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# Hydrothermal synthesis of novel 1-aminoperylene diimide/TiO<sub>2</sub>/ MoS<sub>2</sub> composite with enhanced photocatalytic activity

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1-aminoperylene diimide/TiO<sub>2</sub>/MoS<sub>2</sub> composite (NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub>) with ordered structure was prepared by hydrothermal synthesis method. The composite was characterized by XRD, SEM, FTIR, XPS, BET, DRS, PL, EIS, Raman, photocurrent, and Mott-Schottky plots spectroscopy. The potential positions of the conduction and valence bands, and the band gap energy of the semiconductors were estimated. The composite exhibited higher photocatalytic activity compared with the monocomponent systems. The apparent rate constants (k) were determined as 0.00616, 0.00352, 0.00738, 0.00517, 0.00752, and 0.00806 min<sup>-1</sup> for TiO<sub>2</sub>, NH<sub>2</sub>-PDI, NH<sub>2</sub>-PDI/TiO<sub>2</sub>, MoS<sub>2</sub>, MoS<sub>2</sub>/TiO<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, respectively. The detection of radical scavengers confirmed that superoxide radicals, photogenerated holes, and photogenerated electrons were the main active substances for MB degradation. Between type II- heterojunction mechanism and Z-scheme mechanism, the latter could explain the enhanced photocatalytic activity of the composite better. The Z-scheme mechanism accumulates more electrons at CB level of NH<sub>2</sub>-PDI and hence generates more super oxide radicals.

Water pollution is an important issue of concern in many countries. Many dyes in textile printing or dyeing wastewater contain aromatic compounds, which are chemically stable and harmful to human health<sup>1</sup>. Because of the adverse environmental impacts of these refractory organic compounds, it is necessary to develop new methods to degrade them. Photocatalysis is a promising technique for photodegradation of hazardous chemicals in wastewater<sup>2</sup>. Titanium dioxide (TiO<sub>2</sub>) and TiO<sub>2</sub> based nanostructures are being used for photocatalytic degradation of organic pollutants to protect environment<sup>3</sup>. With the advantages of low cost, large specific surface area, high oxidizing power, and good chemical stability, TiO<sub>2</sub> has become one of the most promising candidates for photocatalysis<sup>4</sup>. However, the catalytic performance of TiO<sub>2</sub> is severely limited by the large band gap (3.2 eV), low photon utilization rate of solar energy (about 5%), and high recombination rate for photogenerated electron–hole pairs<sup>5,6</sup>. On the other hand, the industrial treatment of wastewater containing various organic pollutants using TiO<sub>2</sub>-photocatalyst is not common due to low efficiency of photodegradation<sup>7</sup>. To solve this problem, several methods have been used to improve the photocatalytic efficiency of TiO<sub>2</sub>. It has been reported that in a suitable heterostructured system, the presence of heterojunction changes the energy band positions and their inclination on the interface to accelerate the migration of photogenerated charge carriers, and eventually enhance the efficiency of photocatalyst<sup>8</sup>.

Perylenetetracarboxylic acid diimides (PDIs) are cheap organic dyes with high photothermal stability and strong absorption in the visible region<sup>9,10</sup>. As a typical n-type semiconductor, PDIs have high electron mobility and electron affinity [LUMO] due to its strong  $\pi$ - $\pi$  stacking between the conjugated  $\pi$  bonds<sup>11,12</sup>. PDIs materials have been used in photocatalysis because it can improve the light absorption and decrease the photo-generated electron-hole recombination<sup>13</sup>. PDIs have been introduced into TiO<sub>2</sub>, ZnO, and other photocatalysts to improve the photocatalytic performance<sup>14,15</sup>. However, such composite materials still have some defects such as quick recombination of photoexcited electron-hole pairs and photoetching, which lead to low efficiency of visible light driven photodegradation<sup>16</sup>. Therefore, it is necessary to further improve their photocatalytic activities and stabilities.

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# NH<sub>2</sub>-PDI



Molybdenum disulfide (MoS<sub>2</sub>) is a rapidly rising two-dimensional layered material, which has aroused enormous scientific interest in developing new MoS<sub>2</sub>-based materials for rich potential applications<sup>17</sup>. MoS<sub>2</sub> with different structures such as nanoribbon and nanosheet has been successfully synthesized<sup>18,19</sup>. It has unique physical properties that distinguish it from other materials and has been used in many applications such as lubricant, photovoltaic, and photocatalysis<sup>20,21</sup>. Hu et al. synthesized bulk, nano-slice, and nano-ball MoS<sub>2</sub> by precipitation method, and found that the photocatalytic efficiency of nano-slice MoS<sub>2</sub> towards methyl orange (90%) was higher than others<sup>22</sup>. James et al. synthesized MoS<sub>2</sub> nanoparticles through thermally decomposing method, and the material was used for photocatalytic degradation of methylene blue (MB) with an efficiency of 30%<sup>23</sup>. Xu et al. prepared flower-like MoS<sub>2</sub> nanopowders through hydrothermal method for photocatalytic degradation of Rhodamine B (RhB) and an efficiency of 15% was achieved<sup>24</sup>. In addition, the good conductivity of monolayer MoS<sub>2</sub> can efficiently separate the electron-hole pairs and enhance the photocatalytic activity. Hence, MoS<sub>2</sub> can be used in TiO<sub>2</sub> photocatalyst to improve its photocatalytic performance. For instance, Behzad Pourabbas et al. reported the photo-oxidation of phenol by MoS<sub>2</sub>/TiO<sub>2</sub> hybrids. Suddhasatwa Basu et al. demonstrated the ability to degrade RhB by MoS<sub>2</sub>/TiO<sub>2</sub> nanocomposite<sup>25</sup>. Giri et al. reported the photo-oxidation of RhB by MoS<sub>2</sub>/ TiO<sub>2</sub> hybrids<sup>26</sup>. Du et al. reported a MoS<sub>2</sub>/CdS/TiO<sub>2</sub> photocatalyst which has high photocatalytic activity and stability<sup>27</sup>. This technique produces powerful non-selective oxidants, such as superoxide radicals and hydroxyl radicals, which can degrade and mineralize a wide variety of pollutants<sup>28</sup>. The produced electrons-holes would recombine in pure  $TiO_2$ , which limited its application in the photodegradation. By preventing electron/hole recombination and semiconductors aggregation, multi-component support can improve the photoreactivity of the obtained composite<sup>29</sup>.

In this work, a novel ternary photocatalyst:  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> was synthesized by hydrothermal method (Scheme 1). The CB/VB (conduction band/valence band) of TiO<sub>2</sub>,  $NH_2$ -PDI, and MoS<sub>2</sub> are evaluated to be - 1.26/1.72 eV, - 1.56/0.8 eV, and -0.75/1.00 eV, respectively. These matched potentials are suitable for rapid charge separation between these semiconductors. The composite showed enhanced activity in the photodegradation of Methylene Blue (MB) and amoxicillin (AMX).

#### Experimental

Materials and methods. All other chemicals were analytically pure, purchased from commercial sources and used without further purification. Distilled water was used throughout the experiment. FT-IR spectrum has been studied by a Bruker Tensor-27 spectrophotometer. The <sup>1</sup>H NMR and mass spectrum of NH<sub>2</sub>-PDI were measured on a Bruker Advance 400 spectrometer and a Bruker Maxis UHR-TOF mass spectrometer, respectively. Scanning electron microscopy (SEM) images were obtained on a Sigma, Zeiss microscope equipped with an energy dispersive X-ray (EDX) spectrometer. The specific surface area of the catalyst has been analyzed using nitrogen adsorption at 77 K applying the Brunauer-Emmett-Teller (BET) method using a micrometrics ASAP 2020 V3.00 H. X-ray diffraction (XRD) measurements have been performed using a Rigaku R-AXIS RAPID X-ray diffractometer. Crystallinity and phase composition of the as-synthesized composite have been confirmed from the Lab Ram HR800, Jobin Yvon micro-Raman measurement. The ultraviolet-visible diffuse reflectance absorption spectra (DRS) have been measured on a Shimadzu model UV-Vis diffuse reflectance spectrometer. X-ray photoelectron spectroscopy (XPS) has been performed on a KRATOS model XSAM800 instrument. Fluorescence measurements were performed at a Hitachi FL-4500 spectrometer with 290 nm excitation wavelength at room temperature. The Mott-Schottky plots, photocurrent, and electrochemical impedance spectroscopy (EIS) were measured on a CHI760E electrochemical workstation using a three-electrode system. Platinum wire as counter electrode, ITO deposited by photocatalyst was used as working electrodes (50 mg of photocatalyst was dispersed in 2 mL of ethanol and grounded for 20 min, then it was uniformly dropped onto the FTO glass and stay still overnight to dry off the ethanol.), saturated calomel as reference electrode, and  $0.1 \text{ M } \text{Na}_2\text{SO}_4$ aqueous solution was used as electrolyte. The EIS was performed at an open circuit potential at a frequency of 0-10,000 Hz, and the photoelectric response of the sample was measured at 0.0 V. Photoelectrochemical properties were measured with a 300 W xenon lamp as the light source.

**Synthesis and characterization of NH<sub>2</sub>-PDI.** The compound  $NH_2$ -PDI was synthesized according to literature method<sup>30</sup>. The synthetic route along with the characterization data of  $NH_2$ -PDI (Fig. S-1) are reported in the Supplementary Methods.

**Preparation of TiO<sub>2</sub> and NH<sub>2</sub>-PDI/TiO<sub>2</sub>.** The NH<sub>2</sub>-PDI/TiO<sub>2</sub> composite was prepared by the hydrothermal synthesis method. First, tetrabutyl titanate (10.0 mL) was dissolved in anhydrous ethanol (20.0 mL). Then, 5.0 mL of distilled water was slowly added under vigorous stirring. Later, NH<sub>2</sub>-PDI (0.01 g) was dissolved in dichloromethane (5.0 mL) and added to the solution under sonication. The resulting mixture was stirred for 12 h and then transfer to the hydrothermal kettle for 3 h at 200 °C. The precipitate was filtered, washed thoroughly with distilled water, and dried in an air oven at 100 °C for 4 h. This catalyst contained 0.1 wt% of NH<sub>2</sub>-PDI. Pure TiO<sub>2</sub> was prepared by the same method.

**Preparation of MoS<sub>2</sub>.** MoS<sub>2</sub> was synthesized by hydrothermal route using Na<sub>2</sub>MoO<sub>4</sub>·2H<sub>2</sub>O and NH<sub>2</sub>CSNH<sub>2</sub> as Mo and S sources, respectively. Briefly, Na<sub>2</sub>MoO<sub>4</sub>·2H<sub>2</sub>O (2.0 g) and NH<sub>2</sub>CSNH<sub>2</sub> (8.0 g) was dissolved in 60 mL of water under vigorous stirring for 30 min. The mixture was kept in a hydrothermal kettle at 200 °C for 24 h. The resulting black precipitate was washed several times with ethanol and water to further remove impurities and contaminants followed by a centrifugation and drying process at 60 °C for 6 h to obtain MoS<sub>2</sub> nanoarchitectures.

**Preparation of MoS<sub>2</sub>/TiO<sub>2</sub> and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub>.** The NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composite was fabricated by loading MoS<sub>2</sub> onto the preformed NH<sub>2</sub>-PDI/TiO<sub>2</sub> solid solutions. Hydrothermal synthesis of NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composite was prepared as follows: 3.0 g NH<sub>2</sub>-PDI/TiO<sub>2</sub> and 0.06 g of MoS<sub>2</sub> were dispersed in 5 mL water. The resulting mixture was stirred for 24 h and then transferred to a hydrothermal kettle at 200 °C for 4 h. The prepared composite material was filtered, washed thoroughly with distilled water, and dried in an air oven at 100 °C for 4 h. NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> was formed as a gray powder. This catalyst contained 0.1 wt% NH<sub>2</sub>-PDI and 2.0 wt% MoS<sub>2</sub>. MoS<sub>2</sub>/TiO<sub>2</sub>, NH<sub>2</sub>-PDI/TiO<sub>2</sub>/1%MoS<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/3%MoS<sub>2</sub> were prepared using the same procedure.

**Photocatalytic activity tests.** The photocatalytic experiments were carried out in a photochemical reactor (PhchemIII, Beijing China NBeT) consisting of a 500 W xenon lamp (XE-JY500). The reaction chamber is equipped with 50 mL capacity reaction glass tubes and a magnetic stirrer. The specially designed reflector was made of highly polished aluminum and a built-in cooling fan. The light exposure length is 230 mm.

Methylene Blue (MB) and amoxicillin (AMX) were selected to test the photodegradation activity of the prepared photocatalysts. First, 50 mg of synthesized sample was suspended in 50 ml of 10 mg/L MB aqueous solution. The visible light source was obtained by a 500 W xenon lamp. Before irradiation, the suspension solutions were stirred magnetically for 120 min in the dark to achieve adsorption–desorption equilibrium. The samples were withdrawn at given time intervals and the photocatalyst was removed by centrifugation. A UV–vis spectrophotometer at 664 nm and 198 nm was used to measure the absorbance of MB and AMX solutions, respectivly.

#### **Results and discussion**

Figure 1 shows SEM images of  $TiO_2$ ,  $NH_2$ -PDI,  $NH_2$ -PDI/ $TiO_2$ ,  $MoS_2$ ,  $MoS_2/TiO_2$ , and  $NH_2$ -PDI/ $TiO_2/MoS_2$ . Figure 1(a) shows that  $TiO_2$  forms uniform elongated spherical particles with a length of about 300 nm and a width of about 200 nm. As can be seen from Fig. 1(b),  $NH_2$ -PDI forms nanorods structure. The average width is 1 µm, and the length is in the range of 1–3 µm. Figure 1(c) depicts that  $NH_2$ -PDI/ $TiO_2$  particles formed an agglomerated spherical structure. As shown in Fig. 1d, the average diameter of  $MoS_2$  nanoflowers is 0.5–1 µm. These nanoflowers have massive petals on the surfaces, which are free to gather closely. Figure 1(e) and (f) shows the morphology of the synthesized  $MoS_2/TiO_2$  heterostructure characterized by SEM. It can be seen that  $MoS_2$  and  $TiO_2$  mixed together. In Fig. 1(h) and (i), a peculiar morphology of  $NH_2$ -PDI/ $TiO_2/MoS_2$  is shown. The  $NH_2$ -PDI and  $MoS_2$  mixed with  $TiO_2$ , and they can provide more active sites and mass charge transport pathways in the catalytic system. The color of  $MoS_2/TiO_2$  and  $NH_2$ -PDI/ $TiO_2/MoS_2$  are shown in Fig. 1(g) and (j), respectively.  $MoS_2/TiO_2$  was light grey while  $NH_2$ -PDI/ $TiO_2/MoS_2$  sample showed a red grey color.

The composition of the NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> was further investigated by an EDX attached to SEM (Fig. 2). Figure 2(b)-(h) shows the element mappings in a selected area of the composite (Fig. 2a). The homogeneous distributions of the elements Mo, S, C, N, Ti, and O can be clearly seen from the graphs. As shown in Fig. 2(i) and Fig. S-2, EDX analysis reveals that the composite contains Ti and O (in the case of TiO<sub>2</sub>), Mo and S (in the case of MoS<sub>2</sub>), or C and N (in the case of NH<sub>2</sub>-PDI). The atomic ratio of Ti, Mo and C equals to 25.2: 1: 1.3, meaning that the molar ratio of TiO<sub>2</sub>, MoS<sub>2</sub>, and NH<sub>2</sub>-PDI is about 95.3: 1.89: 0.68. The high proportion of C is due to the elemental carbon and carbonate species adsorbed on the TiO<sub>2</sub> surface.

XRD measurements were performed to determine the crystalline structures of TiO<sub>2</sub>, NH<sub>2</sub>-PDI, NH<sub>2</sub>-PDI/ TiO<sub>2</sub>, MoS<sub>2</sub>, MoS<sub>2</sub>/TiO<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composites (Fig. 3). The crystal phase of TiO<sub>2</sub> was consistent with anatase phase TiO<sub>2</sub> (JCPDS NO. 71–1167) (Fig. 3a). Diffraction peaks appeared at  $2\theta = 25.3^{\circ}$ , 37.7°, 48.0°, 53.8°, 55.0°, 62.6°, 58.6°, 70.6°, and 75.2°, corresponding to (101), (311), (200), (105), (211), (204), (116), (220), and (241) planes of TiO<sub>2</sub>, respectively<sup>31</sup>. The NH<sub>2</sub>-PDI appeared in the corresponding d-spacing as 10.35 Å, 9.22 Å, 5.24 Å, and 3.43 Å (Fig. 3b). The peak at  $2\theta = 25.74^{\circ}$  (*d* spacing 3.43 Å) can be attributed to the  $\pi$ - $\pi$  stacking of the adjacent NH<sub>2</sub>-PDI, since the distance of  $\pi$ - $\pi$  stacking between the perylene nuclei was about 3.5 Å<sup>32</sup>. It was well known that PDIs exhibit a typical sharp diffraction peak at about 8–9°, which was centered at 9.57° (*d* spacing 9.22 Å)<sup>33</sup>. Additionally, X-ray powder diffraction measurements on NH<sub>2</sub>-PDI showed that the first diffraction peak appeared at  $2\theta$ = 8.53° (*d* spacing 10.35 Å) and the second-order diffraction peak was assigned as (002)  $\alpha$ -form crystal diffraction, which has been reported by Miyata<sup>34</sup>. The multiple orders of reflection indicated that the self-assembled structures of NH<sub>2</sub>-PDI are well-ordered and layered microstructures. For the pure 3D flower-like spherical MoS<sub>2</sub> nanostructure, the peaks correspond to (002), (100), (103), and (110) diffraction planes of 2H-MoS<sub>2</sub> (JCPDS: 37-1492), which is consistent with previous studies (Fig. 3d)<sup>35</sup>. For MoS<sub>2</sub>/TiO<sub>2</sub>, the



**Figure 1.** SEM images of  $TiO_2$  (**a**),  $NH_2$ -PDI (**b**),  $NH_2$ -PDI/ $TiO_2$  (**c**),  $MoS_2$  (**d**),  $MoS_2/TiO_2$  (**e**,**f**),  $NH_2$ -PDI/ $TiO_2/MoS_2$  (**h**,**i**), and the appearance of  $MoS_2/TiO_2$  (**g**),  $NH_2$ -PDI/ $TiO_2/MoS_2$  (**j**).

addition of MoS<sub>2</sub> did not change the diffraction peak positions of TiO<sub>2</sub> obviously, indicating that MoS<sub>2</sub> was not incorporated into the TiO<sub>2</sub> lattice. Obviously, both NH<sub>2</sub>-PDI/TiO<sub>2</sub> and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> formed ordered structures. Comparison of the XRD patterns in Fig. 3a,c,e,f confirms that the XRD patterns of the NH<sub>2</sub>-PDI/TiO<sub>2</sub>, MoS<sub>2</sub>/TiO<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> nanoparticles have good consistency with the XRD data of anatase phase TiO<sub>2</sub> (JCPDS NO. 71–1167) at 20 degrees of 25.3°, 37.7°, 48.0°, 53.8°, 55.0°, 62.6°, 58.6°, 70.6°, and 75.2°<sup>36</sup>. Compare with pure TiO<sub>2</sub>, the crystallization properties of the two composites were slightly weakened, which should be due to the binding of NH<sub>2</sub>-PDI (Fig. 3c,f). Since the NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composite was loaded with very little NH<sub>2</sub>-PDI and MoS<sub>2</sub>, the characteristic peaks of NH<sub>2</sub>-PDI and MoS<sub>2</sub> did not appear in its XRD patterns (Fig. 3f). Additionally, compared with pure TiO<sub>2</sub>, NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> exhibited a widened peak width. This





may be caused by the formation of heterojunction between  $TiO_2$ ,  $NH_2$ -PDI and  $MoS_2$ , and the heterojunction lead to lattice distortion of  $TiO_2$ . The average particle sizes of  $TiO_2$ ,  $NH_2$ -PDI/ $TiO_2$ ,  $MOS_2/TiO_2$ , and  $NH_2$ -PDI/ $TiO_2/MOS_2$  calculated by Scherrer's equation were 10.2 nm, 12.5 nm, 11.3 nm, and 13.6 nm, respectively<sup>37</sup>.

Raman spectroscopy was applied to further check the phase and formation of the  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> nanocomposite. As shown in Fig. S-3, the pristine TiO<sub>2</sub> nanoparticles exhibit five peaks at 145.1 ± 0.2, 196.6 ± 0.1, 396.1 ± 0.2, 513.7 ± 0.2, and 638.1 ± 0.1 cm<sup>-1</sup>, which belong to the  $E_g$ ,  $E_g$ ,  $B_{1g}$ ,  $A_{1g}$ , and  $E_g$  anatase tetragonal vibration modes of TiO<sub>2</sub>, respectively<sup>38</sup>. The  $E_g$ ,  $B_{1g}$ , and  $A_{1g}$  peaks correspond to symmetric stretching, symmetric bending vibrations of O-Ti–O, respectively. The results demonstrate the existence of typical anatase TiO<sub>2</sub> phase, and it is consistent with the XRD results. Characteristic peaks of MoS<sub>2</sub> ( $A_{1g}$  and  $E^{1}_{2g}$  modes) were observed in the Raman spectra of NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> photocatalyst, indicating that MoS<sub>2</sub> was loaded on the surface of TiO<sub>2</sub><sup>39</sup>. Compared with pure TiO<sub>2</sub>, NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> photocatalyst exhibited red shifts of  $E_g$ ,  $E_g$ ,  $A_{1g}$ , and  $E_g$  modes (145.6 ± 0.3, 197.8 ± 0.1, 514.1 ± 0.1, and 638.4 ± 0.1 cm<sup>-1</sup>). This indicates that there are strong intimate interactions between TiO<sub>2</sub>, MoS<sub>2</sub>, and NH<sub>2</sub>-PDI.

FT-IR measurements revealed the connection between TiO<sub>2</sub>, NH<sub>2</sub>-PDI and MoS<sub>2</sub> in NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> nanocomposite (Fig. S-4). As shown in Fig. S-4a, for pure TiO<sub>2</sub>, the vibration modes observed at 617 cm<sup>-1</sup> and 749 cm<sup>-1</sup> are due to the O–Ti–O bending and Ti–O stretching vibrations, respectively. For MoS<sub>2</sub>, the vibration mode at 1389 cm<sup>-1</sup> is assigned to Mo–S vibration, while the vibration mode at 1550 cm<sup>-1</sup> may be S–O asymmetric stretching (Fig. S-4d)<sup>40</sup>. In MoS<sub>2</sub>/TiO<sub>2</sub> and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, all characteristic vibration modes were present but with very low Mo–S and S–O vibration modes, which indicates that the MoS<sub>2</sub>/TiO<sub>2</sub> and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> contain very little amount of MoS<sub>2</sub> (Fig. S-4e, f). Remarkably, the bending vibration of C-H at 2926–2817 cm<sup>-1</sup> observed in NH<sub>2</sub>-PDI shifted to 2968–2875 cm<sup>-1</sup> in NH<sub>2</sub>-PDI/TiO<sub>2</sub> and to 2992–2889 cm<sup>-1</sup> in NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, which may be an indicative of the binding between NH<sub>2</sub>-PDI and TiO<sub>2</sub> or MoS<sub>2</sub>. In addition, the stretching vibration of C = C in NH<sub>2</sub>-PDI at 1645 cm<sup>-1</sup> shifted to 1639 cm<sup>-1</sup> in NH<sub>2</sub>-PDI/TiO<sub>2</sub> and to 1627 cm<sup>-1</sup> in NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> was blue shifted, indicating that the π-π interaction within NH<sub>2</sub>-PDI decreased. The absorption peak around 3300 cm<sup>-1</sup> was caused by the stretching vibration of O–H bond, which was related to the atmospheric



**Figure 3.** XRD patterns of  $TiO_2$  (**a**), NH<sub>2</sub>-PDI (**b**), NH<sub>2</sub>-PDI/TiO<sub>2</sub> (**c**), MoS<sub>2</sub> (**d**), MoS<sub>2</sub>/TiO<sub>2</sub> (**e**), NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> (**f**).



Figure 4. Graphical representation of the investigated adsorption mode of  $NH_2$ -PDI onto the  $TiO_2$  and  $MoS_2$  surface.

water adsorbed on the catalyst surface. As shown in Fig. 4, the Ti…O bond and S…H bond can be formed when  $NH_2$ -PDI meets with TiO<sub>2</sub> and MoS<sub>2</sub>. After the addition of deionized water,  $NH_2$ -PDI self-assembled through intermolecular  $\pi$ - $\pi$  stacking interactions. For  $NH_2$ -PDI, dichloromethane is a "good" solvent, while water is a "poor" solvent. When the "poor" solvent was added, the solubility of originally dissolved  $NH_2$ -PDI was limited, and  $NH_2$ -PDI self-assembled through non-covalent interactions and precipitated into solid.

The chemical composition and valence state of  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> sample were studied by XPS (Fig. 5). According to Fig. 5a,  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composite contains Ti, O, C, Mo, and S elements. For  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> nanocomposite, the positions of Ti2p1/2 and Ti2p3/2 were observed at 464.9 eV and 459.1 eV, respectively. While for pure TiO<sub>2</sub> nanoparticles, Ti2p1/2 and Ti2p3/2 peaks were observed at 464.3 eV and 458.5 eV, respectively (Fig. 5b)<sup>41</sup>. The O1s peak positions of  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> and pure TiO<sub>2</sub> were observed at 530.2 eV and 529.8 eV, respectively (Fig. 5c). These results indicate that after adding  $NH_2$ -PDI and  $MoS_2$  to TiO<sub>2</sub> nanoparticles, Ti2p peaks move to a higher energy by 0.6 eV than pure TiO<sub>2</sub>. Also, the peak position of O1s moved 0.4 eV towards high energy. The shifted spectrum implies the presence of more defects or adsorbed hydroxyl groups on the surface of  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub><sup>42</sup>. These defect states may serve as shallow donors to enhance charge transfer at the multiple interfaces and thus improve the overall degradation efficiency towards dyes. In  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> sample, The C1s spectra (Fig. 5d) shows the corresponding peak at 284.8 eV, which can be identified as C-C/C=C/C-H functional groups. The positions of the Mo3d5/2 and Mo3d3/2 peaks were observed at 232.4 eV and 228.5 eV, respectively. Similarly, the position of S2p was observed at 161.9 eV<sup>43</sup>.

The surface areas of TiO<sub>2</sub>, NH<sub>2</sub>-PDI, NH<sub>2</sub>-PDI/TiO<sub>2</sub>, MoS<sub>2</sub>, MoS<sub>2</sub>/TiO<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> were determined by nitrogen adsorption method. The BET surface area of NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> (39 m<sup>2</sup>/g) is higher than that of TiO<sub>2</sub> (21 m<sup>2</sup>/g), NH<sub>2</sub>-PDI/TiO<sub>2</sub> (37 m<sup>2</sup>/g), MoS<sub>2</sub> (23 m<sup>2</sup>/g), or MoS<sub>2</sub>/TiO<sub>2</sub> (26 m<sup>2</sup>/g). Therefore, it can be speculated that the large surface area of NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> could promote the photocatalytic activity.



**Figure 5.** XPS spectra of (**a**) survey spectra in  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, (**b**) Ti2p in pure TiO<sub>2</sub> and  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, (**c**) O1s in pure TiO<sub>2</sub> and  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, (**d**) C1s in  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, (**e**) Mo3d in  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, and (**f**) S2p in  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub>.

The UV–vis diffuse reflection spectra (DRS) and the corresponding Tauc plots of samples are shown in Fig. 6. Pure TiO<sub>2</sub> absorbs only UV light and exhibits absorption below 380 nm. Adding NH<sub>2</sub>-PDI and MoS<sub>2</sub> nanoflakes to TiO<sub>2</sub> nanoparticles can result in more light-harvesting in visible region (Fig. 6a). The absorption intensity of NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> in visible light region was obviously enhanced, which could be attributed to the visible light absorption characteristics of NH<sub>2</sub>-PDI and MoS<sub>2</sub> nanoflakes. The band gap energy of the samples can be estimated by the formula Eg (eV) = 1240/ $\lambda_{AE}$  (nm) using the position of the absorption edge ( $\lambda_{AE}$ )<sup>44</sup>. The downward slopes of the absorption curves are extrapolated to cross the X-axis and the  $\lambda_{AE}$ -values of TiO<sub>2</sub>, MoS<sub>2</sub>/TiO<sub>2</sub>, NH<sub>2</sub>-PDI/TiO<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> samples were estimated to be at 389 nm, 423 nm, 459 nm, and 467 nm, respectively (Fig. S-5). The Eg values of TiO<sub>2</sub>, MoS<sub>2</sub>/TiO<sub>2</sub>, NH<sub>2</sub>-PDI/TiO<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> samples were estimated to be at 3.18, 2.93, 2.70, and 2.66 eV, respectively.

The band gaps (Eg, eV) of the samples are also calculated by the Tauc equation and Kubelka-Munk function<sup>45</sup>:

$$[F(R)h\upsilon]^0.5 = A(h\upsilon - Eg)$$
<sup>(1)</sup>

$$F(R) = (1 - R)^2 / 2R$$
(2)

where, R is the calibrated reflection of samples with  $BaSO_4$  reflection, and F(R) is proportional to the absorption constant. h and v represent the Planck constant and frequency, while A and Eg represent constant and band gap



Figure 6. (a) UV-vis diffuse reflection spectra (DRS) and (b) The corresponding Tauc plot of samples.

energy, respectively. The  $[F(R)hv]^{0.5}$  versus hv is shown in Fig. 6b. Extrapolation of the linear portion at  $[F(R)hv]^{0.5} = 0$  provides Eg value of the samples. The band gaps were found to be 2.98, 2.75, 2.47, and 2.27 eV for TiO<sub>2</sub>,  $MoS_2/TiO_2$ ,  $NH_2$ -PDI/TiO<sub>2</sub>, and  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, respectively. The increase of UV-visible light absorption and decrease of band gap energy enhanced the photodegradation efficiency of  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> towards MB.

The activity diagram of MB degradation by different catalysts in visible light and the first order kinetics curve fitting of MB degradation by different catalysts are shown in Fig. 7. Prior to the illumination, each catalyst was dispersed in the dye solution. After 120 min of vigorous magnetic stirring under dark condition,  $51.3 \pm 1.5\%$  of MB was absorbed by NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, whereas pure TiO<sub>2</sub> only absorbed  $8.2 \pm 0.7\%$  of MB. It can be seen that the spongy NH<sub>2</sub>-PDI clusters and multi-layer MoS<sub>2</sub> nanoflowers decorated TiO<sub>2</sub> exhibited extremely high adsorption efficiency under dark condition. Both MoS<sub>2</sub> and TiO<sub>2</sub>/MoS<sub>2</sub> have strong adsorption capacity, which may be due to the flower-like MoS<sub>2</sub> with negative surface charge promoted the adsorption of cationic dye MB. It can be noted that after 120 min of irradiation, the degradation rate of dye was  $99.0 \pm 1.0\%$  for NH<sub>2</sub>-PDI/TiO<sub>2</sub>/ MoS<sub>2</sub>, while the corresponding data was  $4.0 \pm 0.3\%$ ,  $31.3 \pm 1.5\%$ ,  $71.7 \pm 1.6\%$ ,  $78.7 \pm 1.3\%$ , and  $87.6 \pm 1.5\%$  for NH<sub>2</sub>-PDI, TiO<sub>2</sub>, NH<sub>2</sub>-PDI/TiO<sub>2</sub>, MOS<sub>2</sub>, and MOS<sub>2</sub>/TiO<sub>2</sub>, respectively (Fig. 7A). Though TiO<sub>2</sub> has no visible light absorption, it has been found to be solar active. This may be due to the dye sensitization mechanism of MB to TiO<sub>2</sub>. For better comparison, NH<sub>2</sub>-PDI/TiO<sub>2</sub>/1%MoS<sub>2</sub> or NH<sub>2</sub>-PDI/TiO<sub>2</sub>/3%MoS<sub>2</sub> photocatalysts was used for dye degradation under identical conditions. As a result,  $97.3 \pm 0.6\%$  or  $94.0 \pm 1.1\%$  of dye were degraded respectively under visible light irradiation in 120 min. The results show that 0.1%NH<sub>2</sub>-PDI/TiO<sub>2</sub>/2%MoS<sub>2</sub> had higher efficiency in MB degradation than other catalysts. The kinetics plot shows that  $\ln (C_0/C)$  has linear relationships with reaction time, indicating that the photodegradation of MB follows first-order kinetics (Fig. 7B). The apparent rate constants (k) were determined as 0.00616, 0.00352, 0.00738, 0.00517, 0.00752, and 0.00806 min<sup>-1</sup> for TiO<sub>2</sub>, NH<sub>2</sub>-PDI, NH<sub>2</sub>-PDI/TiO<sub>2</sub>, MoS<sub>2</sub>, MoS<sub>2</sub>/TiO<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub>, respectively. These results confirmed the high photocatalytic activity of NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub>. Fig. S-6 shows the decolorization of MB dye (0.01 g/L) within 90 min in the presence of NH2-PDI/TiO2/MoS2 catalyst. The decrease of absorption spectrum indicates decolorization of the dye under the applied conditions. There were no additional peaks in the UV-Vis spectrum, indicating that the dye was completely degraded. It is reported that the products of the decolorization process are  $H_2O$ ,  $CO_2$ ,  $NO_2$ , and  $SO_2^{46}$ , and the mechanism of MB degradation is shown in Fig. S-7.

Chemical oxygen demand (COD) is an indicator that reflecting the required oxygen for the oxidation of organic matters present in solution<sup>47</sup>. A smaller COD value means a lower pollution of the sample. The



**Figure 7.** (A) The activity diagram of MB degradation by different catalysts in visible light, (B) The first order kinetics curve fitting of MB degradation by different catalysts. (a)  $TiO_2$ , (b)  $NH_2$ -PDI, (c)  $NH_2$ -PDI/ $TiO_2$ , (d)  $MoS_2$ , (e)  $MoS_2/TiO_2$ , (f)  $NH_2$ -PDI/ $TiO_2/2\%MoS_2$ , (g)  $NH_2$ -PDI/ $TiO_2/1\%MoS_2$ , and (h)  $NH_2$ -PDI/ $TiO_2/3\%MoS_2$ . [MB] = 0.01 g/L, pH = 7, catalyst suspended = 1 g/L.

mineralization degree of MB was calculated by COD result, and the kinetic results shown that the k value of  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> was 0.0076 min<sup>-1</sup> (Fig. S-8). The value correspond to  $t_{1/2}$  values of 85.9 and 91.2 min (based on  $t_{1/2} = 0.693/k$ ) for the degradation and mineralization of MB molecules, respectively<sup>48</sup>. These confirmed that the photodegradation of MB was 1.06 times faster than its mineralization. The results confirmed a relatively smaller mineralization extent for MB molecules than its photodegradation extent.

The photonic efficiency of MB dye degradation by  $N\dot{H}_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> under optimum condition was calculated using a reported method<sup>49</sup>. The quantum yield of reaction (Ø) is defined as follows (Eq. (1)):

 $\emptyset = \frac{\text{Number of molecules decomposed}}{\text{Number of photons of light absorbed}}$ 

The photo-degradation rate constants ( $k^2$ ) of MB dyes under monochromatic light source can also be used to calculate the reaction quantum yield (Eq. (2)).

$$\mathcal{D} = \frac{k'}{303I\varepsilon l}$$

where Ø is the reaction of quantum yield. *I* is the light intensity range in 400–800 nm ( $2.312 \times 10^{-3}$  Einstein).  $\varepsilon$  is the molar absorptivity of MB at 664 nm ( $25,440 \text{ cm}^{-1} \text{ M}^{-1}$ ). *l* is the path length of reaction tube, which is 0.23 m for 50 mL solution. The degradation quantum yields obtained with TiO<sub>2</sub>, NH<sub>2</sub>-PDI, NH<sub>2</sub>-PDI/TiO<sub>2</sub>, MoS<sub>2</sub>, MoS<sub>2</sub>/TiO<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> were 0.000152, 0.000869, 0.000182, 0.000128 and 0.000186, respectively. These results indicate that the quantum yield of NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> catalyzes process was higher than that of other processes.

During the process of MB degradation, a large amount of MB was adsorbed on the surface of catalysts. The visible light driven catalysis of the  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> may be enhanced due to the effect of dye sensitization. Therefore, amoxicillin (AMX, a colorless medical antibiotic) was selected to evaluate the photocatalytic activity of  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> nanoparticles under the same conditions. The photocatalytic degradation results of AMX are shown in Fig. S-9.  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> exhibited higher visible photocatalytic activity than  $NH_2$ -PDI/TiO<sub>2</sub> or  $MoS_2/TiO_2$ . When AMX solution was mixed with  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> in the absence of light, the removal rate of AMX was 48%. When  $NH_2$ -PDI/TiO<sub>2</sub> or  $MoS_2/TiO_2$  was used under visible light irradiation for 120 min, the degradation rates of AMX were 56.9% or 71.2%, respectively. Comparatively,  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> exhibited superior photocatalytic activity of  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> reached a maximum value of 0.00986 min<sup>-1</sup>, which was about 1.67 times and 1.79 times higher than  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> has strong visible-light driven photocatalytic activity.

The reusability and stability of a composite photocatalyst are important factor affecting its application (Fig. S-10). The NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> was used for the degradation of different solutions of MB for six consecutive experiments under the same reaction conditions. After each experiment, the catalyst was recovered by centrifugation and dried in an air oven. The degradation efficiency of NH2-PDI/TiO2/MoS2 reduced from 99±0.9% to  $89 \pm 2.3\%$  after six cycles of reuse, which was probably due to a small loss of catalyst during recycle (Fig. S-10a). NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> was prepared by hydrothermal synthesis method, which made the contact between three monomers more compacted and reduced the dissolution of MoS<sub>2</sub> or NH<sub>2</sub>-PDI molecule. Therefore, the composite catalyst exhibits high stability in the degradation process. In addition, the composite material packaged TiO<sub>2</sub> by NH<sub>2</sub>-PDI and MoS<sub>2</sub>. The contact area between the three semiconductors increases due to the point-tosurface contact. Due to the close contact interface, the interactions between the three semiconductors would also be more intense. The  $\pi$ -conjugated electrons in NH<sub>2</sub>-PDI can be quickly transferred from the interior of the semiconductor to the surface, and then to the surface of TiO<sub>2</sub> or MoS<sub>2</sub>. This reduces electrons and holes recombination and improves the photocatalytic activity. Fig. S-10b shows the FT-IR spectra of the NH<sub>2</sub>-PDI/ TiO<sub>2</sub>/MoS<sub>2</sub> photocatalyst before and after six cycles of photocatalytic degradation of MB. It can be seen that the spectrum of the regenerated photocatalyst was basically the same as that of the fresh photocatalyst. There is no spectrum of MB, which proves that MB has been completely degraded rather than adsorbed on the catalyst surface during the illumination process. The result of UV-vis spectra shows that the concentration of organic matter in the washing solution of catalyst was very low. Few intermediate species and NH<sub>2</sub>-PDI were detected, indicating that the dissolution of NH<sub>2</sub>-PDI in the composite structure was very few and MB was mineralized to CO<sub>2</sub>. These results show that the NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> catalyst is stable and reusable.

The photoluminescence (PL) spectrum was used to analyze the recombination of electron hole pairs. In general, a higher peak indicates a more rapid charge recombination. As shown in Fig. S-11, the peak at 399 nm is attributed to a direct transition from the conduction band to the valence band. Moreover, the luminescence peaks at 440, 451, and 469 nm are caused by inter-band transitions, and the peaks at 483 and 492 nm are caused by intra-band transitions within the energy level traps or surface states<sup>50</sup>. The peak intensity was arranged as follow:  $TiO_2 > NH_2$ -PDI/ $TiO_2 > NM_2$ -PDI/ $TiO_2 > NH_2$ -PDI/ $TiO_2 > NH_2$ -PDI/ $TiO_2 > NH_2$ -PDI/ $TiO_2 = NH_2$ -PDI and  $MOS_2 = NH_2$ -PDI acted as short and fast channels for migrating the photogenerated charge carriers from NH\_2-PDI to  $TiO_2$  and  $MOS_2$ . Based on the above results, the introduction of  $NH_2$ -PDI and  $MOS_2$  can greatly accelerate the charge transfer process.

The mechanism of enhanced photocatalytic activity of  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composite was further studied by photoelectrochemistry method. The separation of photogenerated electrons and holes plays an important role in the photocatalytic decomposition of organic pollutants, which can be evaluated by transient photocurrent responses (*I-t*) and electrochemical impedance spectra (*EIS*) (Fig. 8). Figure 8a shows the *I-t* curves of TiO<sub>2</sub>,  $NH_2$ -PDI,  $NH_2$ -PDI/TiO<sub>2</sub>,  $MOS_2$ ,  $MOS_2$ /TiO<sub>2</sub>, and  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> under several intermittent visible light irradiation cycles. It can be seen that the photocurrent of all samples exhibited good repeatability when the light was turned on and off. The light current response of pure TiO<sub>2</sub> and  $NH_2$ -PDI were very weak (only 0.119 and 0.035  $\mu$ A cm<sup>-2</sup>, respectively).  $NH_2$ -PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composite had the largest photocurrent response, and the stable photocurrent was about 1.007  $\mu$ A cm<sup>-2</sup>, which was higher than that of MoS<sub>2</sub> (0.372  $\mu$ A cm<sup>-2</sup>),  $MOS_2/TiO_2$ (0.522  $\mu$ A cm<sup>-2</sup>), and  $NH_2$ -PDI/TiO<sub>2</sub> (0.846  $\mu$ A cm<sup>-2</sup>). The enhanced photocurrent response indicates that the separation efficiency of photogenerated carriers and the photocatalytic performance were improved. Figure 8b displays the EIS Nyquist plots of as-prepared samples. The radius of the arc on the EIS spectrum reflects the



**Figure 8.** (a) Photocurrent transient responses and (b) EIS spectra of  $TiO_2$ ,  $NH_2$ -PDI,  $NH_2$ -PDI/ $TiO_2$ ,  $MoS_2$ ,  $MoS_2/TiO_2$ , and  $NH_2$ -PDI/ $TiO_2/MoS_2$  at a constant potential of 0.4 V with simulated solar light at 300 mW/cm<sup>2</sup>. frequency: 0–10,000 Hz, electrolyte: 0.1 M Na<sub>2</sub>SO<sub>4</sub>, amplitude: 10 mV.



**Figure 9.** Schematic illustration of the photocatalytic mechanism over the NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> heterostructure. (**a**) Type II-Heterojunction and (**b**) Z-scheme pathways.

surface charge transfer resistance and the solid interface delamination resistance. The smaller the semicircle arc is, the easier the charge transfer proceeds<sup>51</sup>. Well discussion on EIS spectra and the resulted Bodes' plots has been presented in previous work<sup>52</sup>. Here, the following trend was obtained for the arc radius of the Nyquist plots, confirming that the NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composite have better charge transfer capability.

 $NH_2 - PDI > TiO_2 > MoS_2 > NH_2 - PDI/TiO_2 > MoS_2/TiO_2 > NH_2 - PDI/TiO_2/MoS_2.$ 

The Mott-Schottky plots were used to estimate the conduction band (CB) of the TiO<sub>2</sub>, NH<sub>2</sub>-PDI, and MoS<sub>2</sub>, which were -1.26 eV, -1.61 eV, and -0.75 eV, respectively (Fig. S-12). Combining with their bandgaps (TiO<sub>2</sub> at 2.98 eV [Fig. 5b], NH<sub>2</sub>-PDI at 2.36 eV<sup>30</sup>, and MoS<sub>2</sub> at 1.75 eV<sup>26</sup>), the CB/VB (conduction band/valence band) of TiO<sub>2</sub>, NH<sub>2</sub>-PDI, and MoS<sub>2</sub> were evaluated to be -1.26/1.72 eV, -1.56/0.8 eV, and -0.75/1.00 eV, respectively. It can be found that the CB position of singlet oxygen ( $\cdot$ O<sub>2</sub><sup>-1</sup>) (-0.18 eV) is shallower than that of NH<sub>2</sub>-PDI, TiO<sub>2</sub>, and MoS<sub>2</sub>. This result indicates that  $\cdot$ O<sub>2</sub><sup>-1</sup> may be the main active substance in NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composite for photocatalytic degradation of MB. The VB positions of MoS<sub>2</sub> was shallower than that of hydroxyl radical ( $\cdot$ OH) (2.40 eV)<sup>33</sup>, so it is possible that the  $\cdot$ OH cannot be generated by NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub>.

The active species in the photocatalytic process was studied by free radical capture experiment<sup>53</sup>. As can be seen from Fig. S-13, with the addition of p-BQ scavenger, the degradation rate decreased from 99.9% to 33%, indicating that  $\cdot O_2^{-}$  is one of the main active species in the photocatalytic degradation process. In the presence of EDTA-2Na, the degradation efficiency was significantly reduced, indicating that h<sup>+</sup> is also one of the major active radicals. After the addition of IPA, the degradation activity of the catalyst did not decrease, indicating that  $\cdot OH$  is not the main active species in MB degradation.

Based on the results above, two mechanisms including 'type II- heterojunction' and 'Z-scheme' were suggested to illustrate the photodegradation pathways occurred in this work (Fig. 9)<sup>54</sup>. In 'type II- heterojunction' mechanism (Fig. 9a), both the NH<sub>2</sub>-PDI and MoS<sub>2</sub> semiconductors can be excited by the arrived photons under visible light irradiation, resulting in production of e<sup>-</sup>/h<sup>+</sup> pairs in both semiconductors. As the CB and VB of NH<sub>2</sub>-PDI are located at higher energy levels than those of TiO2 and MoS2, the photogenerated electrons in the CB of NH2-PDI can transfer to the CB of TiO<sub>2</sub>, and then to the CB of MoS<sub>2</sub>, while holes travel in the opposite direction in the VB. These charge carriers' transfer processes result in the accumulation of the photogenerated electron in the CB level of MoS<sub>2</sub> and the photogenerated holes in the VB level of NH<sub>2</sub>-PDI. While in the 'Z-scheme' mechanism (Fig. 9b), the photogenerated electrons at the CB level of MoS<sub>2</sub> can migrate to the VB level of NH<sub>2</sub>-PDI based on the standard potentials, and thus, the 'Z-scheme' pathway accumulates the photogenerated electrons at the CB level of NH<sub>2</sub>-PDI and the photogenerated holes at the VB level of MoS<sub>2</sub>. The accumulated electrons at the CB level of NH<sub>2</sub>-PDI have the potential of -1.61 eV, which is more negative than the potential of MoS<sub>2</sub> (-0.75 eV). Therefore, the accumulated electrons at the CB level of NH<sub>2</sub>-PDI can effectively reduce dissolved oxygen and generate enough  $O_2^-$  as an efficient reactive species for the degradation of MB. These holes are also stronger oxidizing agents than the accumulated holes in the VB level of NH2-PDI (accumulated in type II mechanism) that can directly oxidize MB molecules. Therefore, the 'Z-scheme' mechanism is a better illustration for the enhanced photocatalytic activity of NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composite in MB degradation than the 'type II-heterojunction' mechanism. Nevertheless, in both mechanisms, TiO<sub>2</sub> facilitates the electron transfer between the conduction bands, reduces photoinduced charge carrier recombination and increases photocatalytic activity.

#### Conclusions

In conclusion, NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> ternary composite was successfully synthesized by hydrothermal synthesis method. Morphological and chemical features of TiO<sub>2</sub>, NH<sub>2</sub>-PDI, NH<sub>2</sub>-PDI/TiO<sub>2</sub>, MoS<sub>2</sub>, MoS<sub>2</sub>/TiO<sub>2</sub>, and NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> composites were examined by SEM, XRD, FTIR, XPS, and DRS. SEM results demonstrate that NH<sub>2</sub>-PDI nanorods and MoS<sub>2</sub> nanoflowers mixed with TiO<sub>2</sub>, which can provide more active sites and mass charge transport pathways in the catalytic system. The photoelectrochemical measurement results show that the photocurrent performance of the ternary composite catalyst was superior to that of binary composites or pure NH<sub>2</sub>-PDI, MoS<sub>2</sub>, or TiO<sub>2</sub>. The NH<sub>2</sub>-PDI/TiO<sub>2</sub>/MoS<sub>2</sub> exhibits excellent photocatalytic activity and high stability. The 'Z-scheme' accumulates more electrons at the CB level of NH<sub>2</sub>-PDI and photogenerated holes at the VB level of MoS<sub>2</sub>. The accumulated electrons can effectively reduce dissolved oxygen and generate enough  $\cdot$ O<sub>2</sub><sup>-</sup> as an efficient reactive species for the degradation of MB. These holes are also strong oxidizing agents that can directly oxidize MB molecules. TiO<sub>2</sub> could reduce photoinduced charge carrier recombination and increases photocatalytic activity. This study highlights the potential application of organic and inorganic photocatalysts, and provides a feasible strategy for the photodegradation of dyes.

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Guarantor of integrity of entire study—Y.M. Study concepts—Y.M.Study design—Y.M. Literature research— F.Z. experimental studies—F.Z. Data acquisition—Y.W. Data analysis/interpretation—F.Z.Statistical analysis— T.J. Manuscript preparation—Y.M. Manuscript definition of intellectual content—Y.M. Manuscript editing— X.L.Manuscript revision/review—Y.Z. Manuscript fnal version approval—Y.M.

## Competing interests

The authors declare no competing interests.

## Additional information

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