SCIENTIFIC REPORTS

Received: 23 June 2017 Accepted: 4 September 2017 Published online: 20 September 2017

OPEN Characterization of Methane Excess and Absolute Adsorption in Various **Clay Nanopores from Molecular** Simulation

Yuanyuan Tian^{1,2,3}, Changhui Yan^{1,2} & Zhehui Jin³

In this work, we use grand canonical Monte Carlo (GCMC) simulation to study methane adsorption in various clay nanopores and analyze different approaches to characterize the absolute adsorption. As an important constituent of shale, clay minerals can have significant amount of nanopores, which greatly contribute to the gas-in-place in shale. In previous works, absolute adsorption is often calculated from the excess adsorption and bulk liquid phase density of absorbate. We find that methane adsorbed phase density keeps increasing with pressure up to 80 MPa. Even with updated adsorbed phase density from GCMC, there is a significant error in absolute adsorption calculation. Thus, we propose to use the excess adsorption and adsorbed phase volume to calculate absolute adsorption and reduce the discrepancy to less than 3% at high pressure conditions. We also find that the supercritical Dubinin-Radushkevich (SDR) fitting method which is commonly used in experiments to convert the excess adsorption to absolute adsorption may not have a solid physical foundation for methane adsorption. The methane excess and absolute adsorptions per specific surface area are similar for different clay minerals in line with previous experimental data. In mesopores, the excess and absolute adsorptions per specific surface area become insensitive to pore size. Our work should provide important fundamental understandings and insights into accurate estimation of gas-in-place in shale reservoirs.

Coupling with the growing global energy demands and continuous depletion of conventional energy resources, as an important unconventional energy supply, shale gas has garnered more and more attentions in recent years¹⁻⁴. Comparing to conventional reservoirs, shale has a heterogeneous composition consisting of organic and inorganic matters. Organic matter is mainly composed of kerogen, while inorganic materials consist of clay minerals, quartz, and calcite, etc. Due to large amount of nanoscale pores, adsorbed gas can provide a significant source for gas-in-place in shale. It is reported that the adsorbed gas may reach 85% of gas-in-place in shale reservoir³. Both kerogen and clay minerals can have significant amount of nanopores⁵. According to shale component analysis, the amount of clay minerals can be in the range from 25% to 70%⁶. Recently, it is reported that clay minerals provide significant amount of nanopores and specific surface area for the transitional shales in China⁷. Experimental researches have shown that clay minerals can significantly contribute to adsorbed gas capacity⁸⁻¹⁰.

There have been a number of experimental studies on gas adsorption in clay minerals and its contribution to gas-in-place in shale. Ross and Bustin compared adsorption capacities of various clay minerals on both dry and moisture basis¹¹. They reported that illite and montmorillonite have larger adsorption capacities than kaolinite at dry condition, but opposite is true for moisturized condition. Liu and his coworkers found that montmorillonite has larger adsorption capacity than kaolinite, while illite has the smallest adsorption capacity¹², which is in line with Zhang et al.¹³ and Ji et al.¹⁴. In addition, Ji et al.¹⁴ found that the specific surface area (SSA) is the main control on methane adsorption capacity in various clay minerals.

In experiment, there are two commonly used methods to study gas adsorption: gravimetric and volumetric method. Gravimetric method uses magnetic suspension balance to obtain adsorption isotherms¹⁵. It measures the

¹State Key Laboratory of Oil and Gas Reservoir Geology and Exploitation (Chengdu University of Technology), Chengdu, 610059, Sichuan, P.R. China. ²College of Energy, Chengdu University of Technology, Chengdu, 610059, Sichuan, P.R. China. ³School of Mining and Petroleum Engineering, Faculty of Engineering, University of Alberta, Edmonton, T6G 1H9, Canada. Correspondence and requests for materials should be addressed to Z.J. (email: zhehui2@ualberta.ca)



Figure 1. The schematic representation of free gas, excess adsorption, and absolute adsorption of methane in slit-nanopores. The yellow area represents adsorbed phase and the green area depicts free gas phase, respectively.

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excess adsorption capacity m_{ex} based on the difference between gravity and buoyancy¹⁶. In practice, it is difficult to assess the buoyancy accurately. As a result, gravimetric method may become unsuitable to obtain the total gas uptake m_{tot} . On the other hand, volumetric method can measure m_{tot} in porous media¹⁷ and m_{ex} is obtained by subtracting the amount of free gas in total accessible pore volume V_p from m_{tot} , ^{18,19}:

$$m_{ex} = m_{tot} - \rho_b V_p, \tag{1}$$

where ρ_b is bulk gas density. Methane adsorption in nanopores is generally considered as single-layered adsorption as we will show in section 3.2²⁰. When pore size is large, methane density profile in the middle of pores approaches bulk density²¹. As a result, one can assume that methane adsorption in nanopore can be divided into adsorbed and free gas regions as shown in Fig. 1. While free gas has a density of ρ_b , adsorbed phase density ρ_a can be higher than ρ_b . Absolute adsorption m_{abs} is defined as the adsorbed amount in the adsorbed phase (yellow area in Fig. 1)²². Based on the adsorption model shown in Fig. 1, only the adsorbed phase contributes to m_{abs} and m_{ex} . With these assumptions, m_{abs} can be obtained from m_{ex} and $\rho_a^{4,23,24}$,

$$n_{abs} = \frac{m_{ex}}{1 - \rho_b / \rho_a}.$$
(2)

The key to calculate m_{abs} in Eq. (2) is to accurately obtain ρ_a . In previous works, ρ_a was assumed to be a constant as the liquid density of methane at normal boiling point, 420 kg/m^{3 25-27}. However, it is well known that adsorbed phase density changes with pressure and temperature²¹. Gensterblum *et al.*²⁸⁻³⁰ obtained ρ_a by using a least-squares minimization procedure based on the modified Langmuir equation and Dubinin-Radushkevitch (DR) equation to match excess adsorption data.

Recently, Xiong *et al.*³¹ used Langmuir, Supercritical DR (SDR), and Ono-Kondo model to correct the excess adsorption to the absolute adsorption assuming constant ρ_a as the density of liquid methane at boiling temperature, 420 kg/m³ or the methane density at the critical point, 373 kg/m³. However, these methods are rather a pure curve fitting without reliable physical mechanisms. Another way to obtain m_{abs} is based on the adsorbed phase volume m_{ex} and $V_a^{4,32}$,

$$m_{abs} = m_{ex} + \rho_b V_a. \tag{3}$$

Zhang *et al.*³³ assumed that absolute adsorption stays constant at high pressures and calculated V_a by taking derivative of excess adsorption with respect to the bulk density. However, there is no rigorous proof of constant methane absolute adsorption even at high pressure conditions.

Comparing to experimental measurements, molecular simulation can explore wider range of pressure and temperature conditions³⁴ and provide underlying mechanisms of gas adsorption from molecular perspective. There have been a number of molecular simulation works on methane adsorption in clay minerals and the calculation of excess and absolute adsorption. Jin and Firoozabadi studied methane adsorption in montmorillonite clay and found that adsorption is mainly dominated by surface area²¹. Chen *et al.*³⁴ and Xiong *et al.*³⁵ found that excess adsorption capacity of methane in various clay nanopores decreases with pore size. In another work, Chen *et al.*³⁶ reported that CO₂ and N₂ excess adsorption in mesopores does not change with pore size. On the other hand, Zhang *et al.*³³ reported that at high pressures, methane excess adsorption can be negative. They also defined the saturated adsorbed phase density $\rho_{a,s}$ obtained from the linear intercept of the excess adsorption with the bulk gas density at pressure *P_s*. They assumed that ρ_a becomes constant when the pressure is higher than *P_s*. When pressure is lower than *P_s*, ρ_a is obtained from the Langmuir curve fitting to ρ_a . However, the excess

adsorption is very sensitive to pore volume³⁷. Varying excess adsorption can be obtained based on different pore volume characterizations³⁸. Thus, the accurate characterization of pore volume is essential for molecular simulation to compare with experimental measurement. In addition, calculation of the adsorbed phase density/volume and absolute adsorption still remains as a daunting challenge for scientists and engineers.

In this work, we use grand canonical Monte Carlo (GCMC) simulations^{21,34,36,38} to study methane adsorption in illite, montmorillonite, and kaolinite nanopores. X-ray diffraction reveals that they are the main constituent of clay minerals in shale^{12,39}. These three clay minerals are represented by full atomistic models and methane molecules are depicted by the single-site Lennard Jones (LJ) particles. We explicitly consider intermolecular interactions between methane molecules and clay atoms. We assume that the inter-pore interactions are negligible and methane adsorbs in nanometer slit-like pores²¹, which is one of the main pore shapes in shale reservoirs⁴⁰⁻⁴². Based on low pressure nitrogen adsorption^{43,44}, in addition to slit-like pores, shale can have other shaped pores such as ink-bottle^{45,46}. Although the adsorption behavior and density distributions may be different in various pore geometries, the focus of this work is to characterize the excess and absolute adsorption in a slit geometry.

We calculate the methane excess adsorption in a similar way to experimental volumetric method: 1) by using helium adsorption in nanopores from GCMC simulation, we obtain the effective pore volume V_p . The details of effective pore volume calculation are shown in Supplementary Information. 2) Then, subtract the free gas occupied by V_p from m_{tot} , which is calculated from GCMC simulations for given temperature and pressure conditions, as in Eq. (1). We will also compare the absolute adsorption from m_{ex} and V_a to that from m_{ex} and ρ_a . In fact, using m_{ex} and ρ_a in Eq. (2) may bring significant error in the calculation of m_{abs} .

In addition, we will assess SDR model³¹ used in experiments to correct the excess adsorption to the absolute adsorption to provide insights into experimental fitting. Then, we will compare methane excess and absolute adsorptions in illite, montmorillonite and kaolinite nanopores.

The remainder of this paper is organized as follows. In section 2, we introduce the molecular simulation method and define the molecular models. In section 3, we first calibrate our GCMC simulation by comparing to experimental measurements and other molecular simulation works. Then, we compare different approaches to calculate absolute adsorption and assess SDR model. In section 4, we investigate methane adsorption in various clay-like slit pores with various pore sizes and temperatures. In section 5, we summarize key findings and implications.

Molecular Model and Simulation

Clay Minerals. In this subsection, we introduce the molecular configurations of illite, montmorillonite, and kaolinite nanopores. The clay atoms are fixed throughout our simulation.

Illite. Illite is one type of 2:1 clays consisting of two Si-O tetrahedral layers and one Al-O octahedral layer⁴⁷. The original illite unit cell is Si₂AlO₅(OH), and the coordinate of each atom is from Pyrophyllite-1Tc powder diffraction^{42,47,48}. The unit cell parameters are a = 0.51602 nm, b = 0.89663 nm, c = 0.93476 nm, $\alpha = 91.184^{\circ}$, $\beta = 100.464^{\circ}$, and $\gamma = 89.752^{\circ 49}$. We duplicate the original unit cell in the (-x, -y, -z) direction and then the resulting structure is duplicated by displacing it a distance a in the x direction. The formula for the resulting 40-atom unit cell is Si₈Al₄O₂₀(OH)₄. The simulation cell consists of two clay sheets and each sheet contains 32 unit cells which is the result of replicating unit cells as $8 \times 4 \times 1$. They have the dimension of $L_x = 4.128 \text{ nm}$ and $L_y = 3.584 \text{ nm}$ in x direction and y direction, respectively. Two clay sheets are separated by a fixed distance to represent illite nanopore. The pore size W is defined as the distance between the center of mass of oxygen atoms in the inner planes of the two sheets. In each 40-atom unit cell, one silica atom in tetrahedral layers is replaced by aluminum atom so that the clay sheet has a negative charge. The negative charge is neutralized by potassium ions distributed in pore space³⁶. Unlike the clay atoms, these potassium ions are mobile in our simulation. With cation exchange, the unit cell formula is K(Si₇Al)Al₄O₂₀(OH)³³. The resulting structure of illite nanopore is shown in Fig. 2a.

Montmorillonite. Montmorillonite is also 2:1 clay consisting of one Al-O layer and two Si-O layers. The neutral montmorillonite clay has a unit cell formula as $Si_8Al_4O_{20}(OH)_4^{50}$. We adopted the atomic coordinates reported by Skipper *et al.*^{21,51,52}. We use two clay sheets with 32 unit cells (8 × 4 × 1) in each to form montmorillonite clay nanopores with a patch of $L_x = 4.224$ nm and $L_y = 3.656$ nm in x - y plane and thickness of 0.656 nm. Montmorillonite clay also has cation exchange capability with one aluminum atom replaced by magnesium atom in every 8 aluminum atoms in the octahedral sheet, and one silica atom replaced by aluminum atolm in every 32 silica atoms in tetrahedral sheet⁵². In our work, the negative charge is compensated by sodium ions. Similar to K-illite, these sodium ions are mobile in our simulation. The unit cell formula of Na-montmorillonite is $Na_{0.75}(Si_{7.75}Al_{0.25})(Al_{3.5}Mg_{0.5})O_{20}(OH)_4$. The pore size *W* is defined as the distance between the inner planes of the two sheets²¹. The resulting structure of montmorillonite nanopore is shown in Fig. 2b.

Kaolinite. Unlike illite and montmorillonite, kaolinite is one type of 1:1 clays composed of one Si-O tetrahedral and one Al-O octahedral sheet. X-ray diffraction analysis has shown that kaolinite has unit cell parameters as a = 0.5153 nm, b = 0.8941 nm, c = 0.7403 nm, $\alpha = 91.692^{\circ}$, $\beta = 104.86^{\circ}$, and $\gamma = 89.822^{\circ}5^{3}$. The formula of kaolinite unit cell is Al₄Si₄O₁₀(OH)₈⁵⁴. We replicate the unit cell 8 times in *x* direction and 4 times in *y* direction. The resulting structure has a surface area as 4.1232 nm × 3.5768 nm. Two clay sheets are separated by a fixed distance to represent kaolinite nanopore and the Al-O plane is the inner plane of pore. The schematic representation of kaolinite nanopore is shown in Fig. 2c.



Figure 2. The schematic representation of clay nanopores (**a**) K-illite; (**b**) Na-montmorillonite; (**c**) kaolinite. Red spheres are Si atoms, green spheres are Al atoms, yellow spheres are O atoms, magenta spheres are H atoms, cyan spheres are Mg atoms, orange spheres are K⁺ ions and blue spheres are Na⁺ ions. The *z* direction is perpendicular to the clay surfaces. The origin of the Cartesian coordinate is placed at the center of the simulation box.

Molecular Model and Force Fields. In our simulation, we use a single site model to describe methane and helium molecules. The TraPPE force field is used to represent the methane intermolecular interactions⁵⁵. The interactions between methane/helium and clay atoms, and interlayer ions, and other methane/helium molecules are described by the pairwise-additive LJ 12-6 potentials:

$$u_{LJ}(r_{ij}) = 4\varepsilon_{ij} \left[\left(\frac{\sigma_{ij}}{r_{ij}} \right)^{12} - \left(\frac{\sigma_{ij}}{r_{ij}} \right)^{6} \right], \tag{4}$$

where r_{ij} , ε_{ij} , and σ_{ij} are the separation, LJ well depth, and LJ size, respectively. The cross interactions between the unlike atoms and molecules, *i* and *j*, are computed using the standard Lorentz-Berthelot combining rules⁵⁶:

$$\sigma_{ij} = (\sigma_{ii} + \sigma_{jj})/2, \tag{5}$$

$$\varepsilon_{ij} = \sqrt{\varepsilon_{ii}\varepsilon_{jj}}.$$
(6)

The ε and σ are 148.0 K and 0.373 nm, respectively, for methane molecules⁵⁵, and 10.9 K and 0.264 nm, respectively, for helium molecules⁵⁷. The interaction between interlayer ions and clay atoms are described by the sum of LJ and electrostatic interaction,

$$u(r_{ij}) = u^{LJ} + u^{C} = 4\varepsilon_{ij} \left[\left(\frac{\sigma_{ij}}{r_{ij}} \right)^{12} - \left(\frac{\sigma_{ij}}{r_{ij}} \right)^{6} \right] + \frac{q_{i}q_{j}}{4\pi\varepsilon_{0}r_{ij}},$$
(7)

in which q_i is the partial charge of the site. We use CLAYFF force field⁵⁸ to describe clay atoms and interlayer ions. The short-range LJ interactions are truncated at a distance of 1.07 nm without shift. Similar to our previous works^{21,59}, to account for the long-range electrostatic interactions and the slab geometry that is periodic in x - y plane and has a finite length in z direction, we place a slab of vacuum in the simulation cell along the z direction with a length much larger than L_x or L_y and use the standard three-dimensional *Ewald* summation with a correction term^{60,61}.

Simulations. Methane adsorption in various clay minerals are performed in the grand canonical (μVT) ensemble with simulation cell as a rectangular box with periodicity in *x* and *y* directions. The box sizes in *x* and *y* directions are represented by L_x and L_y , respectively. The length in the *z* direction is determined by the pore size of the clay and the vacuum²¹.

For simulations of methane molecules in clay nanopores, in each MC cycle, a trial random displacement is applied to randomly selected methane molecules and a methane molecule is randomly removed from or inserted into the simulation box at equal probability depending on the chemical potential of the methane reservoir outside²¹. The chemical potential of methane molecules is obtained from the Widom's particle insertion method⁶² in canonical (*NVT*) ensemble without confinement. The bulk densities at given pressure and temperature are obtained from National Institute of Standards and Technology (NIST) Chemistry Webbook. The MC moves are implemented by the Metropolis algorithm⁶³. The simulation consists of 0.2 million MC cycles per absorbate molecules for sampling density profiles.

Effective Pore Volume. The calculation of effective pore volume is essential to the determination of the excess adsorption and consequently the absolute adsorption. In volumetric method, the effective pore volume V_p is obtained by the helium adsorption^{13,14}, based on the assumption that helium adsorption in nanopores is negligible and overall uptake is mainly dominated by pore filling. Zhang *et al.*¹³ obtained the void pore volume of different clay minerals from helium adsorption at pressure ranging from 6.9 to 150 bar with the average of five measurements. They postulated that the void volume remains constant over various temperatures. Similar to their experiment, in this work, we use helium adsorption at 333.15 K to calculate V_p given as

$$V_p = \frac{\langle N_{He} \rangle}{N_A \rho_{He,b}^m},\tag{8}$$

where $\langle N_{He} \rangle$ is the ensemble averaged number of helium molecules in given nanopores, N_A is the Avogadro constant, and $\rho_{He,b}^m$ is the bulk molar density of helium at given pressures. We use five bulk pressures of 2, 4, 6, 8, and 10 MPa. The effective pore volume is the average of these five pressure conditions. We find that V_p is independent of pressure and temperature. Due to the finite size of helium molecules, V_p is less than simply multiplying the pore width W and the surface area S_A . Chen *et al.*³⁸ claimed that using $W \times S_A$ as pore volume may underestimate the excess adsorption. Details about the helium adsorption and pore volume calculations are presented in the Supplementary Information.

Results and Discussion

Calibration of GCMC Simulation. To calibrate our GCMC simulation, we compare the methane excess adsorption in illite, montmorillonite and kaolinite from our simulations to experimental measurements and other molecular simulation works^{14,22,34,64}. Most of experimental data was reported as per unit mass of the adsorbent (i.e., mmol/g). In a recent work, Chen *et al.*³⁸ claimed that to have a fair comparison between experiment and molecular simulation, one need to use per unit surface area of the adsorbent (i.e., mmol/m²). In experiment, SSA is generally obtained from the nitrogen adsorption¹⁴. In our simulation, we use the area of two x - y plane to describe SSA. The excess adsorption per surface area m_{ex} is given as

$$m_{ex} = \frac{\langle N_{C_1} \rangle / N_A - V_p \rho_{C_1, b}^m}{2S_A}.$$
(9)

where $\langle N_{C_1} \rangle$ is the ensemble averaged number of methane molecules in the given nanopore and $\rho_{C_1,b}^m$ is the bulk molar density of methane at a given pressure.

In Fig. 3, we present mer from our GCMC simulation and experimental data as well as other molecular simulation in various clay minerals. We use 41 m²/g⁶⁵, 71.5 m²/g¹⁴, and 23.5 m²/g⁶⁶ as SSA of illite, montmorillonite, and kaolinite from experiments, respectively. Heller and Zoback²² originally reported the absolute adsorption, assuming ρ_c as saturated liquid density of 420 kg/m^{3 27}. We convert their absolute adsorption data to excess adsorption."Overall, our simulation result is in a general agreement with experimental data. There are some discrepancies probably due to following reasons: (1) SSA in experiment is obtained from low pressure nitrogen adsorption. Low pressure nitrogen adsorption is considered to explore the mesopores (pore size from 2 to 50 nm) in porous media⁴¹, while nitrogen molecules may not penetrate into micropores (pore size less than 2 nm). As a result, SSA might be underestimated⁶⁷. In addition, SSA from nitrogen adsorption is affected by many subjective factors, such as crushed sample particle size¹⁴ and fitting range in the BET model⁴². (2) On the other hand, methane molecules can be adsorbed in both micropores and mesopores and surface adsorption in micropores can be significant²⁰. (3) We simulated idealized condition as every pore is accessible to methane and helium molecules. However, in experiments, some of pores may be accessible to helium but not to methane molecules³⁸. Chalmers et $al.^{41}$ reported that shales can have large amount of nanopores with diameters down to 0.3 nm. While helium molecules may adsorb in such small nanopores, methane may not penetrate into. Considering these factors, our simulation is in a reasonable agreement with experimental data. In addition, our simulation shows excellent agreement with GCMC simulation by Chen *et al.*³⁴ in 2 nm illite pores at 363.15 K, which is obtained from their total adsorption with V_p from our calculations.

We observe that when the pore size is larger than 2 nm, m_{ex} becomes insensitive to W. In other GCMC simulation works by Chen *et al.*³⁴ and Xiong *et al.*³⁵, the excess adsorption per SSA decreases with W. Interestingly, in another work, Chen and his workers found that excess adsorption per specific surface area of CO₂ and N₂ does not change with pore size when $W \ge 2 \text{ nm}^{36}$. In fact, the excess adsorption is very sensitive to the calculation of V_p^{38} . Do *et al.*³⁷ have shown that a small change in V_p can have significant effect on excess adsorption. Our calculation reveals that if V_p is obtained from helium adsorption, the excess adsorption becomes similar when $W \ge 2 \text{ nm}$. In entire pressure range, unlike other simulation works^{33,38}, the excess adsorption is always positive. As a result, the calculation of the adsorbed layer density from the intercept of excess adsorption³³ may not be applicable.

Density profiles. To better understand methane adsorption behavior in clay nanopores, in Fig. 4, we present the methane density distributions in 4-nm illite, montmorillonite, and kaolinite nanopores at various bulk pressures and 333.15 K. For all bulk pressure conditions, methane forms a strong adsorption layer and the density in the middle of the pores approaches bulk. In general, as bulk pressure increases, the adsorption layer density increases. As a result, using a constant adsorbed layer density²² may not be justifiable to predict the absolute adsorption in Eq. (2). At high pressure conditions, e.g. bulk pressure of 50 MPa, methane may form a weak second



Figure 3. Excess adsorption from our GCMC simulations, experimental measurements by Ji *et al.*¹⁴, Fan *et al.*⁶⁴, and Heller and Zoback²² as well as GCMC simulation by Chen *et al.*³⁴ in (**a**) illite; (**b**) montmorillonite; (**c**) kaolinite.

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adsorption layer. However, as we will discuss later, such second adsorption layer may be "averaged out" by the saddle point between the first and second adsorption layers and the averaged density in the second adsorption layer becomes comparable to the bulk density. At relatively lower pressures, e.g. bulk pressure of 10 MPa, after the first adsorption layer, the density is slightly higher than the bulk density, which may indicate a transition zone in density distributions⁶⁸. Such transition zone may significantly affect absolute adsorption calculation.

To understand the effect of pore size on methane adsorption, we present the density distributions in illite nanopores of varying pore widths at 333.15 K in Fig. 5. At P = 50 MPa, except W = 1 nm, methane can form two adsorption layers on the surface. Within 1 nm pores, due to limited pore space, methane can form only one adsorption layer on the surface. When $W \ge 4$ nm, the density in the middle of the pores approaches bulk, while methane shows varying density distributions in the pores for $W \le 2$ nm. As a result, for $W \le 2$ nm, the adsorption model proposed in Fig. 1 may become inapplicable. On the other hand, at P = 5 MPa, methane can form one



Figure 4. Methane density profiles in (**a**) illite; (**b**) montmorillonite; (**c**) kaolinite nanopores of W = 4 nm at 333.15 K with varying bulk pressures. The solid lines are density distributions from GCMC simulation, dashed lines are the guidelines for bulk densities from NIST Chemistry Webbook, and the dotted lines are the guidelines for the first and second adsorption layers.

adsorption layer on the surface and transition zone is observed for 4 and 8 nm pores. It has been debated in the past on whether methane adsorption mechanism in shale is single layer adsorption or micropore filling^{11,69-73}. Our results indicate that adsorption mechanism varies by pore size. In mesopores defined by IUPAC⁷⁴, methane can form strong adsorption layer on the surface and density in the middle of the approaches bulk density. In micropores, due to limited pore space, methane can only form adsorption layer on the surface. It means that there is no free gas region in micropores, and absolute adsorption capacity is the same as the total methane capacity. These results agree with the past works^{4,75,76} that under in-situ shale reservoir condition, methane fills micropores most, and monolayer adsorption occurs in larger pores.

Absolute adsorption. In this subsection, we use methane adsorption in illite at 333.15 K as an example to compare different approaches to calculate the absolute adsorption.

As shown in Figs 4 and 5, in micropores, the surface adsorption dominates and the density in the middle of pore does not approach bulk. In mesopores, however, methane can have a strong adsorption layer on the surface and the density in the middle of the pores approaches bulk. Based on such adsorption mechanism, in mesopores, one can define the adsorbed and free gas phases, respectively, as shown in Fig. 6. In addition, V_p obtained from the helium adsorption is depicted as the region between Point A and A', $z_{AA'}$, which is defined as $z_{AA'} = V_p/S_A$. The adsorbed phase is defined as the region between the Point A and B, which is the saddle point between the first and



Figure 5. Methane density distributions in illite nanopores of varying pore sizes at 333.15 K and (a) P = 50 MPa; (b) P = 5 MPa. The solid lines represent the density profiles from GCMC simulations and dashed lines are guidelines for ρ_h from NIST Chemistry Webbook.



Figure 6. The schematic representation of adsorbed and free gas phases for methane adsorption in illite nanopore of W = 4 nm at 333.15 K and 50 MPa. The heights of adsorbed and free gas phases are depicted from $\rho_a = \int_A^B \rho(z) dz / z_{AB}$ and ρ_b from NIST Chemistry Webbook, respectively.

second adsorption layers. The width of AB in the z direction, z_{AB} , is around 0.38 nm, similar to the LJ diameter of methane molecules, while methane adsorption is considered to be single-layered. Previous theoretical calculations on LJ fluid adsorption in nanopores reveals that the width of adsorption layer is equal to the diameter of absorbate molecules⁷⁷. In addition, Didar and Akkutlu⁶⁸ pointed out that the width of adsorbed phase is equal to the molecular diameter. As shown in Fig. 4, at high pressure conditions, the Point B does not change with pressure. We use the Point B at P = 50 MPa as the boundary for adsorbed phase and assume a constant adsorbed phase volume $V_a = 2S_A z_{AB}$ for various pressure conditions. The adsorbed phase covers the strong surface adsorption layer, where the adsorbed phase density $\rho_a = \int_A^B \rho(z) dz/z_{AB}$ can be higher than ρ_b . The amount in the adsorbed phase is the absolute adsorption, $m_{abs} = \rho_a V_a$. In other words, once the adsorbed phase is defined, the absolute adsorption can be readily obtained from density distributions. The free gas phase is between Point B and B', which covers the weak second adsorption layer. We depict the average density of free gas phase



Figure 7. Comparison of $\rho_f^a = \int_B^{B'} \rho(z) dz / z_{BB'}$ from density distributions and ρ_b from NIST Chemistry Webbook and the variance $\delta = \left(\rho_f^a - \rho_b\right) / \rho_b$ in illite nanopore of W = 4 nm at 333.15 K.



Figure 8. The schematic representation of transition zone in illite nanopores of W = 4 nm at 333.15 K and 10 MPa. The heights of adsorbed and free gas phases are depicted from $\rho_a = \int_A^B \rho(z) dz / z_{AB}$ and ρ_b from NIST Chemistry Webbook, respectively.

 $\rho_f^a = \int_B^{B'} \rho(z) dz/z_{BB'}(z_{BB'}(z_{BB'})$ is the distance between Point B and B') and ρ_b from NIST Chemistry Webbook in Fig. 7. The difference between ρ_f^a and ρ_b becomes negligible at high pressures. At 50 MPa, the variance $\delta = (\rho_f^a - \rho_b)/\rho_b$ is around 0.5%. At higher pressures up to 80 MPa, δ is less than 0.5% (shown in Supplementary Information). It shows that our classification agrees excellently with the adsorption model shown in Fig. 1. At low pressure conditions, there is some discrepancy between ρ_f^a and ρ_b , e.g. δ around 9% at 10 MPa. It is due to the presence of transition zone⁶⁸ as shown in Fig. 8. To have a better fit with the adsorption model shown in Fig. 1, the transition zone is less influenced by the wall and may not behave as adsorbed phase. Due to transition zone, ρ_f^a is higher than ρ_b .

In Fig. 9, we present the adsorbed phase density ρ_a in various clay nanopores of 4 nm at 333.15 K. ρ_a continuously increases with pressure up to 50 MPa. With a constant adsorbed phase volume V_a , absolute adsorption continuously increases with pressure (pressure up to 80 MPa shown in Supplementary Information). For different clay minerals, ρ_a is similar. We also present the effect of temperature on ρ_a in Fig. 10. As temperature increases, ρ_a decreases due to weaker fluid-surface interactions. With the definition of adsorbed and free gas phases, in Fig. 11, we present the absolute adsorption based on m_{ex} and ρ_a , and m_{ex} and V_a as in Eqs (2) and (3), respectively. We also depict the absolute adsorption given as $m_{abs} = \rho_a V_a$, which is from the density distributions. It shows that the absolute adsorption from m_{ex} and ρ_a is overestimated ($m_{abs,2} = m_{ex}/(1 - \rho_b/\rho_a)$ in Fig. 11), especially at low pressure conditions. It is due to the presence of transition zone. The transition zone contributes to m_{ex} , while ρ_a only takes into account the excess adsorption amount in adsorbed phase. On the other hand, the absolute adsorption from $m_{ex} + \rho_b V_a$ in Fig. 11) shows an excellent agreement with m_{abs} . The agreement is better



Figure 9. ρ_a in illite, montmorillonite and kaolinite nanopores of W = 4 nm at 333.15 K.



Figure 10. ρ_a in illite nanopores of W = 4 nm at different temperatures.



Figure 11. Absolute adsorption in illite nanopores of W = 4 nm at 333.15 K. $m_{abs} = \rho_a V_a$ is based on density distribution; $m_{abs,1} = m_{ex} + \rho_b V_a$ is obtained from m_{ex} and V_a ; $m_{abs,2} = m_{ex}/(1 - \rho_b/\rho_a)$ is obtained from m_{ex} and ρ_a .

at higher pressures as shown in Supplementary Information. At low pressure conditions, due the presence of transition zone, there is a small discrepancy and the adsorption model shown in Fig. 1 is less suitable to describe the adsorption behavior. Using both varying ρ_a and V_a would make the adsorption model a pure fitting without rigorous physical foundation. As pressure increases, the deficiency becomes negligible.



Figure 12. The variances of absolute adsorption $m_{abs,1}$ obtained by m_{ex} and V_a ; $m_{abs,2}$ obtained by m_{ex} and ρ_a from $m_{abs} = \rho_a V_a$. We also present $\rho_a / (\rho_a - \rho_b)$ versus pressure.



Figure 13. Excess adsorption versus ρ_b in illite nanopore of W = 4 nm at 333.15 K. The solid line is linear fitting at high pressures to calculate V_a from the slope method by Zhang *et al.*³³.

Based on Eq. (3), difference between m_{abs} and $m_{abs,1}$, Δm_1 is

$$\Delta m_1 = m_{abs,1} - m_{abs} = m_{ex} - (\rho_a - \rho_b) V_a, \tag{10}$$

while according to Eq. (2), difference between m_{abs} and $m_{abs,2}$, Δm_2 is

$$\Delta m_2 = m_{abs,2} - m_{abs} = \frac{\rho_a}{\rho_a - \rho_b} \cdot [m_{ex} - (\rho_a - \rho_b)V_a].$$
(11)

As shown in Fig. 12, $\Delta m_1/m_{abs}$ continuously decreases with pressure beyond 15 MPa. Especially at high pressure, $\Delta m_1/m_{abs}$ is only around 3%. On the other hand, $\Delta m_2/m_{abs}$ is around 30% at high pressure. The difference between Eqs (10) and (11) is the coefficient $\rho_a/(\rho_a - \rho_b)$, which continuously increases with pressure. As a result, $m_{abs,1}$ obtained from m_{ex} and V_a is more accurate than $m_{abs,2}$ from m_{ex} and ρ_a for a given adsorbed phase.

Zhang *et al.*³³ used the slope of the excess adsorption with respect to the corresponding bulk density as adsorbed phase volume at high pressure conditions. Based on the slope method as depicted in Fig. 13, the calculated V_a is around 6.315 nm³ for 4 nm-illite at 333.15 K. This results in a width of adsorbed phase of 0.213 nm, much less than methane diameter. The slope method is considered to be valid only when the absolute adsorption becomes constant as depicted in Fig. 14. However, as shown in Fig. 11 and Figure SI.B1, absolute adsorption keeps increasing with pressure. As a result, the slope method may not be applicable.

Assessment of Supercritical Dubinin-Radushkevich (SDR) Fitting Method. Recently, a number of fitting methods, such as Langmuir⁷⁸, SDR⁷⁹, and Ono-Kondo (OK)⁸⁰ models have been used to fit the excess adsorption and then correct to the absolute adsorption. Among them, SDR model based on pore-filling theory, is a popular choice to obtain the sorption capacity of porous materials³¹. The SDR model for absolute adsorption is given as⁷⁹,



bulk phase density

Figure 14. Ideal excess and absolute adsorption isotherms suitable for determining V_a from the slope method by Zhang et al.³³.

$$m_{abs}^{SDR} = m_{abs, \max}^{SDR} \exp\left\{-C\left[\ln\left(\rho_{a, \max}^{SDR}/\rho_{b}\right)RT\right]^{2}\right\},$$
(12)

in which m_{abs}^{SDR} is the absolute adsorption of a given pressure, $m_{abs, \max}^{SDR}$ is the maximum absolute adsorption, $\rho_{a, \max}^{SDR}$ is the maximum adsorbed phase density, *C* is the interaction constant, *R* is the gas constant, and *T* is the absolute temperature. Note that m_{abs}^{SDR} , $m_{abs, \max}^{SDR}$, and $\rho_{a, \max}^{SDR}$ are all from the SDR fitting, not from density distributions. To fit with m_{ex} , two options have been used for the modification⁴, either (*) from the constant adsorbed phase volume $V_a^{SDR,*}$,

$$m_{ex} = m_{abs}^{\text{SDR}} - \rho_b V_a^{\text{SDR},*},\tag{13}$$

or (**) from the constant adsorbed phase density $\rho_a^{SDR,**}$,

$$m_{ex} = m_{abs}^{SDR} \left(1 - \frac{\rho_b}{\rho_a^{SDR,^{**}}} \right). \tag{14}$$

In the following subsections, we will assess SDR models assuming constant adsorbed phase volume or density by fitting with our m_{ex} at a given temperature.

SDR Fitting with Constant Adsorbed Phase Volume. With constant $V_a^{SDR,*}$, $m_{abs, \max}^{SDR,*}$ can be expressed in terms of $m_{abs, \max}^{SDR} = \rho_{a, \max}^{SDR,*} V_a^{SDR,*}$ and Eq. (12) can be rewritten as

$$m_{abs}^{SDR,*} = \rho_{a,\max}^{SDR,*} V_a^{SDR,*} \exp\left\{-C^* \left[\ln\left(\rho_{a,\max}^{SDR,*}/\rho_b\right) RT\right]^2\right\},\tag{15}$$

and Eq. (13) can be expressed as

$$m_{ex} = \rho_{a, \max}^{SDR,*} V_a^{SDR,*} \exp\left\{-C^* \left[\ln\left(\rho_{a, \max}^{SDR,*} / \rho_b\right) RT\right]^2\right\} - \rho_b V_a^{SDR,*}.$$
(16)

Comparing to the adsorption model shown in Fig. 1, the adsorbed phase density $\rho_s^{SDR,*}$ for a given pressure can be calculated from

$$\rho_a^{SDR,*} = \rho_{a,\max}^{SDR,*} \exp\left\{-C^* \left[\ln\left(\rho_{a,\max}^{SDR,*}/\rho_b\right)RT\right]^2\right\}.$$
(17)

Our excess adsorption is fitted by the least-square method with all free parameters in Eq. (16) varying over the following ranges: $0 < \rho_{a,\max}^{SDR,*} < 500 \text{ kg/m}^3, V_a^{SDR,*} > 0 \text{ nm}^3, \text{ and } 0 < C^* < 0.05 \text{ mol}^2/\text{kJ}^{231}$. Excess adsorption in illite nanopores of W = 4 nm at 333.15 K is used for SDR fitting with constant $V_a^{SDR,*}$ as in Fig. 15. The fitting parameters are $\rho_{a,\max}^{SDR,*} = 280.65067 \text{ kg/m}^3, V_a^{SDR,*} = 8.56702 \text{ nm}^3, \text{ and } C^* = 0.02564 \text{ mol}^2/\text{kJ}^2$, respectively. The corresponding adsorbed phase width is around 0.29 nm, which is less than the methane molecule diameter. With the fitted $V_a^{SDR,*}$, based on adsorption model shown in Fig. 1, adsorbed phase density ρ_a^* can be obtained from the integration of density distributions within adsorbed phase as $\rho_a^* = \int_A^{B^*} \rho(z) dz/z_{AB^{**}}$ in which B^* is the boundary of adsorbed phase with volume $V_a^{SDR,*}$ and $z_{AB^*} = V_a^{SDR,*}/2S_A$ represents the width of the corresponding



Figure 15. SDR Fitting with constant $V_a^{SDR,*}$ to the excess adsorption in illite nanopores of W = 4 nm at 333.15 K with $R^2 = 0.9975$.



Figure 16. ρ_a^* from the integration of density distributions within adsorbed phase with volume $V_a^{SDR,*}$ and $\rho_a^{SDR,*}$ from SDR fitting as shown in Eq. (17).

adsorbed phase. In Fig. 16, we depict ρ_a^* and $\rho_a^{SDR,*}$ from Eq. (17). For all pressure conditions, $\rho_a^{SDR,*}$ is higher than ρ_a^* . The difference first increases with pressure, then decreases. We depict the absolute adsorption $m_{abs}^* = \rho_a^* V_a^{SDR,*}$ from density distributions and $m_{abs}^{SDR,*} = m_{ex} + \rho_b V_a^{SDR,*}$ from the SDR model with constant $V_a^{SDR,*}$ in Fig. 17. Similar to Fig. 11, there is some discrepancy when pressure is relatively low and agreement becomes better at high pressure conditions. The difference is due to the presence of transition zone. In Fig. 17, We also present the variance of absolute adsorption, $\Delta m_{abs}^*/m_{abs}^* = (m_{abs}^{SDR,*} - m_{abs}^*)/m_{abs}^*$. Comparing to our model as shown in Fig. 12, the variance is larger. It is because in SDR model with constant V_a^{US} , more transition zone is included in the free gas phase.

SDR Fitting with Constant Adsorbed Phase Density. With constant $\rho_a^{\text{SDR},^{**}}$, $\rho_{a, \max}^{\text{SDR},^{**}}$ is the same as $\rho_a^{\text{SDR},^{**}}$. $m_{abs, \max}^{\text{SDR},^{**}}$ can be expressed in terms of $m_{abs,\max}^{\text{SDR},^{**}} = \rho_a^{\text{SDR},^{**}} V_{a,\max}^{\text{SDR},^{**}}$ and Eq. (12) can be rewritten as

$$m_{abs}^{SDR,^{**}} = \rho_a^{SDR,^{**}} V_{a,\max}^{SDR,^{**}} \exp\left\{-C^{**} \left[\ln(\rho_a^{SDR,^{**}}/\rho_b)RT\right]^2\right\},\tag{18}$$

and Eq. (14) can be expressed as

1

$$m_{ex} = \rho_a^{SDR,^{**}} V_{a,\max}^{SDR,^{**}} \exp\left\{-C^{**} \left[\ln(\rho_a^{SDR,^{**}}/\rho_b)RT\right]^2\right\} \left[1 - \frac{\rho_b}{\rho_a^{SDR,^{**}}}\right]$$
(19)

Comparing to the adsorption model shown in Fig. 1, the adsorbed phase volume $V_a^{SDR,**}$ for a given pressure can be calculated from



Figure 17. Absolute adsorption $m_{abs}^* = \rho_a^* V_a^{SDR,*}$ and $m_{abs}^{SDR,*} = m_{ex} + \rho_b V_a^{SDR,*}$ from SDR model with constant $V_a^{SDR,*}$ as well as the variance of absolute adsorption $\Delta m_{abs}^*/m_{abs}^*$.



Figure 18. SDR Fitting with constant $\rho_a^{SDR,**}$ to the excess adsorption in illite nanopores of W = 4 nm at 333.15 K with $R^2 = 0.9962$.

$$V_a^{SDR,**} = V_{a,\max}^{SDR,**} \exp\left\{-C^{**} \left[\ln(\rho_a^{SDR,**}/\rho_b)RT\right]^2\right\}.$$
(20)

Our excess adsorption is fitted by the least-square method with all free parameters in Eq. (19) varying over the following ranges: $0 < \rho_a^{SDR,*} < 500 \text{ kg/m}^3, V_{a, \max}^{SDR,*} > 0 \text{ nm}^3$, and $0 < C^{**} < 0.05 \text{ mol}^2/\text{kJ}^{231}$. Excess adsorption in illite nanopores of W = 4 nm at 333.15 K is used for SDR fitting with constant $\rho_a^{SDR,**}$ as shown in Fig. 18. The fitting parameters are $\rho_a^{SDR,**} = 286.09381 \text{ kg/m}^3, V_{a,\max}^{SDR,*} = 7.5316 \text{ nm}^3$, and $C^{**} = 0.02848 \text{ mol}^2/\text{kJ}^2$, respectively. The fitted $\rho_a^{SDR,**}$ is less than the density of liquid methane at boiling temperature, 420 kg/m^3 or the methane density at the critical point, 373 kg/m^3 , which were used by Xiong *et al.*³¹.

density at the critical point, 373 kg/m³, which were used by Xiong *et al.*³¹. With constant $\rho_a^{SDR,**}$, the adsorbed phase volume $V_a^{SDR,**}$ for a given pressure from SDR fitting varies with pressure (bulk density) as in Eq. (20). We present $V_a^{SDR,**}$ for a given pressure in Fig. 19. It is observed that $V_a^{SDR,**}$ monotonically increases with pressure. At P = 5 MPa, the calculated $V_a^{SDR,**}$ is 2.5224 nm³ and the width of adsorbed phase is only around 0.0852 nm. Such small adsorbed phase is unphysical, since at P = 5 MPa, the width of adsorbed phase is much larger than that as shown in Fig. 8.

width of adsorbed phase is much larger than that as shown in Fig. 8. With the fitted $V_a^{SDR,**}$, we present the $\rho_a^{**} = \int_A^{B^{**}} \rho(z) dz/z_{AB^{**}}$ with varying adsorbed phase boundary $B^{**}(B^{**})$ is obtained from $V_a^{SDR,**}$ in Eq. (20)) and adsorbed phase width $z_{AB^{**}} = V_a^{SDR,**}/2S_A$ and $\rho_a^{SDR,**}$ in Fig. 20. Due to small $V_a^{SDR,**}$ at low pressure conditions, ρ_a^{**} is much smaller than the fitted value $\rho_a^{SDR,**}$. In addition, ρ_a^{**} varies with pressure, while $\rho_a^{SDR,**}$ is constant. In other words, $\rho_a^{SDR,**}$ from SDR model with constant adsorbed phase density is not self-consistent with methane density distributions.

density is not self-consistent with methane density distributions. We also present the absolute adsorption $m_{abs}^{**} = \rho_a^{**}V_a^{SDR,**}$ from the density distributions and $m_{abs}^{SDR,**} = m_{ex}/(1 - \rho_b/\rho_a^{SDR,**})$ from the SDR model with constant $\rho_a^{SDR,**}$ as well as the variance of the absolute adsorption $\Delta m_{abs}^{*BR,**} = (m_{abs}^{SDR,**} - m_{abs}^{**})/m_{abs}^{**}$ in Fig. 21. At low pressures, m_{abs}^{**} is much less than $m_{abs}^{SDR,**}$ due to small adsorbed phase volume $V_a^{SDR,**}$ and ρ_a^{**} as shown in Figs 19 and 20. The variance can be as large as 100% at 5 MPa. As pressure increases, the agreement becomes better.







Figure 20. $\rho_a^{**} = \int_A^{B^{**}} \rho(z) dz / z_{AB^{**}}$ and $\rho_a^{SDR,^{**}}$ from SDR model with constant adsorbed phase density.



Figure 21. Absolute adsorption $m_{abs}^{**} = \rho_a^{**}V_a^{SDR,**}$ from the density distributions and $m_{abs}^{SDR,**} = m_{ex}/(1 - \rho_b/\rho_a^{SDR,**})$ from the SDR model with constant $\rho_a^{SDR,**}$ as well as the variance of the absolute adsorption $\Delta m_{abs}^{**}/m_{abs}^{**}$.

Comparison between SDR Fittings with Our Model. In our model, we define the adsorbed phase width z_{AB} as 0.38 nm which is close to the diameter of methane molecules. In SDR model with constant $V_a^{SDR,*}$, $V_a^{SDR,*}$ is obtained from the fitting with m_{ex} and the resulting adsorbed phase width is $z_{AB^*} = 0.29$ nm. In SDR model with



Figure 22. The schematic representation of adsorbed and free gas regions for methane adsorption in illite nanopore of W = 4 nm at 333.15 K and 10 MPa. The yellow area presents the adsorbed phase from our model, the orange area presents the adsorbed phase from SDR model with constant $V_a^{SDR,*}$, the blue area depicts the adsorbed phase from SDR model with constant $\rho_a^{SDR,**}$ with corresponding $V_a^{SDR,**}$ at 10 MPa, and the green area depicts the free gas phase from our model. B, B*, and B** represent the boundaries of adsorbed phase from our model, SDR model with constant $V_a^{SDR,**}$, and SDR model with constant $\rho_a^{SDR,**}$, respectively. The heights of adsorbed and free gas phases from our model are depicted from $\rho_a = \int_A^B \rho(z) dz/z_{AB}$ and ρ_b from NIST Chemistry Webbook, respectively. The heights of adsorbed phase from SDR model with constant $V_a^{SDR,**}$ are depicted from $\rho_a^* = \int_A^{B^*} \rho(z) dz/z_{AB^*}$ and $\rho_a^{SDR,**}$ are depicted from $\rho_a^* = \int_A^{B^*} \rho(z) dz/z_{AB^*}$, respectively. The adsorbed phase density $\rho_a^{SDR,**}$ from SDR model with constant $V_a^{SDR,**}$ is from Eq. (17), and $\rho_a^{SDR,**} = 286.09381 \text{ kg/m}^3$ is from the SDR fitting with constant $\rho_a^{SDR,**}$.

constant $\rho_a^{SDR,**}$, $V_a^{SDR,**}$ varies with pressure, while the maximum adsorbed phase volume $V_{a, \max}^{SDR,**}$ is obtained from the fitting with m_{ex} and the resulting maximum adsorbed phase width is $z_{AB^{**}} = 0.25$ nm. The adsorbed phase volumes from both fitting approaches are less than methane diameter, which is considered as the width of adsorbed phase⁶⁸.

In Fig. 22, we present the schematic representation of adsorbed phases from different approaches and the corresponding adsorbed phase density and volume of methane density distributions in illite nanopore of W = 4 nm at 333.15 K and P = 10 MPa. It can be seen that due to small $V_a^{SDR,**}$ from SDR model with constant $\rho_a^{SDR,**}$, the adsorbed phase from fitting can only cover part of adsorbion layer. The actual adsorbed phase density between $m_{abs}^{**} = \int_A^{B^{**}} \rho(z) dz/z_{AB^{**}}$ and $m_{abs}^{SDR,**} = m_{ex}/(1 - \rho_b/\rho_a^{SDR,**})$ as shown in Fig. 21. Comparing to $V_a^{SDR,**}$ from SDR model with constant $\rho_a^{SDR,**}$ from SDR model with constant $\rho_a^{SDR,**}$ from constant adsorbed phase volume fitting is larger and the calculated ρ_a^* is closer to $\rho_a^{SDR,**}$. However, due to smaller adsorbed phase volume, comparing to our model, it has larger transition zone in free gas phase, which negatively affects the absolute adsorption calculation.

Similar to Fig. 7, in Fig. 23, we present the variance of the average density in the free gas phase from ρ_b in our model and SDR fittings. It is observed that due to small $V_a^{SDR,**}$, SDR model with constant $\rho_a^{SDR,**}$ has the largest variance. The variance can be as much as 50% at low pressure conditions. Comparing to SDR model with constant $V_a^{SDR,*}$, the variance in our model is smaller. In other words, the effect of transition zone is less significant in our model.

Overall, SDR model with constant adsorbed phase volume performs better than that with constant adsorbed phase density. The latter may not have solid physical foundation. The adsorbed phase densities from SDR method are inconsistent with density distributions. Comparing to both fitting methods, our model agrees better with the adsorption model shown in Fig. 1.

Comparison of Excess and Absolute Adsorption in Various Clay Minerals. In this subsection, we will compare the excess and absolute adsorption in various clay nanopores.

In Fig. 24, we present the methane excess adsorption in illite, montmorillonite, and kaolinite nanopores at 333.15 K. As shown in Fig. 3, when $W \ge 2$ nm, m_{ex} per SSA becomes insensitive to the pore size. Thus, we only present the results for 4 nm pore. We observe that m_{ex} per SSA is similar for different clay minerals at given pressures. By using volumetric method, Ji *et al.*¹⁴ found that methane excess adsorption for various types of clay minerals correlates well with SSA. For various clay minerals, excess adsorption capacity has a maximum around 15 MPa. As pressure further increases, excess adsorption decreases due to higher ρ_b . In Fig. 25, we present the corresponding absolute adsorption in various clay minerals. The adsorbed phase region in montmorillonite and kaolinite nanopores is defined in the same way as illite at 50 MPa as shown in Fig. 6. The absolute adsorption is obtained from m_{ex} and V_a as in Eq. (3). Similar to the excess adsorption, the absolute adsorptions per SSA in various clay nanopores are similar.



Figure 23. $\delta = \left(\int_{B}^{B'} \rho(z)dz/z_{BB'} - \rho_{b}\right)/\rho_{b}$ from our model, $\delta^{*} = \left(\int_{B^{*}}^{B'^{*}} \rho(z)dz/z_{B^{*}B^{*}} - \rho_{b}\right)/\rho_{b}$ from SDR fitting with constant $V_{a}^{SDR,*}$, and $\delta^{**} = \left(\int_{B^{**}}^{B'^{**}} \rho(z)dz/z_{B^{**}B^{**}} - \rho_{b}\right)/\rho_{b}$ from SDR fitting with constant $\rho_{a}^{SDR,**}$, respectively, in illite nanopore of W = 4 nm at 333.15 K. $z_{BB'}, z_{B'}, z_{B'}, z_{B'}, z_{B'}$ and $z_{B'^{**}B'^{**}}$ represent the width of free gas phase from our model, SDR model with constant $V_{a}^{SDR,*}$, and SDR model with constant $\rho_{a}^{SDR,**}$, respectively.



Figure 24. Excess adsorption in illite, montmorillonite and kaolinite nanopores of W = 4 nm at 333.15 K.



Figure 25. Absolute adsorption in illite, montmorillonite and kaolinite nanopores of W = 4 nm at 333.15 K.



Figure 26. Excess adsorption in illite nanopores of W = 4 nm at various temperatures.



Figure 27. The same as Fig. 26, but for absolute adsorption.

Effect of Temperatures on Excess and Absolute Adsorption. In Fig. 26, we present the excess adsorption isotherms at various temperatures in illite nanopores of 4 nm pores. We observe that in the low pressure region, as temperature increases, the excess adsorption amount decreases. But in high pressure region, the excess adsorption becomes comparable for various temperatures. However, the absolute adsorption decreases with temperature as shown in Fig. 27. It is because at higher temperature, due to weaker fluid-surface interactions, surface adsorption becomes less significant.

Conclusion

In this work, we use GCMC simulations to study methane adsorption in illite, montmorillonite and kaolinite nanopores of varying pore sizes and temperatures. Clay minerals can have micro-to-mesopore structures which can greatly enhance gas-in-place in shale reservoirs. We employ full atomistic models to describe clay nanopore structures and explicitly consider the intermolecular interactions. To match with experimental data¹⁴, we use helium adsorption to determine the effective pore volume and the corresponding excess adsorption.

Our simulation shows a good agreement with experimental data on the methane excess adsorption. When $W \ge 2$ nm, the excess adsorption per SSA becomes insensitive to the pore size. We find that SSA plays a dominant role in the methane excess and absolute adsorption capacity in various clay minerals in line with previous experimental data¹⁴. The methane density distributions in various clay minerals indicate that adsorbed phase density is not only dependent on temperature but also pressure. As a result, using a constant liquid phase density to calculate absolute adsorption and adsorbed phase density may bring a significant error. Instead, we propose to use the excess adsorption and adsorbed phase volume. While experiments may only get total or excess adsorption, molecular simulation can provide V_a of the specific adsorbate based on the density distribution.

We also show that the SDR method used in experiment to correct the excess adsorption to the absolute adsorption may result in unphysical values for adsorbed phase density or volume, especially for SDR model with constant adsorbed phase density assumption.

This work should shed important insights into the evaluation of absolute adsorption and provide fundamental understandings toward the underlying mechanisms of methane adsorption in clay nanopores.

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Acknowledgements

This research was enabled in part by support provided by Westgrid (www.westgrid.ca) and Compute Canada (www.computecanada.ca). All authors acknowledge an Open Fund (PLC201704) provided by State Key Laboratory for Oil and Gas Reservoir Geology and Exploitation (Chengdu University of Technology). All authors acknowledge the National Science and Technology Major Project (No. 2017ZX05036003-007) and the National Science and Technology Major Project (No. 2017ZX05036004-006). The third author acknowledges a Discovery Grant by Natural Sciences and Engineering Research Council of Canada (NSERC RGPIN-2017-05080).

Author Contributions

Yuanyuan Tian performed the GCMC simulations and drafted the main manuscript. Changhui Yan supervised the project. Zhehui Jin provided the main idea, defined the statement of problem and helped draft the manuscript. All authors reviewed the manuscript.

Additional Information

Supplementary information accompanies this paper at https://doi.org/10.1038/s41598-017-12123-x.

Competing Interests: The authors declare that they have no competing interests.

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