# Polysaccharide-based wrinkled surfaces induced by polyion complex skin layers upon drying 

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#### Abstract

We introduce a novel surface wrinkling system based on the fabrication of a polyion complex (PIC) skin layer. PIC skin layers are fabricated on the surface of chitosan (CS) films via immersion in sodium alginate (AG) solutions. After removal of excess adsorbed AG, microscopic wrinkles appear upon drying. The wrinkle size can be controlled by the immersion temperature during the formation of the skin layer because of the differences in skin layer thicknesses. In addition, we demonstrate that there is only a two-fold difference in the elastic moduli between the PIC skin layer and the CS film. Furthermore, wrinkled films can be made water resistant by thermal crosslinking of the PIC skin layer. The thermally crosslinked wrinkled surfaces composed of covalently crosslinked CS and AG demonstrate the low cell adhesiveness that is required for implantable materials. Thus, this method that combines surface wrinkling induced by a PIC skin layer and subsequent thermal crosslinking can provide a variety of wrinkled biomaterials.


## Introduction

Surface wrinkling has attracted increased interest because it is generally easy to produce, requires low cost, and is environmentally benign, while the number of potential applications are large in fields such as optical [1] and electronic devices [2], surface-enhanced Raman spectroscopy substrates [3], promotion of tunable wettability [4] and adhesion [5], and the synthesis of cell culture scaffolds [6]. The wrinkles result from inhomogeneous changes that are induced by a skin layer that is a thin hard surface layer adhering to a substrate. In most wrinkling systems, the skin layers are fabricated on an elastic film by chemical vapor deposition [7], photocrosslinking [8], $\mathrm{UV} / \mathrm{O}_{3}$ [9] or plasma

[^0][10] oxidation, or chemical reactions [11]. The characteristics of the skin layer strongly affect wrinkle morphology. In common wrinkles, the wavelength ( $\lambda$ ) of the wrinkle is dependent on the skin thickness (d), and the mechanical properties of the film can be described by the following equation [12, 13]:
$\lambda=2 \pi d\left(\frac{\bar{E}_{\mathrm{s}}}{3 \bar{E}_{\mathrm{f}}}\right)^{\frac{1}{3}}$,
where $\bar{E}$ is the plane-strain modulus given by $E /\left(1-\nu^{2}\right)$, the subscripts s and f refer to the skin layer and the foundation (substrate), respectively, $E$ is the elastic modulus, and $\nu$ is the Poisson ratio. In addition, the amplitude (A) of the wrinkle is described by the following equation [12]:
\[

$$
\begin{equation*}
A=d\left(\frac{\varepsilon}{\varepsilon_{\mathrm{c}}}-1\right)^{\frac{1}{2}}, \tag{2}
\end{equation*}
$$

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where $\varepsilon$ and $\varepsilon_{\mathrm{c}}$ are the applied and critical strain, respectively. $\varepsilon_{\mathrm{c}}$ is dependent on $\bar{E}$ as follows [12]:
$\varepsilon_{\mathrm{c}}=\frac{1}{4}\left(\frac{3 \bar{E}_{\mathrm{f}}}{\bar{E}_{\mathrm{s}}}\right)^{\frac{2}{3}}$.

Thus, the design of the skin layer, which includes material selection and control of the thickness and elastic modulus, is a key factor in the development of wrinkled surfaces with desirable morphologies.

A trigger is required to initiate the inhomogeneous changes that are necessary for the generation of wrinkles [14]. The application of external stress [9] or heat [7] are common approaches to induce/control surface wrinkling [13]. Alternatively, although surface wrinkling due to water evaporation is ubiquitous in nature [14-16], it has barely been considered during film consolidation to trigger surface wrinkling [17, 18]. We considered the possibility that this was because there is no common design concept for surface wrinkling upon drying, and recently reported an original approach to surface wrinkling using drying [19]. In this method, microscopic wrinkles were fabricated on a chitosan (CS) film by immersion in a phenolic acid solution, followed by a horseradish peroxidase-catalyzed surface reaction and drying [20]. We demonstrated that the crosslinking structure capable of skin-layer formation consisted of ionic bonds between the CS and oligomeric phenolic acid residues on the CS [21]. This evidence provided inspiration for further research. We hypothesized that a skin layer composed of ionic crosslinks would promote surface wrinkling upon drying; i.e., a polyion complex (PIC) skin layer prepared by simple immersion of a CS film into an anionic polymer solution would induce surface wrinkling. Establishing this simple concept based on the PIC skin layer would accelerate the development of wrinkled surfaces induced by drying. In addition, we expected that this method would be favorable for producing biopolymer-based wrinkled surfaces that were applicable for biomaterials, because with such an approach, it is not necessary to apply external stresses to induce surface wrinkling.

Alginate (AG) is an anionic polysaccharide that is extracted from brown algae [22]. AG salts dissolve in water to produce viscous liquids or hydrogels. AG is therefore widely used in a broad range of industries, as a rheological control agent in aqueous systems, and as a stabilizer for emulsions and suspensions [23]. Furthermore, AG has attracted much attention in the development of biomaterials by virtue of its biocompatibility and sustainability [24]. The carboxy groups on AG can interact electrostatically with the amino groups on CS to form gels, particles, and film materials for biomedical applications
[24-31]. Thus, wrinkled surfaces prepared from AG and CS may provide unique biointerfaces with microscopic wrinkles.

Here, we introduce a new concept for developing surface wrinkling upon drying (Fig. 1). We fabricated a PIC skin layer on a CS film by immersion in an AG solution. After washing away excess adsorbed AG by soaking the film in water, the film with the PIC skin layer was dried. We then investigated the effect of the AG molecular weight, the AG concentration, and the immersion temperature in the skin layer formation process on the surface morphology. Our analysis revealed differences in the elastic moduli of the PIC skin layer and the CS substrate. Furthermore, we developed a methodology for preparing a water-resistant wrinkled surface via a thermal dehydration condensation reaction of the PIC skin layer. Finally, 3 T 3 cells were cultured on the thermally crosslinked wrinkled films.

## Experimental procedures

## Materials

CS was supplied by the Koyo Chemical Co., Ltd (Tottori, Japan). The $M_{n}$ estimated by gel permeation chromatography analysis using Pullulan standards was $64.1 \times 10^{3}$ ( $M_{W} / M_{n}=2.03$ ), and the degree of deacetylation estimated by elemental analysis was $76.5 \%$. Sodium alginates (80-120, 300-400, and $500-600 \mathrm{cps}, 10 \mathrm{~g} / \mathrm{L}$ at $25^{\circ} \mathrm{C}$ ) and $0.25 \mathrm{w} / \mathrm{v} \%$ Trypsin $1 \mathrm{mmol} / \mathrm{L}$ EDTA•4Na solution with phenol red were purchased from Wako Pure Chemical Industries (Osaka, Japan). Phosphate-buffered saline (PBS) ( $\mathrm{pH} 7.4,10 \times$; Gibco ${ }^{\mathrm{TM}}$ ), penicillin-streptomycin-neomycin (PSN) antibiotic mixture (Gibco ${ }^{\text {TM }}$ ), Dulbecco's modified Eagle's medium, high glucose (DMEM; Gibco ${ }^{\text {TM }}$ ), and Trypan blue (Gibco ${ }^{\text {TM }}$ ) were purchased from Thermo Fisher Scientific (Waltham, MA, USA). Other reagents were obtained as commercial grades and used without further purification.

## Measurements

SEM images were obtained using a TM303Plus instrument (Hitachi High-Technologies Corporation, Japan) without coatings. The wrinkle amplitudes of the films were measured using a NanoCute-NanoNavi IIs system (Seiko Instruments, Japan). Optical microscopic images were obtained to evaluate the stability of the wrinkles in water and for cell culture experiments using BIX53 (Olympus, Japan) and CKX53 instruments (Olympus, Japan), respectively. Infrared (IR) spectra of the samples were obtained with a Spectrum 65 spectrometer (Perkin-Elmer Japan Co.,


Fig. 1 Illustration of a surface wrinkling system based on the fabrication of a PIC skin layer followed by thermal crosslinking of the layer

Ltd, Japan) equipped with an ATR attachment. Film weights were obtained using an XS3DUV balance (METTLER TOLEDO, USA). Elastic moduli were measured with a universal testing instrument (AG-10KNX; Shimadzu, Japan).

## Preparation of CS films

CS ( 2.0 g ) was dissolved in 100 mL of an acidic aqueous solution containing 0.5 mL of acetic acid. Then, 10 mL of the CS solution was added to a Teflon Petri dish ( $\varphi=$ 50 mm ) and degassed under reduced pressure. The CS solution was heated at $50^{\circ} \mathrm{C}$ for 24 h to yield a CS film after evaporation. The film was then heated at $50^{\circ} \mathrm{C}$ under reduced pressure for 12 h . The inhomogeneous edge of the film was cut with scissors, and the film was soaked in methanol for 2 days. It was then dried under reduced pressure; the weight and thickness of the CS film were ca. 0.15 g and ca. $110 \mu \mathrm{~m}$, respectively. The air side of the CS film that formed during evaporation was used for this study.

## Preparation of wrinkled films via soaking processes

In a typical experiment, a CS film was immersed in water for 5 min at room temperature. The swollen film was removed and soaked in 20 mL of an AG aqueous solution $(1.0 \mathrm{wt} \%, \mathrm{pH}$ adjusted to 6.5 with 1.0 M HCl$)$ at $30^{\circ} \mathrm{C}$ for 24 h , after which the film was removed and gently washed with water. The film was soaked in 100 mL of water at $30^{\circ} \mathrm{C}$ for 6 h , after which it was removed and dried at $40^{\circ} \mathrm{C}$ for 12 h .

## Preparation of PIC films

CS ( 2.0 g ) was dissolved in 100 mL of an acidic aqueous solution containing 1.0 mL of acetic acid and, separately, AG ( 2.0 g ) was dissolved in 100 mL of water. The AG solution was added dropwise to the CS solution, and the PIC solution was vigorously stirred for 2 h at room temperature. Then, 10 mL of the PIC solution was added to a Teflon Petri dish ( $\varphi=50 \mathrm{~mm}$ ) and degassed under reduced pressure. The PIC solution was heated at $50^{\circ} \mathrm{C}$ for 24 h to yield a PIC film after evaporation.

## Tensile tests of the PIC, CS, and AG films

The PIC, CS, and AG films were dried under reduced pressure at $25^{\circ} \mathrm{C}$. Stress-strain curves were obtained for rectangular samples $(4.5 \mathrm{~cm} \times 1.0 \mathrm{~cm})$ at a crosshead speed of $1 \mathrm{~mm} / \mathrm{min}$ and a gauge length of 2.5 cm . The elastic moduli (stress/strain) were calculated from the straight line portions of the stress-strain curves.

## Dehydration condensation of PIC skin layer

The wrinkled film was placed in a vacuum oven (ca 10 Pa ) that was gradually heated $\left(10^{\circ} \mathrm{C} / \mathrm{h}\right)$ to $160^{\circ} \mathrm{C}$. The film was then cooled to room temperature under reduced pressure.

## Film drying under stress

The CS film with a PIC skin layer that was prepared via immersion treatments with a $1.0 \mathrm{wt} \%$ AG solution at $30^{\circ} \mathrm{C}$ for 24 h was cut into a rectangle $(4.5 \mathrm{~cm} \times 3.0 \mathrm{~cm})$. The film
was clamped, a weight ( 50 g ) was added to one end, and the film was set to hang for 12 h in air at $40^{\circ} \mathrm{C}$.

## 3T3 cell cultures on wrinkled films

The wrinkled CS films (diameter: 14 mm ) were immersed in PBS $(50 \mathrm{~mL})$ containing a PSN antibiotic mixture $(0.5 \mathrm{~mL})$ for 1 h . These films were then placed in wells of 24 -well culture plates (FALCON ${ }^{\oplus}$, Corning, NY, USA). Each film was fixed with a silicon ring (outside diameter: 16 mm ; inside diameter: 13 mm ), and 0.5 mL of a 3 T 3 cell solution ( $3.4 \times 10^{4}$ cells $/ \mathrm{mL}$, 3 T 3 cells were diluted with DMEM) was seeded and cultured for 2 days ( $37^{\circ} \mathrm{C}, 5 \% \mathrm{CO}_{2}$ ). The cells were then washed twice with 0.5 mL of PBS and then peeled with trypsin. The peeled cells $(10 \mu \mathrm{~L})$ were mixed with trypan blue $(10 \mu \mathrm{~L})$, and the number of cells was calculated by a hemocytometer and an optical microscope.

## Results and discussion

## Preparation of wrinkled CS films

We investigated the effect of AG concentrations on wrinkle morphology using low-viscosity AG ( $80-120 \mathrm{cps}$ ) solutions. PIC skin layers were fabricated via an immersion treatment of CS in $0.25,0.50$, or $1.00 \mathrm{wt} \%$ AG solutions at $30^{\circ} \mathrm{C}$ for 24 h . Films were then washed by soaking in water at $30^{\circ} \mathrm{C}$ for 6 h . The resulting films were dried at $40^{\circ} \mathrm{C}$. The duration for film washing was optimized (see Supplemental Material). Figure 2b-d shows SEM images of the films that were obtained. Microscopic wrinkles were observed for the 0.50 and $1.00 \mathrm{wt} \%$ treatments, indicating that the PIC skin layer induced surface wrinkling as expected. In contrast, semispherical bumps were observed for the $0.25 \mathrm{wt} \%$ treatment. Such surfaces are induced by a thin and/or softer skin layer $[32,33]$. In this system, the depth that the AG permeated into a CS film corresponded to the thickness of the skin layer. This should be proportional to the amount of AG adsorbed. Therefore, we measured the amount of AG that was adsorbed from the weight differences of the films before and after treatments. The adsorbed amounts from the $0.25 \mathrm{wt} \%$ treatment were $0.43 \pm 0.10 \mathrm{mg} / \mathrm{cm}^{2}$, significantly lower than those obtained from 0.50 and $1.00 \mathrm{wt} \%$ treatments ( $0.85 \pm 0.14$ and $0.93 \pm 0.20 \mathrm{mg} / \mathrm{cm}^{2}$, respectively), suggesting that the semispherical bumps were induced by a thinner skin layer that was caused by the lower osmotic pressure of the $0.25 \mathrm{wt} \%$ AG solution. These results indicate that the production of wrinkles and semispherical bumps could be controlled by the concentration of AG.

The effect of immersion temperature on wrinkle morphology was investigated with PIC skin layers that were prepared via immersion treatments with $1.0 \mathrm{wt} \%$ AG


Fig. 2 SEM images of the original CS film surface (a) and film surfaces obtained via the skin-layer formation process at $30^{\circ} \mathrm{C}$ for 24 h with AG solutions of $0.25,0.50$ and $1.00 \mathrm{wt} \% ~(\mathbf{b}, \mathbf{c}$, and $\mathbf{d}$, respectively)
solutions at $30,40,50$, and $60^{\circ} \mathrm{C}$ for 24 h . Figure $3 \mathrm{a}-\mathrm{e}$ shows SEM images of the surfaces of the films that were obtained and their detailed characterization. Wrinkle size gradually increased with immersion temperature. Mean wrinkle wavelengths and amplitudes after treatment at 30 , 40,50 , and $60^{\circ} \mathrm{C}$ were $1.1 \pm 0.2,2.2 \pm 0.6,4.0 \pm 0.8$, and $5.1 \pm 0.8 \mu \mathrm{~m}$, respectively, and $0.3 \pm 0.1,0.5 \pm 0.2,0.9 \pm$ 0.3 , and $1.0 \pm 0.3 \mu \mathrm{~m}$, respectively. This result indicated that temperature was a suitable parameter for controlling the wrinkle size during skin layer formation. The amounts of AG adsorbed under treatment at $30,40,50$, and $60^{\circ} \mathrm{C}$ were $0.93 \pm 0.20,0.98 \pm 0.16,1.18 \pm 0.13$, and $1.22 \pm$ $0.19 \mathrm{mg} / \mathrm{cm}^{2}$, respectively. Thus, the amounts gradually increased with immersion temperature. When the CS film was immersed in water at $30,40,50$, or $60^{\circ} \mathrm{C}$, the CS weight gradually decreased with increases in immersion temperature $(-0.32 \pm 0.10,-0.63 \pm 0.21,-0.08 \pm 0.33$, and $-2.32 \pm 0.19 \mathrm{mg} / \mathrm{cm}^{2}$, respectively), probably due to elution of soluble CS. Thus, the actual amounts of AG that were adsorbed were much higher than the amounts described above. The amounts of AG adsorbed and CS lost were 1.25, $1.61,2.26$, and $3.54 \mathrm{mg} / \mathrm{cm}^{2}$ in the case of $30,40,50$, and $60^{\circ} \mathrm{C}$, respectively. These results suggest that the AG permeation depth increased with increasing immersion temperatures, and in turn formed a broader skin layer. The broader skin layers induced larger wrinkles as well as the relations shown in Eqs. (1) and (2).

We also evaluated the process using solutions with higher molecular weights of AG (300-400 and 500-600 cps). When the PIC skin layers were prepared via the immersion treatments with $1.0 \mathrm{wt} \%$ AG solutions at $30^{\circ} \mathrm{C}$ for 24 h , semispherical bumps were observed (Fig. 4). The amounts of AG that were adsorbed using 300-400 and 500-600 cps solutions were $0.43 \pm 0.21$ and $0.55 \pm 0.18 \mathrm{mg} / \mathrm{cm}^{2}$, respectively. These values are comparable to those that were obtained using a $0.25 \mathrm{wt} \% \mathrm{AG}(80-120 \mathrm{cps})$ solution, where semispherical bumps formed as described above (Fig. 2). This result suggests that the production of thinner skin layers is


Fig. 3 SEM images of the surfaces of films obtained via the skin layer formation process at $30,40,50$, and $60^{\circ} \mathrm{C}$ for 24 h with a $1.0 \mathrm{wt} \% \mathrm{AG}$ solution ( $\mathbf{a}, \mathbf{b}, \mathbf{c}, \mathbf{d}$, respectively), and the mean wrinkle wavelength and amplitude of wrinkles (e)


Fig. 4 SEM images of the surfaces of films obtained via skin layer formation at $30^{\circ} \mathrm{C}$ for 24 h with $1.0 \mathrm{wt} \%$ solutions of higher molecular weight AG (300-400 or 500-600 cps) (a and b, respectively)
due to the difficulty of high molecular weight AG permeating into the CS film. To confirm the production of thinner skin layers, we conducted an optical microscopic analysis of the cross-sections of the films with wrinkles and bumps (Fig. 5). The image shows the topmost layer, which is considered to be the skin layer, with an $80-120 \mathrm{cps}$ solution of AG. This topmost layer was approximately $0.7-\mu m$-thick. On the other hand, using the $300-400 \mathrm{cps}$ solution of AG , the topmost layer could not be distinguished. These results indicate that semispherical bumps can be produced by using a higher molecular weight of AG because of the production of a much thinner skin layer.

When a higher molecular weight CS was used ( $M_{n}=$ $\left.162.4 \times 10^{3}\left(M_{W} / M_{n}=1.98\right)\right)$, microstructured surfaces, including wrinkles and bumps, did not form. We suspect that AG could not permeate into films that were created by higher molecular weight CS because they absorb very little water, probably due to strong aggregation. Indeed, the lower molecular weight CS films swelled 2.6 -fold in weight by


Fig. 5 Optical microscopic images of the cross-sections of films with wrinkles and bumps that were prepared with $80-120 \mathrm{cps}$ (a) and $300-400 \mathrm{cps}$ (b) of AGs
immersion in water, whereas those films prepared with a higher molecular weight CS exhibited minimal swelling (1.3-fold in weight). Hence, CS films prepared using lower molecular weight CS, which have moderate water miscibility, are suitable for this system.

## Elastic modulus of the PIC skin layer

Although the difference in the elastic moduli of the skin layer and substrate is important for wrinkle development, surface wrinkling upon drying has never been reported. The elastic moduli of the PIC skin layer can be estimated from the elastic moduli of the bulk PIC films. Thus, the elastic moduli of the PIC films composed of CS and AG were measured by tensile tests. In the PIC skin layer, it is unlikely that the amount of AG will exceed that of CS, because AG penetrates into the CS film. Accordingly, PIC films were prepared with $1 / 1,1 / 2$, and $1 / 3 \mathrm{AG} / \mathrm{CS}$ ratios by weights. Figure 6 shows the elastic moduli of the PIC, CS and AG films. The elastic modulus of the CS film was $1.19 \pm 0.07$


Fig. 6 Elastic moduli of the PIC films (AG/CS $=1 / 1,1 / 2,1 / 3$ ), the CS film (0/1), and the AG film (1/0)

GPa. The elastic moduli of the PIC films prepared with $1 / 1$, $1 / 2$, and $1 / 3 \mathrm{AG} / \mathrm{CS}$ ratios were $2.43 \pm 0.14,2.53 \pm 0.26$, and $2.18 \pm 0.24 \mathrm{GPa}$, respectively; these enhanced values were due to ionic crosslinking with AG. In the widely used surface wrinkling system based on fabrication of a skin layer on strained poly(dimethylsiloxane) (PDMS), the elastic moduli of the skin layers $(\sim 1 \mathrm{GPa})$ are ca. 1000-fold higher than that of PDMS ( $\sim 1 \mathrm{MPa}$ ) [13, 34]. Surprisingly, in our system, the difference was only two-fold. This finding suggests that the surface wrinkling upon drying does not require a large difference in elastic moduli between the skin layer and the substrate.

## Preparation of wrinkled films with a thermally crosslinked PIC skin layer

Wrinkled films must be stable under aqueous conditions for use in biomedical applications. Therefore, we evaluated the stability of wrinkled films that were prepared via skin layer formation at $30^{\circ} \mathrm{C}$ for 24 h with a $1.0 \mathrm{wt} \% \mathrm{AG}$ solution. Figure 7a, b shows the optical microscopic images of wrinkled films after soaking in water at $37^{\circ} \mathrm{C}$ for 24 and 48 h , respectively. Although the wrinkles were relaxed by hydration of the skin layer, they remained after 24 h . However, after 48 h of hydration, the wrinkles completely disappeared. Therefore, we investigated dehydration condensation of the PIC skin layer by a thermal treatment under reduced pressure to stabilize the wrinkle morphology. The film that was obtained via a thermal treatment under ca. 10 Pa at $120^{\circ} \mathrm{C}$ for 12 h was extremely brittle, probably due to hydrolysis that was promoted by adsorbed water molecules. Therefore, the wrinkled film was gradually heated from room temperature to $160^{\circ} \mathrm{C}\left(10^{\circ} \mathrm{C} / \mathrm{h}\right)$ under ca. 10 Pa , so that adsorbed water molecules would be removed at lower temperatures. Figure 7e-d' shows the IR spectrum of the wrinkled films after thermal treatment. In the spectrum, the absorption peak attributed to the $\mathrm{C}=\mathrm{O}$ (ester) stretching vibration is present at $1770 \mathrm{~cm}^{-1}$. Generally, primary amines have greater nucleophilicity than primary alcohols. Therefore, we suspect that both esters and amide bonds
were actually present in the skin layer, even though the absorption peak attributed to the amide bond at approximately $1650 \mathrm{~cm}^{-1}$ was overlapped. Figure 7c, d shows optical microscopic images of the thermally crosslinked wrinkled film after soaking for 24 h and 1 week, respectively. Wrinkle morphologies were preserved after both soaking durations, with mean wrinkle wavelengths after 24 h and 1 week of $2.1 \pm 0.5$ and $3.0 \pm 0.4 \mu \mathrm{~m}$, respectively. This result indicates that the wrinkles induced by the PIC skin layers can be made water resistant by thermal crosslinking.

## 3T3 cell cultures on wrinkled surfaces

The effect of wrinkled surfaces and thermally crosslinked PIC skin layers on cell culture warrants investigation, because we previously reported that microscopic wrinkled surfaces composed of CS and oligo(ferulic acid) showed good performance as a cell culture substrate [35]. To evaluate the effects of wrinkled surfaces, 3T3 cells were cultured on a surface obtained via the skin layer formation method at $30^{\circ} \mathrm{C}$ for 24 h with a $1.0 \mathrm{wt} \% \mathrm{AG}$ solution, and its thermally crosslinked surface. The crosslinked anisotropic wrinkled surface (Figure S2) was also examined in order to evaluate the effect of wrinkle morphology. Figure 8 a shows the number of 3 T 3 cells that were peeled from the substrates by trypsin treatment after 2 days of incubation. Cell numbers on the wrinkled and crosslinked wrinkled surfaces were $16.0 \times 10^{3}$ and $5.0 \times 10^{3} \mathrm{~mL}^{-1}$, respectively, which were lower than that obtained using polystyrene cell culture dishes.

In our previous report, the cell numbers obtained on surfaces with a similar wrinkle size were comparable to those using polystyrene. However, the number of cells on the crosslinked wrinkled surface was much lower than that obtained with a wrinkled surface. This indicated that wrinkled surfaces composed of crosslinked CS and AG were less effective for 3 T 3 cell adhesion compared to wrinkled surfaces composed of CS and oligo(ferulic acid). This was probably due to a decrease of positive charges caused by amide formation. Indeed, the adhesion of various cells on CS films was decreased by N -acetylation [36, 37]. Interestingly, on the crosslinked anisotropic wrinkled surface, the cell numbers were below $0.5 \times 10^{3} \mathrm{~mL}^{-1}$, suggesting that 3 T 3 cells adhered weakly to this surface. Indeed, as shown in Fig. 8b, no 3T3 cells were observed on the surface. (On the polystyrene plate, 3T3 cells were nearly confluent, as shown in Fig. 8c.) This finding demonstrates that thermally crosslinked wrinkled surfaces constitute a low cell-adhesion substrate. Moreover, cell adhesion was further weakened by wrinkle alignment. Wrinkled films that exhibit low cell adhesion and good biocompatibility may be suitable for use in an implantable device [38]. Clarification


Fig. 7 Rendering the wrinkled CS film with a PIC skin layer water resistant. Optical microscopic images of the wrinkled film after soaking in water for $24 \mathrm{~h}(\mathbf{a})$ and $48 \mathrm{~h}(\mathbf{b})$; of the thermally crosslinked wrinkled films after soaking for $24 \mathrm{~h}(\mathbf{c})$ and 1 week (d); and IR spectra of CS ( $\mathbf{a}^{\prime}$ ), AG ( $\mathbf{b}^{\prime}$ ), the wrinkled film (c'), and the thermally crosslinked wrinkled film (d') (e)


Fig. 8 Cell cultures on wrinkled films. The number of 3 T3 cells that were peeled from the surface by trypsin treatment after incubation for 2 days (a). Optical microscopic images of the thermally crosslinked anisotropic wrinkled surface (b) and a polystyrene plate (c) after 2 days
of the mechanisms involved in weakened cell adhesion is now in progress.

## Conclusions

We investigated a novel surface wrinkling system based on the fabrication of a PIC skin layer. PIC skin layers were fabricated on the surface of CS films using a simple procedure that involved immersion in a sodium alginate (AG) solution and subsequent immersion in water to remove excess AG. Upon drying, microscopic wrinkles appeared. The wrinkle size could be controlled within a range of ca. $1-5 \mu \mathrm{~m}$ by the immersion temperature during skin layer formation because of the differences in the thicknesses of the skin layers. We also observed that the formation of thinner PIC skin layers using a low concentration AG solution or high molecular
weight AG led to the formation of bumps. Furthermore, we demonstrated that there was only a two-fold difference in the elastic moduli between the PIC skin layer and the CS film. This result was very different from that obtained with a surface wrinkling system that was induced by mechanical stress. Thus, a design principle far from that generally used was necessary to develop a surface wrinkling system upon drying. Indeed, although the PIC between CS and AG is widely used, this wrinkling phenomenon has never been reported. This study is the first to demonstrate that the PIC skin layer is an effective tool for inducing surface wrinkling upon drying. This design concept could enable fabrication of various microscopic surfaces and simplification of surface wrinkling upon drying.

Moreover, we successfully rendered the wrinkled films water resistant by thermal crosslinking of the PIC skin layer. We then investigated the growth of 3 T 3 cells on the
film. The thermally crosslinked wrinkled surfaces composed of covalently crosslinked CS and AG exhibited low cell adhesiveness, which is required for implantable materials. This method, involving surface wrinkling induced by a PIC skin layer and subsequent thermal crosslinking, will provide various wrinkled biomaterials that are applicable for wound healing [39], biological adhesives [40], and implantable drug delivery systems [41].

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## Compliance with ethical standards

Conflict of interest The authors declare that they have no conflict of interest.

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