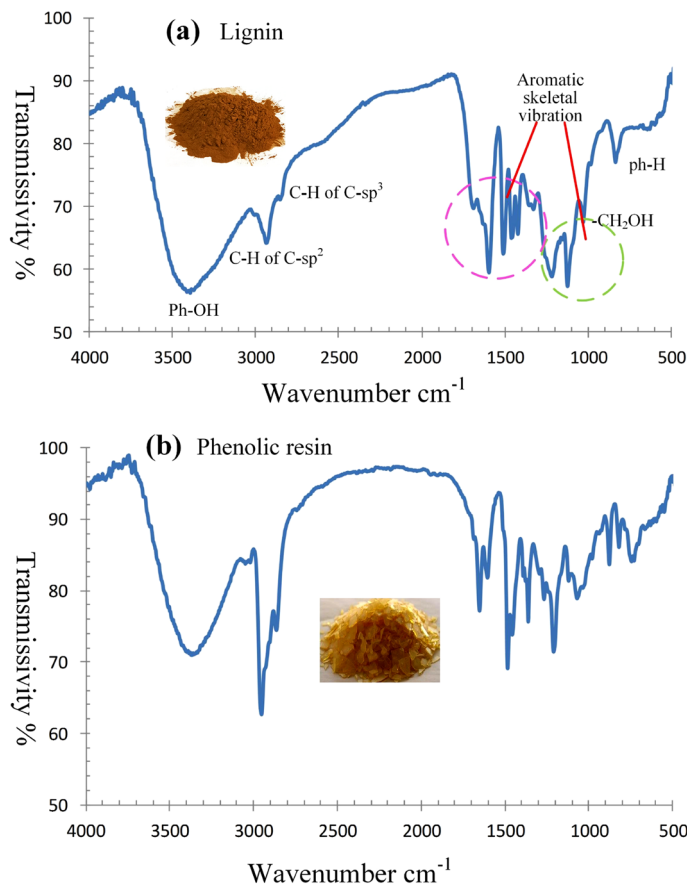


Figure 2. Comparison of FTIR spectrums between lignin and phenolic resin. (a) The FTIR spectrum of lignin. (b) The FTIR spectrum of phenolic resin. Lignin has several functional groups, alcohol hydroxyl, phenol hydroxyl, carbonyl, and ether group. The FTIR spectrums of lignin and phenolic resin are very similar.

using method 1, are like solid cement slags in dimension larger than 100 microns; the particles of lignin-b in Fig. 4b, prepared using method 2, are like softened cotton in dimension around 20 microns; and the particles of lignin-c in Fig. 4c, solidified from alcohol solution, are rough balls with diameters from 10 to 20 microns; and Fig. 4d is the SEM image of lignin film-d prepared using electrospinning technique, and the thickness of lignin film is around 1 μm .

Results and discussion

The sp^2 - sp^3 hybrid carbon products from lignin

In this section four hard carbon materials derived from the four lignin precursors are reported and discussed.

Lignin derived 3D graphenes

The carbon products, derived from lignin precursors, are strongly impacted by the structures (or architectures) of its precursors. In a published paper of Du's group³² a 3D graphene product was reported that was derived from precursors lignin-a, possessing wood-ear like architecture, is labeled as 3D-graphene-a in this study. In this study a different 3D-graphene is fabricated from precursor lignin-b at 1200 $^{\circ}\text{C}$ for 1 h in nitrogen atmosphere, and is labeled as 3D-graphene-b. Figure 5a is an SEM image of the 3D-graphene-b, in which the 3D graphene particles are like a broken porcelain shop, having different shapes, such as American football, broken earthen jar, and rose petal like sheets, and so on. On the other hand the SEM images of 3D-graphene-a in reference³², the 3D graphene particles, fabricated from precursor lignin-a, have wood-ear like architecture. An obvious difference between 3D-graphene-a and 3D-graphene-b is that the curved graphene sheets in 3D-graphene-a merge each other and form dihedral angles with sharp edges; while the graphene sheets of the 3D-graphene-b have smooth and curved surfaces, and have no the edges of dihedral angles, different from those in the former. Figure 5b and c are the HRTEM images of 3D-graphene-b, in which the particles have the graphere and fullerene-like carbon crystal structures^{54,55}. It is obvious the architectures of both 3D-graphene-a and 3D-graphene-b are strongly influenced by their precursors lignin-a and lignin-b.

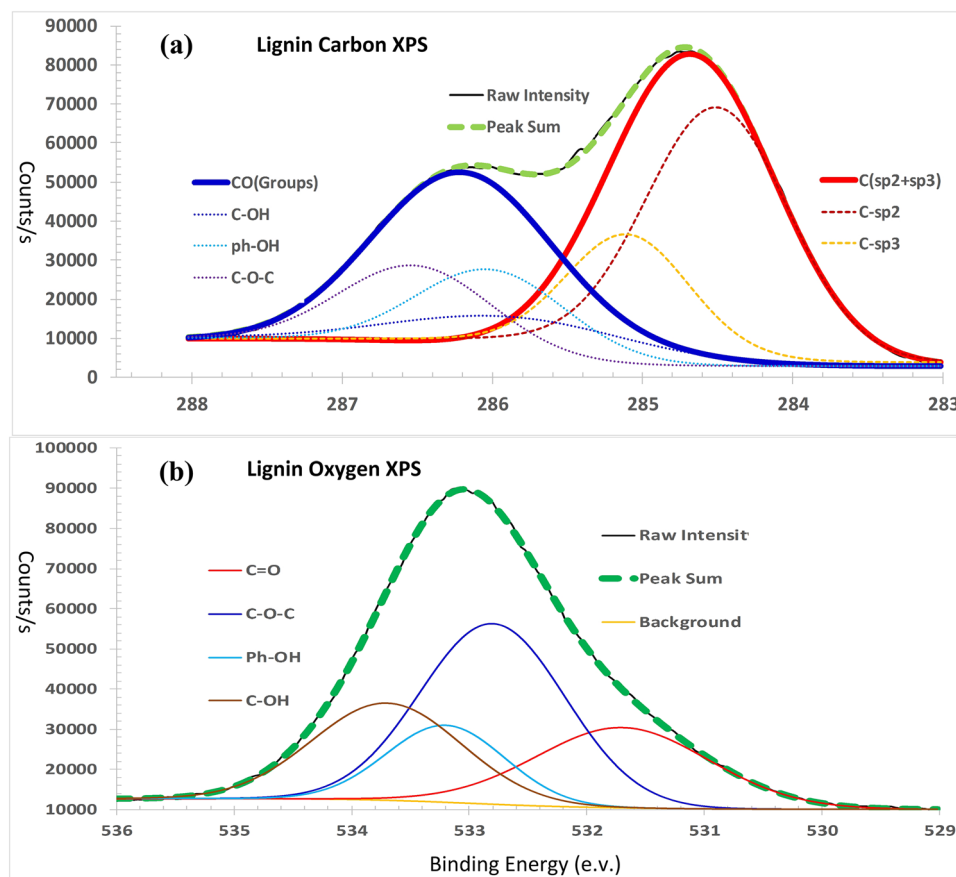


Figure 3. The XPS spectra of carbon and oxygen in lignin. **(a)** The XPS spectrum of carbon atoms in lignin. There are two overlapped peaks, the red one is the peak of C_{1s} carbon atoms centered at 284.7 eV, composed by two sub-peaks sp^2 (284.5 eV) and sp^3 (285.1 eV) of carbon atoms; and the blue one is the peak of carbon atoms in oxygen groups, centered at 286.2 eV, composed by the sub-peaks of 3 carbon–oxygen groups. **(b)** XPS spectrum of oxygen in lignin. The oxygen XPS spectrum is composed by four oxygen–carbon groups.

	Atoms	Atomic electron configuration	Position (eV)	Area	Percent
Lignin	C and C in groups	$C(sp^2 + sp^3)$	284.7	106,937	62.0
		CO groups	286.2	65,413	38.0
Carbon	C atoms	$C-sp^2$	284.5	74,006	72.7
		$C-sp^3$	285.1	27,778	27.3
	C in groups	C–OH	285.8	18,351	26.0
		Ph–OH	286.0	26,716	37.8
		C–O–C	286.5	25,582	36.2
Oxygen	O in groups	C=O	531.6	30,993	18.1
		C–O–C	532.6	45,015	26.3
		Ph–OH	533.0	51,253	29.9
		C–OH	533.6	44,222	25.8

Table 1. The components of carbon and carbon–oxygen groups in lignin derived from XPS spectrum.

Graphene microcrystalline derived from lignin

The graphene microcrystalline (GMC), first reported in reference³⁵ of Du's group, actually is a type of hard carbon^{56,57}. In this study GMC was produced from the precursor lignin-c, which was solidified using spray-drying technique from the lignin-ethylene glycol solution. In current investigation the powder of lignin-c was put in graphite crucibles, carbonized in a tube furnace in nitrogen atmosphere at temperature 1200 °C for 1 h, very hard carbon blocks were produced, shown in Fig. 6b and c. The resolution 5 nm HRTEM image of the hard carbon product is shown in Fig. 6a, in which the 3–5 parallel lines in length around 3–5 nm are the microcrystals of

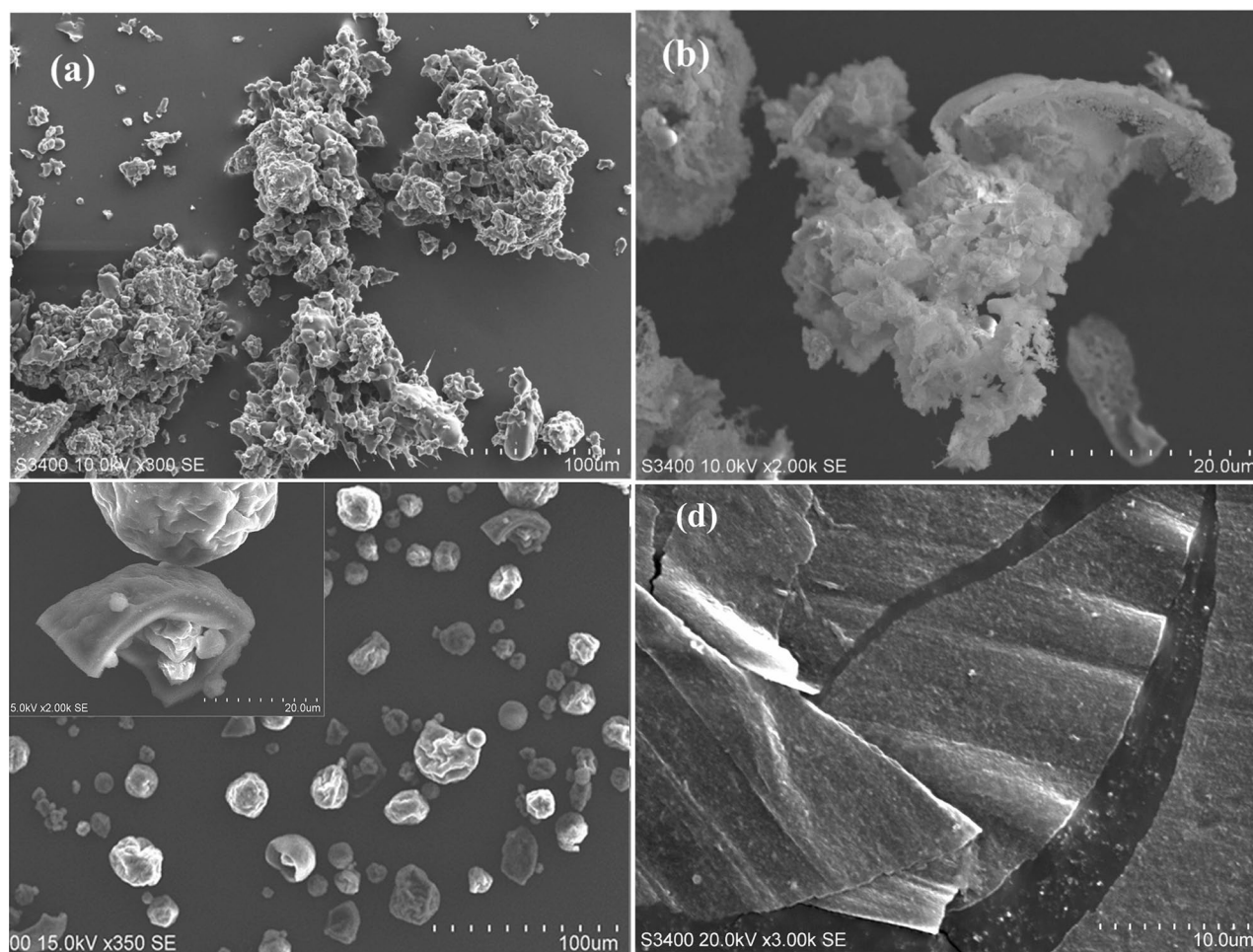


Figure 4. The SEM (scanning electron microscope) images of the lignin particles prepared using four methods. (a) The particles of lignin-a, prepared using method 1, are like solid cement slags; (b) The particles of lignin-b, prepared using method 2, are like soft cotton; (c) The particles of lignin-c, solidified from alcohol solution, are rough balls with diameters from 10 to 20 microns. (d) The lignin film-d prepared from lignin-ethanol solution by means of electrospinning technique with thickness around 1 μm .

graphene. In Fig. 6a the microcrystal units of graphene, arranged in different directions, are jointed by the sp^3 carbon atoms, forming the structure of graphene microcrystalline (GMC). Figure 6c is the SEM image of GMC, where the sharp edges of the GMC particles are very like the glass slags. Figure 6d is a model structural picture of GMC, in which the short parallel lines are the microcrystal units of graphene, and the join points are the sp^3 carbon atoms, indicated by pink circles.

The Brunauer–Emmett–Teller (BET) test was performed for GMC, and the specific surface area (SSA) was almost zero measured using MBET method, and 39 m^2/g using DFT method, and the total pore volume is merely 0.088 ml/g . All SSA of GMC distribute in the range larger than 5 nm, almost no micro pores, as shown in Fig. 8d. The graphene interlayer spacing of GMC ranges from 0.37 to 0.40 nm⁵⁸, depending on carbonization temperature and lignin precursors, and may be affected by co-reagents. The very hard substrate and the adjustable interlayer spacing of GMC makes it an excellent candidate for anode hard carbon of sodium ion battery^{58–61}.

Carbon film fabricated from lignin

The lignin membrane precursor lignin-d, shown in Fig. 4d, was put in graphite crucibles, carbonized in a tube furnace in nitrogen atmosphere at temperature 1200 $^\circ\text{C}$ for 1 h, then carbon films were produced. Figure 7a is a SEM image of the carbon film that keeps the membrane shape of its precursor lignin-d, and Fig. 7b is a HRTEM image of the carbon film in 5 nm resolution, consisting of graphene microcrystal sheets arranged in different directions, where the graphene hexagonal lattices can be seen clearly. In a recent reference⁶¹ bio-derived hard carbon nanosheets found application in sodium ion battery.

Porous carbon fabricated from alkali lignin

The GMC hard carbon, produced from lignin precursor lignin-c in section (2), has very small specific surface area, and almost no micro pore volume smaller than 5 nm. On the other hand, with the help of activating reagents⁶², porous hard carbon could be fabricated from lignin precursors. In this section alkali lignin, provided

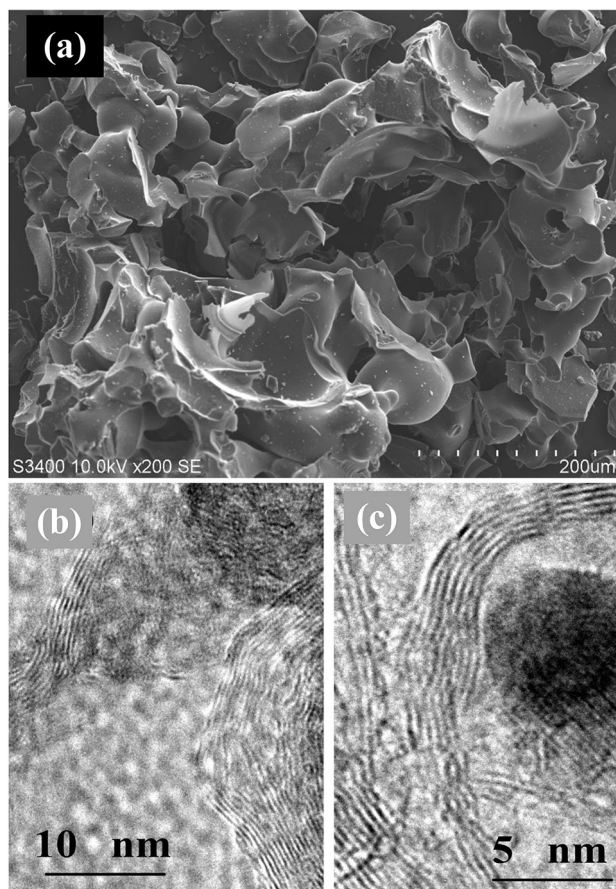


Figure 5. The SEM and HRTEM images of 3D-graphene-b. (a) The SEM image of 3D-graphene-b derived from lignin-b. The particles are like variety store: American football, broken earthen jar, and rose petals. (b) and (c) The HRTEM images of 3D-graphene-b, where the particles of 3D-graphene-b have the fullerene-like carbon crystal structures.

by Yuanfu Co.Ltd., was used to manufacture porous carbon, in which lignin is the carbon source and alkali is the activating reagent. The powder of alkali lignin was put in graphite crucibles, carbonized and activated in a box-type furnace in nitrogen atmosphere at temperature 750 °C for 1 h. The output porous carbon mixed with molten sodium hydroxide was soaked in clean water, then was filtered for separation of alkaline solution and carbon. The alkaline solution can be directly used for cooking pulp, and porous carbon was washed by 2 M hydrochloric acid, then washed with deionized water to neutral. BET surface adsorption test was performed for porous carbon, and results are shown in Fig. 8a. The BET SSA of porous carbon from alkali lignin is 564 m²/g, much larger than that of GMC in Fig. 8d.

In order to promote the micro pores in porous carbon, which is very important for carbon molecular sieves for gas separation, polyethylene glycol (PEG)⁶³ was used as an co-activating reagent. Alkali lignin powder was mixed with PEG-4000 in ratio 10:1, then was carbonized and activated using the same methods as above, and the results were shown in Fig. 8b. Comparing with Fig. 8a, after co-activating reagent PEG-4000 was used, the BET SSA, particularly the micro pores smaller than 1 nm, increase remarkably.

The SSA of porous carbon, produced directly from alkali lignin, is not very high, because sodium hydroxide is not a powerful activating agent comparing with expensive potassium hydroxide⁶⁴. However, a remarkable merit is that in the process of making porous carbon from alkali lignin, the alkali consumed by pulping is recovered and can be recycled for pulp production.

In order to produce hard porous carbon with large surface area for supercapacitors, activation reagent potassium carbonate (K₂CO₃) and co-activation reagent urea were added in alkali lignin in ratio 1:0.4:0.2. Using the same procedure as above and with the help of K₂CO₃ and urea, the SSA of alkali lignin derived hard porous carbon reached 2619 m²/g, and the total pore volume was 1.332 ml/g, as shown in Fig. 8c.

Theoretical analysis of lignin-derived hard carbon

In this section the *sp*²-*sp*³ hybrid structures of lignin-derived carbon materials, the carbonization mechanism of lignin, and the new activating technique for lignin porous carbon are theoretically analyzed and discussed.

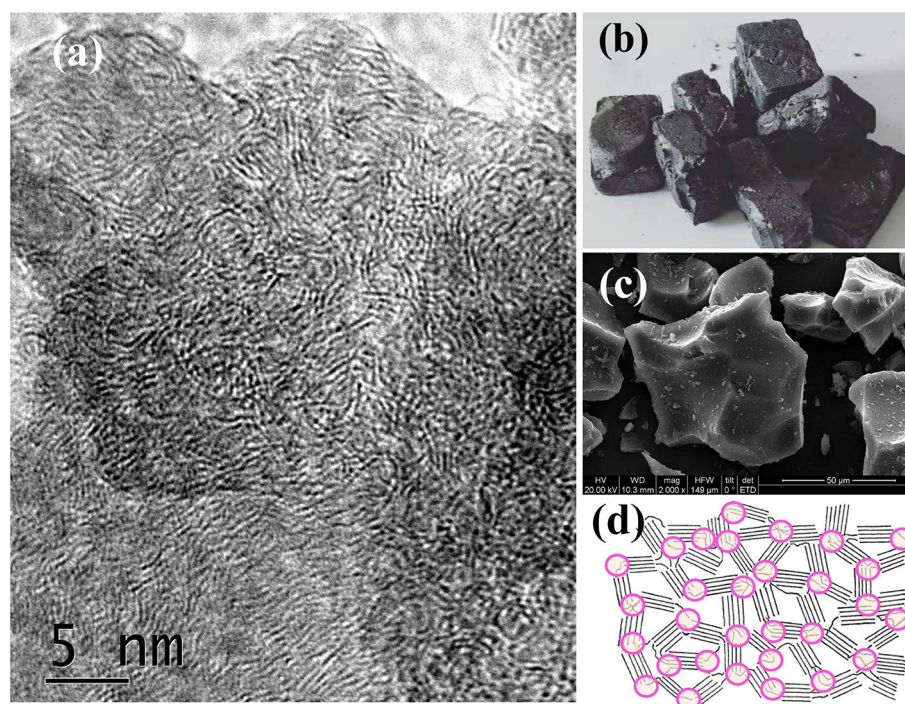


Figure 6. The SEM and HRTEM images of GMC (graphene microcrystalline). **(a)** The 5 nm HRTEM image of GMC. The 3 to 5 parallel lines in length around 3–5 nm are the microcrystals of graphene. **(b)** The photograph of GMC. The hard GMC blocks are like cement slurry. **(c)** The SEM image of GMC. The sharp edges of the GMC particles are very like the glass slags. **(d)** A model structural picture of GMC. The short parallel lines are the microcrystals of graphene, and the joint points are the sp^3 carbon atoms, indicated by pink circles.

The sp^2 – sp^3 hybrid structures

The Raman spectrum of typical graphene⁶⁵ has a D-band at 1350 cm^{-1} , a G-band at 1580 cm^{-1} and a 2D band at 2680 cm^{-1} . The substrate of the four lignin derived hard carbon materials is the graphene microcrystal. Therefore the Raman spectrums of lignin derived hard carbon products are similar to that of graphene, however, disturbed by the sp^3 carbon atoms to different degrees. The Raman spectrums of the four lignin derived hard carbon materials are shown in Fig. 9. In the Raman spectrum of GMC (Fig. 9b) there are all the three Raman bands (D, G and 2D), indicating its finer structure of microcrystal graphene, while other three lignin derived carbon products have no clear 2D-band. Particularly in the Raman spectrum of porous carbon (Fig. 9d) the Raman bands (D and G) are lower than others because of its less finer crystal structure of microcrystal graphene.

The XPS surveys of lignin and 4 lignin-derived hard carbon materials are shown in Fig. 10, where Fig. 10f is the portions of carbon, oxygen and other elements. The peak-fitting technique⁶⁶ was used to separate the overlap peaks of carbon and oxygen in the XPS spectrums of four lignin-derived carbon materials, the results are shown in Fig. 11, and the electron configuration portions of carbon atoms and oxygen atoms are summarized in Table 2.

Based on Fig. 11 and Table 2 all the 4 lignin-derived hard carbon materials are sp^2 – sp^3 hybrid carbon products, and the ratios (65.2–74.8%) of sp^2 carbon atoms and the portions (25.2–34.8%) of sp^3 carbon atoms are close to the ratio in lignin, 72.7% of sp^2 and 27.3% of sp^3 , indicating that during carbonization of lignin at higher temperature most carbon atoms keep their original electron configurations.

Carbonization mechanism of lignin

A well known phenomenon is that the appearances and architectures of biomass carbon products are very like their biomass precursors, there must be some kind of relationship between bio-precursors and carbon products. The traditional point of view is that carbonization of biomass precursors follows the mechanism “carbonization in situ”^{67–70}, it means during the carbonization of biomass organics the carbon atoms stay in their original positions, and the carbon products keep the shapes and architectures of their biomass precursors. In this study the carbonization reactions of lignin precursors follow the same mechanism, and the four hard carbon products keep the architectures of their precursor (lignin-a, b, c, and d). Consequently the architectures and shapes of lignin derived hard carbon materials could be designed by pre-shaping the architectures of lignin precursors.

In this study all the four lignin derived hard carbon products are sp^2 – sp^3 hybrid structure, like their lignin precursor. It follows from this phenomenon that in carbonization reaction of lignin, the benzene rings of sp^2 carbon atoms form the graphene fragments, and the sp^3 carbon atoms of short carbon chains joint the nearby graphene units, forming graphene microcrystal units, and most carbon atoms keep their original electron configuration in lignin. Consequently, all carbon products derived from lignin are sp^2 – sp^3 hybrid carbon materials with hard graphene microcrystal structures.

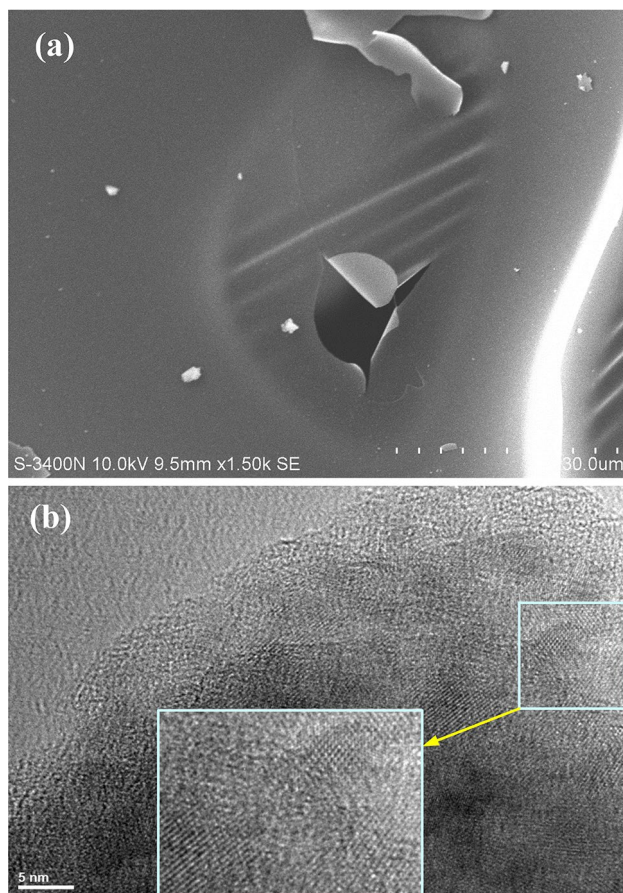


Figure 7. The SEM and HRTEM images of carbon film derived from precursor lignin-d. **(a)** The SEM image of carbon film keeps the film shape of its precursor lignin-d. **(b)** The HRTEM image of the carbon film in 5 nm resolution, consisting of graphene microcrystal units arranged in different directions, where the graphene hexagonal lattices can be seen clearly.

Activation of lignin derived porous carbon

The traditional production of porous carbon from biomass precursors needs two steps. The first step is the carbonization of precursors, and the second step is the activation of carbon materials using physical or chemical techniques, which are well developed and very familiar for researchers. However, the high carbon ratio and active functional groups in lignin provide the possibility to develop new activation techniques for lignin derived porous carbon production⁶². The activating reagent potassium hydroxide or potassium carbonate can chemically combine with the phenolic hydroxyl groups of lignin, forming alkali-lignin complex, making the activation reaction accurately and efficiently. In addition various co-activating reagents can be used for lignin derived porous carbon, such as urea⁶² and polymer PEG that help pore developing and controlling pore size. A merit of lignin derived porous carbon over other biomass derived carbon products is that the very hard GMC substrate allows lignin porous carbon bearing more holes and larger specific surface area without collapsing than other biomass derived porous carbon products.

Conclusion

The conclusion points from this study are summarized as follows. (1) The lignin derived hard carbon products consist of microcrystal units of sp^2 graphene fragments, jointed by sp^3 carbon atoms and forming sp^2 - sp^3 hybrid hard carbon family. (2) From the lignin precursor to the sp^2 - sp^3 hybrid hard carbon products, most carbon atoms retain their original electron configurations (sp^2 or sp^3) and keep their composition in lignin. (3) The architectures of lignin-derived hard carbon materials are closely dependent on the forms of their lignin precursors, and could be performed by different pretreatment techniques. (4) The carbonization of lignin precursors follows the mechanism “carbonization in situ and recombination nearby”. (5) Due to the high carbon ratio and abundant active functional groups in lignin, new activation techniques could be developed for control of pore size and pore volume. (6) The remaining oxygen atoms in the lignin-derived hard carbon materials are in the forms C=O and C-OH and in the range 5–15%, depending on the carbonization temperature and catalysts, which may provide Faradaic pseudocapacitance and improve hydrophilicity.

In general lignin is an excellent raw material for sp^2 - sp^3 hybrid hard carbon products, a green and sustainable alternative resource for phenolic resin. Based on our experiments the carbon yields from lignin to hard carbon

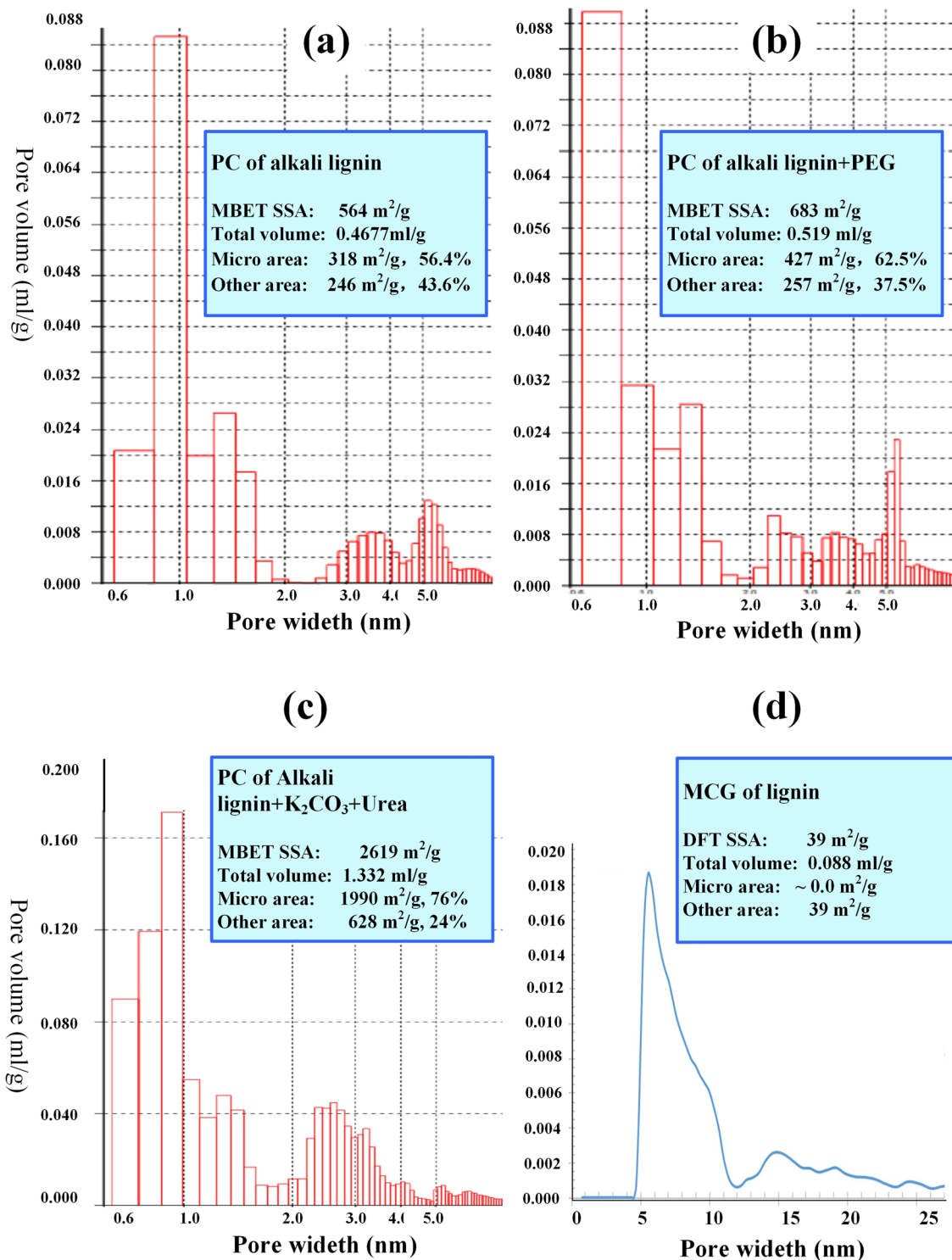


Figure 8. The BET (Brunauer–Emmett–Teller) measurement results of GMC (graphene microcrystalline) and porous carbon derived from lignin. (a) The pore volume vs pore width of porous carbon of alkali lignin. (b) The pore volume vs pore width of porous carbon of alkali lignin + PEG-4000. (c) The pore volume vs pore width of porous carbon of alkali lignin + K₂CO₃ + urea. (d) The pore volume vs pore width of GMC.

products are around 40–60%, depending on reaction time and temperature. The yield of lignin-derived porous carbon is 30–40%. Consequently, industrial production for lignin-derived hard carbon materials would be feasible, also beneficial to ecological and environmental protection, and also to sustainable economy.

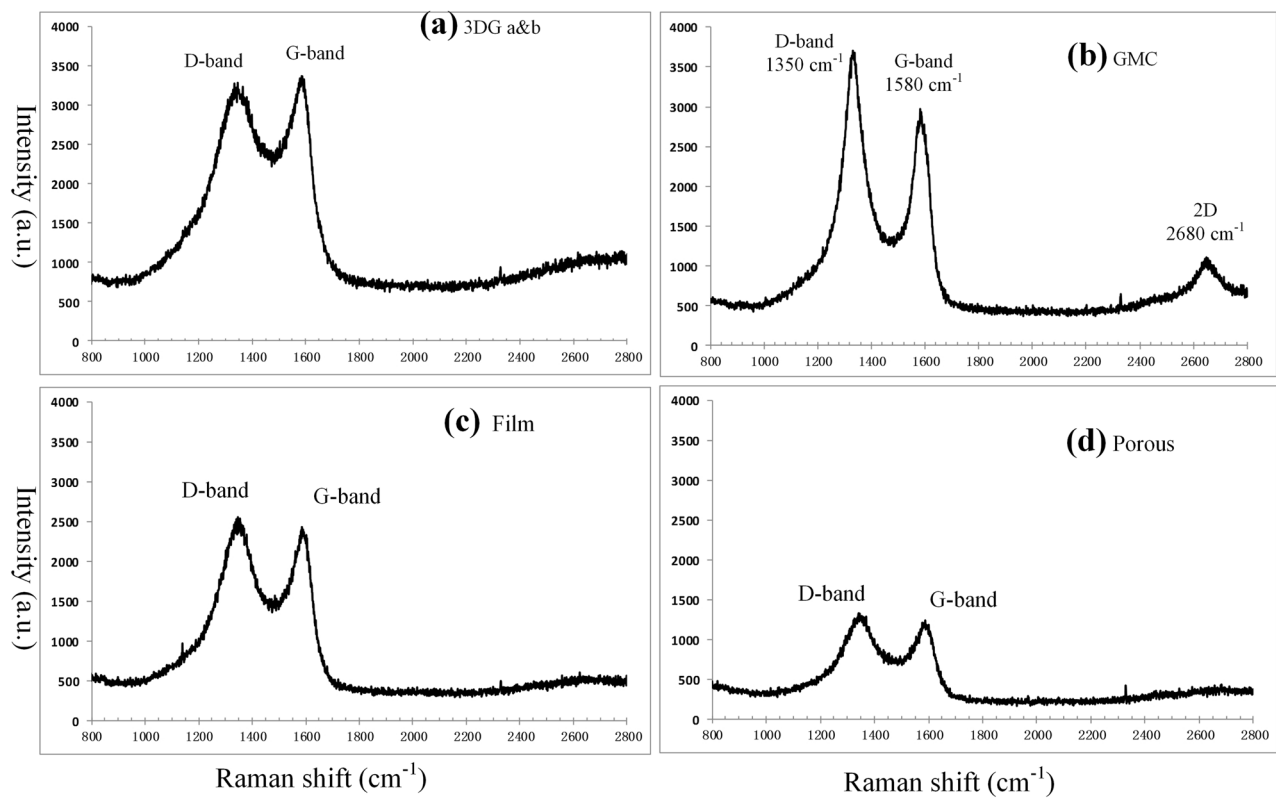


Figure 9. The Raman spectra of 4 lignin derived hard carbon materials. (a) Raman spectrum of 3DG (3D graphene). (b) Raman spectrum of GMC (graphene microcrystal). (c) Raman spectrum of carbon film. (d) Raman spectrum of porous carbon.

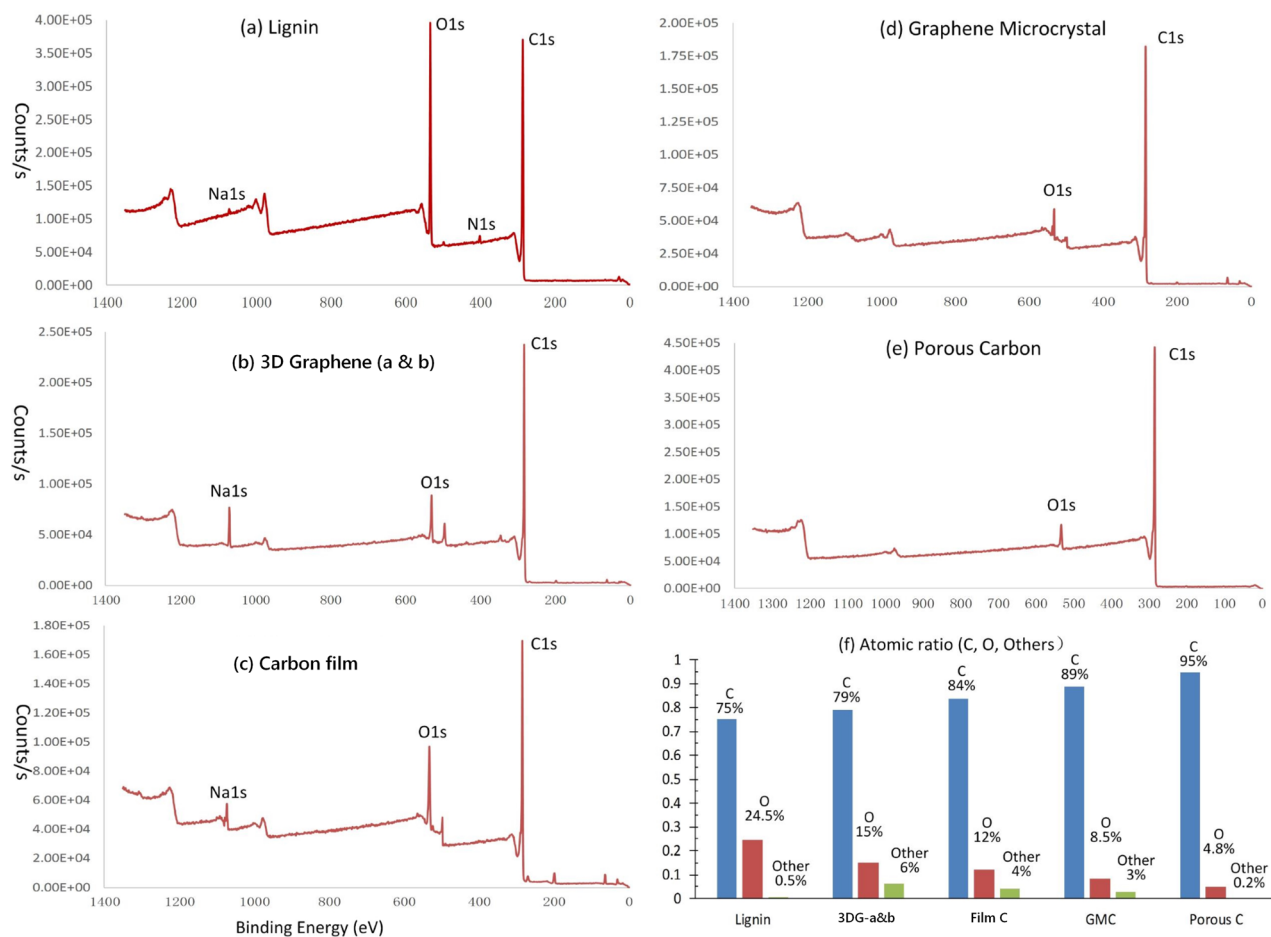


Figure 10. The XPS surveys of lignin and 4 carbon materials derived from lignin precursors. (a) XPS survey of carbon in lignin. (b) XPS survey of 3D graphene a & b. (c) XPS survey of carbon film. (d) XPS survey of graphene microcrystal. (e) XPS survey of porous carbon. (f) The portions of carbon, oxygen and other elements in lignin and four carbon materials.

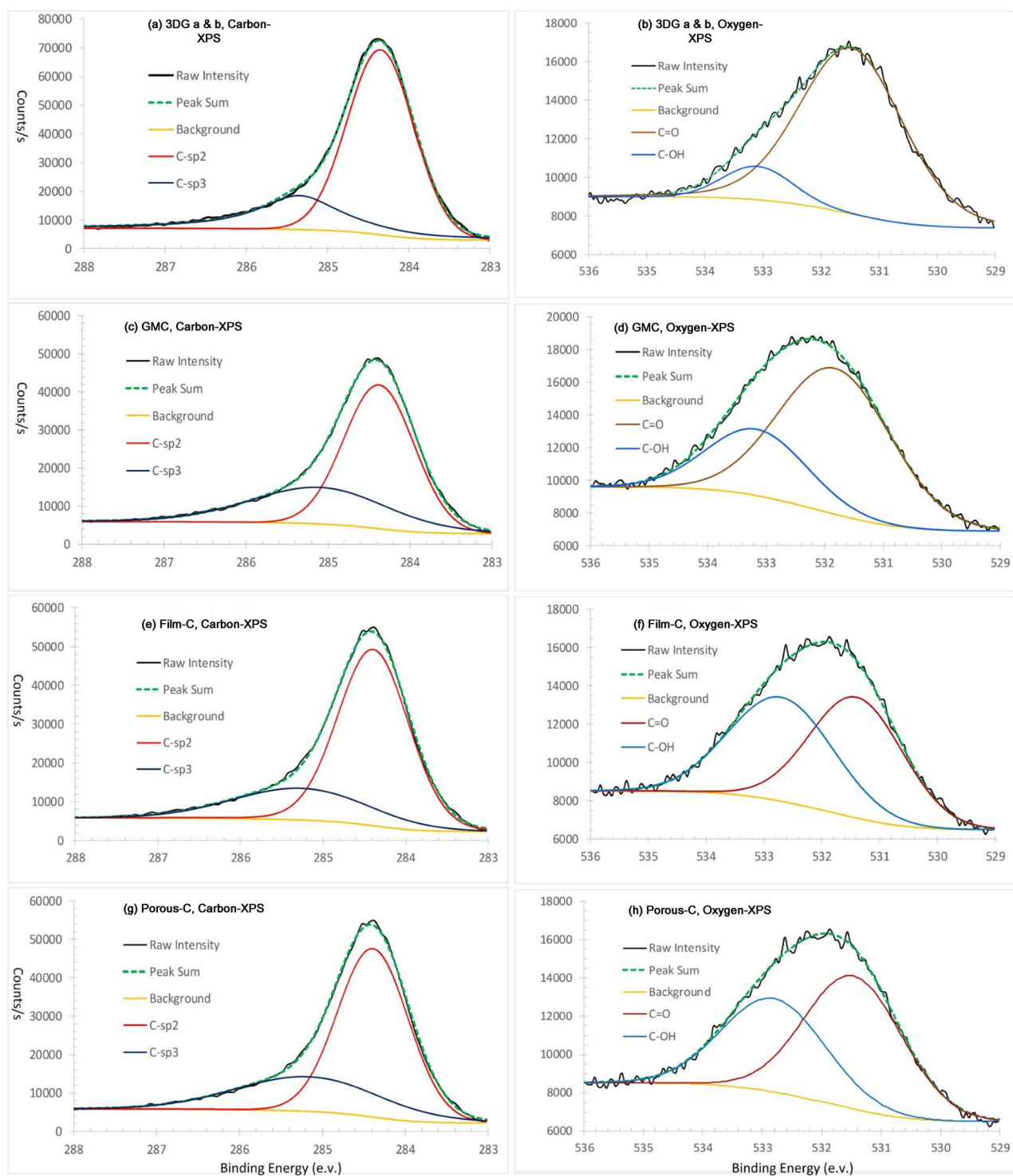


Figure 11. Computational decompositions of XPS spectrums of carbon and oxygen atoms in four carbon materials derived from lignin precursors. **(a)** Carbon XPS of 3D graphene a & b. **(b)** Oxygen XPS of 3D graphene a & b. **(c)** Carbon XPS of graphene microcrystal (GMC). **(d)** Oxygen XPS of graphene microcrystal (GMC). **(e)** Carbon XPS of carbon film. **(f)** Oxygen XPS of carbon film. **(g)** Carbon XPS of porous carbon. **(h)** Oxygen XPS of porous carbon.

	Atom	Electron configuration	Position (eV)	Area	Percent (%)
3DG-a	C atoms	C- sp^2	284.4	66,219	74.6
		C- sp^3	285.3	22,509	25.4
	O in groups	C=O	531.5	20,549	88.9
		C-OH	533.1	2567	11.1
3DG-b	C atoms	C- sp^2	284.4	40,341	65.2
		C- sp^3	285.1	21,494	34.8
	O in groups	C=O	531.8	21,350	71.2
		C-OH	533.2	8652	28.8
GMC	C atoms	C- sp^2	284.4	50,422	74.8
		C- sp^3	285.2	16,964	25.2
	O in groups	C=O	531.4	12,714	51.0
		C-OH	532.7	12,214	49.0
Porous carbon	C atoms	C- sp^2	284.4	46,751	70.0
		C- sp^3	285.2	20,043	30.0
	O in groups	C=O	531.5	14,359	57.6
		C-OH	532.8	10,569	42.4

Table 2. The components of carbon and carbon–oxygen groups in lignin-related carbon materials derived from XPS spectrums.

Data availability

All data that support the findings in this study are in the article, and asking for detailed experimental data is possible depending on further communication.

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References

- Oberlin, A. & Terriere, G. Graphitization studies of anthracites by high-resolution electron-microscopy. *Carbon* **13**, 367–376. [https://doi.org/10.1016/0008-6223\(75\)90004-4](https://doi.org/10.1016/0008-6223(75)90004-4) (1975).
- Marsh, H., Rodriguez-Reinoso, F., Marsh, H. & RodriguezReinoso, F. *Activated Carbon (Origins)* (Elsevier, 2006).
- Xu, H. Q. *et al.* Strengthening synergistic effects between hard carbon and soft carbon enabled by connecting precursors at molecular level towards high-performance potassium ion batteries. *Nano Res.* **16**, 10985–10991. <https://doi.org/10.1007/s12274-023-5853-1> (2023).
- Liu, X. *et al.* Catalytic synthesis of hard/soft carbon hybrids with heteroatom doping for enhanced sodium storage. *Chemistryselect* **4**, 3551–3558. <https://doi.org/10.1002/slct.201900501> (2019).
- Yoshio, M., Wang, H., Fukuda, K., Abe, T. & Ogumi, Z. Soft carbon-coated hard carbon beads as a lithium-ion battery anode material. *Chem. Lett.* **32**, 1130–1131 (2003).
- Zhang, X., Han, S., Fan, C., Li, L. & Zhang, W. Hard carbon enveloped with graphene networks as lithium ion battery anode. *Mater. Lett.* **138**, 259–261 (2015).
- Raj, K. A., Panda, M. R., Dutta, D. P. & Mitra, S. Bio-derived mesoporous disordered carbon: An excellent anode in sodium-ion battery and full-cell lab prototype. *Carbon* **143**, 402–412. <https://doi.org/10.1016/j.carbon.2018.11.038> (2019).
- Stevens, D. & Dahn, J. High capacity anode materials for rechargeable sodium-ion batteries. *J. Electrochem. Soc.* **147**, 1271 (2000).
- Bommier, C., Surta, T. W., Dolgos, M. & Ji, X. New mechanistic insights on Na-ion storage in nongraphitizable carbon. *Nano Lett.* **15**, 5888–5892 (2015).
- Qiu, S. *et al.* Manipulating adsorption–insertion mechanisms in nanostructured carbon materials for high-efficiency sodium ion storage. *Adv. Energy Mater.* **7**, 1700403 (2017).
- Sun, D. *et al.* Engineering the trap effect of residual oxygen atoms and defects in hard carbon anode towards high initial Coulombic efficiency. *Nano Energy* **64**, 103937 (2019).
- Sun, N. *et al.* Extended “adsorption–insertion” model: A new insight into the sodium storage mechanism of hard carbons. *Adv. Energy Mater.* **9**, 1901351 (2019).
- Zhou, J. *et al.* Nitrogen-doped hierarchical porous carbon materials prepared from meta-aminophenol formaldehyde resin for supercapacitor with high rate performance. *Electrochim. Acta* **153**, 68–75 (2015).
- Ghosh, S. *et al.* Natural biomass derived hard carbon and activated carbons as electrochemical supercapacitor electrodes. *Sci. Rep.* **9**, 1–15 (2019).
- Ragupathi, V., Rishi Praneash, K. B., Panigrahi, P. & Subramaniam, N. G. Porous hard carbon as high-performance electrode material for supercapacitors: Towards sustainable approach. *ECS J. Solid State Sci. Technol.* **11**, 041010 (2022).
- Katsuyama, Y. *et al.* Hard carbon: Quinone-based redox supercapacitor using highly conductive hard carbon derived from oak wood. *Adv. Sustain. Syst.* **3**, 1970023 (2019).
- Ryoo, R., Joo, S. H. & Jun, S. Synthesis of highly ordered carbon molecular sieves via template-mediated structural transformation. *J. Phys. Chem. B* **103**, 7743–7746 (1999).
- Vu, D. Q., Koros, W. J. & Miller, S. J. Effect of condensable impurity in CO₂/CH₄ gas feeds on performance of mixed matrix membranes using carbon molecular sieves. *J. Membr. Sci.* **221**, 233–239 (2003).
- Shirley, A. & Lemcoff, N. Air separation by carbon molecular sieves. *Adsorption* **8**, 147–155 (2002).
- Cazorla-Amoros, D., Alcaniz-Monge, J., de la Casa-Lillo, M. A. & Linares-Solano, A. CO₂ as an adsorptive to characterize carbon molecular sieves and activated carbons. *Langmuir* **14**, 4589–4596. <https://doi.org/10.1021/la980198p> (1998).
- Abbasi, T. & Abbasi, S. A. Decarbonization of fossil fuels as a strategy to control global warming. *Renew. Sustain. Energy Rev.* **15**, 1828–1834. <https://doi.org/10.1016/j.rser.2010.11.049> (2011).

22. Field, C. B., Behrenfeld, M. J., Randerson, J. T. & Falkowski, P. Primary production of the biosphere: integrating terrestrial and oceanic components. *Science* **281**, 237–240 (1998).
23. Jones, C. D. & Cox, P. M. Constraints on the temperature sensitivity of global soil respiration from the observed interannual variability in atmospheric CO₂. *Atmos. Sci. Lett.* **2**, 166–172 (2001).
24. Schlesinger, W. H. & Andrews, J. A. Soil respiration and the global carbon cycle. *Biogeochemistry* **48**, 7–20 (2000).
25. Wu, X. Y., Leonard, D. P. & Ji, X. L. Emerging non-aqueous potassium-ion batteries: challenges and opportunities. *Chem. Mater.* **29**, 5031–5042. <https://doi.org/10.1021/acs.chemmater.7b01764> (2017).
26. Datta, M. K. & Kumta, P. N. Silicon, graphite and resin based hard carbon nanocomposite anodes for lithium ion batteries. *J. Power Sources* **165**, 368–378. <https://doi.org/10.1016/j.jpowsour.2006.12.013> (2007).
27. Yasuda, E., Park, S., Akatsu, T., Tanabe, Y. & Kakihana, M. Porous surface layer in furan resin-derived carbon. *Andrologie* **17**, 255–279. <https://doi.org/10.1007/BF03040737> (1995).
28. Zhang, S. Y., Song, H. H., Zheng, M. S. & Chen, X. H. Preparation of phenolic resin-derived pyrolytic carbon materials and their electrochemical performances. *J. Beijing Univ. Chem. Technol.* **31**, 34–38 (2004).
29. Larcher, D. & Tarascon, J. M. Towards greener and more sustainable batteries for electrical energy storage. *Nat. Chem.* **7**, 19–29. <https://doi.org/10.1038/nchem.2085> (2015).
30. Wang, Q. Q. *et al.* Rice husk-derived hard carbons as high-performance anode materials for sodium-ion batteries. *Carbon* **127**, 658–666. <https://doi.org/10.1016/j.carbon.2017.11.054> (2018).
31. Liu, J. Y. *et al.* Facile synthesis of high quality hard carbon anode from Eucalyptus wood for sodium-ion batteries. *Chem. Pap.* **76**, 7465–7473. <https://doi.org/10.1007/s11696-022-02397-5> (2022).
32. Du, Q.-S. *et al.* Graphene like porous carbon with wood-ear architecture prepared from specially pretreated lignin precursor. *Diam. Relat. Mater.* **90**, 109–115 (2018).
33. Du, F.-L. *et al.* A comparative study for the organic byproducts from hydrothermal carbonizations of sugarcane bagasse and its bio-refined components cellulose and lignin. *PLoS ONE* **13**, e0197188 (2018).
34. Long, S.-Y. *et al.* Graphene two-dimensional crystal prepared from cellulose two-dimensional crystal hydrolysed from sustainable biomass sugarcane bagasse. *J. Clean. Prod.* **241**, 118209 (2019).
35. Tang, P.-D. *et al.* Fabrication and characterization of graphene microcrystal prepared from lignin refined from sugarcane bagasse. *Nanomaterials* **8**, 565 (2018).
36. Qu, W. D., Xue, Y., Gao, Y. W., Rover, M. & Bai, X. L. Repolymerization of pyrolytic lignin for producing carbon fiber with improved properties. *Biomass Bioenergy* **95**, 19–26. <https://doi.org/10.1016/j.biombioe.2016.09.013> (2016).
37. Zhang, M. & Ogale, A. A. Effect of temperature and concentration of acetylated-lignin solutions on dry-spinning of carbon fiber precursors. *J. Appl. Polym. Sci.* <https://doi.org/10.1002/app.43663> (2016).
38. Zhao, Y., Lu, C., Yu, F. R. & Zhou, H. F. Investigation of black liquor-derived carbon for removal of Cr(VI): Comparison with lignin-derived carbon. *Colloids Surf. A Physicochem. Eng. Asp.* <https://doi.org/10.1016/j.colsurfa.2022.128730> (2022).
39. Hodasova, L., Jablonsky, M., Skulcova, A. & Haz, A. Lignin, potential products and their market value. *Wood Res.* **60**, 973–986 (2015).
40. Xu, R. *et al.* Lignin depolymerization and utilization by bacteria. *Bioresour. Technol.* **269**, 557–566 (2018).
41. Serilevy, A. & Avnir, D. The Brunauer–Emmett–Teller equation and the effects of lateral interactions—A simulation study. *Langmuir* **9**, 2523–2529. <https://doi.org/10.1021/la00034a009> (1993).
42. Liu, Y. B., Peng, Y. L., An, B. H., Li, L. C. & Liu, Y. Effect of molecular structure on the adsorption affinity of sulfonamides onto CNTs: Batch experiments and DFT calculations. *Chemosphere* **246**, 125778. <https://doi.org/10.1016/j.chemosphere.2019.125778> (2020).
43. Haerle, R., Riedo, E., Pasquarello, A. & Baldereschi, A. sp(2)/sp(3) hybridization ratio in amorphous carbon from C 1s core-level shifts: X-ray photoelectron spectroscopy and first-principles calculation. *Phys. Rev. B* **65**, 045101. <https://doi.org/10.1103/PhysRevB.65.045101> (2002).
44. Krastev, V. *et al.* X-ray photoelectron spectroscopy study of carbon nitride films. *Surf. Coat. Technol.* **125**, 313–316. [https://doi.org/10.1016/s0257-8972\(99\)00583-6](https://doi.org/10.1016/s0257-8972(99)00583-6) (2000).
45. Faix, O., Grunwald, C. & Beinhoff, O. Determination of phenolic hydroxyl group content of milled wood lignins (MWLs) from different botanical origins using selective aminolysis, FTIR, 1H-NMR, and UV spectroscopy. *Holzforschung* **46**, 425–432. <https://doi.org/10.1515/hfsg.1992.46.5.425> (1992).
46. Humpert, D., Ebrahimi, M. & Czermak, P. Membrane technology for the recovery of lignin: A review. *Membranes* **6**, 42 (2016).
47. Mendes, S. F. *et al.* Forward black liquor acid precipitation: Lignin fractionation by ultrafiltration. *Appl. Biochem. Biotechnol.* **193**, 3079–3097 (2021).
48. Stiefel, S., Lölsberg, J., Kipshagen, L., Möller-Gulland, R. & Wessling, M. Controlled depolymerization of lignin in an electrochemical membrane reactor. *Electrochem. Commun.* **61**, 49–52 (2015).
49. Smith, C., Utley, J. H., Petrescu, M. & Viertel, H. Biomass electrochemistry: anodic oxidation of an organo-solv lignin in the presence of nitroaromatics. *J. Appl. Electrochem.* **19**, 535–539 (1989).
50. Pardini, V. L., Smith, C. Z., Utley, J. H. P., Vargas, R. R. & Viertel, H. Electroorganic reactions. 38. Mechanism of electrooxidative cleavage of lignin model dimers. *J. Organ. Chem.* **56**, 7305–7313. <https://doi.org/10.1021/jo00026a022> (1991).
51. Singh, A. K., Bhushan, M. & Shahi, V. K. Alkaline stable thermal responsive cross-linked anion exchange membrane for the recovery of NaOH by electrodialysis. *Desalination* **494**, 114651. <https://doi.org/10.1016/j.desal.2020.114651> (2020).
52. Bahi, A., Shao, J. Z., Mohseni, M. & Ko, F. K. Membranes based on electrospun lignin-zeolite composite nanofibers. *Sep. Purif. Technol.* **187**, 207–213. <https://doi.org/10.1016/j.seppur.2017.06.015> (2017).
53. Xiao, W. *et al.* An integrated separator/anode assembly based on electrospinning technique for advanced lithium-ion batteries. *Electrochim. Acta* **389**, 138776. <https://doi.org/10.1016/j.electacta.2021.138776> (2021).
54. Smalley, R. E. & Yakobson, B. I. The future of the fullerenes. *Solid State Commun.* **107**, 597–606. [https://doi.org/10.1016/s0038-1098\(98\)00210-5](https://doi.org/10.1016/s0038-1098(98)00210-5) (1998).
55. Zhao, Z. S. *et al.* Nanoarchitected materials composed of fullerene-like spheroids and disordered graphene layers with tunable mechanical properties. *Nat. Commun.* **6**, 6212. <https://doi.org/10.1038/ncomms7212> (2015).
56. Bai, P. X. *et al.* Elucidation of the sodium-storage mechanism in hard carbons. *Adv. Energy Mater.* **8**, 9. <https://doi.org/10.1002/aenm.201703217> (2018).
57. Xiao, B., Rojo, T. & Li, X. A Minireview on hard carbon as Na-ion battery anodes: progresses and challenges. *ChemSusChem* **12**, 133–144 (2018).
58. Asfaw, H. D., Gond, R., Kotronia, A., Tai, C.-W. & Younesi, R. Bio-derived hard carbon nanosheets with high rate sodium-ion storage characteristics. *Sustain. Mater. Technol.* **32**, e00407 (2022).
59. Lin, X., Liu, Y., Tan, H. & Zhang, B. Advanced lignin-derived hard carbon for Na-ion batteries and a comparison with Li and K ion storage. *Carbon* **157**, 316–323 (2020).
60. Chen, L. *et al.* Readiness level of sodium-ion battery technology: A materials review. *Adv. Sustain. Syst.* **2**, 1700153 (2018).
61. Xiao, B., Rojo, T. & Li, X. Hard carbon as sodium-ion battery anodes: Progress and challenges. *ChemSusChem* **12**, 133–144 (2019).
62. Tsubouchi, N., Nishio, M. & Mochizuki, Y. Role of nitrogen in pore development in activated carbon prepared by potassium carbonate activation of lignin. *Appl. Surf. Sci.* **371**, 301–306. <https://doi.org/10.1016/j.apsusc.2016.02.200> (2016).

63. Feng, L. *et al.* Preparation and characterization of polyethylene glycol/active carbon composites as shape-stabilized phase change materials. *Sol. Energy Mater. Sol. Cells* **95**(2), 644–650 (2011).
64. Manyala, N. *et al.* Coniferous pine biomass: A novel insight into sustainable carbon materials for supercapacitors electrode. *Mater. Chem. Phys.* **182**, 139–147. <https://doi.org/10.1016/j.matchemphys.2016.07.015> (2016).
65. Hulman, M. Raman spectroscopy of graphene. *Graphene* **2**, 381–411 (2021).
66. Siokou, A. *et al.* Surface refinement and electronic properties of graphene layers grown on copper substrate: An XPS, UPS and EELS study. *Appl. Surf. Sci.* **257**, 9785–9790. <https://doi.org/10.1016/j.apsusc.2011.06.017> (2011).
67. Liu, S., Zhao, S. C., Yao, Y. C., Dong, P. & Yang, C. Crystallined hybrid carbon synthesized by catalytic carbonization of biomass and in-situ growth of carbon nanofibers. *J. Mater. Sci. Technol.* **30**, 466–472. <https://doi.org/10.1016/j.jmst.2013.11.004> (2014).
68. Yuan, B. *et al.* Design of NiCo₂O₄ nanoparticles in-situ grown on lignin-derived porous carbon and MWCNTS composites for supercapacitors. *Diam. Relat. Mater.* **136**, 110079. <https://doi.org/10.1016/j.diamond.2023.110079> (2023).
69. Besocke, K. *et al.* Characterization of films deposited by in-situ wall carbonization in TEXTOR. *J. Nucl. Mater.* **145–147**, 651–654. [https://doi.org/10.1016/0022-3115\(87\)90417-X](https://doi.org/10.1016/0022-3115(87)90417-X) (1987).
70. Cui, X. *et al.* An In situ carbonization-replication method to synthesize mesostructured WO₃/C composite as nonprecious-metal anode catalyst in PEMFC. *Chem. Asian J.* **8**, 429–436. <https://doi.org/10.1002/asia.201200902> (2013).

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Author contributions

Q.S.D., S.Y.L., Y.Q. and Q.Y.W. conceived the study project and designed the experiments. S.Y.L., Y.Q., J.L.L., X.Q.X., L.Q.Z. and W.D.L. performed the experiments. S.Y.L., Y.Q., L.Q.Z. and W.D.L. analyzed the data of samples. Y.Q., J.L.L., P.D.T. and X.Q.X. prepared the materials and instruments. Q.S.D., S.Y.L., Y.Q. and Q.Y.W. wrote the article.

Competing interests

The authors declare no competing interests.

Additional information

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