SCIENTIFIC REPORTS natureresearch

OPEN

Rapid solvent-evaporation strategy for three-dimensional cobalt-based complex hierarchical architectures as catalysts for water oxidation

Hong Jiang^{1,3}, Hao Zhang^{2,3}, Qiaoling Kang², Haifeng Ma², Yinlin Tong¹, Feng Gao ^{1*} & Qingyi Lu^{2*}

It is a challenging task to seek a highly-efficient electrocatalyst for oxygen evolution reaction (OER) of water splitting. Non-noble Co-based nanomaterials are considered as earth-abundant and effective catalysts to lower overpotential and increase polarization current density of OER. In this work, we reported, for the first time, a "rapid solvent-evaporation" strategy for the synthesis of threedimensional (3D) cobalt complex hierarchical architectures constructed by two-dimensional (2D) nanosheets. The 3D structured cobalt complexes have excellent performances in catalyzing OER with lower onset potential, overpotential, Tafel slope and better stability than commercial IrO₂. Superior electrochemical performances would be beneficial from the unique 3D structure. This extremely simple method for 3D Co complex with good OER activities makes the complex be promising commercial OER catalyst to replace earth-rare and expensive IrO₂.

Hydrogen is considered as a promising energy carrier to address issues of global energy security, environmental emissions and sustainability^{1–3}. Electrocatalytic water splitting is considered an efficient technology for hydroten production, which consists of hydrogen evolution reaction (HER) on the cathode, and oxygen evolution reaction (OER) on the anode⁴. In the process of OER, large overpotentials are required to promote the reaction, which increases the cost of hydrogen production. So far expensive Ru- and Ir-based compounds, such as Ru, Ir, RuO₂ and IrO₂ have reported to efficiently catalyze the OER process⁵. However, high cost and poor durability of these noble catalysts restricts their large-scale application⁶. It is imperative to develop highly efficient and low-cost electrocatalysts to lower the overpotential and accelerate the OER reaction.

Co-based compounds have emerged as a kind of non-noble metal catalysts for OER since Nocera reported that cobalt (II)/phosphate could be an efficient oxygen-evolving catalyst in 2008⁷. Until now, considerable efforts have been devoted to Co-based OER electrocatalysts, including Co oxides^{8,9}, Co-based layered metal hydroxides¹⁰⁻¹², Co phosphides/phosphates^{13,14}, Co sulfides/selenides^{15,16}, Co oxyhydroxides^{17,18} and cobalt nitrides¹⁹⁻²¹. Previous reports prove that the active sites for OER catalysis are likely from amorphous overlayers comprising cobaltate aggregates instead of the native oxide^{22,23}. However, these reported catalysts usually focus on the inorganic cobalt compounds, rare on the cobalt complexes, though the complexes have structural topology and potential functions for gas adsorption, catalytic action and selective adsorption²⁴.

Recently, multifunctional three-dimensional (3D) architectures assembled by low-dimensional blocks, such as 0D nanoparticles²⁵, 1D nanorods²⁶, and 2D nanosheets^{27,28}, have drawn wide attention owing to the novel properties produced by synergistic effects of building blocks. Assembling 2D nanosheets into 3D structure not only maintains the intrinsic performances of the 2D nanostructures but also synergistically avoids the accumulation of 2D nanosheets that would dramatically weakens the excellent properties of nanounits²⁹. However, according to previous reports, the construction of 2D nanosheets-assembled 3D architectures usually depends on the guide of small molecules or macromolecules, and requires rigorous conditions and techniques, which

¹Department of Materials Science and Engineering, Collaborative Innovation Center of Advanced Microstructures, Nanjing University, Nanjing, 210093, P. R. China. ²State Key Laboratory of Coordination Chemistry, Coordination Chemistry Institute, Collaborative Innovation Center of Advanced Microstructures, Nanjing National Laboratory of Microstructures, School of Chemistry and Chemical Engineering, Nanjing University, Nanjing, 210093, P. R. China. ³These authors contributed equally: Hong Jiang and Hao Zhang. *email: fgao@nju.edu.cn; gylu@nju.edu.cn restrict their wide application as practical methods^{30,31}. Therefore, it is highly necessary to develop a facile and efficient strategy for preparing novel 3D structures. In this work, by developing a facile and efficient method called "rapid solvent-evaporation" (RSE) strategy, we synthesized 3D cobalt complex hierarchical architectures assembled by 2D nanosheets. This facile RSE method evaporates solvent quickly during reaction process under high-temperature, resulting in the formation of novel 3D cobalt complex hierarchical architectures, which are completely different from those obtained at slow evaporation processes. More importantly, as a new type Co-based catalyst, the resulted 3D cobalt complexes show excellent catalyzing OER performances in water oxidation compared to commercial IrO₂.

Experimental

Materials. All reagents were of analytical grade and used without further purification. Cobalt acetate and urea were obtained from Sinopharm Chemical Reagent Co. Ltd. Ethanol was purchased from Xilong Scientific Co. Ltd. High purity nitrogen was obtained from Nanjing Shangyuan Industrial Gas Plant. Distilled water was utilized in all experimental procedures.

Synthesis of the 3D Co complex architectures. Typically, 50 mg of urea was mixed with 1 mL of cobalt acetate aqueous solution (40 mg/mL) in a 4 mL glass bottle. Then, the uncovered bottle was transferred into a drying oven and maintained at 200 °C. The solvent evaporated quickly in 5 min. After the solid was further kept at 200 °C for 1.5 h, blue Co complex (CC-B) was collected at the bottom of the bottle. Final green 3D Co complex (CC-G) was obtained after CC-B was washed alternately with deionized water and ethanol and dried overnight at 80 °C. Schematic diagram of experimental preparation process is presented in Fig. S1.

Synthesis of SE-100 and SE-150. Comparatively, the bottles with the mixing solution of urea and cobalt acetate were put in drying oven at 100 °C and 150 °C for 1.5 h for slow evaporation. The obtained pink products were washed alternately with deionized water and ethanol, dried overnight at 80 °C and designed as SE-100 and SE-150, respectively.

Synthesis of CC-air and CC-N₂. CC-air and CC-N₂ were prepared through heat annealing treatment of 3D CC-G at the temperature of 600 °C for 3 h in the air or N₂ atmosphere with a heating rate of 2 °C/min.

Characterizations. Scanning electron microscopy (SEM) was performed on Hitachi S-4800 at 10 kV. Transmission electron microscopy (TEM) images were obtained by using a JEOL JEM-2100 transmission electron microscope operating at 200 kV. Powder X-ray diffraction (XRD) patterns were collected by using a Bruker D8 ADVANCE diffractometer with CuK α radiation ($\lambda = 1.5418$ Å). Infrared spectra were obtained on Fourier transform infrared spectroscopy (FT-IR, Nicolet 6700, Thermo Company). Thermogravimetric analysis of the powders were preformed in air on Pyris Diamond TG/ DTA (Perkin-Elmer). The flow rate of air was set at 120 mL/min and the temperature was increased from 30 to 800 °C at a rate of 10 °C/min. X-Ray photoelectron spectroscopy (XPS) was collected on an ESCALab MKII X-ray photoelectron spectrometer, using non-monochromatized Al_{K α} X-ray as excitation source. The binding energies were corrected for specimen charging by calibrating the C_{1s} peak to 284.6 eV. Element analysis (EA) was characterized by element analyzer (German Heraeus Company, CHN-0-Rapid). The percentage composition of cobalt was analyzed by Inductively Coupled Plasma (America PE Company, Optima 5300 DV).

Electrochemical measurements. The electrochemical measurements were performed in a conventional three electrode cell using a CHI760D (Shanghai Chenhua, China) electrochemical workstation with catalyst-coated glassy carbon (GC) as the working electrode, a Pt wire as the counter electrode and a saturated Ag/AgCl electrode as the reference electrode. The catalyst ink was prepared by blending 5 mg of the catalyst with $50 \,\mu$ L of Nafion solution (4 wt %) and $950 \,\mu$ L of water/isopropanol solution (3:1) via sonication. An amount of $5 \,\mu$ L of the dispersion was transferred onto the GC electrode with a catalyst loading of about 0.18 mg/cm². Then, the prepared catalyst electrode was dried at room temperature. With this electrode as the working electrode, electrochemical measurements were conducted in 1 M KOH solution. Polarization curves were obtained at a scan rate of 1 mV/s. Accelerated degradation measurement was conducted for 3000 cyclic voltammetry (CV) cycles at a scan rate of 50 mV/s. In all measurements, the reference electrode was calibrated with respect to reversible hydrogen electrode (RHE). All Polarization data were without iR-corrected.

Results and Discussion

Figures 1a and S1 briefly depict the facile synthesis process of the 3D Co complex. Figure 1b,c show SEM images of the product, revealing that the sample consists of plate-like 3D superstructures. The 3D plate is assembled by a great number of 2D nanosheets, which are curled and perpendicular to the plate surface, resulting in a lot of porous channels. TEM images in Fig. 1d,e also confirm the 3D porous superstructure of the product constructed by thin 2D nanosheets. Figure S2 shows the adsorption/desorption isothermals and the pore size distribution of the obtained Co-complex architecture, which clearly confirm the mesoporous structure with the pore sizes in the range of 3~12 nm. Since the solvent evaporation process is very quick and there is no enough time for slow crystallization in solution, no obvious diffraction peaks can be detected in the XRD patterns (Fig. 2a), suggesting the amorphous nature of the obtained Co product. Fourier transform infrared spectra (FTIR) is employed to analyze the information of functional groups. Figure 2b shows the FTIR spectra of the synthesized products and pure urea. It is obviously observed that both CC-B and CC-G have similar curves to urea, suggesting the remaining existence of urea molecules in the product, which would serve as ligands to coordinate with Co²⁺. Newly emerged peak at 2184.8 cm⁻¹ indicates the coordination bonding between Co²⁺ and ligands (urea). Compared to that of urea, stretching vibration of C-N in the product nearly disappears, suggesting possible interaction between



Figure 1. (a) The facile synthesis process of the 3D Co complex architecture; (b,c) SEM and (d,e) TEM images of the as-synthesized product.

 Co^{2+} and amino group that weakens the vibration of C-N. The specific peak at 1746.4 cm⁻¹ in CC-G may result from coordination bond of Co-O by the hydrolysis of CC-B and water. Energy dispersive spectrum (EDS) in Fig. 2c further proves the co-existence of multiple elements including C, O, N and Co (Si from substrate and Au from gold sputtering), confirming the obtained product could be a Co-based complex with the organic ligand. Elemental mappings (Fig. 2d~h) verify that the product is made up of C, O, N and Co in accordance with the EDS analysis, and these elements are uniformly dispersed in the product, suggesting that the product obtained through the fast solvent evaporation process is not cobalt oxides or hydroxides. It can be concluded by combining the XRD, IR and EDS results that the as-prepared product would be amorphous Co complex through the coordination between Co²⁺ and urea.

Since the OER mainly occurs on the surface of catalyst, it is necessary to study the surface properties with X-ray photoelectron spectroscopy (XPS). All of the binding energies in XPS were corrected for specimen charging by referencing to the C1s peak (set at 284.6 eV). Co, C, N and O were all detected from the overall XPS spectrum of the 3D Co complex in Fig. 3a. Figure 3b shows the fine-scanned C1s spectrum, in which the peaks of C1s locates at around 288.9 eV and 285.4 eV contributing to the carbonyl and C-N. Figure 3c shows that the sample has a obvious peak located at 399.9 eV, which should be attributed to the Co-N according to the NIST XPS data. The Co2p XPS spectrum is shown in Fig. 3d, in which the peaks of Co $2p^{3/2}$ and $Co2p^{1/2}$ are at around 781.0 eV and 796.9 eV, respectively, with a $\Delta E_{3/2-1/2}$ of 15.9 eV, indicating that the state of cobalt could be Co^{2+} . The O1s XPS spectrum in Fig. 3e shows three peaks locating at 532.7 eV, 531.7 eV, and 531 eV, which indicate the existence



Figure 2. (a) XRD patterns; (b) IR spectra; (c) EDS pattern and (**d**~**h**) elemental mappings of the as-synthesized product.



Figure 3. (a) Survey XPS spectrum; $(b \sim e)$ High resolution XPS spectra of (b) C1s, (c) N1s, (d) Co2p, (e) O1s and (f) TG-DTA and DSC curves of the as-synthesized product.

carbonyl oxygen and hydroxy oxygen. Inductively coupled plasma spectroscopy (ICP) and elemental analyzer (EA) are regarded as efficient means to characterize the content of elements. The ICP measurement demonstrates that the mass fraction of Co in Co complex is about 42.75%, and EA measurement shows the approximate atomic ratio of C:N:H is 1:1:3. Combining the existing urea, hydroxy, and possible carbonate according to the XPS and FTIR data, the structural formula of the Co-based complex can be inferred as $Co_2(CN_2H_4O)(OH)_2CO_3$, which has a Co content of 43.06% in accordance with the ICP and EA results. Figure 3f presents the thermogravimetric analysis (TGA) and differential scanning calorimeter (DSC) curves of the Co complex under air atmosphere. After the decomposition and the oxidation reaction of Co^{2+} to Co^{3+} , the sample has a remaining mass of 57.8% of initial mass, which is in accordance with the result of ICP and demonstrates the rationality of the proposed structural formula of the Co-based complex.

The formation of 3D $Co_2(CN_2H_4O)(OH)_2CO_3$ architecture goes through the following two steps: Firstly, the free Co ions react with urea and CO_3^{2-} produced by the decomposition of urea to form blue bulk Co complex with constant evaporation of solvent at 200 °C under open system, and secondly, the blue bulk Co complex can be transferred to green 3D Co complex architecture assembled by nanosheets through hydrolysis process by washing



Figure 4. (a) Polarization curves of different Co-based products and commercial IrO_2 in 0.1 M KOH aqueous solution; (b) The corresponding Tafel plots derived from polarization curves; (c) Polarization curves of the 3D Co complex architecture and commercial IrO_2 before and after cyclic voltammogram test for 100 cycles in 0.1 M KOH solution; (d) AC impedance spectra of the 3D Co complex architecture and commercial IrO_2 .

the blue Co complex with deionized water. The first step is the key for the formation of the 3D $Co_2(CN_2H_4O)$ $(OH)_2CO_3$ structure. The rapid solvent evaporation under high temperature prevents the total decomposition of the used urea. The remained urea can serve as ligand to coordinate with Co^{2+} in the following procedures. At the same time, the decomposition of part of urea results in the formation of CO_3^{2-} to balance the positive charges. The Co complex tends to form dense bulk in the concentrated system, then the 2D nanosheets produced by hydrolysis directly grow on the blue bulk for final 2D nanosheets-assembled 3D architecture. So, it is highly necessary to set high temperature condition for the rapid evaporation of solvent for the formation of 3D architecture. Conversely, relatively slow solvent-evaporation experiments at 100 °C and 150 °C were also conducted. Figure S3 shows the SEM images of the obtained products at 100 °C (SE-100) and 150 °C (SE-150), which both present disordered distribution of bulk nanoplates. XRD patterns (Fig. S4a) confirm that they are also amorphous in nature and the corresponding XPS spectra (Fig. S4b) demonstrate that the state of cobalt are also Co^{2+}_2 . However, the most of urea, which serve as ligand to form complex in RSE system, gradually decomposes into NH₃ and CO₂ in the relatively long time under low temperatures. Therefore, the 2D nanosheets-assembled 3D $Co_2(CN_2H_4O)$ (OH)₂CO₃ complex cann't be realized under low temperatures because of the absence of ligands.

The OER electrocatalytic activity were investigated in 1 M KOH aqueous solution using a standard three-electrode system. As shown in Fig. 4a, initial linear sweep voltammetry (LSV) curves recorded with the Co complex (CC) reveals a relatively lower overpotential of 360 mV at a current density of 10 mA/cm² than that of commercial IrO2 (400 mV) under the same conditions. Meanwhile, to highlight the superior OER performance, Co_3O_4 (CC-air) and Co_3C_x (CC-N₂) nanostructures were also synthesized by calcining the 3D Co complex under air or N₂ atmosphere. XRD patterns, IR, XPS spectra and SEM images shown in Figs S5 and S6 clearly demonstrate the complete transformation from Co complex to Co_3O_4 and Co_3C_x . The polarization curves of Co_3O_4 (CC-air) and Co_3C_x (CC-N₂) nanostructures are also displayed in Fig. 4a, whose onset potentials and overpotentials are higher than those of Co complex or IrO₂. The Tafel slope derived from polarization (Fig. 4b) was used to evaluate OER kinetics. Co complex has an approximately equivalent Tafel slope to commercial IrO₂, indicating the excellent OER activity at low overpotential. It is worth mentioning that the obtained Co_3O_4 has a lowest Tafel slope, contributing to an even higher enhancement in OER activity at high potential. Worse OER performance of magnetic CoC_x may be caused by the deletion of oxygen-containing functional groups under high temperature condition, which benefits the absorption of H₂O molecules on the catalyst. LSV curves after 100 CV cyclings at a scan rate of $0.1 \text{ V} \cdot \text{s}^{-1}$ were utilized to compare the stability of 3D Co complex and IrO₂. Figure 4c shows the almost completely coincident LSV curves, indicating the excellent stability for Co complex. The corresponding SEM images, XRD pattern and IR spectrum of the catalyst after 100 OER cycles shown in Fig. S8 demonstrate that the microstructure and composition of Co complex are highly stable. In contrast, the LSV of IrO₂ after 100



Figure 5. (a) CVs of CC in 1 M KOH at different scan rates from 10 to 50 mV/s; (b) Charging current density differences $(\Delta j = j_a - j_c)$ at 1.20 V vs. RHE plotted against the scan rate. The linear slope (equivalent to twice of the C_{dl}) was used to represent the ECSA.

CV cyclings has a big deviation. Figure S9a shows the LSV curves of 3D Co complex, IrO₂ and amorphous SE-100 after 1000 CV cyclings, which indicates the 3D Co complex has a remarkably superior performance for OER than IrO₂ or SE-100 in the long-term cycle testing, though it has a fine deviation compared to initial cycle that might result from possible fall off of catalyst on the surface of electrode. AC impedance spectra for Co complex and IrO_2 displayed in Fig. 4d indicate the low electron transfer impedance (R_{rl}) for Co complex Impedance data demonstrate that charge transfer and ions diffusion are not advantages versus that of IrO2. It is well accepted that the electrochemically active surface area (ECSA) of the catalysts can be estimated from measurements of the electrochemical double-layer capacitance $(C_{dl})^{12,14}$. The double layer capacitance (C_{dl}) , which can calculated by using cyclic voltammetric method, is proportional to ECSA. The bigger the effective electrochemical surface area, the more of the active sites. Figure 5a shows the CVs of the CC products, which reveals that the sample has an obvious increase in current density according to different scan rates. The calculated EDLC of CC is 12.25 mF/ cm² (Fig. 5b), superior to the reported OER electrocatalysts^{12,14}. The result indicates that the number of electrochemically active sites for water oxidation significantly increases due to the special structure. Compared with the control samples, CC has a highest specific surface area of 142.9 m²/g (Table S1). It is believed that the initial OER process in alkaline conditions involves adsorbed OH and O species on the active sites of catalysts based on the following scheme³².

$$OH^- + * \to OH^* + e^- \tag{1}$$

$$OH^* + OH^- \rightarrow O^* + H_2O + e^-$$
⁽²⁾

$$O^* + OH^- \to OOH^* + e^- \tag{3}$$

$$OOH^* + OH^- \rightarrow O_2 + H_2O + e^-$$
⁽⁴⁾

Summary: $4OH^- \rightarrow O_2 + 2H_2O + 4e^-$

Step (1) and (2) play an important role to determine the overall OER rate, while reactions (3), (4) for O_2 production are inversible and fast³³. Therefore, the adsorption energy of OH⁻ greatly affects the OER process. 2D nanosheets arrays with abundant hydrophilic functional groups have high specific surface area for the contact of catalyst and OH⁻ or H₂O. Meanwhile, amorphous structure supplied abundant defect sites for OER³⁴⁻³⁶. More importantly, the self-assembly of 2D nanosheets for stable 3D architecture that avoids the undesirable re-stack and condense of 2D nanosheets can remarkably improve the duration performance of catalyst for OER.

Conclusions

In summary, we firstly proposed an extremely facile route called "rapid solvent-evaporation strategy" to prepare 3D Co complex architectures constructed by 2D nanosheet blocks as highly efficient catalyst of OER. The new-type Co-based catalyst exhibited excellent electrocatalytic activity of OER with a low onset potential and a lower overpotential of 360 mV at a current density of 10 mA·cm⁻² than that of commercial IrO₂. Abundant active sites and hydrophilic group of 2D nanosheets, amorphous and stable 3D structure, cooperatively achieved the superior catalyzing OER performance. It is expected that the 3D Co complex prepared by simple, green and low-cost method would be as promising OER catalyst for large-scale application to replace expensive and earth-rare IrO₂. Meanwhile, it is believed that through the simple "rapid solvent-evaporation" strategy other novel 3D structures with great application potential could be obtained in the future. Received: 19 December 2018; Accepted: 11 October 2019; Published online: 30 October 2019

References

- 1. Zare, M. *et al.* Evolution of rough-surface geometry and crystalline structures of aligned TiO₂ nanotubes for photoelectrochemical water splitting. *Sci. Rep.* **8**, 10870 (2018).
- Naseri, N. et al. Microstructure, morphology and electrochemical properties of Co nanoflake water oxidation electrocatalyst at micro- and nanoscale. RSC Adv. 7, 12923–12930 (2017).
- 3. Naseri, N. *et al.* How morphological surface parameters are correlated with electrocatalytic performance of cobalt-based nanostructures. *J. Ind. Eng. Chem.* **57**, 97–103 (2018).
- 4. Wang, J. H. *et al.* Recent progress in cobalt-based heterogeneous catalysts for electrochemical water splitting. *Adv. Mater.* 28, 215–230 (2016).
- 5. Sun, Y. F. *et al.* Atomically-thin non-layered cobalt oxide porous sheets for highly efficient oxygen-evolving electrocatalysts. *Chem. Sci.* **5**, 3976–3982 (2014).
- 6. Yuan, C. Z. *et al.* Cobalt phosphate nanoparticles decorated with nitrogen-doped carbon layers as highly active and stable electrocatalysts for the oxygen evolution reaction. *J. Mater. Chem. A* **4**, 8155–8160 (2016).
- Kanan, M. W. & Nocera, D. G. In situ formation of an oxygen-evolving catalyst in neutral water containing phosphate and Co²⁺. Science 321, 1072–1075 (2008).
- Xu, L. et al. Plasma-engraved Co₃O₄ nanosheets with oxygen vacancies and high surface area for the oxygen evolution reaction. Angew. Chem. Int. Ed. 55, 5277-5281 (2016).
- Rosen, J., Hutchings, G. S. & Jiao, F. Ordered mesoporous cobalt oxide as highly efficient oxygen evolution catalyst. J. Am. Chem. Soc. 135, 4516–4521 (2013).
- Zou, X. X., Goswami, A. & Asefa, T. Efficient noble metal-free (electro)catalysis of water and alcohol oxidations by zinc-cobalt layered double hydroxide. J. Am. Chem. Soc. 135, 17242–17245 (2013).
- Zhao, Z. L., Wu, H. X., He, H. L., Xu, X. L. & Jin, Y. D. A high-performance binary Ni-Co hydroxide-based water oxidation electrode with three-dimensional coaxial nanotube array structure. Adv. Funct. Mater. 24, 4698–4705 (2014).
- 12. Jiang, Y. M., Li, X., Wang, T. X. & Wang, C. M. Enhanced electrocatalytic oxygen evolution of α -Co(OH)₂ nanosheets on carbon nanotube/polyimide films. *Nanoscale* **8**, 9667–9675 (2016).
- 13. Popczun, E. J., Read, C. G., Roske, C. W., Lewis, N. S. & Schaak, R. E. Highly active electrocatalysis of the hydrogen evolution reaction by cobalt phosphide nanoparticles. *Angew. Chem. Int. Ed.* **53**, 5427–5430 (2014).
- 14. Liu, Y. Y. *et al.* Electrochemical tuning of olivine-type lithium transition-metal phosphates as efficient water oxidation catalysts. *Energy Environ. Sci.* **8**, 1719–1724 (2015).
- Sun, Y. J. et al. Electrodeposited cobalt-sulfide catalyst for electrochemical and photoelectrochemical hydrogen generation from water. J. Am. Chem. Soc. 135, 17699–17702 (2013).
- Liu, Y. W. et al. Low overpotential in vacancy-rich ultrathin CoSe₂ nanosheets for water oxidation. J. Am. Chem. Soc. 136, 15670–15675 (2014).
- 17. Huang, J. H. et al. CoOOH nanosheets with high mass activity for water oxidation. Angew. Chem. Int. Ed. 54, 8722-8727 (2015).
- 18. Zhang, B. et al. Homogeneously dispersed multimetal oxygen-evolving catalysts. Science 352, 333-337 (2016).
- Jiang, M., Li, Y. J., Lu, Z. Y., Sun, X. M. & Duan, X. Binary nickel-iron nitride nanoarrays as bifunctional electrocatalysts for overall water splitting. *Inorg. Chem. Front.* 3, 630–634 (2016).
- Zhong, X. et al. Synergistic effect of nitrogen in cobalt nitride and nitrogen-doped hollow carbon spheres for the oxygen reduction reaction. ChemCatChem 7, 1826–1832 (2015).
- Yoon, K. R. et al. Brush-like cobalt nitride anchored carbon nanofiber membrane: Current collector-catalyst integrated cathode for long cycle Li-O₂ batteries. ACS Nano 12, 128–139 (2018).
- Ullman, A. M., Brodsky, C. N., Li, N., Zheng, S. L. & Nocera, D. G. Probing edge site reactivity of oxidic cobalt water oxidation catalysts. J. Am. Chem. Soc. 138, 4229–4236 (2016).
- Bergmann, A., Martinez-Moreno, E., Teschner, D., Chernev, P. & Gliech, M. Reversible amorphization and the catalytically active state of crystalline Co₃O₄ during oxygen evolution. *Nat. Commun.* 6, 8625 (2015).
- 24. Kataoka, Y. *et al.* Photocatalytic hydrogen production from water using porous material [Ru₂(p-BDC)₂]_n. *Energy Environ. Sci.* **2**, 397–400 (2009).
- Levchenko, T. I. et al. Controlled solvothermal routes to hierarchical 3D superparticles of nanoscopic CdS. Chem. Mater. 27, 3666–3682 (2015).
- 26. Ming, T. et al. Ordered gold nanostructure assemblies formed by droplet evaporation. Angew. Chem. Int. Ed. 47, 9685–9690 (2008).
- 27. Wang, P. P. et al. Zinc sulfide nanosheet-based hybrid superlattices with tunable architectures showing enhanced photoelectrochemical properties. Small 11, 3909-3915 (2015).
- Dang, L. Y., Wei, C. Z., Ma, H. F., Lu, Q. Y. & Gao, F. Three-dimensional honeycomb-like networks of birnessite manganese oxide assembled by ultrathin two-dimensional nanosheets with enhanced Li-ion battery performances. *Nanoscale* 7, 8101–8109 (2015).
- 29. Li, H. Y. & Wang, X. Three-dimensional architectures constructed using two-dimensional nanosheets. Sci. China Chem. 58, 1792–1799 (2015).
- Liao, C. W., Lin, Y. S., Chanda, K., Song, Y. F. & Huang, M. H. Formation of diverse supercrystals from self-assembly of a variety of polyhedral gold nanocrystals. J. Am. Chem. Soc. 135, 2684–2693 (2013).
- Zhao, Y. et al. Small-molecule-directed nanoparticle assembly towards stimuli-responsive nanocomposites. Nat. Mater. 8, 979–985 (2009).
- Huang, W., Zuo, Z. J., Han, P. D., Li, Z. H. & Zhao, T. D. XPS and XRD investigation of Co/Pd/TiO₂ catalysts by different preparation methods. J. Electron. Spectrosc. 173, 88–95 (2009).
- Bajdich, M., García-Mota, M., Vojvodic, A., Nørskov, J. K. & Bell, A. T. Theoretical investigation of the activity of cobalt oxides for the electrochemical oxidation of water. J. Am. Chem. Soc. 135, 13521–13530 (2013).
- 34. Xu, K. et al. Metallic nickel nitride nanosheets realizing enhanced electrochemical water oxidation. J. Am. Chem. Soc. 137, 4119–4125 (2015).
- Yang, Y., Fei, H. L., Ruan, G. D., Xiang, C. S. & Tour, J. M. Efficient electrocatalytic oxygen evolution on amorphous nickel-cobalt binary oxide nanoporous layers. ACS Nano 8, 9518–9523 (2014).
- 36. Smith, R. D. L. *et al.* Photochemical route for accessing amorphous metal oxide materials for water oxidation catalysis. *Science* **340**, 60–63 (2013).

Acknowledgements

This work is supported by the National Natural Science Foundation of China (Grant No. 21471076, 21871130 and U1832137) and a Project Funded by the Priority Academic Program Development of Jiangsu Higher Education Institutions.

Author contributions

Q.L. and F.G. guided the entire project, carried out data analyses and co-wrote the manuscript. H.J. and H.Z. performed the experiments, XRD characterizations, and SEM investigations. H.M. and Y.T. performed OER experiments. Q.K. performed OER experiments and characterized the samples after OER. All the co-authors discussed the results and commented on the manuscript.

Competing interests

The authors declare no competing interests.

Additional information

Supplementary information is available for this paper at https://doi.org/10.1038/s41598-019-51979-z.

Correspondence and requests for materials should be addressed to F.G. or Q.L.

Reprints and permissions information is available at www.nature.com/reprints.

Publisher's note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit http://creativecommons.org/licenses/by/4.0/.

© The Author(s) 2019