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Nonlinear optical properties of arsenic telluride and its use in ultrafast fiber lasers

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We report the first investigation results of the nonlinear optical properties of As_2Te_3 . More specifically, the nonlinear optical absorption properties of the prepared $\alpha\text{-As}_2\text{Te}_3$ were investigated at wavelengths of 1.56 and 1.9 μm using the open-aperture (OA) Z-scan technique. Using the OA Z-scan technique, the nonlinear absorption coefficients (β) of $\alpha\text{-As}_2\text{Te}_3$ were estimated in a range from $(-54.8 \pm 3.4) \times 10^4 \text{ cm/GW}$ to $(-4.9 \pm 0.4) \times 10^4 \text{ cm/GW}$ depending on the irradiance of the input beam at 1.56 μm , whereas the values did from $(-19.8 \pm 0.8) \times 10^4 \text{ cm/GW}$ to $(-3.2 \pm 0.1) \times 10^4 \text{ cm/GW}$ at 1.9 μm . In particular, the β value at 1.56 μm is an order of magnitude larger than the previously reported values of other group-15 sesquichalcogenides such as Bi_2Se_3 , Bi_2Te_3 , and Bi_2TeSe_2 . Furthermore, this is the first time report on β value of a group-15 sesquichalcogenide at a 1.9- μm wavelength. The density functional theory (DFT) calculations of the electronic band structures of $\alpha\text{-As}_2\text{Te}_3$ were also conducted to obtain a better understanding of their energy band structure. The DFT calculations indicated that $\alpha\text{-As}_2\text{Te}_3$ possess sufficient optical absorption in a wide wavelength region, including 1.5 μm , 1.9 μm , and beyond (up to 3.7 μm). Using both the measured nonlinear absorption coefficients and the theoretically obtained refractive indices from the DFT calculations, the imaginary parts of the third-order optical susceptibilities ($\text{Im } \chi^{(3)}$) of As_2Te_3 were estimated and they were found to vary from $(-39 \pm 2.4) \times 10^{-19} \text{ m}^2/\text{V}^2$ to $(-3.5 \pm 0.3) \times 10^{-19} \text{ m}^2/\text{V}^2$ at 1.56 μm and $(-16.5 \pm 0.7) \times 10^{-19} \text{ m}^2/\text{V}^2$ to $(-2.7 \pm 0.1) \times 10^{-19} \text{ m}^2/\text{V}^2$ at 1.9 μm , respectively, depending on the irradiance of the input beam. Finally, the feasibility of using $\alpha\text{-As}_2\text{Te}_3$ for SAs was investigated, and the prepared SAs were thus tested by incorporating them into an erbium (Er)-doped fiber cavity and a thulium–holmium (Tm–Ho) co-doped fiber cavity for both 1.5 and 1.9 μm operation.

Nonlinear optical responses of materials have always been of high technical interest in the area of optics and photonics due to their usefulness in various applications such as optical switching¹, wavelength conversion², second harmonic generation³, and saturable absorption⁴. The commonly investigated nonlinear optical responses of materials include $\chi^{(2)}$ effects⁵, $\chi^{(3)}$ effects (also called Kerr effects)⁶, and nonlinear absorption effects^{7–9}. Among the aforementioned nonlinear optical properties, nonlinear absorption effects are technical interesting since they can be applied to implementing absorption-related functional devices such as saturable absorbers (SAs)⁴ and multi-photon absorption devices¹⁰.

Recently, a huge number of investigations have been conducted regarding saturable absorption properties of various emerging materials, more specifically nano-materials. It is well-known that nonlinear saturable absorption effect usually occurs due to the Pauli's blocking principle within semiconducting materials¹¹. Note that the continued rapid growth in pulsed laser-related industries such as material processing, medicine, gas sensing, LIDAR, and free-space communication^{12–15} demands the development of high performance pulsed lasers, which can be realized only if an efficient SA should be used. The ever-increasing technical demand for highly efficient, cost-effective SAs has generated enormous interest in searching for new and novel nonlinear absorption materials.

Until now, most of commercially available SAs have been implemented using III–V compound semiconductors¹⁶ due to their proven performance and reliability, even if they have the fundamental limitations of limited operating bandwidth and the need for sophisticated/expensive facilities. A large number of investigations

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into alternative nonlinear optical materials that could overcome these limitations have been conducted so far. A variety of materials, which are mostly nano-structured materials, have been identified as saturable absorption materials suitable for SA implementation. These include: carbon nanotubes (CNTs)^{4,17–20}, graphene^{21–26}, graphene oxide (GO)^{27,28}, topological insulators (TIs)^{29–38}, topological semimetal³⁹, transition metal dichalcogenides (TMDCs)^{40–53}, transition metal monochalcogenides (TMMCs)⁵⁴, filled skutterudites (FSs)⁵⁵, black phosphorus (BPs)^{56–58}, gold nano-particles^{59–61}, and MXenes^{62,63}.

Our research group has investigated the nonlinear saturable absorption properties of a range of materials such as TIs, MXene, and TMDCs. In particular, we conducted a series of investigations into the ultimate potential of various TIs such as Bi₂Te₃, Bi₂Se₃, and CoSb₃ as saturable absorption materials. We showed that TIs possess good nonlinear saturable absorption properties whether the materials are nano-structured or bulk-structured.

As an ongoing study, our group has focused on arsenic telluride (As₂Te₃), which is another bulk form of group-15 sesquichalcogenides with a generic formula A₂B₃ (A = As, Sb, Bi; B = S, Se, Te). As₂Te₃ is known to have lattice constants similar to Bi₂Se₃. As₂Te₃ exists in two crystallographic forms: α - and β -As₂Te₃. α -As₂Te₃ at ambient pressure is known to have a monoclinic structure with a C2/m space group^{64,65}, and it exhibits lower thermoelectric figure of merit (ZT) than Sb- and Bi-based tellurides⁶⁶. On the other hand, β -As₂Te₃ with a rhombohedral *R3m* symmetry, which is known to have outstanding thermoelectric properties^{67,68}, has been reported to have topologically protected surface states as a three-dimensional TI when the uniaxial strain is along the *c*-axis of the rhombohedral crystal structure^{69,70}.

A number of investigations into the physical properties of group-15 sesquichalcogenides have been conducted in terms of thermoelectric, electronic, vibrational, and optical properties following the recent technical interest in those materials. However, only a few investigations on As₂Te₃ have been conducted so far. Those limited investigations were mostly focused on thermoelectric, vibrational, and electronic properties^{71–75} and thus there has been no research on the optical properties of As₂Te₃, to the best of our knowledge.

In this work, the nonlinear optical absorption properties of α -As₂Te₃ were investigated in both theoretical and experimental ways. First, a series of measurements including scanning electron microscopy (SEM), energy dispersive spectroscopy (EDS), Raman spectrum, and X-ray photoelectron spectroscopy (XPS), were carried out for As₂Te₃ particles, to determine the material properties. Second, the electronic and optical properties such as energy band structure, absorption spectrum, and wavelength dependent refractive index values were theoretically calculated with the density functional theory (DFT) calculation method. It is shown that α -As₂Te₃ is a semiconducting material with a wide linear absorption bandwidth that can cover the 2- μ m wavelength region. Third, the nonlinear absorption coefficients (β) of α -As₂Te₃ were measured using the open aperture (OA) Z-scan measurement method at wavelengths of 1,560 nm and 1,900 nm. Subsequently, the imaginary parts of the third-order optical susceptibilities ($\text{Im } \chi^{(3)}$) of As₂Te₃ were estimated at 1,560 and 1,900 nm, using both the measured nonlinear absorption coefficients and the theoretically obtained refractive indices from the DFT calculations. The nonlinear absorption coefficients (β) of α -As₂Te₃ varied in a range from $(-54.8 \pm 3.4) \times 10^4$ cm/GW to $(-4.9 \pm 0.4) \times 10^4$ cm/GW depending on the irradiance of the input beam at 1,560 nm, whereas the value did from $(-19.8 \pm 0.8) \times 10^4$ cm/GW to $(-3.2 \pm 0.1) \times 10^4$ cm/GW at 1,900 nm. The estimated imaginary parts of the third-order optical susceptibilities ($\text{Im } \chi^{(3)}$) of As₂Te₃ were found to vary from $(-39 \pm 2.4) \times 10^{-19}$ m²/V² to $(-3.5 \pm 0.3) \times 10^{-19}$ m²/V² at 1,560 nm and $(-16.5 \pm 0.7) \times 10^{-19}$ m²/V² to $(-2.8 \pm 0.1) \times 10^{-19}$ m²/V² at 1,900 nm, respectively, depending on the irradiance of the input beam. Fourth, the feasibility of using As₂Te₃ as a base material for implementation of a broadband SA was investigated by fabricating an SA on a fiber ferrule-based sandwich structure platform. Using the prepared α -As₂Te₃-based SA within an erbium (Er) fiber-based ring cavity, stable mode-locked pulses with a temporal width of ~ 858 fs were generated at a wavelength of $\sim 1,559.8$ nm. Additionally, mode-locked pulses with a temporal width of ~ 1.34 ps were readily obtained at a wavelength of 1,911.4 nm by using another As₂Te₃-based SA within a thulium–holmium (Tm–Ho) co-doped fiber laser ring cavity.

Results

Experimental and theoretical investigation of material properties. Commercially available As₂Te₃ crystals (LTS Research Lab., 99.99%) were used as a starting material in the present experiment. A bath type ultrasonicator was used to synthesize the α -As₂Te₃ particles. The measured energy dispersive spectroscopy (EDS) spectrum of the α -As₂Te₃ particles is shown in Fig. 1a. A small amount of α -As₂Te₃ particle solution was dropped on top of a slide glass and dried for 24 h for the EDS measurement. The spectrum shows strong peaks corresponding to As and Te. A scanning electron microscope (SEM) image of the α -As₂Te₃ particles dried on a silicon substrate is shown in the inset of Fig. 1a. The size of the α -As₂Te₃ particles ranged from tens of nanometers to about few micrometers. Figure 1b shows the measured Raman spectrum of the prepared α -As₂Te₃ particles. The A¹ mode E'' mode of Te are shown at 125 cm⁻¹ and 141 cm⁻¹, and two high-frequency modes were observable at 171 cm⁻¹ and 200 cm⁻¹, respectively⁷⁵. XPS measurements were conducted to analyze the stoichiometry of the α -As₂Te₃ particles. Figure 1c shows the As 2p spectrum, whereas the Te 3d spectrum is shown in Fig. 1d. The peak at $\sim 1,323.9$ eV in the As 2p region of Fig. 1c is consistent with the reported binding energy value of the As 2p_{3/2}⁷⁶, while the two peak at ~ 572.5 and ~ 583 eV in the Te 3d region of Fig. 1d are consistent with those of Te 3d_{5/2} and Te 3d_{3/2}⁷⁸. An additional peak was located at ~ 1326.5 eV in the As 2p region, whereas they were at ~ 576 and 586.4 eV in the Te 3d region. The existence of those additional peaks can be attributed to the oxidation of As and Te atoms on the surface^{76–78}.

For a deeper investigation of the electrical and optical properties of α -As₂Te₃, we performed density functional theory (DFT) calculations of the electronic band structures as well as optical refractive and/or absorption spectra of the α -As₂Te₃ crystal. The optimized geometry of α -As₂Te₃ is shown in Fig. 2a, which belongs to space group C2/m. The calculation showed that the electronic band gap of α -As₂Te₃ is around 0.34 eV, indicating a good agreement with previous ab-initio calculations (0.32 eV)⁶⁵ and readily allowing 1,550–1,900-nm saturable

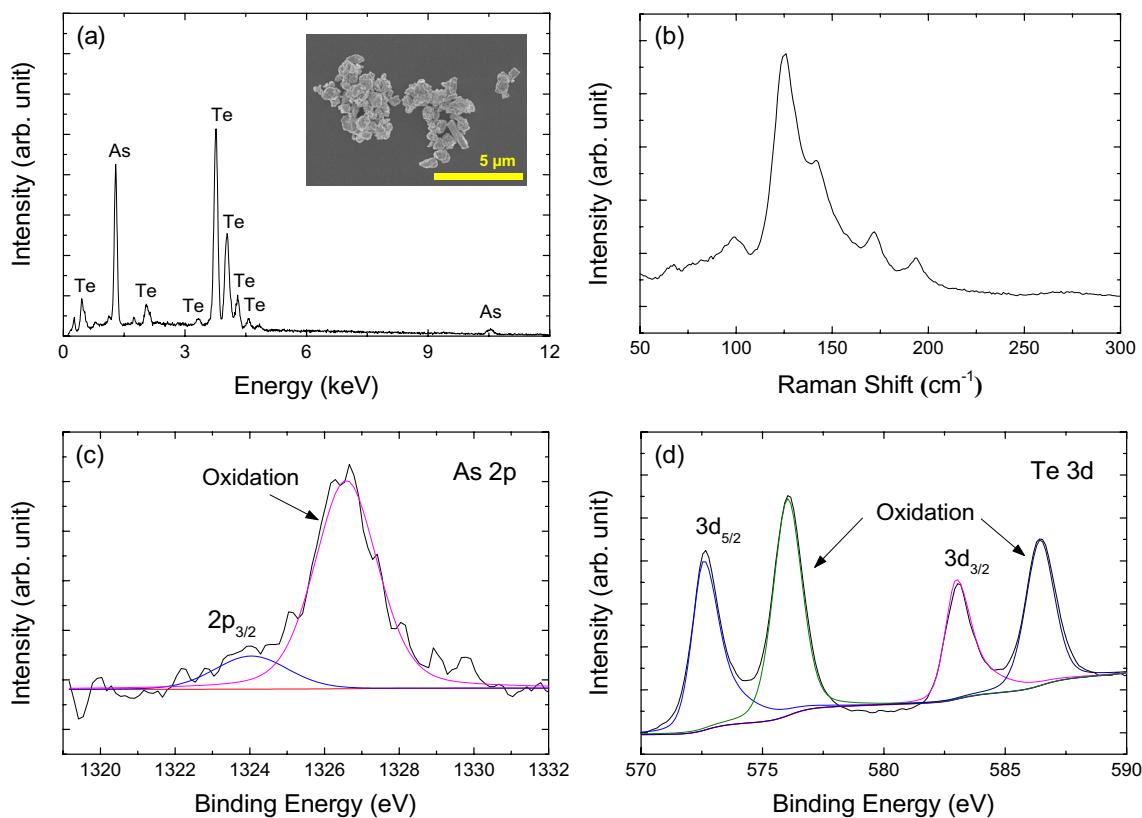


Figure 1. (a) EDS spectrum of the As_2Te_3 particles. Inset: SEM image of the As_2Te_3 particles. (b) Raman spectrum of the As_2Te_3 particles and XPS spectra of the (c) As 2p core level, and (d) Se 3d core level.

absorption (Fig. 2b). To make this point clear, we further investigated the optical refractive and/or absorption properties of $\alpha\text{-As}_2\text{Te}_3$ using the Kubo–Greenwood formula and were able to obtain consistent results (Fig. 2c). Meanwhile, the previous experiment showed that the bandgap should be 0.46 eV⁷⁹, while the estimation based on temperature-dependent thermoelectric activation underestimated the bandgap to be around 0.13 eV⁸⁰. Considering our calculation results, $\alpha\text{-As}_2\text{Te}_3$ might be a good SA even for applications requiring longer wavelengths than that of 0.34 eV due to exciton formation, edge-states, and structural defects. In order to check the linear optical absorption of the As_2Te_3 , the linear absorption measurement was conducted for the $\alpha\text{-As}_2\text{Te}_3$ /polyvinylpyrrolidone (PVP) composite deposited onto a glass slide, and the linear absorption of the $\alpha\text{-As}_2\text{Te}_3$ /PVP composite was measured using a spectrophotometer (UV-3600PLUS, Shimadzu). Figure 2d clearly shows broadband absorption of the $\alpha\text{-As}_2\text{Te}_3$ /PVP composite over a wide spectral range from 1,000 to 3,300 nm. The pure PVP film has a small absorption over a range of 1,000 nm–2,700 nm with two small peaks at 1946 nm and 2,280 nm, respectively, which are caused by the PVP vibration⁸¹. The interesting fact is that there is a strong peak at $\sim 2,950$ nm (wavenumber = $\sim 3,400$ cm^{-1}) in the absorption spectrum of the $\alpha\text{-As}_2\text{Te}_3$ /PVP film, which corresponds to the O–H stretching vibration of PVP^{82,83}. Since the uniformity of our prepared sample is not good, it is thus very difficult to figure out the precise thickness value of our sample at a particular position, at which the incident laser beam is focused. Therefore, it is very hard to obtain a precise linear absorption coefficient of our sample. In general, the linear absorption increases as the sample thickness is enlarged, according to Beer–Lambert law (i.e., $T \sim \exp(-\alpha \times d)$ where T is the transmittance, α is the absorption coefficient, and d is the sample thickness). The sample-thickness-dependent property variations have been reported for various saturable absorption materials such as TIs, TMDCs, and BP^{84–89}. The bandgap energies of the materials are known to change depending on the layer number. Considering the fact that a material with not less than six layers can be regarded as bulk, our used As_2Te_3 sample is bulk-structured⁸⁷ since the minimum size of the As_2Te_3 particles used in this investigation is tens of nanometers. This indicates that the non-uniformity of our As_2Te_3 particles could not induce any associated bandgap change due to its bulk nature.

Open aperture (OA) Z-scan measurements were performed at 1,560 nm and 1,900 nm to investigate the nonlinear optical properties of the $\alpha\text{-As}_2\text{Te}_3$, as shown in Fig. 4. In the step, a mode-locked fiber laser beam was focused through a plano-convex lens (lens 1 in Fig. 3) onto an $\alpha\text{-As}_2\text{Te}_3$ sample mounted on a motorized translation stage. The incident laser pulses at 1,560 nm were obtained from a ~ 22.26 MHz, ~ 300 fs mode-locked fiber laser and the input laser pulses at 1,900 nm were obtained from a ~ 36.94 MHz, ~ 691 fs mode-locked fiber laser. Note that the intensity of the transmitted beam is dependent on the sample position, and it was focused using the photodetector reading of the transmittance change of the incident beam. As shown in Fig. 4a for the Z-scan curve at 1,560 nm, the normalized transmittance gradually increased as the As_2Te_3 sample approached the focal point ($Z = 0$), which was caused by the saturable absorption response. Furthermore, as the input peak intensity increased from 0.2 to 27.68 MW/cm^2 , the peaks of the OA Z-scan curves increased. These results indicate that

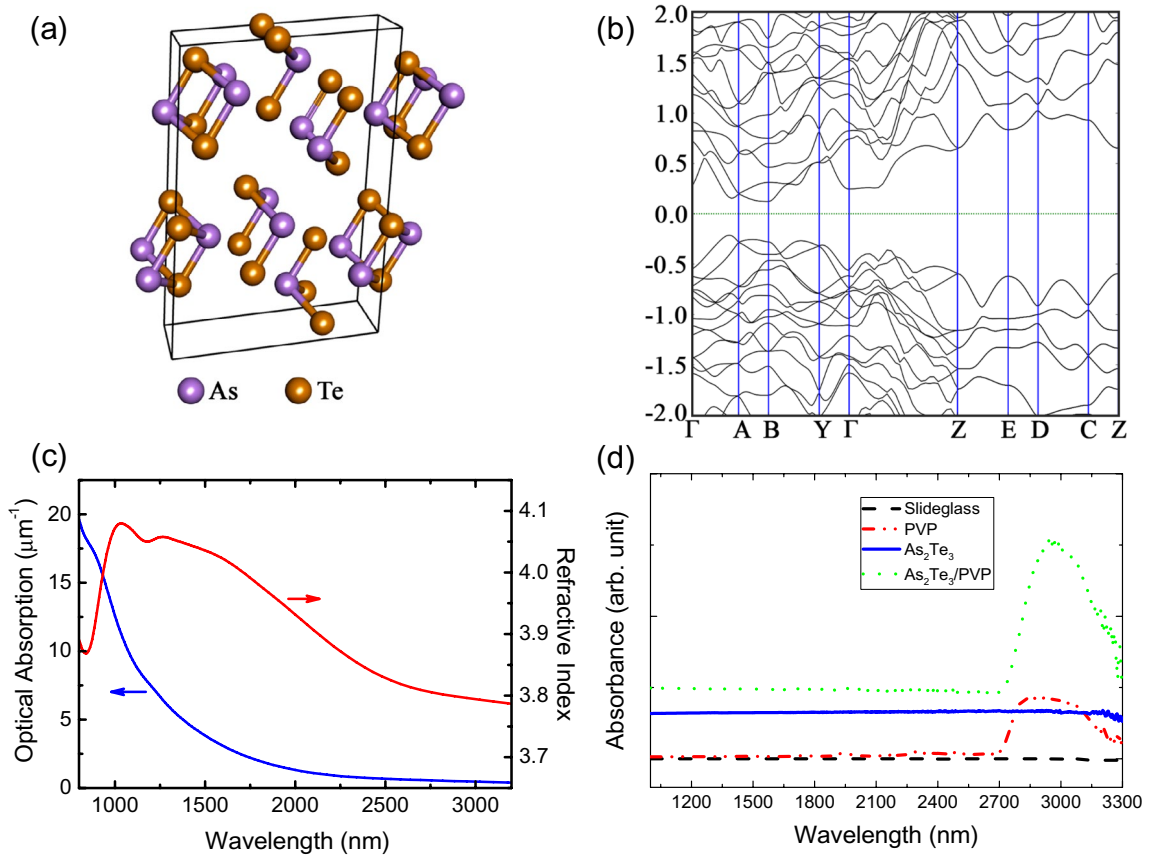


Figure 2. (a) Structure of the α -As₂Te₃ crystal. (b) The calculated electronic band structure of α -As₂Te₃. (c) The calculated optical absorption and refractive index spectra of α -As₂Te₃. (d) Measured linear absorption spectrum of α -As₂Te₃/PVP composite.

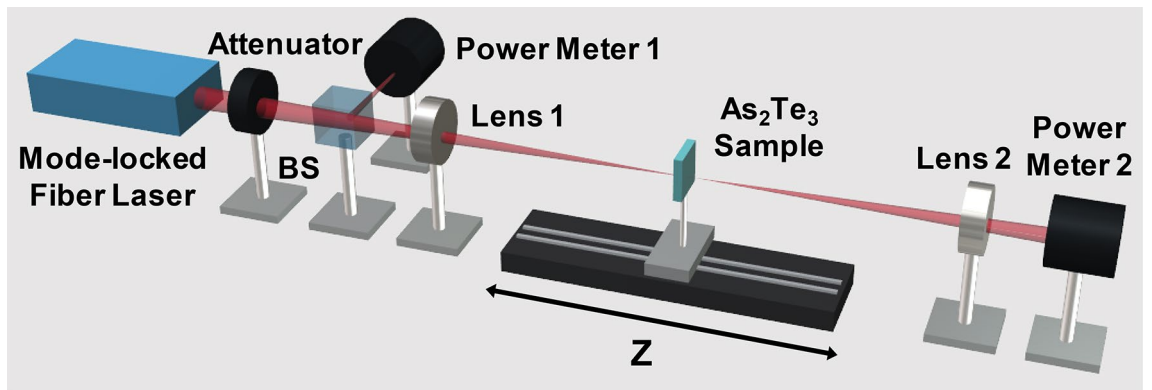


Figure 3. Schematic diagram of the Z-scan experimental setup. BS: Beam splitter.

the nonlinear optical absorption of α -As₂Te₃ indeed came from α -As₂Te₃ itself. The measured Z-scan curves were fitted with the following approximate equation^{90,91}

$$T(z) = \sum_{n=0}^{\infty} (-\beta I_0 L_{eff})^n / (1 + z^2/z_0^2)^n (n + 1)^{3/2} \approx 1 - \beta I_0 L_{eff} / 2^{3/2} (1 + z^2/z_0^2) \quad (1)$$

where $T(z)$ is the normalized transmittance, β is the nonlinear absorption coefficient, I_0 is the peak on axis-intensity at the focus, L_{eff} is the effective length, z is the position of the sample, and z_0 is the Rayleigh length.

Figure 4b shows the measured nonlinear absorption coefficient β value as a function of the excitation irradiance. The nonlinear absorption coefficient values of the prepared α -As₂Te₃ sample vary from $(-54.8 \pm 3.4) \times 10^4$ cm/GW to $(-4.9 \pm 0.4) \times 10^4$ cm/GW as the irradiance of the input beam was increased, as shown in Fig. 4b. The experimental results showed that, the nonlinear absorption coefficient was obviously dependent on the

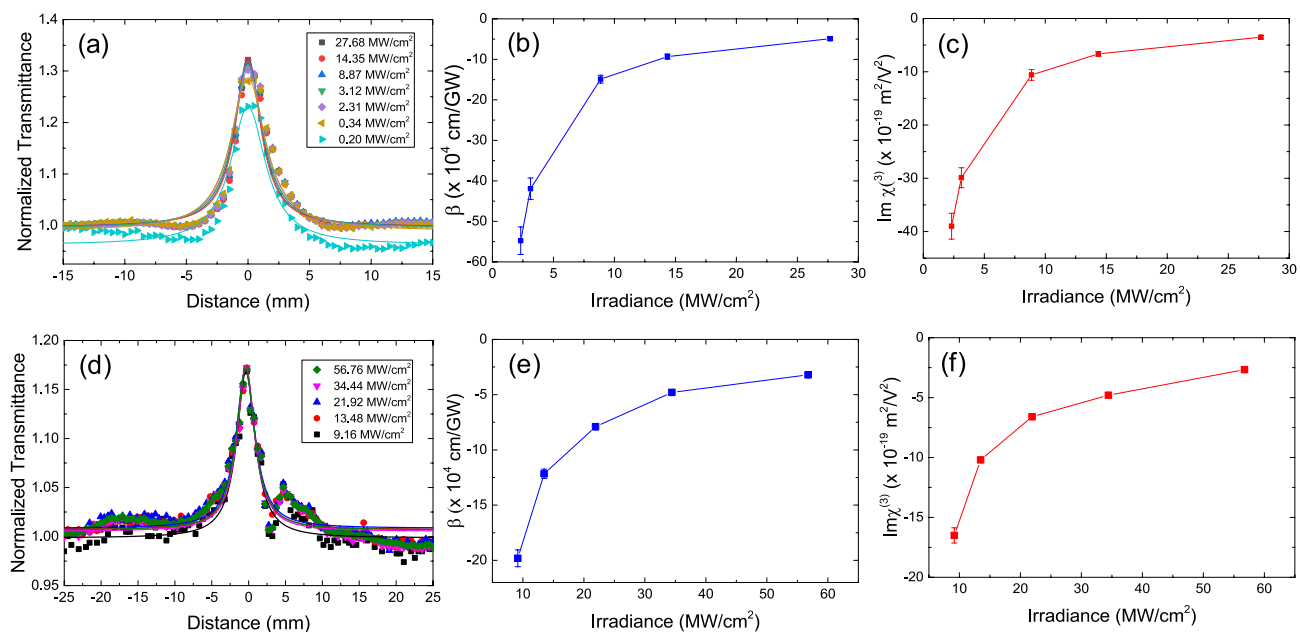


Figure 4. (a) Open aperture Z-scan curves of the α -As₂Te₃ film at 1,560 nm. (b) Nonlinear absorption coefficient (β) and (c) imaginary part of the third-order optical susceptibility ($\text{Im}\chi^{(3)}$) as a function of pulse irradiance at 1,560 nm. (d) Open aperture Z-scan curves of the α -As₂Te₃ film at 1,900 nm. (e) Nonlinear absorption coefficient (β) and (f) the imaginary part of the third-order optical susceptibility ($\text{Im}\chi^{(3)}$) as a function of pulse irradiance at 1,900 nm.

irradiance. This phenomenon was also observed in the previous reports on topological insulators of Bi₂Te₃ and Bi₂Se₃, which also belong to group-15⁹⁷. It was also reported in MXenes, SnSe₂, and gold nanoparticles^{63,92,93}. Figure 4d shows the measured Z-scan curves with fitting curves from the approximate equation^{90,91} at 1,900 nm. The nonlinear absorption coefficient of the sample was measured to be from $(-19.8 \pm 0.8) \times 10^4$ cm/GW to $(-3.2 \pm 0.1) \times 10^4$ cm/GW at 1,900 nm. The asymmetric shape of the measured data in Fig. 4d could be attributed to the surface non-uniformity of the deposited sample. Sample distortions, wedges, or tilting of the sample during Z-scan translation can cause unwanted fluctuations in the detected signal of the Z-scan measurement setup^{98–100}. In order to avoid the unwanted fluctuations in Z-scan curve, a uniform surface of the sample is essential. A further improvement for our sample preparation process is required and will be done for the future.

It is well-known that the OA Z-scan technique is a standard method for measuring nonlinear optical responses associated with saturable absorption and two-photon absorption, even if the results could be influenced by the thermal effects caused by the input pulse laser temporal characteristics. Saturable absorption and two-photon absorption are known to occur under different conditions, even if their coexistence of the two phenomena was reported^{101–105}. In general, the threshold of two-photon absorption is higher than that of saturable absorption. When a high intensity beam is launched into a nonlinear optical material, saturable absorption first occurs due to the Pauli's blocking principle¹¹. As the incident beam intensity was enlarged, the carriers in the valence band absorb two photons simultaneously and the two-photon absorption process begins to appear^{103–105}. However, it would be possible to induce two-photon absorption before saturable absorption occurs^{101,102}. In the case of coexistence of saturable absorption and two-photon absorption, the OA Z-scan curve exhibits symmetrical dips on both sides around the center peak. Note that our measured OA Z-scan curve does not have such symmetrical dips despite of a dip on a single side. This indicates that the contribution of two-photon absorption for our measured OA Z-scan curve is not evident. Furthermore, regarding the thermal effects, which might result in a wrong interpretation on OA Z-scan measurement results, it is believed that we could rule the possibility of the intra-pulse and pulse-to-pulse cumulative thermal effects out in our measurements due to use of low pulse-energy, 703-fs pulses as an input beam even if the depletion rate is quite high (22.26 MHz)^{101,106}. However, further investigations into the thermal effects needs to be conducted for the future.

It is well known that nonlinear absorption properties vary depending on the polarization status of the incident beam and the OA Z-scan measurement thus exhibits polarization-dependent curve variations^{107–110}. However, we have not observed any polarization-dependent variations on the OA Z-scan curves for our prepared sample. This phenomenon could be attributed to the fact that our used As₂Te₃ particles are polycrystalline and bulk-structured. Note that the polarization dependence of nonlinear absorption was mostly observed in the case of crystalline or 2-D structured materials.

Table 1 summarizes the nonlinear absorption coefficients of different group-15 sesquichalcogenides and the α -As₂Te₃. It should be noticed that the nonlinear absorption coefficient of the α -As₂Te₃ is an order of magnitude larger than the values at of Bi₂Se₃, Bi₂Te₃, and Bi₂TeSe₂ at a 1.5- μ m wavelength^{96,97}. Since our measurements were conducted at 1.56 and 1.9 μ m wavelengths, only, the comparison was limited to the measured values at 1.5- μ m wavelengths. To the best of the authors' knowledge no Z-scan measurement at a wavelength of 1.9 μ m has been

Materials	Wavelength (nm)	Nonlinear absorption coefficient ($\times 10^4$ cm/GW)	References
Sb ₂ Te ₃	632.8	-6.63×10^5	94
Bi ₂ Se ₃	800	$-(84 \pm 1.5)$	95
Bi ₂ Se ₃	532	$-(2.0 \pm 0.4)$	97
Bi ₂ Se ₃	800	$-(6.5 \pm 1.3)$	97
Bi ₂ Se ₃	1,050	$-(5.5 \pm 1.1)$	97
Bi ₂ Se ₃	1,550	$-(2.3 \pm 0.5)$	97
Bi ₂ Te ₃	1,550	-1	96
Bi ₂ Te ₃	532	$-(3.2 \pm 0.6)$	97
Bi ₂ Te ₃	800	$-(2.1 \pm 0.4)$	97
Bi ₂ Te ₃	1,050	$-(4.7 \pm 0.9)$	97
Bi ₂ Te ₃	1,550	$-(3.9 \pm 0.8)$	97
Bi ₂ TeSe ₂	532	$-(3.4 \pm 0.7)$	97
Bi ₂ TeSe ₂	800	$-(5.6 \pm 1.1)$	97
Bi ₂ TeSe ₂	1,050	$-(8.7 \pm 1.7)$	97
Bi ₂ TeSe ₂	1,550	$-(5.9 \pm 1.2)$	97
Bi ₂ Te ₂ Se	532	$-(2.6 \pm 0.5)$	97
Bi ₂ Te ₂ Se	800	$-(3.7 \pm 0.7)$	97
Bi ₂ Te ₂ Se	1,050	$-(9.3 \pm 1.9)$	97
Bi ₂ Te ₂ Se	1,550	$-(7.7 \pm 0.8)$	97
α -As ₂ Te ₃	1,560	$(-54.8 \pm 3.4) \sim (-4.9 \pm 0.4)$	This work
α -As ₂ Te ₃	1,900	$(-19.8 \pm 0.8) \sim (-3.2 \pm 0.1)$	This work

Table 1. Nonlinear absorption coefficients (β) of the various TI materials and the As₂Te₃.

reported on group-15 sesquichalcogenides until now. Our Z-scan results imply that the α -As₂Te₃ possesses nonlinear optical response large enough for the practical implementation of as SAs.

Also, the imaginary part of the third-order optical susceptibility ($\text{Im}\chi^{(3)}$) was calculated using the following equation^{63,111}:

$$\text{Im}\chi^{(3)} = \frac{2\varepsilon_0 c^2 n_0^2}{3\omega} \beta \quad (2)$$

where c is the light speed, ε_0 is the vacuum permittivity, ω is the angular frequency, and n_0 is the refractive index. Note that the refractive index (n_0) used for this calculation was obtained from our DFT calculation results in Fig. 2c. As shown in Fig. 2c, the theoretical refractive indices of As₂Te₃ at a wavelength of 1.56 and 1.9 μm are 4.03 and 3.95, respectively. The imaginary parts of the third-order optical susceptibilities of As₂Te₃ were estimated to vary from $(-39 \pm 2.4) \times 10^{-19} \text{ m}^2/\text{V}^2$ to $(-3.5 \pm 0.3) \times 10^{-19} \text{ m}^2/\text{V}^2$ at 1,560 nm and $(-16.5 \pm 0.7) \times 10^{-19} \text{ m}^2/\text{V}^2$ to $(-2.8 \pm 0.1) \times 10^{-19} \text{ m}^2/\text{V}^2$ at 1,900 nm, respectively, depending on the irradiance of the input beam.

It was reported that the precise estimation for the imaginary part of the third-order optical susceptibility ($\text{Im}\chi^{(3)}$) requires not only the nonlinear absorption coefficient but also the additional information of nonlinear refractive index, which can be obtained with the CA Z-scan technique¹¹². However, a reasonable estimation of the third-order optical susceptibility ($\text{Im}\chi^{(3)}$) without loss of generality, is known to be still obtainable only from OA Z-scan measurement results^{63,111,113}.

Fabrication and characterization of a saturable absorber. Two all-fiberized SAs based on α -As₂Te₃ were fabricated using simple sandwich structures of fiber ferrules: one is for 1.5- μm operation and the other for 1.9- μm . The prepared α -As₂Te₃ solution was first mixed with PVP to facilitate the formation of a film when it was dropped onto a flat surface. More precisely, 1 g of the As₂Te₃ bulk was grinded in a mortar to obtain α -As₂Te₃ powders. The α -As₂Te₃ particle solution was prepared using a sonication in 30 ml of distilled water without centrifugation after ultrasonication for 8 h. In order to form a composite of the As₂Te₃ particles and the PVP, 500 mg of PVP was mixed with the α -As₂Te₃ particle solution. A small amount of α -As₂Te₃/PVP composite solution was directly deposited onto the end surface of a FC/APC fiber ferrule and was connected to another FC/APC fiber ferrule to form a sandwich-structured SA. The insertion loss of the α -As₂Te₃/PVP-based SA operating at a wavelength of 1,560 nm was measured to be ~ 2.8 dB, while the value of the SA at 1,900 nm was ~ 2.5 dB.

Then, the nonlinear saturable absorption properties of both of the fabricated SAs were measured to determine their nonlinear transmission properties, as shown in Fig. 5. In these measurements, we used our custom-made mode-locked fiber lasers. The pulse width and repetition rate of our 1.56- μm mode-locked fiber laser were ~ 730 fs and ~ 22.26 MHz, respectively, whereas those of the 1.9- μm mode-locked fiber laser were ~ 703 fs and ~ 36.94 MHz, respectively. The measurement setup is shown in Fig. 5a. Figure 5b shows the measured nonlinear transmission curve of the α -As₂Te₃/PVP-based SA operating at 1.56 μm as a function of the incident peak power, while Figure 5c shows the curve of the SA at 1.9 μm . The modulation depths and saturation intensity were

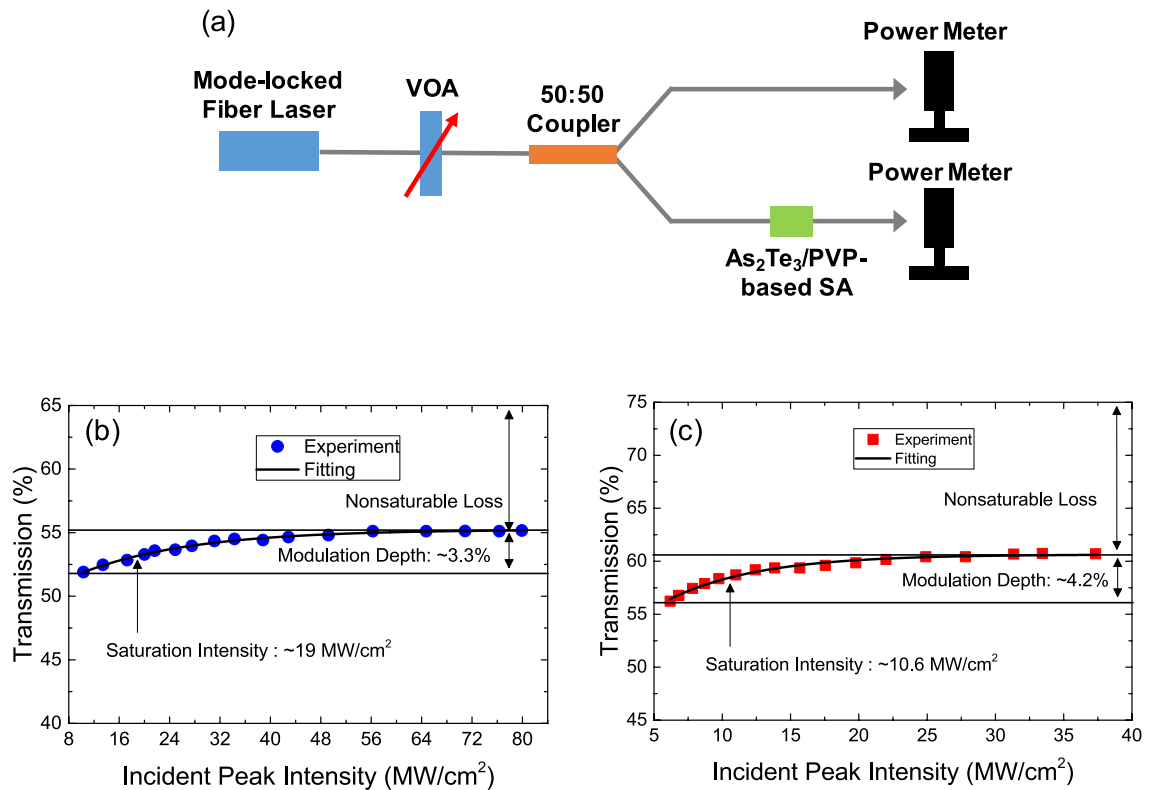


Figure 5. (a) Measured setup for nonlinear transmission of the α -As₂Te₃-based SA. Measured nonlinear transmission curve as a function of incident peak intensities at (b) 1,560 nm and (c) 1,900 nm.

measured to be $\sim 3.3\%$ and ~ 19 MW/cm² at 1.56 μ m, and $\sim 4.3\%$ and ~ 10.6 MW/cm² at 1.9 μ m, respectively. The following formula, which is commonly used to fit the SA, was used for curve fitting¹¹⁴:

$$T(I) = 1 - \Delta T \cdot \exp\left(\frac{-I}{I_{sat}}\right) - T_{ns} \quad (3)$$

where $T(I)$ is the transmission, ΔT is the modulation depth, I is the input pulse energy, I_{sat} is the saturation energy, and T_{ns} is the nonsaturable loss.

Implementation of pulsed fiber lasers. The experimental schematic of an Er-doped fiber laser with an α -As₂Te₃/PVP-based SA is shown in Fig. 6a. The α -As₂Te₃/PVP-based SA with an insertion loss of ~ 2.8 dB was incorporated into the cavity produce mode-locked pulses from an Er-doped fiber laser cavity. Using this fiber laser configuration, mode-locked pulses were obtainable by increasing the pump power with suitable adjustment of the polarization controller. When the pump power was above 18 mW, stable mode-locked pulses were successfully obtained in which the average output power was ~ 0.1 mW. The repetition rate and temporal period of the output mode-locked pulses were measured to be 14.35 MHz and 69.7 ns, respectively, which correspond to the fundamental resonance frequency and round-trip time of the fiberized cavity (Fig. 6b).

Figure 6c shows the optical spectrum of the output pulses together with its sech² fitting curve. The center wavelength and 3-dB bandwidth were measured to be ~ 1559.9 nm and ~ 3.46 nm, respectively. Kelly sidebands were clearly observed on the optical spectrum, indicating that the mode-locked fiber laser operated in the soliton regime¹¹⁵. Next, we conducted an autocorrelation measurement by using a two-photon absorption-based autocorrelator. The temporal width of the output pulses was measured to be ~ 858 fs, as shown in Fig. 6d, and the time-bandwidth product was estimated to be 0.366, indicating that the output pulses were slightly chirped. We observed a sharp and strong peak in the electrical spectrum of the output pulses having a fundamental repetition rate of 14.35 MHz and a peak-to-background ratio of ~ 58 dB (Fig. 6e).

As shown in our theoretical investigation and linear absorption measurement, the optical absorption of α -As₂Te₃ was in the mid-infrared wavelength region. Therefore, to verify the applicability of the α -As₂Te₃/PVP-based SA to mid-infrared wavelength lasers, the prepared SA was incorporated in a Tm-Ho co-doped fiber ring cavity to generate mode-locked pulses in the 1.9 μ m wavelength region. A fiber laser setup similar to Fig. 6a was used for this particular experiment. A 1-m long Tm-Ho co-doped fiber (TH512, CorActive) with 13 dB/m absorption at 1,550 nm was used as a gain medium. A 1,550-nm laser diode with a maximum pump power of ~ 297 mW was used as a pump source, which was coupled into the gain medium through a 1,550/2,000 nm wavelength division multiplexer (WDM). Stable mode-locked pulses were readily obtained when the pump power was set to be ~ 155 mW.

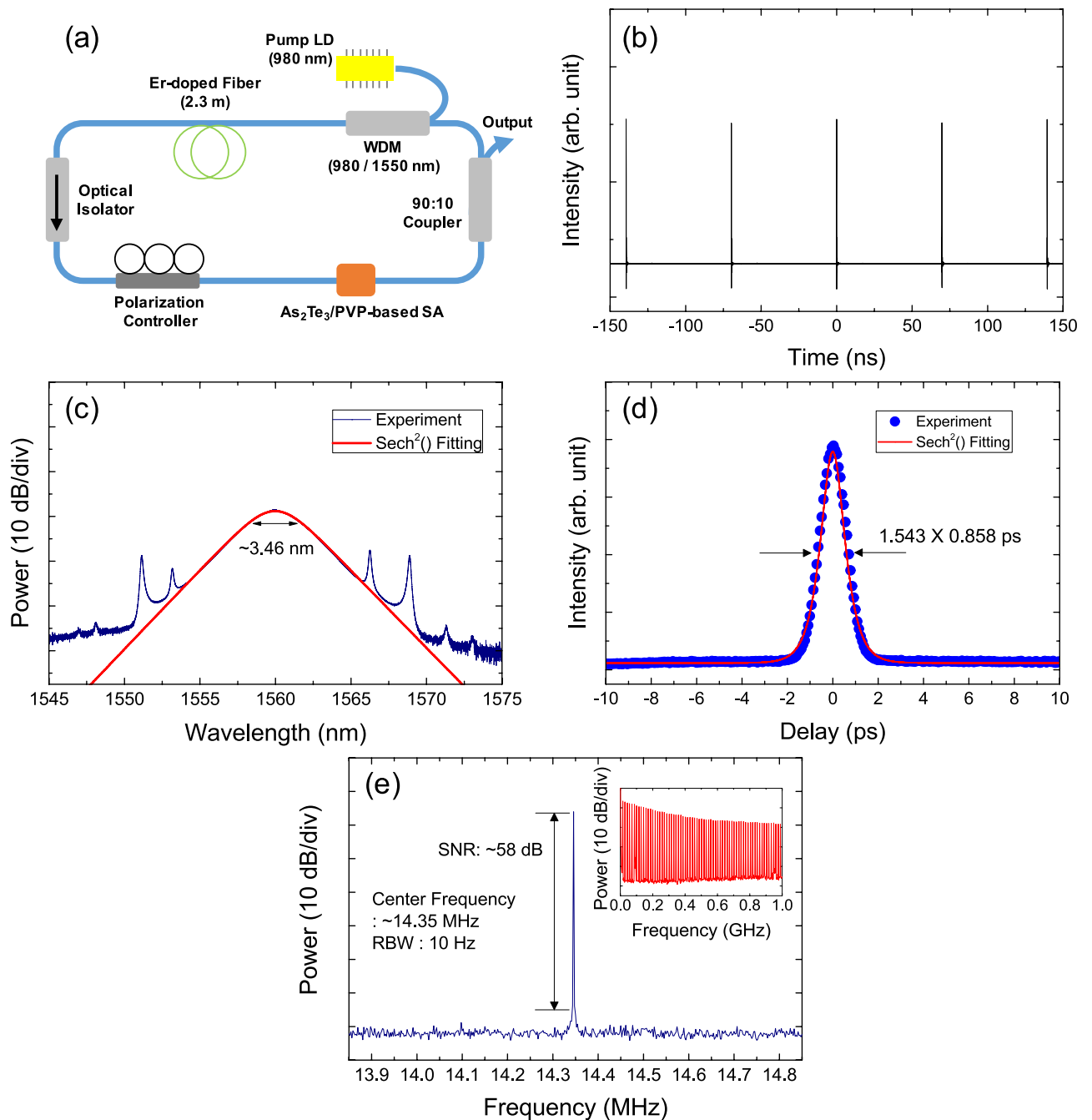


Figure 6. (a) Configuration of an erbium-doped fiber (EDF) laser cavity. Measured (b) oscilloscope trace, (c) optical spectrum, (d) autocorrelation trace, and (e) electrical spectrum of the output pulses at 1.56 μm . Inset: measured electrical spectrum over a 1-GHz span.

Figure 7a shows the measured oscilloscope trace of the output pulses with a combination of a 16 GHz real-time oscilloscope and a 12.5-GHz photodetector. The repetition rate of the output pulses was measured to be 17.5 MHz. Figure 7b shows the measured optical spectrum of the mode-locked pulses. The center wavelength and 3-dB bandwidth were measured to be ~ 1911.4 nm and 3.12 nm, respectively. Next, an autocorrelation measurement was conducted for the output pulses. The measured temporal width of the mode-locked pulses was ~ 1.34 ps, as shown in Fig. 7c, and the time-bandwidth product was estimated to be 0.343, indicating that the output pulses were slightly chirped. Finally, the electrical spectrum was measured to check the phase noise of mode-locked pulses as shown in Fig. 7d. A strong signal peak with an electrical signal-to-noise ratio (SNR) of ~ 59 dB was clearly observed at a fundamental frequency of ~ 17.5 MHz.

We checked the long-term stability of the fabricated SAs for few days, however, we could not find any performance degradation of the SAs. We also checked the long-term stability of the mode-locked fiber lasers by monitoring the laser output for several hours but could not find any stability problem. Furthermore, we launched

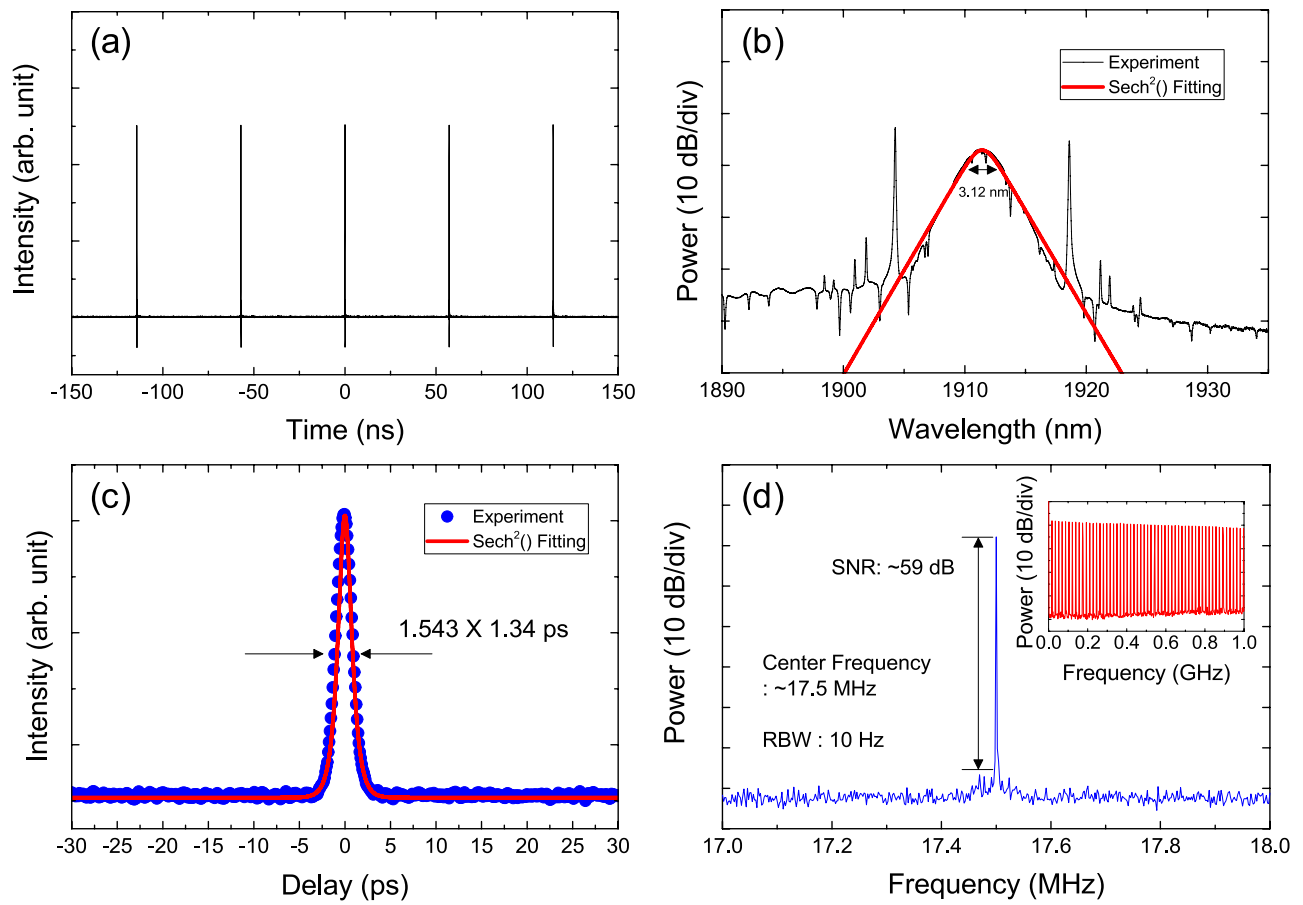


Figure 7. Measured (a) oscilloscope trace, (b) optical spectrum, (c) autocorrelation trace, and (d) electrical spectrum of the output pulses at 1.9 μm . Inset: measured electrical spectrum over a 1-GHz span.

continuous-wave, amplified laser beams with powers of 1 W and 100 mW at 1,550 nm and 1,900 nm, respectively into our prepared As_2Te_3 -based SAs to measure their damage thresholds. We have not observed any damage of the prepared SAs within the power levels. Therefore, it is believed that the damage threshold values of the prepared As_2Te_3 -based SAs must be larger than 1 W at 1,550 nm and 100 mW at 1,900 nm, each. However, it was impossible to measure the precise damage thresholds due to the limited availability of high power lasers in our laboratory.

Conclusion

Here, we have conducted the first investigation of the nonlinear saturable absorption properties of $\alpha\text{-As}_2\text{Te}_3$ using both experimental and theoretical techniques. Nonlinear absorption coefficients (β) of $\alpha\text{-As}_2\text{Te}_3$ were measured using the OA Z-scan technique at a wavelength of 1,560 nm and 1,900 nm. They were found to vary in a range from $(-54.8 \pm 3.4) \times 10^4 \text{ cm/GW}$ to $(-4.9 \pm 0.4) \times 10^4 \text{ cm/GW}$ at 1,560 nm and from $(-19.8 \pm 0.8) \times 10^4 \text{ cm/GW}$ to $(-3.2 \pm 0.1) \times 10^4 \text{ cm/GW}$ at 1,900 nm, respectively. Interestingly, the nonlinear absorption coefficient of $\alpha\text{-As}_2\text{Te}_3$ at 1.56 μm was found to be an order of magnitude larger than the previously reported values of other group-15 sesquichalcogenides such as Bi_2Se_3 , Bi_2Te_3 , and Bi_2TeSe_2 . Also, the estimated imaginary parts of the third-order optical susceptibilities ($\text{Im}\chi^{(3)}$) of the prepared As_2Te_3 -sample vary from $(-39 \pm 2.4) \times 10^{-19} \text{ m}^2/\text{V}^2$ to $(-3.5 \pm 0.3) \times 10^{-19} \text{ m}^2/\text{V}^2$ at 1,560 nm and $(-16.5 \pm 0.7) \times 10^{-19} \text{ m}^2/\text{V}^2$ to $(-2.7 \pm 0.1) \times 10^{-19} \text{ m}^2/\text{V}^2$ at 1,900 nm, respectively. Furthermore, it was also shown that $\alpha\text{-As}_2\text{Te}_3$ can serve as a base material for near-infrared SAs to generate ultrafast mode-locked pulses that can cover a bandwidth of 1.5–1.9 μm . The electric band structure analysis showed that the operating bandwidth was expandable to $\sim 3.7 \mu\text{m}$, indicating that it can cover a wide range of wavelength, including the mid-infrared region.

Arsenic telluride is known to be a good thermoelectric material but is still an unknown material in the field of optics and photonics. We believe that our work reveals the significant potential of $\alpha\text{-As}_2\text{Te}_3$ in the field of optics and photonics. Furthermore, our work is believed to provide a meaningful database for nonlinear optical materials in the fields of lasers and photonics and suggest a new, nonlinear saturable absorption material for the implementation of ultrafast lasers.

Methods

Analysis. The elemental composition of a prepared sample was analyzed using an EDS measurement (VEGA3, TESCAM). Raman spectroscopy measurement was performed using a LabRam Aramis (Horiba Jovin Yvon) at room temperature under excitation by 532 nm laser with output power of ~0.5 mW. XPS measurements were carried out on a K-alpha (Thermo Scientific Inc., UK) using an Al Ka μ -focused monochromator (1,486.6 eV) with a 400 μ m spot size and energy step size of 0.1 eV. The FWHM of energy resolution is 0.7 eV measured in the Ag 3d^{5/2} peaks, and an 180° double focusing hemispherical analyzer with a 128-channel detector was employed. The energy range was from 100 to 4,000 eV, and the base pressure was 2.9×10^{-9} mbar.

Density functional theory calculations. We performed the DFT calculation with spin-orbit coupling by employing gradient-density approximation and Perdew–Burke–Ernzerh of exchange–correlation parameterization as implemented in the Atomistic Toolkit package^{116,117}. The density mesh cutoff was 150 hartree and the $2 \times 7 \times 3$ Monkhorst–Pack grid was used for k-space sampling to calculate the electronic band structure and dielectric constant of α -As₂Te₃. To obtain optical absorption spectrum, the susceptibility tensor was calculated using the Kubo–Greenwood formula shown below¹¹⁸:

$$\chi_{ij}(\omega) = -\frac{e^2 \hbar^4}{m^2 \varepsilon_0 V \omega^2} \sum_{n,m} \frac{f(E_m) - f(E_n)}{E_{nm} - \hbar\omega - i\hbar\Gamma} \pi_{nm}^i \pi_{nm}^j \quad (4)$$

where f is the Fermi distribution function, Γ is the broadening parameter, π_{nm}^i is the i -component of the dipole matrix element between states n and m , and V is the volume of the considered system. The frequency-dependent dielectric constant was estimated using the relation of $\varepsilon(\omega) = 1 + \chi(\omega)$ and the optical absorption spectrum was finally obtained from the imaginary part of the dielectric constants.

Linear absorption measurement. The linear absorption of the α -As₂Te₃/PVP composite was measured by using a spectrophotometer (UV-3600PLUS, Shimadzu) from 1000 to 3300 nm.

Z-scan measurement. As the input light source, homemade mode-locked fiber lasers operating at 1.56 and 1.9 μ m were used. The beam splitter was used to split the input beam into two parts. One of the two parts was directed to a power meter to monitor the reference beam (Power meter 1), while the other was focused through a plan-convex lens and is vertically directed to the α -As₂Te₃ sample. The α -As₂Te₃ sample was placed on the translation stage and moved gradually across in the propagation direction. The transmittance through the sample was measured using a power meter (Power meter 2)⁵³.

Nonlinear saturable absorption measurement. Homemade mode-locked fiber lasers (center wavelength = 1.56 μ m, repetition rate = 22.26 MHz, and temporal width = 730 fs/center wavelength = 1.9 μ m, repetition rate = 36.94 MHz, and temporal width = 703 fs) are used to measure the nonlinear transmission curve of the prepared SA. A fiber-based variable optical attenuator was used to adjust the optical power of the input mode-locked pulses. An input mode-locked pulses were divided into two ports using a 3-dB coupler. One of the two ports was connected to the prepared α -As₂Te₃/PVP-based SA, while the other was directly connected to a power meter to monitor the input power of the SA. Another power meter was used to measure the output power from the α -As₂Te₃/PVP-based SA for comparison with the input power^{48,53}.

Fabrication of α -As₂Te₃-based saturable absorber. Bulk As₂Te₃ crystals were grinded in a mortar to obtain α -As₂Te₃ powders. The α -As₂Te₃ particle solution was prepared using a sonication in a distilled water without centrifugation after ultrasonication for 8 h. Then, the PVP was mixed into the α -As₂Te₃ particle solution. A small amount of α -As₂Te₃/PVP composite solution was dropped on an FC/APC fiber ferrule and then dried at room temperature for 8 h¹¹⁹.

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References

- Asobe, M., Kanamori, T. & Kubodera, K. Ultrafast all-optical switching using highly nonlinear chalcogenide glass fiber. *IEEE Photon. Technol. Lett.* **4**, 362–365 (1992).
- Li, L., Abdurkerim, N. & Rochette, M. Mid-infrared wavelength conversion from As₂Se₃ microwires. *Opt. Lett.* **42**, 639–642 (2017).
- Langrock, C., Kumar, S., McGeehan, J. E., Willner, A. E. & Fejer, M. M. All-optical signal processing using $\chi^{(2)}$ nonlinearities in guided-wave devices. *J. Lightw. Technol.* **24**, 2579–2592 (2006).
- Set, S. Y., Yaguchi, H., Tanaka, Y. & Jablonski, M. Laser mode locking using a saturable absorber incorporating carbon nanotubes. *J. Lightw. Technol.* **22**, 51–56 (2014).
- Makarov, S. V. *et al.* Efficient second-harmonic generation in nanocrystalline silicon nanoparticles. *Nano Lett.* **17**, 3047–3053 (2017).
- Lee, J. H. *et al.* Experimental comparison of a Kerr nonlinearity figure of merit including the stimulated Brillouin scattering threshold for state-of-the-art nonlinear optical fibers. *Opt. Lett.* **30**, 1698–1700 (2005).
- Rumi, M. & Perry, J. W. Two-photon absorption: An overview of measurements and principles. *Adv. Opt. Photon.* **2**, 451–518 (2010).
- Bao, Q. *et al.* Monolayer graphene as a saturable absorber in a mode-locked laser. *Nano Res.* **4**, 297–307 (2011).
- Lee, J. *et al.* Numerical investigation of the impact of the saturable absorber recovery time on the mode-locking performance of fiber lasers. *J. Lightw. Technol.* **1**, 20. <https://doi.org/10.1109/JLT.2020.2985718> (2020).

10. Manolatu, C. & Lipson, M. All-optical silicon modulators based on carrier injection by two-photon absorption. *J. Lightw. Technol.* **24**, 1433–1439 (2006).
11. Zitter, R. N. Saturated optical absorption through band filling in semiconductors. *Appl. Phys. Lett.* **14**, 73 (1969).
12. Sugioka, K. & Cheng, Y. Ultrafast lasers-reliable tools for advanced materials processing. *Light Sci. Appl.* **3**, e149 (2014).
13. Ploder, K. & Bruce, S. Treatment of melisma using a novel 1,927-nm fractional thulium fiber laser: A pilot study. *Dermatol. Surg.* **38**, 199–206 (2012).
14. De Young, R. J. & Barnes, N. P. Profiling atmospheric water vapor using a fiber laser lidar system. *Appl. Opt.* **49**, 562–567 (2010).
15. Guilhot, D. & Ribes-Pleguezuelo, P. Laser technology in photonic applications for space. *Instruments* **3**, 55 (2019).
16. Keller, U. *et al.* Semiconductor saturable absorber mirrors (SESAMs) for femtosecond to nanosecond pulse generation in solid-state lasers. *IEEE J. Sel. Top. Quantum Electron.* **2**, 435–453 (1996).
17. Solodyankin, M. A. *et al.* Mode-locked 1.93 μm thulium fiber laser with a carbon nanotube absorber. *Opt. Lett.* **33**, 1336–1338 (2008).
18. Kieu, K. & Wise, F. W. “Soliton thulium-doped fiber laser with carbon nanotube saturable absorber. *IEEE Photon. Technol. Lett.* **21**, 128–130 (2009).
19. Chernysheva, M. A. *et al.* Thulium-doped mode-locked all-fiber laser based on NALM and carbon nanotube saturable absorber. *Opt. Express* **20**, B124–B130 (2012).
20. Wang, F. *et al.* Wideband-tuneable, nanotube mode-locked, fiber laser. *Nat. Nanotechnol.* **3**, 738–742 (2008).
21. Bao, Q. *et al.* Atomic-layer graphene as a saturable absorber for ultrafast pulsed laser. *Adv. Funct. Mater.* **19**, 3077–3083 (2009).
22. Zhang, M. *et al.* Tm-doped fiber laser mode-locked by graphene-polymer composite. *Opt. Express* **20**, 25077–25084 (2012).
23. Sobon, G. *et al.* Thulium-doped all-fiber laser mode-locked by CVD-graphene/PMMA saturable absorber. *Opt. Express* **21**, 127971–127976 (2013).
24. Wang, Q. *et al.* All-fiber passively mode-locked thulium-doped fiber ring laser using optically deposited graphene saturable absorber. *Appl. Phys. Lett.* **102**, 131117 (2013).
25. Sobon, G. *et al.* All-polarization maintaining, graphene-based femtosecond Tm-doped all-fiber laser. *Opt. Express* **23**, 9339–9346 (2015).
26. Wang, G. *et al.* Ultrafast nonlinear optical properties of a graphene saturable mirror in the 2 μm wavelength region. *Laser Photon. Rev.* **11**, 1700166 (2017).
27. Ko, S. *et al.* Chemical wet etching of an optical fiber using a hydrogen fluoride-free solution for a saturable absorber based on the evanescent field interaction. *J. Lightw. Technol.* **34**, 3776–3784 (2016).
28. Jung, M. *et al.* Mode-locked pulse generation from an all-fiberized, Tm-Ho-codoped fiber laser incorporating a graphene oxide-deposited side-polished fiber. *Opt. Express* **21**, 20062–20072 (2013).
29. Zhao, C. *et al.* Ultra-short pulse generation by a topological insulator based saturable absorber. *Appl. Phys. Lett.* **101**, 211106 (2012).
30. Luo, Z. *et al.* Topological-insulator passively Q-switched double-clad fiber laser at 2 μm wavelength. *IEEE J. Sel. Top. Quantum Electron.* **20**, 0902708 (2014).
31. Lee, J., Koo, J., Jhon, Y. M. & Lee, J. H. A femtosecond pulse erbium fiber laser incorporating a saturable absorber based on a bulk-structured Bi_2Te_3 topological insulator. *Opt. Express* **22**, 6165–6173 (2014).
32. Jung, M. *et al.* A femtosecond pulse fiber laser at 1935 nm using a bulk-structured Bi_2Te_3 topological insulator. *Opt. Express* **22**, 7865–7874 (2014).
33. Liu, H. *et al.* Femtosecond pulse generation from a topological insulator mode-locked fiber laser. *Opt. Express* **22**, 6868–6873 (2014).
34. Yin, K. *et al.* Soliton mode-locked fiber laser based on topological insulator Bi_2Te_3 nanosheets at 2 μm . *Photon. Res.* **3**, 72–76 (2015).
35. Lin, Y.-H. *et al.* Using n- and p- type Bi_2Te_3 topological insulator nanoparticles to enable controlled femtosecond mode-locking of fiber lasers. *ACS Photon.* **2**, 481–490 (2015).
36. Qiao, J. *et al.* High-quality 2- μm Q-switched pulsed solid-state lasers using spin-coating-corededuction approach synthesized Bi_2Te_3 topological insulators. *Photon. Res.* **6**, 314–320 (2018).
37. Wang, Y.-R. *et al.* Optical nonlinearity engineering of a bismuth telluride saturable absorber and application of a pulsed solid state laser therein. *Nanoscale* **9**, 19100–19107 (2017).
38. Lee, J., Kim, Y., Lee, K. & Lee, J. H. Femtosecond mode-locking of a fiber laser using a CoSb_3 -skutterudite-based saturable absorber. *Photon. Res.* **6**, C36–C43 (2018).
39. Zhu, C. *et al.* A robust and tuneable mid-infrared optical switch enabled by bulk Dirac fermions. *Nat. Commun.* **8**, 14111 (2017).
40. Zhang, H. *et al.* Molybdenum disulfide (MoS_2) as a broadband saturable absorber for ultra-fast photonics. *Opt. Express* **22**, 7249–7260 (2014).
41. Liu, H. *et al.* Femtosecond pulse erbium-doped fiber laser by a few-layer MoS_2 saturable absorber. *Opt. Lett.* **39**, 4591–4594 (2014).
42. Wu, K., Zhang, X., Wang, J., Li, X. & Chen, J. WS_2 as a saturable absorber for ultrafast photonic applications of mode-locked and Q-switched lasers. *Opt. Express* **23**, 11453–11461 (2015).
43. Li, S. *et al.* A microfiber knot incorporating a tungsten disulfide saturable absorber based multi-wavelength mode-locked erbium-doped fiber laser. *J. Lightw. Technol.* **36**, 5633–5639 (2018).
44. Mao, D. *et al.* Erbium-doped fiber laser passively mode locked with few-layer $\text{WSe}_2/\text{MoSe}_2$ nanosheets. *Sci. Rep.* **6**, 23583 (2015).
45. Luo, Z. *et al.* Nonlinear optical absorption of few-layer molybdenum diselenide (MoSe_2) for passively mode-locked soliton fiber laser [Invited]. *Photon. Res.* **3**, A79–A86 (2015).
46. Jung, M. *et al.* Mode-locked, 1.94- μm , all-fiberized laser using WS_2 -based evanescent field interaction. *Opt. Express* **23**, 19996–20006 (2015).
47. Wang, J. *et al.* Ultrafast thulium-doped fiber laser mode locked by monolayer WSe_2 . *IEEE Sel. Top. Quantum Electron.* **24**, 1100706 (2018).
48. Lee, J., Koo, J., Lee, J., Jhon, Y. M. & Lee, J. H. All-fiberized, femtosecond laser at 1912 nm using a bulk-like MoSe_2 saturable absorber. *Opt. Mater. Express* **7**, 2968–2979 (2017).
49. Mao, D. *et al.* Nonlinear saturable absorption of liquid-exfoliated molybdenum/tungsten ditelluride nanosheets. *Small* **12**, 1489–1497 (2016).
50. Koo, J. *et al.* Near-infrared saturable absorption of defective bulk-structured WTe_2 for femtosecond laser mode-locking. *Adv. Funct. Mater.* **26**, 7454–7461 (2016).
51. Wang, J. *et al.* Magnetron-sputtering deposited WTe_2 for an ultrafast thulium-doped fiber laser. *Opt. Lett.* **42**, 5010–5013 (2017).
52. Huang, B. *et al.* Bulk-structured PtSe_2 for femtosecond fiber laser mode-locking. *Opt. Express* **27**, 2604–2611 (2019).
53. Lee, L., Lee, K., Kwon, S., Soo, B. & Lee, J. H. Investigation of nonlinear optical properties of rhenium diselenide and its application as a femtosecond mode-locker. *Photon. Res.* **7**, 984–993 (2019).
54. Jhon, Y. I., Lee, J., Seo, M., Lee, J. H. & van der Jhon, Y. M. Waals layered tin selenide as highly nonlinear ultrafast saturable absorber. *Adv. Opt. Mater.* **7**, 1801745 (2019).
55. Lee, J. *et al.* Filled skutterudites broadband saturable absorbers. *Adv. Opt. Mater.* **5**, 1700096 (2017).

56. Park, K. *et al.* Black phosphorus saturable absorber for ultrafast mode-locked pulse laser via evanescent field interaction. *Ann. Phys.* **527**, 770–776 (2015).
57. Sotor, J. *et al.* Ultrafast thulium-doped fiber laser mode locked with black phosphorus. *Opt. Lett.* **40**, 3885–3888 (2015).
58. Hanlon, D. *et al.* Liquid exfoliation of solvent-stabilized few-layer black phosphorus for applications beyond electronics. *Nat. Commun.* **6**, 8563 (2015).
59. Jiang, T. *et al.* Passively Q-switching induced by gold nanocrystals. *Appl. Phys. Lett.* **101**, 151122 (2012).
60. Wang, X.-D. *et al.* Microfiber-based gold nanorods as saturable absorber for femtosecond pulse generation in a fiber laser. *Appl. Phys. Lett.* **105**, 161107 (2014).
61. Kang, Z. *et al.* Mode-locked thulium-doped fiber laser at 1982 nm by using a gold nanorods saturable absorber. *Laser Phys. Lett.* **12**, 045105 (2015).
62. Jhon, Y. I. *et al.* Metallic MXene saturable absorber for femtosecond mode-locked lasers. *Adv. Mater.* **29b**, 1702496 (2017).
63. Jiang, X. *et al.* Broadband nonlinear photonics in few-layer MXene $\text{Ti}_3\text{C}_2\text{T}_x$ ($T = \text{F, O, or OH}$). *Laser Photon. Rev.* **12**, 1700229 (2018).
64. Carron, G. J. The crystal structure and powder data for arsenic telluride. *Acta. Cryst.* **16**, 338–343 (1963).
65. Deng, H. Theoretical prediction of the structural, electronic, mechanical and thermodynamic properties of the binary $\alpha\text{-As}_2\text{Te}_3$ and $\beta\text{-As}_2\text{Te}_3$. *J. Alloys Compd.* **656**, 695–701 (2016).
66. Scheidemantel, T. J., Meng, J. F. & Badding, J. V. Thermoelectric power and phase transition of polycrystalline As_2Te_3 under pressure. *J. Phys. Chem. Solids* **66**, 1744–1747 (2005).
67. Scheidemantel, T. J. & Badding, J. V. Electronic structure of $\beta\text{-As}_2\text{Te}_3$. *Solid State Commun.* **127**, 667–670 (2003).
68. Vaney, J.-B. *et al.* High-temperature thermoelectric properties of Sn-doped $\beta\text{-As}_2\text{Te}_3$. *Adv. Electron. Mater.* **1**, 1400008 (2015).
69. Pal, K. & Waghmare, U. V. Strain induced Z_2 topological insulating state of $\beta\text{-As}_2\text{Te}_3$. *Appl. Phys. Lett.* **105**, 062105 (2014).
70. Pal, K., Anand, S. & Waghmare, U. V. Thermoelectric properties of materials with nontrivial electronic topology. *J. Mater. Chem. C* **3**, 12130 (2015).
71. Nolas, G. S., Sharp, J. & Goldsmid, J. *Thermoelectrics; Basic Principles and New Materials Development* (Springer, Berlin, 2001).
72. Ovsyannikov, S. V. & Shchennikov, V. V. High-pressure routes in the thermoelectricity or how one can improve a performance of thermoelectrics. *Chem. Mater.* **22**, 635–647 (2010).
73. Platakis, N. S. & Gatos, H. C. Threshold and memory switching in crystalline chalcogenide materials. *Phys. Status Solidi. A* **13**, K1–K4 (1972).
74. Badalov, S. V., Kandemir, A. & Sahin, H. Monolayer AsTe_2 : Stable robust metal in 2D, 1D and 0D. *ChemPhysChem* **19**, 2176–2182 (2018).
75. Cuenca-Gotor, V. P. *et al.* Structural, vibrational, and electronic study of $\alpha\text{-As}_2\text{Te}_3$ under compression. *J. Phys. Chem. C* **120**, 19340–19352 (2016).
76. Bahl, M. K., Woodall, R. O., Watson, R. L. & Irgolic, K. J. Relaxation during photoemission and LMM Auger decay in arsenic and some its compounds. *J. Chem. Phys.* **64**, 1210 (1976).
77. Taylor, J. A. An XPS study of the oxidation of ALAs thin films grown by MBE. *J. Vac. Sci. Technol.* **20**, 751 (1982).
78. Bando, H. *et al.* The time-dependent process of oxidation of the surface of Bi_2Te_3 studied by X-ray photoelectron spectroscopy. *J. Phys. Condens. Matter* **12**, 5607–5616 (2000).
79. Popescu, M. A. *Non-crystalline Chalcogenides* (Springer, Berlin, 2000).
80. Vaney, J. B. Thermoelectric properties of the $\alpha\text{-As}_2\text{Te}_3$ crystalline phase. *J. Electron. Mater.* **45**, 1447–1452 (2016).
81. Li, S. *et al.* Gold nanowires with surface plasmon resonance as saturable absorbers for passively Q-switched fiber laser at $2\ \mu\text{m}$. *Opt. Mater. Express* **9**, 2406–2414 (2019).
82. Koczur, K. M., Mourdikoudis, S., Polavarapu, L. & Skrabalak, S. E. Polyvinylpyrrolidone (PVP) in nanoparticle syntheses. *Dalton Trans.* **44**, 17883 (2015).
83. Bhavsar, V. & Tripathi, D. Structural, optical, and aging studies of biocompatible PVC-PVP blend films. *J. Polym. Eng.* **38**, 419–426 (2018).
84. Yazyev, O. V., Kioupakis, E., Moore, J. E. & Louie, S. G. Quasiparticle effect in the bulk and surface-state bands of Bi_2Se_3 and Bi_2Te_3 topological insulators. *Phys. Rev. B* **85**, 161101 (2012).
85. Reid, T. K., Alpay, S. P., Balatsky, A. V. & Nayak, S. K. First-principles modeling of binary layered topological insulators: Structural optimization and exchange-correlation functionals. *Phys. Rev. B* **101**, 085140 (2020).
86. Tran, V., Soklaski, R., Liang, Y. & Yang, L. Layer-controlled band gap and anisotropic excitons in few-layer black phosphorus. *Phys. Rev. B* **89**, 235319 (2014).
87. Kim, Y. *et al.* Anomalous Raman scattering and lattice dynamics in mono- and few-layer WTe_2 . *Nanoscale* **8**, 2309 (2016).
88. Ansari, L. *et al.* Quantum confinement-induced semimetal-to-semiconductor evolution in large-area ultra-thin PtSe_2 films grown at $400\ ^\circ\text{C}$ Npj 2D. *Mater. Appl.* **3**, 33 (2019).
89. Zhao, Z.-Y. & Liu, Q.-L. Study of the layer-dependent properties of MoS_2 nanosheets with different crystal structures by DFT calculations. *Catal. Sci. Technol.* **8**, 1867 (2018).
90. Sheik-Bahae, M., Said, A. A., Wei, T.-H., Hagan, D. J. & Van Stryland, E. W. Sensitive measurement of optical nonlinearities using a single beam. *IEEE J. Quantum Electron.* **26**, 760–769 (1990).
91. Ge, Y. *et al.* Broadband nonlinear photoreponse of 2D TiS_2 for ultrashort pulse generation and all-optical thresholding devices. *Adv. Opt. Mater.* **6**, 1701166 (2018).
92. Cheng, C., Li, Z., Dong, N., Wang, J. & Chen, F. Tin diselenide as a new saturable absorber for generation of laser pulses at $1\ \mu\text{m}$. *Opt. Express* **25**, 6131–6140 (2017).
93. Pang, C. *et al.* Lithium niobate crystal with embedded Au nanoparticles: A new saturable absorber for efficient mode-locking of ultrafast laser pulses at $1\ \mu\text{m}$. *Adv. Opt. Mater.* **6**, 1800357 (2018).
94. Liu, J., Liu, S. & Wei, J. Origin of the giant optical nonlinearity of Sb_2Te_3 phase change materials. *Appl. Phys. Lett.* **97**, 261903 (2010).
95. Xu, Y. *et al.* Bilayer bismuth selenide nanoplatelets based saturable absorber for ultra-short pulse generation (Invited). *Opt. Commun.* **395**, 55–60 (2017).
96. Wang, Y. *et al.* Observation of large nonlinear responses in a graphene- Bi_2Te_3 heterostructure at a telecommunication wavelength. *Appl. Phys. Lett.* **108**, 221901 (2016).
97. Wang, Y. *et al.* Ultra-broadband nonlinear saturable absorption for two-dimensional $\text{Bi}_2\text{Te}_3\text{Se}_{3-x}$ nanosheets. *Sci. Rep.* **6**, 33070 (2016).
98. Liu, Z.-B., Yan, X.-Q., Tian, J.-G., Zhou, W.-Y. & Zang, W.-P. Nonlinear ellipse rotation modified Z-scan measurements of third-order nonlinear susceptibility tensor. *Opt. Express* **15**, 13351–13359 (2007).
99. Umarani, P., Kalainathan, S. & Jagannathan, K. Third order nonlinear properties of pure and Na doped L-arginine single crystal by Z-scan technique. *Orient. J. Chem.* **32**, 213–218 (2016).
100. Sandeep, K. M., Shreeha, B., Dharmaparakash, S. M., Patil, P. S. & Byrappa, K. Defect assisted saturable absorption characteristics in Al and Li doped ZnO thin films. *J. Appl. Phys.* **120**, 123107 (2016).
101. Torres-Torres, C. *et al.* Coexistence of two-photon absorption and saturable absorption in ion-implanted platinum nanoparticles in silica plates. *J. Opt. Soc. Am. B* **35**, 1295–1300 (2018).

102. Gaur, A., Syed, H., Yendeti, B. & Soma, V. R. Experimental evidence of two-photon absorption and its saturation in malachite green oxalate: A femtosecond Z-scan study. *J. Opt. Soc. Am. B* **35**, 2906–2914 (2018).
103. Wang, J., Gu, B., Wang, H.-T. & Ni, X.-W. Z-scan analytical theory for material with saturable absorption and two-photon absorption. *Opt. Commun.* **283**, 3525–3528 (2010).
104. Qiao, J. *et al.* Two-photon absorption within layered Bi₂Te₃ topological insulators and the role of nonlinear transmittance therein. *J. Mater. Chem. C* **7**, 7027–7034 (2017).
105. Yang, H. *et al.* Giant two-photon absorption in bilayer graphene. *Nano Lett.* **11**, 2622–2627 (2011).
106. Gnoli, A., Razzari, L. & Righini, M. Z-scan measurements using high repetition rate lasers: How to manage thermal effects. *Opt. Express* **13**, 7976–7981 (2005).
107. He, W.-Q., Gu, C.-M. & Shen, W.-Z. Direct evidence of Kerr-like nonlinearity by femtosecond Z-scan technique. *Opt. Express* **14**, 5476–5483 (2006).
108. Garcia-Ramirez, E. V. *et al.* Third-order nonlinear optical properties of colloidal Au nanorods systems: Saturable and reverse-saturable absorption. *Opt. Express* **24**, A154–A167 (2015).
109. Cesca, T. *et al.* Wavelength- and polarization-dependent nonlinear optical properties of plasmonic nanoprisms arrays. *Proc. SPIE* **9894**, 98941A (2016).
110. Cesca, T. *et al.* Dichroic nonlinear absorption response of silver nanoprisms arrays. *RSC Adv.* **7**, 17741 (2017).
111. de Araújo, C. B., Gomes, A. S. L. & Boudebs, G. Techniques for nonlinear optical characterization of materials: A review. *Rep. Prog. Phys.* **79**, 036401 (2016).
112. del Coso, R. & Solis, J. Relation between nonlinear refractive index and third-order susceptibility in absorbing media. *J. Opt. Soc. Am. B* **21**, 640–644 (2004).
113. Solati, E. & Dorrani, D. Nonlinear optical properties of the mixture of ZnO nanoparticles and graphene nanosheets. *Appl. Phys. B* **122**, 76 (2016).
114. Wu, K. *et al.* High-performance mode-locked and Q-switched fiber lasers based on novel 2D materials for topological insulators, transition metal dichalcogenides and black phosphorus: Review and perspective (invited). *Opt. Commun.* **406**, 214–229 (2018).
115. Kelly, S. M. J. Characteristic sideband instability of periodically amplified average soliton. *Electron. Lett.* **28**, 806–807 (1992).
116. Perdew, J. P., Burke, K. & Ernzerhof, M. Generalized gradient approximation made simple. *Phys. Rev. Lett.* **77**, 3865–3868 (1996).
117. Atomistix Toolkit version 205.3, Quantum Wise A/S, Copenhagen, Denmark.
118. Harrison, W. A. *Solid State Theory* (McGraw-Hill Press, New York, 1970).
119. Lee, J., Lee, J., Koo, J., Chung, H. & Lee, J. H. Linearly polarized, Q-switched, erbium-doped fiber laser incorporating a bulk-structured bismuth telluride/polyvinyl alcohol saturable absorber. *Opt. Eng.* **55**, 076109 (2016).

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Author contributions

J.L. and J.H.L. conceived the idea of the experiment, J.L. and K.L. conducted the measurements, Y.I.J. and Y.M.J. performed the numerical simulations. J.L., K.L. and J.H.L. analyzed the results, J.L., Y.I.J., and J.H.L. wrote the manuscript. All authors reviewed the manuscript.

Competing interests

The authors declare no competing interests.

Additional information

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