SCIENTIFIC REPORTS

natureresearch

OPEN Cerium/diethyldithiocarbamate complex as a novel corrosion inhibitive pigment for AA2024-T3

Iman Mohammadi¹, Taghi Shahrabi¹, Mohammad Mahdavian² & Mazdak Izadi¹

In this work, cerium-diethyldithiocarbamate (Ce-DEDTC) complex was synthesized as a novel anticorrosion pigment. The structure of the synthesized pigment was characterized by employing Fourier transfer infrared spectroscopy, X-ray diffraction, thermogravimetric analysis, inductively coupled plasma optical emission spectroscopy, and ultraviolet-visible spectroscopy. All of the characterization techniques showed that the Ce-DEDTC pigment was successfully produced. The electrochemical tests were used to investigate the subsequence effect of the synthesized complex on the corrosion behavior of the AA2024-T3. AA2024-T3 showed a wide passive range in the presence of the Ce-DEDTC pigment. Scanning electron microscopy, optical microscopy, X-ray photoelectron spectroscopy, and contact angle tests were employed to investigate the effect of the synthesized pigment on aluminum surface properties. The result illustrated that the existence of the Ce-DEDTC complex led to the creation of a thin film on the AA2024-T3 surface, which was significantly inhibited the localized corrosion of the aluminum alloy.

Due to high strength to weight ratio and low density, aluminum, and its alloys have found various industrial applications¹⁻³. AA2024-T3 which is the precipitation hardening alloy, has various engineering applications in aeronautics and aerospace industries⁴. S-phase Al₂MgCu and Al-Cu-Fe-Mn are the main intermetallic compounds in the structure of the 2024-T3 aluminum alloy^{4,5}. In the presence of these intermetallic compounds, the substrate is locally degraded, especially in the presence of halide species⁵. The utilization of coatings, especially organic coatings, is a promising method to improve the corrosion performance of aluminum alloy against aggressive solutions⁶. But, it is well-known that the organic coatings alone don't have the long-term protective behavior against the aggressive ions. Due to the permeation of the oxygen, water, and corrosive ions through the organic coatings, as the main corrosive species for the metallic substrates, the metal is degraded even in the presence of the organic coatings⁷. Therefore, for improving the ability of the organic coatings against the destructive species, the use of other materials is a requirement.

Inhibitive pigments are the main additives used to improve the corrosion protection functioning of the organic coatings in corrosive media. Pigments are utilized in different types and they have different protection mechanisms⁸⁻¹⁰. It is generally believed that the pigments are the fine particle size and insoluble materials used as colloidal dispersions which are added to the organic coatings for various purposes¹¹. Providing the color, modifying the applications of the coatings, reducing the cost and improving the performance of the coatings are the main pigment applications¹¹. Enhancing the organic coatings corrosion inhibition is the most important application of the pigments. Although it was mentioned that the pigments are the insoluble materials, the corrosion-inhibiting pigments should be somewhat soluble in water¹².

Chromium-based compounds are the main inorganic pigments, which have the capability to release the corrosion inhibitors providing anti-corrosion performance of the organic coatings. It is well known that these complexes release the hexavalent species in contact with water. In this situation, the metallic substrate passivates, and a thin film is formed by the reaction of the base substrate and the chromium-containing inhibitors. Therefore, the corrosion properties of the metallic substrates are significantly improved by employing the chromium-based pigments¹³⁻¹⁵. Although chromium-based systems still remain as a low-cost and effective source for improving the corrosion properties of the metallic substrates in the aeronautics and aerospace industry, the toxicity and carcinogenic effects of these compounds have begun severe restrictions on their use by international environmental standards¹⁶.

¹Department of Materials Engineering, Faculty of Engineering, Tarbiat Modares University, P. O. Box 14115-143, Tehran, Iran. ²Surface Coatings and Corrosion Department, Institute for Color Science and Technology, Tehran, Iran. *email: tshahrabi34@modares.ac.ir; mahdavian-m@icrc.ac.ir

Due to the limitation of the chromium-based compounds, various pigments have been employed to improve the corrosion performance of metallic substrates^{17–21}. As a result, the research works are being sought to find a more benign corrosion prevention technologies based on eco-friendly pigments^{12,22–24}. Rare earth metal compounds are suggested as alternative materials for chromates to improve the corrosion properties of the metallic substrates^{16,25–31}. Among all of the rare earth compounds, it has been shown that the cerium-based compounds are the most effective materials to provide aluminum alloy protection performance against corrosion^{30,32–35}. It has also been shown that organic inhibitors can be engaged as potentially more benign alternatives. In many cases, these compounds need much higher quantities to be operative^{36–38}. To decrease the amount of organic inhibitors and also improve the corrosion performance of these species, combination of the rare earth metals and organic ligands can create a multifunctional corrosion inhibition pigment^{37,39–42}. It this situation, the mixture of an organic compound and a rare earth metal might provide superior corrosion properties in comparison with the individual constituents at the same concentration^{16,34,41–44}.

Due to the presence of the thiol group in the structure of the DEDTC, which also was found in very active corrosion inhibitors, this compound have a potential application to be used as an inexpensive corrosion inhibitor for the AA2024-T3⁴⁵⁻⁴⁷. Ferrer *et al.*⁴⁸ used the NaY zeolite particles double-doped with cerium and diethyldith-iocarbamate (DEDTC) to improve the corrosion inhabitation of the AA2024-T3 in saline solution. Their results showed that double doping process improved anticorrosive behavior of AA2024-T3 in aqueous saline solution and also suggested potential synergetic effects between Ce and DEDTC.

In the present study, the cerium-diethyldithiocarbamate (Ce-DEDTC) complex was synthesized as a novel inorganic-organic pigment to improve the corrosion protection performance of the 2024-T3 aluminum alloy. For this purpose, a simple blending method was used to synthesize the Ce-DEDTC complex. Also, X-ray diffraction (XRD), thermogravimetric, inductively coupled plasma optical emission spectroscopy (ICP-OES), ultraviolet-visible spectroscopy, and Fourier transformation infrared spectroscopy (FT-IR) analyses were used for characterization of the produced pigment. Potentiodynamic polarization (PDS), and electrochemical impedance spectroscopy (EIS) tests were employed to evaluate the subsequence effect of the synthesized pigments on the electrochemical behavior of the AA2024-T3. To study the surface changes in the presence of the synthesized pigments, scanning electron microscopy equipped with energy dispersive spectroscopy (SEM-EDS), optical microscopy (OM), X-Ray photoelectron spectroscopy (XPS), and water droplet contact angle (CA) tests were employed.

Materials and methods

Materials and synthesis procedure. Cerium (III) nitrate hexahydrate ($Ce(NO_3)_3.6H_2O$), and sodium diethyldithiocarbamate ($C_5H_{10}NS_2Na$) were provided from the Merck Co. Deionized water, NaCl, acetone, and 2024-T3 aluminum alloy were supplied from local companies. The chemical composition of the aluminum alloy was: 0.45% Fe, 0.45% Si, 1.24–1.85% Mg, 0.32–0.94% Mn, 3.88–4.96% Cu, and balanced Al. In order to synthesize the Ce-DEDTC pigments, 5 g/lit (22.2 mM) of DEDTC and 6.55 g/lit (15 mM) of Ce(NO₃)₃.6H₂O were added to the deionized water in the unpressurized reactor. After that, the mixture was stirred by a magnetic stirrer at 50 °C for 24 h. Then, the solid particles were collected by a centrifuged (6000 rpm) apparatus and rinsed 3 times with deionized water to ensure about the elimination of the untreated species. For removing the physical absorption of the water molecules, the produced Ce-DEDTC pigment was dried at 60 °C for 12 h.

Characterization of the synthesized pigments. The thermogravimetric analysis (TGA) was performed using the PerkinElmer apparatus in the nitrogen atmosphere to study the amount of the organic and inorganic compounds in the structure of the synthesized pigments. In this measurement, the temperature range was 25–700 °C, and the heat flow was 10 °C/min. Fourier transformation infrared spectroscopy (FT-IR) (SHIMADZU) in the region of 4000–400 cm⁻¹ and X-ray diffraction (XRD) (X'Pert Pro MPD-PW3040/60 with Cu K α generated at 40 mA and 40 kV) studies were used to assess the chemical composition and crystalline phases of the synthesized pigments, respectively. Furthermore, inductively coupled plasma optical emission spectroscopy (ICP-OES/VISTA-PRO/Varian-Inc) and ultraviolet-visible spectroscopy (UV-vis/OPTIZEN 3220) in the wavenumber of 190–500 nm were employed to study the extent of released inorganic and organic compounds from the hybrid pigment, respectively.

Electrochemical tests. To investigate the electrochemical behavior of the AA2024-T3 in the presence of synthesized pigment, the plates were cut in the dimensions of $50 \times 30 \times 20$ mm. Then, the metallic substrate mechanically polished by the SiC emery paper from the grade of 220 to 1000. To ensure the complete degreasing of the AA2024-T3 samples, they were ultra-sonicated in acetone for 10 min. Finally, the samples rinsed by the mixture of the water and ethanol and dried by the nitrogen flow. In order to evaluate the effect of synthesized pigments on the electrochemical behavior of the samples, they were immersed in the aqueous extract solution of the Ce-DEDTC pigments. For this purpose, 5 g of the synthesized pigments were added to 11 of 3.5 wt.% NaCl solution and the mixture was mixed by a magnetic stirrer (500 rpm) for 24 h. The extract solution was provided after the removing of the solid particles using a centrifuge at 6000 rpm. In all of the electrochemical tests, a three electrodes cell was used in which the 1×1 cm of the samples was considered to be working electrode and the reminder of the surface isolated by the beeswax, the counter electrode was platinum, and the reference electrode was saturated calomel electrode (SCE). In the potentiodynamic polarization test, the sweep rate and the polarization range were 0.5 mV/s and +250 to -750 mV vs. open circuit potential (OCP), respectively. For the EIS measurements, a 10 mV sinusoidal perturbation signal was used in the frequency range of 10 mHz to 100 kHz. For the EIS tests, 5 points per frequency decade were recorded and the curves were fitted employing Zview software. The EIS experimental tests were repeated three times to ensure about the repeatability of the results.



Figure 1. The X-ray diffraction pattern of synthesized Ce-DEDTC pigment.



Figure 2. FT-IR spectra of the parent DEDTC and Ce-DEDTC pigment.

Surface studies. To study the surface changes during the immersion of the aluminum alloy in the presence of the synthesized pigments, the specimens cut in the dimension of 1×1 cm and prepared according to the preparation procedure which described for the electrochemical tests in the previous section. Then, the samples soaked in the Ce-DEDTC pigments extract solution, and the surface changes were recorded versus the soaking time. The corrosion products chemical composition and morphology were studied by employing the scanning electron microscopy (FEI ESEM Quanta 200) equipped energy dispersive spectroscopy (EDAX EDS Silicon drift 2017). To study the localized corrosion of the specimens, optical microscopy (OM/Olympus) images in various magnifications over soaking times were employed. The X-ray photoelectron spectroscopy (XPS/Specs EA 10 plus) test was used to study the film formation during immersion of samples in the extract solution of the pigments. An OCA 15 plus contact angle device was used to determine the static contact angles of the immersed samples over time.

Results and discussion

Characterization of the synthesized pigment. The XRD pattern of the produced pigment is presented in Fig. 1. Diffraction peaks at 13.46, 17.48, 19.48, 21.28, 24.64 and 28.52 clearly show the presence of the organic-inorganic hybrid complex. Due to the large atomic radius and low electro-negativity of the Ce^{3+} cations as the heavy metal, the sulfur atoms in the structure of the DEDTC shared the isolated electron pairs to form coordination bond^{49,50}. This mechanism was studied in various research works and has been showed that the reaction of the DEDTC and heavy metals can lead to the creation of $C_{10}H_{20}MN_2S_4$ complex (the structure of the complex is shown in the protection mechanism section), where M represents the metallic compound. This result was according to the theoretical analysis and experimental works which were carried out to investigate the structure of the carbamate based complex⁵⁰. The CeO₂ phase is observed in the XRD spectra of the synthesized pigments, which shows that the cerium oxide has been also formed.

The FT-IR spectra of the Ce-DEDTC pigment and parent DEDTC are illustrated in Fig. 2. Based on the structure, the IR spectra of the DEDTC can be interpreted by the vibration bands of the C-S, C-N, C=S, and also the C-H in methyl and methylene groups. The C-N vibration band in NCS_2^- group, appears at the wavenumber of 1450–1550 cm⁻¹⁵¹. Furthermore, the vibration band of the C-S in -CSS group and vibration bands of the C-H in the methyl and methylene groups are detected at the region of 950–1000 cm⁻¹ and 2800–3000 cm⁻¹, respectively^{20,51,52}. Based on the literature, the band appears at the regions of 400–500 cm⁻¹ is related to the vibration bands of the M-S, which M shows the metallic compound in the structure of the complex⁵¹. Also, it can be



Figure 3. The TG-DSC analysis of the parent DEDTC and Ce-DEDTC pigment.

mentioned that the dithiocarbamate group can be coordinated by the symmetric or asymmetric vibration bands which are assigned to the both or one of the sulfur atoms, respectively. It is well known that these frequency modes can be used as the diagnostic factors to determine the monodentate or bidentate mode of the dithiocarbamate^{53,54}. The existence of only one strong C–S vibration band can be due to the bidentate coordination of dithiocarbamate. In the other hands, a double peak for the C–S vibration band is anticipated in the case of monodentate coordination⁵⁵. The result of the FT-IR shows that the FT-IR spectrum of the DEDTC consists of a broad double peak in the wavenumber of 950–1000 cm⁻¹ which was related to the monodentate stretching vibration mode of the C–S band. Meanwhile, the Ce-DEDTC pigment has an intensive single peak associated with the bidentate stretching mode of the C–S band. Also, it can be observed that the stretching vibration frequency of the C–S band for the parent DEDTC appears at the wavenumber of 985.15 cm⁻¹ which decreased to 966.86 cm⁻¹ for the Ce-DEDTC complex. Based on the literature, this positive shift in this vibration frequency indicates the formation of chelate⁵⁴.

As can be seen in the FT-IR spectra of the specimens, the stretching vibration frequency of the C–N band in thioureide structure (S_2 C–N stretching mode) for the parent DEDTC and Ce-DEDTC complex are 1477.82 cm⁻¹ and 1497.7 cm⁻¹, respectively⁵¹. Also, as seen that the absorption peak value for the Ce-DEDTC complex is higher than that for the parent DEDTC. This result indicates that the increase in the contribution of the thioureide leads to electrons meso-meric drift from the DEDTC moiety towards the metal center. It also can be seen that the M-S peak in FT-IR spectra of the Ce-DEDTC pigment possesses a negative shift by moving to the lower wavenumber which shows the complex formation⁵¹.

The TG-DCS test in the dynamic inert atmosphere was used to evaluate the thermal behavior of the Ce-DEDTC pigment. The TG-DCS curves for the Ce-DEDTC complex and parent DEDTC are shown in Fig. 3. It is well known that the mass changes in the materials as a function of temperature can be measured by the TG test⁵⁶. While the DCS measures required heat to keep the sample and reference at the same temperature⁵⁷. The TG-DCS of the samples contains two main peaks. The endothermic peak appears at the temperature range of 80–120 and 80–280 °C for the Ce-DEDTC pigment and DEDTC, respectively. The endothermic peak represents the physically absorbed water in the DEDTC structure⁵⁰. The mass loss at this stage for the Ce-DEDTC complex is about 2%, while this value for the parent DEDTC is about 30%. As seen, the amount of absorbed water in DEDTC is higher than that in Ce-DEDTC pigment. The exothermic peak is appeared in the temperature range of 330 and 350 °C for the DEDTC and Ce-DEDTC pigment, respectively. The weight loss at around 230 °C can be attributed to the dihydroxylation of minor Ce(OH)₃ phase in the Ce-DEDTC. It is generally believed that this exothermic peak is associated with the thermal decomposition of the sample⁵⁰. Based on the results, the thermal stability of the DEDTC increased by the creation of Ce-DEDTC complex. The total mass loss for the DEDTC is about 73%, which decreased to 42% through bonding with Ce (Ce-DEDTC pigment). This result indicates that the total Ce in the structure of the Ce-DEDTC is more than 29%.

In order to study the partial solubility of the pigment, UV-vis analysis was used in various dissolving times. The UV-vis spectra for the Ce-DEDTC pigments dissolved in 3.5 wt.% NaCl electrolyte versus dissolution time are illustrated in Fig. 4. UV-vis spectra of the Ce-DEDTC pigment extract and DEDTC solution contain 3 main absorption bands. The first absorption band appears at 200–230 nm is associated with the π - π * transient of the sulfur nonbonding electrons^{51,58}. The second shoulder centered at 230–270 nm is related to the intra-ligand and $\pi \rightarrow \pi$ * transients found on N-C-S and within S-C=S groups^{51,58}. Furthermore, the n $\rightarrow \pi$ * electron transients sited on the sulfur atoms appeared at 270–300 nm^{51,58}. Based on Fig. 4, compared to the UV-Vis of the parent DEDTC, a negative shift is detected in the location of the first peak of the Ce-DEDTC complex. Also, it can be seen that the intensity of the first peak increased for Ce-DEDTC pigments compared to the DEDTC. This absorption has been intensified upon complex formation with Ce cations. Similar behavior has been reported



Figure 4. The UV-vis spectra of the 3.5 wt.% NaCl electrolyte, different concentration of the DEDTC and extract solution of the Ce-DEDTC pigment at different dissolution times: (a) UV-vis spectra of the 3.5 wt.% NaCl solution at difference concentration of the DEDTC, (b) linear regression curve of the DEDTC concentration vs UV-vis absorbance peak at ca. 260 nm extracted from the part (a,c) UV-vis spectra of the Ce-DEDTC extract solution, and (d) the amount on the released DEDTC (measured from the UV-vis result) and Ce (measured by ICP-OES test) from the structure of the Ce-DEDTC complex.

in literature⁵⁹. Moreover, the location of the second and last peak in UV-Vis spectra of the Ce-DEDTC pigment extract possesses a negative shift in comparison with the parent DEDTC. For studying the amount of released DEDTC from the structure of the Ce-DEDTC pigments, the intensity of the peak at 230–270 nm of the parent DEDTC spectra was investigated and compared with the Ce-DEDTC spectra. For calculating the amount of released DEDTC from the structure of the pigment a linear regression curve showed in Fig. 4b was used. As illustrated in Fig. 4d, the released DEDTC from the structure of the Ce-DEDTC pigment in two steps. At the first region, the amount of the released DEDTC increased rapidly by the dissolution time. It is due to the DEDTC moieties which didn't react with the Ce ions and are physically adsorbed on the Ce-DEDTC or CeO₂ pigment surface. At the second region, gradual release of the DEDTC related to the chemical dissociation of Ce-DEDTC complex can be observed.

The ICP-OES test was used to calculate the amount of released Ce ions from the structure of the Ce-DEDTC pigments at difference dissolution times. Figure 4d shows the concentration of the Ce ions which released from the structure of the Ce-DEDTC pigment in 3.5 wt.% NaCl solution versus dissolution time. As can be seen, the released Ce ions have a time-dependence behavior. Also, results indicated that the amount of Ce³⁺ ions in Ce-DEDTC pigment extraction is about 23 ppm after 120 h.

Electrochemical study. To investigate the specimen electrochemical behavior, the AA2024-T3 soaked in 3.5 wt. % NaCl electrolyte and extract saline solution of the Ce-DEDTC pigments and the EIS measurements were conducted over the immersion time. The EIS spectra of the blank sample and specimen exposure in the Ce-DEDTC pigment extract solution are illustrated in Fig. 5. As can be observed from Fig. 5, the impedance value for the sample exposed in the Ce-DEDTC pigment extract solution is significantly greater than that for the blank sample in all of the soaking times. Therefore, it can be concluded that the Ce-DEDTC pigments have inhibitive properties against the corrosion of the AA2024-T3. Also, as can be seen from the EIS curves, the impedance value of the blank sample decreased by an increase in soaking time from 2 to 8 h. It can be due to local corrosion occurring on the surface of the specimens. After that, the impedance value of the blank sample increased. The formation of thin film associated with corrosion production is the main reason for this effect. Conversely, the impedance value for the Ce-DEDTC sample increased during the exposure time. This result shows that an inhibitor film formed at the beginning of the immersion times. Also, result indicated that the thickness and also the inhibitive properties of this layer is promoted versus the immersion time.

To investigate the corrosion parameters of the specimen in more detail, the two and three time-constant circuits were employed for fitting the EIS curves of the blank and Ce-DEDTC specimens, respectively (Fig. 6). The equivalent circuit of the blank sample contains a low-frequency time constant associate with the corrosion process on the surface of the specimen and a high-frequency time constant which is associated with the oxide



Figure 5. Nyquist and Bode plots for the 2024-T3 aluminum alloy at different immersion time soaked in various solutions: (**a,b**) 3.5 wt.% NaCl electrolyte, and (**c,d**) extract saline solution of the Ce-DEDTC complex.



Figure 6. (a) Two-time and (b) three-time constants, used for fitting the impedance curves of the blank and Ce-DEDTC specimens, respectively.

layer naturally created on the AA2024-T3 surface. The equivalent circuit of the Ce-DEDTC contains three-time constants at the low-frequency, middle frequency and high-frequency which are related to the inhibitor film, oxide layer, and double layer, respectively. In these equivalent circuits, the R_s , R_{inh} , R_{OX} , and R_{ct} illustrate the solution resistance, the thin film resistance associated with the inhibitor, oxide layer resistance and charge transfer resistance, respectively. Also, CPE_{inh}, CPE_{ox}, and CPE_{dl} show the constant phase element of the inhibitor film, oxide layer, and charge transfer of the double layer, respectively. In these circuits, the constant phase element was employed instead of the ideal capacitance. It is associated with this fact that the surface of the samples has a non-ideal behavior, due to the surface heterogeneous^{1,2}. To calculate the capacitance of the double layer the following equation (Eq. 1) was used⁶⁰:

$$C_{dl} = Y_{0.dl}^{-1/n} \left(\frac{R_s R_{ct}}{R_s + R_{ct}} \right)^{(1-n)/n}$$
(1)

where n is the exponent of the constant phase element and Y_0 is admittance. The result of the fitted parameters illustrates in Table 1. The Eq. 2 was used for calculating the R_{total} .

	blank*			Ce-DEDTC**			
Immersion time (h)	R_{total} (k $\Omega.cm^2$)	R_{ct} (k Ω .cm ²)	C _{dl} (µF/cm ²)	$\begin{array}{c} R_{total} \\ (k\Omega.cm^2) \end{array}$	R _{ct} (kΩ.cm ²)	C _{dl} (µF/cm ²)	
2	25	9	185.1	526	442	0.0926	
8	61	54	600.6	1108	659	0.0811	
24	25	18	275.9	1293	945	0.0286	
72	20	13	103.6	2597	2360	0.0198	
120	12	6	1932.5	2793	2437	0.0053	

Table 1. The R_{total}, R_{ct}, and C_{dl} values for the blank and Ce-DEDTC specimens versus the immersion time.*Chi-square is between 0.001–0.005. **Chi-square is between 0.0004–0.003.



Figure 7. Potentiodynamic polarization curves of the blank and Ce-DEDTC specimens after 120 h of exposure.

$$R_{total} = R_{inh} + R_{ox} + R_{ct}$$
⁽²⁾

As can be observed from Table 1, the R_{total} for the Ce-DDTS is significantly higher than that for the blank sample. Also, it can be seen that the R_{total} of the Ce-DEDTC increases over the immersion time which shows that the corrosion inhibitor adsorbed on the surface of the specimen.

As can be seen from Table 1, the resistance of the charge transfer for the Ce-DEDTC pigment has a higher value than that for the specimen exposed to neat saline solution. Also, it can be seen that the value of the R_{ct} for the blank sample decrease versus the immersion time. This decrement in R_{ct} values of the blank sample can be due to the destructive effect of the Cl^- anions⁶¹. The bigger R_{total} value and also the R_{ct} increment versus the immersion time of the Ce-DEDTC sample in comparison with the blank sample were associated with the formation of a thin layer during exposure of the sample in extract solution and altering the surface chemical properties of the AA2024-T3. Due to the change in the chemical composition of the surface, the C_{dl} of the Ce-DEDTC sample is much lower than that for the blank sample. Moreover, the C_{dl} for the Ce-DEDTC sample decreases over the immersion time. Conversely, the C_{dl} for the blank sample increases at the prolonged immersion times. The decrement in the C_{dl} values of Ce-DEDTC sample over the immersion time may be associated with the replacing of the water molecules with the corrosion inhibitor presenting in the extract solution of the Ce-DEDTC pigment. Also, considering the capacitance as inversely proportional to the film thickness, the C_{dl} reduction is probably related to an increase in the area covered by the Ce deposition via island growth. Then, a reduction in local dielectric constant is happened in the presence of the Ce-DEDTC pigment^{62,63}.

The potentiodynamic polarization curves of the blank and Ce-DEDTC specimens after 120 h of exposure are illustrated in Fig. 7. The polarization test was repeated for 5 times and the curves provided in this figure are the curves showing average behavior. The study of the local corrosion is performed as an effective method for evaluating the potentiodynamic polarization parameters. Due to the local dissolution of the bare sample, associated with the intermetallic compound, no passivity range can be seen for the bare sample⁵. Also, to calculate the corrosion current density (i_{corr}) of the specimens, the vertical line crossing the corrosion potential (E_{corr}) and interpolation of the cathodic branch was employed and the extracted results are illustrated in Table 2. As can be observed from Fig. 7 and Table 2, the E_{corr} achieved more negative value in the presence of the Ce-DDTC pigment. Also, the i_{corr} of the Ce-DDTC specimen is about 10 times lower than that for the Blank sample. The result illustrates that the passivity range (E_{br} - E_{corr}) of the Ce-DEDTC is about 427.2 ± 12 mV. Existence of such extensive passive area can be due to the thin film associated with the corrosion inhibitor on the surface of the AA2024-T3. All of

Sample	i _{corr} (A/cm ²)	E _{corr} (mV vs. SCE)	E _{br} (mV vs. SCE)	E _{br} -E _{corr} (mV vs. SCE)
Blank	$9.8\pm 0.45\times 10^{-7}$	-624.3 ± 13	_	—
Ce-DDTC	$1.02\pm 0.24\times 10^{-7}$	-981.3 ± 21	1554.1 ± 12	427.2 ± 12

Table 2. Polarization parameters of the Blank and Ce-DDTC samples after 120 h.

Specimen	Immersion time (h)	Contact angle (°)	WA (mJ/m ²)	$\gamma_{\rm sv} ({\rm mJ}/{ m m}^2)$
blank	2	$93.5\pm2\%$	$69.91\pm2\%$	$26.36 \pm 2\%$
	8	$69.8\pm1\%$	98.36±1%	$41.54 \pm 0.02\%$
	24	$31.9\pm3\%$	$134.29\pm3\%$	$63.06 \pm 0.1\%$
	72	$0.0\pm0\%$	$145.07\pm0\%$	$74.12 \pm 0\%$
	120	$0.0\pm0\%$	$145.07\pm0\%$	$74.12 \pm 0\%$
Pigment	2	$98.6\pm3\%$	$60.36\pm3\%$	$19.84 \pm 1\%$
	8	$95.3\pm1\%$	$67.53 \pm 1\%$	$20.45\pm1\%$
	24	$67.4\pm2\%$	$101.56\pm2\%$	$43.26 \pm 0.04\%$
	72	$55.3\pm2\%$	$114.21\pm2\%$	50.42 ± 0.01
	120	$45.8\pm4\%$	$124.05\pm4\%$	$56.33\pm1\%$

Table 3. The values of CA, WA, and γ_{sv} for the AA2024-T3 immersed in 3.5 wt.% NaCl solution (blank) and the Ce-DEDTC pigment extract saline solution (Pigment) after various soaking times.



Figure 8. The contact angle images of the distilled water droplet placed on the AA2024-T3 surface soaked in 3.5 wt.% NaCl solution (blank) and extract solution of the Ce-DEDTC pigment (Pigment) after various immersion times.

these results indicate that the Ce-DEDTC complex can act as an effective anti-corrosion pigment for the 2024-T3 aluminum alloy. The result of the potentiodynamic polarization test is in good agreement with the result of the EIS measurements. Moreover, as can be observed, the current density of the cathodic branch of the polarization curves significantly decreased in the presence of Ce-DEDTC. Therefore, it can be concluded that the Ce-DEDTC acts as a mixed-type with cathodic dominant inhibition suppressing both anodic and cathodic reactions.

Surface study. Figure 8 shows the contact angle test result corresponded to the samples exposed to 3.5 wt.% NaCl electrolyte (blank) and the Ce-DEDTC pigment extract saline solution (Pigment) versus the soaking time. The results of the CA measurement are illustrated in Table 3. In Table 3, the surface energy parameters and work of adhesion (WA) were measured by using Young's equation (Eq. 3) and Neumann's equation (Eq. 4)⁶⁴:

$$WA = \gamma_{lv} (1 + \cos \theta) \tag{3}$$

$$WA = 2(\gamma_{lv}, \gamma_{sv})^{1/2} \exp[-\beta(\gamma_{lv} - \gamma_{sv})^2]$$
(4)

In these equations, the value of the β constant is about 0.000125 \pm 0.00001 (mJ/m²)², γ_{lv} represent the surface tension of water (72 mJ/m²), γ_{sv} shows the surface free energy of the specimens, and θ represent the contact angle of water. As can be observed from Table 3, the value of the CA for all of the specimens decreased versus the immersion time. The WA, γ_{sv} of the samples increase over the exposure period. Also, it can be seen that the values of the CA for the Ce-DEDTC pigment are larger than that for the blank sample in all of the soaking times. In addition, the WA and γ_{sv} of the blank sample are larger than that for the Ce-DEDTC pigment specimen. It's is due to the formation of corrosion product on the AA2024-T3 surface. It is generally believed that the changing in surface chemical composition and roughness are the main reason for the changing in CA of specimens⁶⁵. For the blank sample, as seen in the previous section of this work, the aluminum hydroxide/oxide is fabricated on the specimen surface. Therefore, these changes may lead to changing in roughness as well as the chemical composition of the surface the CA of the sample. Furthermore, it was shown that the surface of



Figure 9. Optical microscopy images for the blank and Ce-DEDTC samples in the various magnifications and immersion times.

the sample immersed in the extract solution of the Ce-DEDTC complex is covered by a thin film. So, it can be concluded that the changing in the chemical composition of the 2024-T3 aluminum alloy surface, in the presence of the Ce-DEDTC pigment, lead to increase the CA of the sample. Therefore, the WA and γ_{sv} of the sample significantly decrease in the presence of the Ce-DEDTC pigment extract solution.

In order to investigate the localized corrosion inhibition performance of the specimens, the optical microscopy images were employed. The OM micrographs of the blank and Ce-DEDTC samples at different magnifications and soaking times are demonstrated in Fig. 9. As can be observed, the surface of the blank sample has been degraded by trenching of the intermetallic compounds. This finding is in good agreement with the previous work conducted to study the corrosion behavior of the aluminum alloy in the presence of intermetallic compounds^{66–68}. The trenching attack is strengthened at prolonged exposure times. Also, it can be observed that the surface of the blank sample was covered with thick corrosion products at longer immersion times. Also, the big pits can be observed behind the corrosion products. It was associated with the pit coalescence. Based on the OM result, it can be seen that there wasn't any pit on the surface of the sample soaked in the Ce-DEDTC pigment extract solution even after 120 h. It seems that most of the intermetallic particles suppressed by a surface film. These results are in good accordance with the result of potentiodynamic polarization test and show the better inhibitive performance of the Ce-DEDTC sample against the localized corrosion.

The SEM-EDS micrographs of the blank and Ce-DEDTC samples in various immersion times and magnifications are shown in Fig. 10. Also, to study the composition of the surface corrosion products formed on the AA2024-T3 in the presence and absence of the Ce-DEDTC extract, the elemental analyses (EDS) was used. The



Figure 10. SEM micrographs of the blank and Ce-DEDTC samples at the various immersion times and magnifications.

results of the EDS test are illustrated in Table 4. According to the SEM-EDS results the surface of the blank sample contains localized corrosion nearby the intermetallic particles (trenching mechanism). Moreover, the surface of the blank sample is covered by corrosion products at longer times which shows severe corrosion attack on the surface of the specimen. Based on the literature, the intermetallic compounds in the structure of the AA2024-T3 are the main reason for the localized corrosion⁵. The increase in the amount of the oxygen on the surface of the blank sample versus the immersion time can be assigned to the aluminum oxide/hydroxide. According to Fig. 10, it seems that the released corrosion inhibitors improve the inhibitive performance of the AA2024-T3 in all of the immersion times. These results indicate the better corrosion properties of the AA2024-T3 in the presence of the Ce-DEDTC complex. Also, the result of the EDS analysis shows that the surface of the Ce-DEDTC specimen contains the C, N, S, and Ce elements as the main compound of the complex showing that the DEDTC species adsorbed on the aluminum surface. Therefore, it seems that a thin film associated with the extract solution of the Ce-DDC pigment is created on the aluminum surface. These results are in good accordance with the EIS test result associated with the thin film formation on the surface of the specimen in the presence of the complex extract solution.

The XPS analysis was used for studying the chelate formation in the presence of the Ce and DEDTC in the extract solution. In order to do that, the aluminum plate immersed in extract solution of the Ce-DEDTC

Specimen	Immersion time (h)	Al	Cu	Mg	o	Ce	с	N	s
Blank	2	82.63	3.77	2.96	10.64	-	-	-	-
	24	68.24	2.63	1.93	27.2	-	-	-	-
	120	55.16	2.63	1.01	49.8	-	-	-	-
Ce-DEDTC	2	89.99	2.62	3.11	1.31	0.15	0.1	0.08	2.64
	24	88.17	3.07	3.14	1.95	0.21	0.15	0.18	3.13
	120	89.85	2.69	3.18	1.31	0.15	0.1	0.08	2.64

Table 4. The composition of corrosion products (atomic %) formed on the surface of the 2024-T3 aluminumalloy soaked in the 3.5 wt.% NaCl solution and Ce-DEDTC extract solution vs. the immersion time.

pigment and the XPS spectra was provided from the surface of the specimen after 120 h. The XPS survey and also the high-resolution spectra of Ce₃d, Al₂p, S₂p, C₁s, and O₁s are presented in Fig. 11. The deconvolution of high-resolution spectra was conducted to study the potential chemical bonds between the species. The high-resolution spectra of the Ce₃d show the peaks around the 899.01, 900.9, 903.9, 907.21 and 916.82 eV representing the spin-orbit states of the Ce₃d_{3/2} assigned to the 3d104f1 electronic state of Ce³⁺ ions⁶⁹⁻⁷³. Also, the 3d104f0 spin-orbit state attributed to the Ce⁴⁺ appeared at the 880.41, 882.6, 885.09, 888.61, and 898.22 eV⁶⁹⁻⁷³. The O1s contains 4 main peaks at around of 529.7, 530.9, 532.22 and 533.43 eV. The peaks at 529.7 eV and 530.9 eV are related to the CeO₂ and Ce₂O₃, respectively^{22,74,75}. The peaks at 532.22 eV and 533.43 eV are due to the Al-O bonded and Ce(OH)₃, respectively^{22,74,75}. The high-resolution spectra of the Al2p contain 3 main peaks at binding energy of 72.8, 75.1 and 78.23 eV assigned to Al, Al-O, and Al-S, respectively^{22,76}. C 1 s peak can be deconvoluted into five species, the peak corresponding to the C-C/C-H bond with hybridization of sp2 (284.56 eV)⁷⁷⁻⁸⁰, the peak at the binding energy of 285.56 eV is associated with the covalent bond of C-S^{79,81}, and the sp2 C-N bond appears at 286.53 eV⁷⁷⁻⁸⁰. Also, based on the literature, the peaks at the binding energy of 287.78 and 290.35 eV could be due to the formation of S-C-N and S-C-S bonds on the aluminum substrate surface, respectively⁸².

As presented in Fig. 11, the S2p deconvoluted into four peaks at 161.88, 163.77, 165.61, and 169.35 eV. The other peaks are satellite. The peaks around the 163.77 and 165.61 eV are assigned to the sulfur (C–S/S–S) in its elemental state in the structure of the DEDTC^{81,83-85}. The peaks around the 169.35 and 161.88 eV can be due to the interaction of the thiol group with the metallic compound^{83,84}. Therefore, it seems that the peak around the 161.88 and 169.35 eV can be associated with the C–S–Al⁸⁶ and CeO_{2–x}S_x⁸⁷, which shows the mechanism of the adsorption of corrosion inhibitors on the AA2024-T3 surface.

Protection mechanism. The schematic illustration of different possible mechanisms which improve the inhibition behavior of the AA2024-T3 is presented in Fig. 12. Based on the characterization results, the synthesized Ce-DEDTC pigment contained the DEDTC and Ce species. Also, it was shown that the Ce ions and DEDTC molecules released from the Ce-DEDTC pigment structure due to its partial dissolution in the saline solution. Therefore, in the presence of the Ce and DEDTC species in the extract solution of the Ce-DEDTC pigment, three main mechanisms can be activated to improve the corrosion behavior of the AA2024-T3. The first mechanism is based on the fact that the thiol sites in the structure of DEDTC can be interacted with the Al^{3+} cations or surface Al oxide film (Fig. 12a). This mechanism leads to deposition of a protective thin film on the active corrosion regions of Al surface. The presence of Al-S bond in XPS result (Fig. 10) proved this mechanism. This mechanism is in good agreement with the result of Forsyth et. al. about the formation of a mixed metal organic species on the surface of the metallic substrate in the condition of presence of cerium salicylate compound in saline solution³⁹. They have shown that the physical adsorption of the complex is performed, when the mentioned compound was used for improving the anti-corrosive behavior of the metallic substrate in corrosive media³⁹. The second mechanism is based on this fact that the Ce ions which released from the structure of the Ce-DEDTC pigment, due to cathodic performance, deposited on the cathodic sites of the substrate. After that, the thiol group in the DEDTC structure is bonded with the deposited structure (Fig. 12b). The presence of the $CeO_{2-x}S_x$ phase in XPS result (Fig. 10) confirmed this mechanism. Moreover, the presence of the CeO₂, Ce₂O₃ and Ce(OH)₃ in the XPS results (Fig. 10) can confirm the deposition of the Ce derivate on the surface of the AA2024-T3. Ferrer et. al. confirmed the cathodic performance of double-doped zeolites with Ce cations and DEDTC molecules for AA2024-T3. Also, their results demonstrated that the Ce-DEDTC complex was created on the intermetallic compounds, when the corrosive media contained these species⁴⁸. The final mechanism is based on the chelate formation (Fig. 12c). The released Ce ions and DEDTC molecules in solution reform complex and deposited on the surface of the aluminum. Based on these results, it could be concluded that the complex physically and chemically adsorbed on the aluminum surface. Therefore, the corrosion behavior of the AA2024-T3 significantly enhanced in the existence of the Ce-DEDTC complex. The results of this work are in good agreement with the previous works conducted to performed the corrosion properties of the metallic substrate in corrosive media in the presence of the Ce-Organic complex. Where, the absorption of the inhibitor via the organic component in the first instance to form a metal-organic bond, followed by the formation of a mixed Ce-organic compound-metal substrate leading to the mixed inhibition^{39,40}.

Conclusion

In this work, the Ce-DEDTC complex was synthesized as a novel anti-corrosion pigment. Also, the subsequent effect of this pigment on the corrosion behavior and also surface properties of the AA2024-T3 was investigated and the following main conclusions were obtained:



Figure 11. The XPS survey and high-resolution spectra of Ce₃d, Al₂p, S₂p, C₁s, and O₁s.



Figure 12. The schematic illustration of different potential mechanisms which improve the inhibition properties of the AA2024-T3 in 3.5 wt.% NaCl electrolyte in the existence of Ce-DEDTC complex.

- The XRD, FT-IR, TG, ICP-OES, and ICP analyzing methods were used to characterize the produced pigment. The result showed that the pigments successfully were synthesized. Also, the result indicated that the total released Ce and DEDTC species from the 3.5 wt.% NaCl solution contained 5 g/L Ce-DEDTC complex after 120 h were about 23 ppm (0.16 mM) and 70 ppm (0.4 mM), respectively.
- The electrochemical behavior of the 2024-T3 aluminum alloy in the existence of the Ce-DEDTC complex
 was investigated by EIS and PDS tests. The result indicated that the inhibition performance of the aluminum
 alloy significantly enhanced in the presence of the Ce-DEDTC pigments. Also, the thin film formation was
 considered as the main protection mechanism of the Ce-DEDTC pigment. The impedance of the AA2024-T3
 was about 100 times larger than that for the blank sample. Also, a wide passivation range was observed in the
 existence of the Ce-DEDTC pigment extract solution.
- The SEM, EDS, OM, XPS, and CA tests were employed to investigate the subsequent effect of the Ce-DEDTC pigments on the AA2024-T3 surface properties. The results indicated that the resistance of the AA2024-T3 against the pit formation significantly enhanced in the presence of the Ce-DEDTC pigment. Moreover, the results showed that a thin film is created on the AA2024-T3 surface in the presence of the Ce-DEDTC pigment extract.

Data availability

The dataset generated or analyzed during the current study are available from the corresponding authors for two years after publication.

Received: 20 August 2019; Accepted: 5 March 2020; Published online: 19 March 2020

References

- 1. Mohammadi, I. & Afshar, A. Modification of nanostructured anodized aluminum coatings by pulse current mode. *Surface and Coatings Technology* 278, 48–55 (2015).
- 2. Mohammadi, I., Afshar, A. & Ahmadi, S. Al₂O₃/Si₃N₄ nanocomposite coating on aluminum alloy by the anodizing route: Fabrication, characterization, mechanical properties and electrochemical behavior. *Ceramics International* **42**, 12105–12114 (2016).
- Mohammadi, I., Ahmadi, S. & Afshar, A. Effect of pulse current parameters on the mechanical and corrosion properties of anodized nanoporous aluminum coatings. *Materials Chemistry and Physics* 183, 490–498 (2016).
- 4. Hashimoto, T. *et al.* Investigation of dealloying of S phase (Al₂CuMg) in AA 2024-T3 aluminium alloy using high resolution 2D and 3D electron imaging. *Corrosion Science* **103**, 157–164 (2016).
- Prakashaiah, B., Kumara, D. V., Pandith, A. A., Shetty, A. N. & Rani, B. A. Corrosion inhibition of 2024-T3 aluminum alloy in 3.5% NaCl by thiosemicarbazone derivatives. *Corrosion Science* 136, 326–338 (2018).
- Williams, G. & McMurray, H. N. Anion-exchange inhibition of filiform corrosion on organic coated AA2024-T3 aluminum alloy by hydrotalcite-like pigments. *Electrochemical and solid-state letters* 6, B9–B11 (2003).
- Hu, J., Zhang, J. & Cao, C. Determination of water uptake and diffusion of Cl- ion in epoxy primer on aluminum alloys in NaCl solution by electrochemical impedance spectroscopy. *Progress in Organic Coatings* 46, 273–279 (2003).
- Yeh, J.-M., Liou, S.-J., Lai, C.-Y., Wu, P.-C. & Tsai, T.-Y. Enhancement of corrosion protection effect in polyaniline via the formation of polyaniline – clay nanocomposite materials. *Chemistry of Materials* 13, 1131–1136 (2001).
- 9. Chang, C.-H. et al. Novel anticorrosion coatings prepared from polyaniline/graphene composites. Carbon 50, 5044–5051 (2012).
- Weng, C.-J. et al. Advanced anticorrosive coatings prepared from electroactive polyimide-TiO₂ hybrid nanocomposite materials. Electrochimica Acta 55, 8430–8438 (2010).
- 11. Herbst, W. & Hunger, K. Industrial organic pigments: production, properties, applications. (John Wiley & Sons, 2006).
- Mousavifard, S., Nouri, P. M., Attar, M. & Ramezanzadeh, B. The effects of zinc aluminum phosphate (ZPA) and zinc aluminum polyphosphate (ZAPP) mixtures on corrosion inhibition performance of epoxy/polyamide coating. *Journal of Industrial and* Engineering Chemistry 19, 1031-1039 (2013).
- Ilevbare, G., Scully, J., Yuan, J. & Kelly, R. Inhibition of pitting corrosion on aluminum alloy 2024-T3: effect of soluble chromate additions vs chromate conversion coating. *Corrosion* 56, 227–242 (2000).
- 14. Kendig, M., Jeanjaquet, S., Addison, R. & Waldrop, J. Role of hexavalent chromium in the inhibition of corrosion of aluminum alloys. *Surface and Coatings Technology* 140, 58–66 (2001).
- Kendig, M. & Buchheit, R. Corrosion inhibition of aluminum and aluminum alloys by soluble chromates, chromate coatings, and chromate-free coatings. Corrosion 59, 379–400 (2003).
- Forsyth, M. et al. Inhibition of corrosion on AA2024-T3 by new environmentally friendly rare earth organophosphate compounds. Corrosion 64, 191–197 (2008).
- 17. Izadi, M., Shahrabi, T. & Ramezanzadeh, B. Electrochemical investigations of the corrosion resistance of a hybrid sol-gel film containing green corrosion inhibitor-encapsulated nanocontainers. *Journal of the Taiwan Institute of Chemical Engineers* **81**, 356–372 (2017).
- Izadi, M., Shahrabi, T. & Ramezanzadeh, B. Active corrosion protection performance of an epoxy coating applied on the mild steel modified with an eco-friendly sol-gel film impregnated with green corrosion inhibitor loaded nanocontainers. *Applied Surface Science* 440, 491–505 (2018).
- Izadi, M., Mohammadi, I., Shahrabi, T., Ramezanzadeh, B. & Fateh, A. Corrosion inhibition performance of novel eco-friendly nanoreservoirs as bi-component active system on mild steel in aqueous chloride solution. *Journal of the Taiwan Institute of Chemical Engineers* 95, 555–568 (2019).
- Mohammadi, I., Izadi, M., Shahrabi, T., Fathi, D. & Fateh, A. Enhanced epoxy coating based on cerium loaded Na-montmorillonite as active anti-corrosive nanoreservoirs for corrosion protection of mild steel: Synthesis, characterization, and electrochemical behavior. *Progress in Organic Coatings* 131, 119–130 (2019).
- Izadi, M., Mohammadi, I., Shahrabi, T., Ramezanzadeh, B. & Fateh, A. The electrochemical behavior of nanocomposite organic coating based on clay nanotubes filled with green corrosion inhibitor through a vacuum-assisted procedure. *Composites Part B: Engineering* (2019).
- Niroumandrad, S., Rostami, M. & Ramezanzadeh, B. Corrosion resistance of flaky aluminum pigment coated with cerium oxides/ hydroxides in chloride and acidic electrolytes. *Applied Surface Science* 357, 2121–2130 (2015).
- Nikravesh, B., Ramezanzadeh, B., Sarabi, A. & Kasiriha, S. Evaluation of the corrosion resistance of an epoxy-polyamide coating containing different ratios of micaceous iron oxide/Al pigments. *Corrosion Science* 53, 1592–1603 (2011).
- Jadhav, N., Vetter, C. A. & Gelling, V. J. The effect of polymer morphology on the performance of a corrosion inhibiting polypyrrole/ aluminum flake composite pigment. *Electrochimica acta* 102, 28–43 (2013).
- Arenas, M., Bethencourt, M., Botana, F., De Damborenea, J. & Marcos, M. Inhibition of 5083 aluminium alloy and galvanised steel by lanthanide salts. *Corrosion Science* 43, 157–170 (2001).
- Muster, T. et al. An investigation of rare earth chloride mixtures: combinatorial optimisation for AA2024-t3 corrosion inhibition. Surface and Interface Analysis: An International Journal devoted to the development and application of techniques for the analysis of surfaces, interfaces and thin films 42, 170–174 (2010).
- 27. Cabral, A. *et al.* The corrosion resistance of hot dip galvanised steel and AA2024-T3 pre-treated with bis-[triethoxysilylpropyl] tetrasulfide solutions doped with Ce (NO₃) 3. *Corrosion Science* **48**, 3740–3758 (2006).
- Muster, T. et al. A combinatorial matrix of rare earth chloride mixtures as corrosion inhibitors of AA2024-T3: optimisation using potentiodynamic polarisation and EIS. Electrochimica Acta 67, 95–103 (2012).
- 29. Hu, T. et al. Cerium tartrate as a corrosion inhibitor for AA 2024-T3. Corrosion Science 95, 152-161 (2015).
- Denissen, P. J. & Garcia, S. J. Cerium-loaded algae exoskeletons for active corrosion protection of coated AA2024-T3. Corrosion Science 128, 164–175 (2017).
- Yasakau, K. A., Zheludkevich, M. L., Lamaka, S. V. & Ferreira, M. G. Mechanism of corrosion inhibition of AA2024 by rare-earth compounds. *The Journal of Physical Chemistry B* 110, 5515–5528 (2006).
- Jakab, M., Presuel-Moreno, F. & Scully, J. Critical concentrations associated with cobalt, cerium, and molybdenum inhibition of AA2024-T3 corrosion: Delivery from Al-Co-Ce (-Mo) alloys. *Corrosion* 61, 246–263 (2005).
- Dias, S. et al. The role of Ce (III)-enriched zeolites on the corrosion protection of AA2024-T3. Electrochimica Acta 112, 549–556 (2013).
- Udoh, I. I., Shi, H., Liu, F. & Han, E.-H. Synergistic Effect of 3-Amino-1, 2, 4-triazole-5-thiol and Cerium Chloride on Corrosion Inhibition of AA2024-T3. *Journal of The Electrochemical Society* 166, C185–C195 (2019).
- Jegdić, B., Bobić, B. & Linić, S. Corrosion behaviour of AA2024 aluminium alloy in different tempers in NaCl solution and with the CeCl3 corrosion inhibitor. *Materials and Corrosion*.
- Yao, J. et al. Extending surface Raman spectroscopy to transition metals for practical applications IV. A study on corrosion inhibition of benzotriazole on bare Fe electrodes. Electrochimica acta 48, 1263–1271 (2003).

- Ralkhal, S., Shahrabi, T., Ramezanzadeh, B. & Bahlakeh, G. A combined electrochemical, molecular dynamics, quantum mechanics and XPS analysis of the mild steel surface protected by a complex film composed of neodymium (III) and benzimidazole. *Applied Surface Science* 464, 178–194 (2019).
- Tabatabaei Majd, M., Bahlakeh, G., Dehghani, A., Ramezanzadeh, B. & Ramezanzadeh, M. A green complex film based on the extract of Persian Echium amoenum and zinc nitrate for mild steel protection in saline solution; Electrochemical and surface explorations besides dynamic simulation. *Journal of Molecular Liquids* 291, 111281, https://doi.org/10.1016/j.molliq.2019.111281 (2019).
- Forsyth, M., Forsyth, C. M., Wilson, K., Behrsing, T. & Deacon, G. B. ATR characterisation of synergistic corrosion inhibition of mild steel surfaces by cerium salicylate. *Corrosion science* 44, 2651–2656 (2002).
- 40. Blin, F. *et al.* Corrosion mitigation of mild steel by new rare earth cinnamate compounds. *Journal of applied electrochemistry* **34**, 591–599 (2004).
- Coelho, L. B., Cossement, D. & Olivier, M. G. Benzotriazole and cerium chloride as corrosion inhibitors for AA2024-T3: An EIS investigation supported by SVET and ToF-SIMS analysis. *Corrosion Science* 130, 177–189, https://doi.org/10.1016/j. corsci.2017.11.004 (2018).
- 42. Ralkhal, S., Ramezanzadeh, B. & Shahrabi, T. Studying dual active/barrier and self-healing reinforcing effects of the Neodymium (III)-Benzimidazole hybrid complex in the epoxy coating/mild steel system. *Journal of Alloys and Compounds* **790**, 141–155 (2019).
- Ralkhal, S., Shahrabi, T. & Ramezanzadeh, B. Synthesis and construction of a highly potent hybrid organic/inorganic anti-corrosive pigment for effective corrosion control of mild steel in simulated seawater. *Construction and Building Materials* 222, 400–413 (2019).
- Majd, M. T., Shahrabi, T. & Ramezanzadeh, B. Production of an eco-friendly anti-corrosion ceramic base nanostructured hybridfilm based on Nd (III)-C₇H₆N₂ on the mild steel surface; Electrochemical and surface studies. *Construction and Building Materials* 221, 456–468, https://doi.org/10.1016/j.conbuildmat.2019.06.122 (2019).
- Williams, G., Coleman, A. J. & McMurray, H. N. Inhibition of aluminium alloy AA2024-T3 pitting corrosion by copper complexing compounds. *Electrochimica Acta* 55, 5947–5958 (2010).
- Harvey, T. *et al.* The effect of inhibitor structure on the corrosion of AA2024 and AA7075. *Corrosion science* 53, 2184–2190 (2011).
 Denissen, P. J. & Garcia, S. J. Reducing subjectivity in EIS interpretation of corrosion and corrosion inhibition processes by *in-situ*
- optical analysis. Electrochimica Acta 293, 514–524, https://doi.org/10.1016/j.electacta.2018.10.018 (2019).
- 48. Ferrer, E. L., Rollon, A. P., Mendoza, H. D., Lafont, U. & Garcia, S. J. Double-doped zeolites for corrosion protection of aluminium alloys. *Microporous and Mesoporous Materials* 188, 8–15 (2014).
- Wang, Z.-Y. et al. Prediction of gas phase thermodynamic function of polychlorinated dibenzo-p-dioxins using DFT. Journal of Molecular Structure: THEOCHEM 672, 97–104 (2004).
- Liu, W., Duan, H., Wei, D., Cui, B. & Wang, X. Stability of diethyl dithiocarbamate chelates with Cu (II), Zn (II) and Mn (II). Journal of Molecular Structure 1184, 375–381 (2019).
- Kane, S. *et al.* Separation of heavy metal from water samples—The study of the synthesis of complex compounds of heavy metal with dithiocarbamates. *Journal of Environmental Science and Health, Part A* 51, 335–340 (2016).
- Awang, N., Baba, I., Yamin, B. M., Othman, M. S. & Farahana, K. N. Synthesis, characterization and biological activities of organotin (IV) methylcyclohexyldithiocarbamate compounds. *American Journal of Applied Sciences* 8, 310–317 (2011).
- Manav, N., Mishra, A. & Kaushik, N. In vitro antitumour and antibacterial studies of some Pt (IV) dithiocarbamate complexes. Spectrochimica Acta Part A: Molecular and Biomolecular Spectroscopy 65, 32–35 (2006).
- 54. Hogarth, G. et al. Functionalised dithiocarbamate complexes: Synthesis and molecular structures of 2-diethylaminoethyl and 3-dimethylaminopropyl dithiocarbamate complexes [M {S2CN (CH2CH2NEt2) 2} n] and [M {S2CN (CH2CH2CH2NMe2) 2} n] (n = 2, M = Ni, Cu, Zn, Pd; n = 3, M = Co). Inorganica Chimica Acta 362, 2020–2026 (2009).
- Jayaraju, A., Ahamad, M. M., Rao, R. M. & Sreeramulu, J. Synthesis, characterization and biological evaluation of novel dithiocarbamate metal complexes. J Der Pharma Chemica 4, 1191–1194 (2012).
- Müsellim, E., Tahir, M. H., Ahmad, M. S. & Ceylan, S. Thermokinetic and TG/DSC-FTIR study of pea waste biomass pyrolysis. *Applied Thermal Engineering* 137, 54–61 (2018).
- Labus, M. & Lempart, M. Studies of Polish Paleozoic shale rocks using FTIR and TG/DSC methods. Journal of Petroleum Science and Engineering 161, 311–318 (2018).
- Mbese, J. Z. & Ajibade, P. A. Synthesis, structural and optical properties of ZnS, CdS and HgS nanoparticles from dithiocarbamato single molecule precursors. *Journal of Sulfur Chemistry* 35, 438–449 (2014).
- Abbas, M. N., Radwan, A.-L. A., Nawwar, G. A., Zine, N. & Errachid, A. A durable solid contact sulfide sensor based on a ceric acrylohydrazide ionophore attached to polyacrylamide with a nanomolar detection limit. *Analytical Methods* 7, 930–942 (2015).
- Samiee, R., Ramezanzadeh, B., Mahdavian, M. & Alibakhshi, E. Corrosion inhibition performance and healing ability of a hybrid silane coating in the presence of praseodymium (III) cations. *Journal of The Electrochemical Society* 165, C777–C786 (2018).
- Zhang, D. et al. Preparation of self-healing hydrophobic coating on AA6061 alloy surface and its anti-corrosion property. Journal of Alloys and Compounds 774, 495–501 (2019).
- Soltani, N., Tavakkoli, N., Khayatkashani, M., Jalali, M. R. & Mosavizade, A. Green approach to corrosion inhibition of 304 stainless steel in hydrochloric acid solution by the extract of Salvia officinalis leaves. *Corrosion Science* 62, 122–135 (2012).
- 63. Outirite, M. *et al.* ac impedance, X-ray photoelectron spectroscopy and density functional theory studies of 3, 5-bis (n-pyridyl)-1, 2, 4-oxadiazoles as efficient corrosion inhibitors for carbon steel surface in hydrochloric acid solution. *Electrochimica Acta* 55, 1670–1681 (2010).
- 64. Mahidashti, Z., Shahrabi, T. & Ramezanzadeh, B. A new strategy for improvement of the corrosion resistance of a green cerium conversion coating through thermal treatment procedure before and after application of epoxy coating. *Applied Surface Science* **390**, 623–632 (2016).
- 65. Ramezanzadeh, B., Vakili, H. & Amini, R. Improved performance of cerium conversion coatings on steel with zinc phosphate posttreatment. *Journal of Industrial and Engineering Chemistry* **30**, 225–233 (2015).
- 66. Kakinuma, H. et al. Mechanism for the Morphological Change from Trenching to Pitting around Intermetallic Particles in AA1050 Aluminum. Journal of The Electrochemical Society 166, C19–C32 (2019).
- Boag, A., Hughes, A., Glenn, A., Muster, T. & McCulloch, D. Corrosion of AA2024-T3 Part I: Localised corrosion of isolated IM particles. *Corrosion Science* 53, 17–26 (2011).
- Schmutz, P. & Frankel, G. Corrosion study of AA2024-T3 by scanning kelvin probe force microscopy and *in situ* atomic force microscopy scratching. *Journal of the Electrochemical Society* 145, 2295–2306 (1998).
- Sánchez-Amaya, J., Blanco, G., Garcia-Garcia, F., Bethencourt, M. & Botana, F. XPS and AES analyses of cerium conversion coatings generated on AA5083 by thermal activation. Surface and Coatings Technology 213, 105–116 (2012).
- Wang, C., Zhang, C., Zhao, Y., Yan, X. & Cao, P. Poisoning Effect of SO₂ on Honeycomb Cordierite-Based Mn–Ce/Al₂O₃Catalysts for NO Reduction with NH₃ at Low Temperature. *Applied Sciences* 8, 95 (2018).
- Smirnov, M. Y. et al. Comparative XPS Study of Al₂O₃ and CeO₂ Sulfation in Reactions with SO₂, SO₂ + O₂, SO₂ + H₂O, and SO₂ + O₂ + H₂O. Kinetics and catalysis 44, 575–583 (2003).
- Eslami, M., Fedel, M., Speranza, G., Deflorian, F. & Zanella, C. Deposition and characterization of cerium-based conversion coating on HPDC low Si content aluminum alloy. *Journal of The Electrochemical Society* 164, C581–C590 (2017).
- Zhu, Y. et al. MnOx decorated CeO₂ nanorods as cathode catalyst for rechargeable lithium-air batteries. Journal of Materials Chemistry A 3, 13563-13567 (2015).

- 74. Dong, S., Xingguo, F., Mingren, S., Xinxin, M. & Guangze, T. Composition and corrosion resistance of cerium conversion films on 2195Al-Li alloy. *Journal of Rare Earths* **30**, 383–387 (2012).
- Martínez, L. et al. Surface study of cerium oxide based coatings obtained by cathodic electrodeposition on zinc. Applied Surface Science 257, 6202–6207 (2011).
- Johansen, H. D., Brett, C. M. & Motheo, A. J. Corrosion protection of aluminium alloy by cerium conversion and conducting polymer duplex coatings. *Corrosion science* 63, 342–350 (2012).
- Akhavan, H., Izadi, M., Mohammadi, I., Shahrabi, T. & Ramezanzadeh, B. The synergistic effect of BTA-Co system on the corrosion inhibition of mild steel in 3.5 wt% NaCl solution. *Journal of The Electrochemical Society* 165, C670–C680 (2018).
- Lee, J.-M. et al. X-ray photoelectron spectroscopy study of cobalt supported multi-walled carbon nanotubes prepared by different precursors. Carbon letters 8, 120–126 (2007).
- Razzaq, A. A. et al. High-performance lithium sulfur batteries enabled by a synergy between sulfur and carbon nanotubes. Energy Storage. Materials 16, 194–202 (2019).
- Guo, J., Wen, Z., Wang, Q., Jin, J. & Ma, G. A conductive selenized polyacrylonitrile cathode material for re-chargeable lithium batteries with long cycle life. *Journal of Materials Chemistry A* 3, 19815–19821 (2015).
- 81. Ye, J. et al. Sulfur/carbon nanocomposite-filled polyacrylonitrile nanofibers as a long life and high capacity cathode for lithium-sulfur batteries. *Journal of Materials Chemistry A* 3, 7406–7412 (2015).
- 82. Parviz, M. et al. Investigation of the Antifouling Properties of Phenyl Phosphorylcholine-Based Modified Gold Surfaces. Electroanalysis 26, 1471–1480 (2014).
- Peng, Z., Jia, D., Al-Enizi, A. M., Elzatahry, A. A. & Zheng, G. From water oxidation to reduction: homologous Ni-Co based nanowires as complementary water splitting electrocatalysts. Advanced Energy. *Materials* 5, 1402031 (2015).
- Xue, Y., Zuo, Z., Li, Y., Liu, H. & Li, Y. Graphdiyne-Supported NiCo2S4 Nanowires: A Highly Active and Stable 3D Bifunctional Electrode Material. Small 13, 1700936 (2017).
- Wang, X. *et al.* Stabilizing high sulfur loading Li–S batteries by chemisorption of polysulfide on three-dimensional current collector. *Nano Energy* 30, 700–708 (2016).
- Wang, F., Wang, Y. & Li, Y. Study of influencing factors and the mechanism of preparing triazinedithiol polymeric nanofilms on aluminum surfaces. *International journal of molecular sciences* 11, 4715–4725 (2010).
- Xiao, Y. et al. Hydrogenated CeO_{2-x} Sx mesoporous hollow spheres for enhanced solar driven water oxidation. Chemical Communications 52, 2521–2524 (2016).

Author contributions

T. Shahrabi and M. Mahdavian conceived the idea. I. Mohammadi ran the experiments and wrote the first draft of manuscript. M. Izadi corrected the first draft and consulted on experimentation. T. Shahrabi and M. Mahdavian supervised the project, provided critical notes and prepared the final version of the manuscript.

Competing interests

The authors declare no competing interests.

Additional information

Correspondence and requests for materials should be addressed to T.S. or M.M.

Reprints and permissions information is available at www.nature.com/reprints.

Publisher's note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit http://creativecommons.org/licenses/by/4.0/.

© The Author(s) 2020