Isothermal crystallization of poly(L-lactide) and poly(butylene adipate) crystalline/crystalline blends

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The isothermal crystallization kinetics, morphology and melting behavior of poly(L-lactide) and poly(butylene adipate) (PLLA/ PBA) crystalline/crystalline blends were studied in the high- and low-temperature regions using differential scanning calorimetry (DSC) and polarized optical microscopy (POM). In the high-temperature region, the PLLA crystallized faster, and the blends had lower equilibrium melting temperatures than neat PLLA due to the high-mobility PBA chain. In the low-temperature region, the PBA crystallization rate was lower in the blends. Additionally, the polymorphic structures and crystallization morphologies of PBA were influenced by PLLA greatly. The PBA in the blends formed α and β crystal structures, whereas neat PBA only formed α crystal structures. POM indicated that the neat PBA formed Maltese-cross spherulites across the studied temperature range. However, banded spherulites formed in the blend at 32 and 34 °C.

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INTRODUCTION

With increasing public awareness of the environmental hazards of polymer disposal, increasing attention has been given to biodegradable polymers in recent years. Of all biodegradable polymers, poly(Llactide) (PLLA) stands out because it is biocompatible and non-toxic to both the environment and the humans. However, its slow crystallization rate, low crystallinity and inherent brittleness limit its applications in many fields.^{1,2}

Blending PLLA with biodegradable polymers is a useful, economical way to improve its properties while maintaining its biodegradability. Many biodegradable aliphatic polyesters, such as poly(butylene succinate),³ poly(ϵ -caprolactone),⁴ poly(3-hydroxybutyrate)⁵ and poly(ethylene oxide),⁶ have been blended with PLLA. An important feature of these aliphatic polyesters is their crystallinity. Therefore, these PLLA blends form crystalline/crystalline systems that can exhibit varying crystallization behaviors in complicated crystallization environments. Thus, the crystalline/crystalline blends not only improve the material usability but also supply many interesting samples that can be used to investigate crystallization behavior. Many works have been published on the miscibility, morphology, crystallization kinetics and melting behavior of PLLA blends, such as poly(butylene succinate)/PLLA,³ poly(3-hydroxybutyrate)/ poly(L-lactic acid)⁵ and poly(ethylene succinate)/PLLA.^{6,7} The morphologies of the crystalline/crystalline blends have been shown to be influenced by the miscibility, crystallization process and unique morphologies that are occasionally formed.8-15 Furthermore, the kinetics and melting behavior were also altered by varying crystalline environments.^{7,8,11,15} These results provide many new choices that can be used to modify the material properties. Thus, the crystalline behavior of the crystalline/crystalline blends is worth studying from both the academic and industrial viewpoints.

Poly(butylene adipate) (PBA) is a biodegradable aliphatic polyester with abundant crystallization phenomena, such as complex crystalline morphologies,¹⁶ multiple melting behaviors and polymorphism.^{17,18} PBA was shown to improve the mechanical properties of PLLA.¹⁹ Thus, PLLA/PBA crystalline/crystalline blends are a valuable system for investigating polymer crystallization behavior. Furthermore, the melting points of PLLA and PBA greatly differ; the melting temperature of PLLA is ~160 °C, while the melting temperature of PBA is below 75 °C. Therefore, their crystallization processes can be studied separately by controlling the temperature. Previously, we investigated the miscibility and morphologies of PLLA/PBA blends in high- and low-temperature regions. These blends were partially miscible and showed various crystallization morphologies.²⁰ However, understanding the isothermal crystallization behaviors of these blends is important to obtain PLLA/PBA blends with desirable properties, which can allow us to design material with proper crystallinity and structures.

Therefore, the present work aims to clarify the interactions between PBA and PLLA by investigating their isothermal crystallization kinetics, melting behavior and crystallization morphologies in different temperature regions. The isothermal crystallization kinetics and melting behavior of the PLLA/PBA blends in the high- and lowtemperature regions were studied using differential scanning

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calorimeter (DSC). The polymorphism structures of the PLLA/PBA blends formed in the low-temperature regions were studied by combining the DSC and wide-angle X-ray diffractometer (WAXD) results. Additionally, the morphology of the blends formed at various temperatures was studied via polarized optical microscope (POM). The obtained results could be helpful for future PLLA/PBA blend applications and for studying the isothermal crystallization and melting behaviors of other polymer blends.

EXPERIMENTAL PROCEDURE

Materials

PLLA with Mw = 250 000 Da was purchased from Nature Works (Blair, NE, USA). PBA with Mw = 68 000 Da was synthesized via the melt polycondensation of 1,4-butanediol (CAS No 110-63-4, supplied by BASF, Ludwigshafen, Germany) and adipic acid (CAS No 124-04-9, supplied by BASF), with tetrabutyl orthotitanate (CAS No 5593-70-4, supplied by Fluka, Zurich, Switzerland) as the catalyst.²¹ The material was purified before use via solution depositing. PLLA and PBA were dissolved in tetrahydrofuran before precipitation from methanol and were dried under vacuum before the investigation.

Blend preparation

The PLLA/PBA blend samples were prepared using the mutual solvent tetrahydrofuran. A total of 10 mg of the polymers (PLLA and PBA) were dissolved in 1 ml of solvent and stirred in a flask. According to the research objectives, the blends were prepared with different compositions. To study the influence of PBA on PLLA crystallization, PLLA/PBA blends with 100/0 and 80/20 weight ratios were prepared. Similarly, to study the effects of PLLA on PBA crystallization behavior, PLLA/PBA blends with the 0/100 and 20/80 weight ratios were prepared. The blend solutions were first cast onto a Petri dish at room temperature. After evaporating the solvent under a controlled stream of air for 3 days, the blend was further dried in a vacuum at 40 °C for 72 h to remove the residual solvent. Cast films were prepared from PLLA/PBA blends with various compositions.

Polarized optical microscopy

The samples used to study the formed morphology via POM were prepared by casting blend solutions on covered glass (20×20 mm in size). After volatilizing the solvent, the samples were melted on a hot stage at 200 °C for 3 min before being transferred to a hot stage at a specific temperature for isothermal crystallization. After a sufficient time, POM pictures of the samples were taken at room temperature using a polarized optical microscope (Nikon Optiphot-2, POM, Nikon, Tokyo, Japan) equipped with a charge-coupled device digital camera.

Differential scanning calorimetry

The isothermal crystallization and thermal behavior of the PLLA/PBA blends were determined using a Mettler-Toledo DSC (DSC1, Zurich, Switzerland) operating under a nitrogen flow of 20 ml per minute. The temperature and heat flow of the instrument were carefully calibrated using standard materials before measurements were taken. Samples of \sim 3–5 mg were encapsulated in the DSC aluminum pan.

Because the PLLA melt crystallization rate is very slow, an isothermal coldcrystallization procedure was adopted for this work. Therefore, the PLLA/PBA samples with weight ratios of 100/0 and 80/20 were first heated to 200 °C for 3 min and then cooled at a rate of 80 °C per minute to a final temperature of 25 °C for their isothermal crystallization. After 3 min, the samples were heated to a given temperature, $T_{\rm c}$ at a rate of 80 °C per minute to finish the isothermal cold-crystallization. After sufficient time passed, the PLLA/PBA blends were heated directly from $T_{\rm c}$ to the melting point at a rate of 10 °C per minute.

Because the PBA crystallizes rapidly, the isothermal crystallization in the low-temperature region was set as an isothermal melt-crystallization. The PLLA/PBA samples with a weight ratio of 0/100 and 20/80 were first heated to 200 °C for 3 min for their isothermal crystallization before cooling at 80 °C per

minute to a given temperature, $T_{\rm cv}$ to finish the isothermal melt-crystallization. After sufficient time had passed, the PLLA/PBA blends were heated directly from 75 °C to the melting point at a rate of 10 °C per minute.

Wide-angle X-ray diffraction

To investigate the polymorphism structures of PBA, PLLA/PBA films with a weight ratio of 0/100 and 20/80 were prepared by first sandwiching the polymer samples between two poly(ethylene terephthalate) sheets, with another poly(ethylene terephthalate) film (0.1 mm in thickness) as a spacer; the samples were then compressed on a hot plate at 200 °C for 2 min at a pressure of 15 MPa. Afterwards, the sandwiched polymer films were rapidly transferred to an oven at the temperature designated for isothermal crystallization. After the complete crystallization, the polymer films were separated for wide-angle X-ray diffraction analysis. The WAXD patterns were recorded at room temperature on a Rigaku D/Max-2500PC diffractometer (Nagano, Japan) using nickel-filtered Cu K α radiation (wavelength $\lambda = 0.154$ nm, 40 kv and 110 mA) across the 2 θ range from 6° to 40°, with a scanning step of 0.02°.

RESULTS AND DISCUSSION

Isothermal crystallization of neat PLLA and PLLA/PBA blends in the high-temperature region

The neat PLLA and PLLA/PBA blend with an 80/20 weight ratio were used to investigate the effects that incorporating PBA had on PLLA crystallization. The isothermal cold crystallization kinetics were first studied in the temperature region from 95 °C to 110 °C using the Avrami equation:^{22,23}

$$1 - X_t = \exp(-kt^n) \tag{1}$$

where X_t is the relative crystallinity, k is the crystallization rate constant associated with nucleation and growth processes and n is the Avrami index, which depends on the nucleation nature and crystal growth spatial dimensions. Figure 1 shows the relative crystallinity (X_t) curves versus the crystallization time for neat PLLA and the PLLA/PBA blend. The observed crystallization time for each sample shortened with increasing crystallization temperature. The blend crystallization time was shorter than that of neat PLLA at the same temperature. For example, neat PLLA requires ~80 min to finish the crystallization process at 95 °C, while the PLLA/PBA blend requires ~50 min.

Avrami plots are shown in Figure 2 to compare the isothermal crystallization of neat PLLA with PLLA/PBA in detail. The parameters K and n were separately obtained from interceptions and the slopes of the plots. The parameter $t_{0.5}$ was calculated according to equation (2). All the parameters including K, n and $t_{0.5}$ are listed in Table 1.

$$t_{0.5} = \left(\frac{\ln 2}{k}\right)^{1/n} \tag{2}$$

The values of *n* for both the neat PLLA and PLLA/PBA blends were between 2.2 and 2.7, which is close to 3. These results indicate that the PLLA crystal grows in three dimensions. Additionally, incorporating PBA did not change the crystallization mechanism of PLLA.

On the basis of the $t_{0.5}$ value, the crystallization rate varied greatly with the crystallization temperature. For neat PLLA and PLLA/ PBA(80/20), the crystallization rate increased with increasing crystallization temperature across the range from 95 to 110 °C. This trend was because the mobility of the molecular chain improved with the increasing crystallization temperature. Comparing the $t_{0.5}$ values for the blended and neat PLLA indicated that the PLLA/PBA blend had a faster crystallization rate than the neat PLLA. As we previously reported,²⁰ PBA has two effects on PLLA crystallization. On one hand, PBA has a diluting effect on PLLA, which reduces the nucleation density. On the other hand, it promotes PLLA mobility and improves the growth rate after nucleation. The combined results

324

Neat PLLA

а

log[-In(1-Xt)]

1

0

-1

-2



Figure 1 Plots of the relative crystallinity degree versus the crystallization temperature. (a) Neat PLLA and (b) PLLA/PBA 80/20. A full color version of this figure is available at Polymer Journal online.

in Table 1 indicate that the second effect is stronger in the PLLA/PBA blend, and PBA loading increases the PLLA crystallization rate.

Figure 3 shows the DSC heating curves after isothermal crystallization for both neat PLLA and PLLA/PBA(80/20). Both the neat PLLA and the PLLA/PBA blend contained two melting peaks: T_{m1} in the low-temperature region and T_{m2} in the high-temperature region. The endothermic peak temperatures for T_{m1} and T_{m2} as a function of $T_{\rm c}$ are shown in Figure 4. The location of $T_{\rm m2}$ is independent of the crystallization temperature, and its intensity decreased with increasing $T_{\rm c}$, while $T_{\rm m1}$ strictly depended on the crystallization temperature.

Therefore, the equilibrium melting temperature (T_m^{0}) can be obtained from the primary crystallization melting peak and crystallization temperature using Hoffman-Weeks's equation:²⁴

$$T_{\rm m} = T_{\rm m}^0 (1 - \frac{1}{r}) + \frac{T_{\rm c}}{r}$$
(3)

where r is a factor dependent on the lamellar thickness. Thus, the equilibrium melting temperature is 163 °C for PLLA and 162 °C for PLLA/PBA. The T_m^0 values were obvious lower than those reported in the literature (206.8 °C).²⁵ This difference is because a commercial PLLA sample was used in this work, and this sample was not 100% pure PLLA. However, the decrease in the equilibrium melting temperature is very small (only 1 °C) for the PLLA/PBA blends, which indicates that the addition of PBA had little effect on the PLLA lamella thickness in the blends.



-⊡- 110°C



Figure 2 Avrami plots for (a) neat PLLA and (b) PLLA/PBA 80/20. A full color version of this figure is available at Polymer Journal online.

Table 1	Isothermal	crystallization	kinetics	parameters	of neat	PLLA
and PLL	A/PBA bleı	nd obtained by	Avrami	fit		

Samples	T_c (°C)	Ν	K (min ⁻ⁿ)	t _{0.5} (min)
Neat PLLA	95	2.7	$3.3 imes 10^{-5}$	40.7
	100	2.8	$3.2 imes10^{-5}$	35.7
	105	2.6	$1.8 imes10^{-4}$	24.9
	110	2.4	4.5×10^{-4}	21.6
PLLA/PBA:80/20	95	2.8	$8.6 imes10^{-5}$	26.2
	100	2.4	$5.2 imes10^{-3}$	19.1
	105	2.2	1.9×10^{-3}	15.4
	110	2.2	2.4×10^{-3}	12.7

Abbreviations: PBA, poly(butylene adipate); PLLA, poly(L-lactide).

Figure 5 shows the POM images for the neat PLLA and PLLA/PBA 80/20 after a sufficient isothermal crystallization time at different temperatures. Both the neat PLLA and the PLLA/PBA blend formed Maltese-cross spherulites. However, the spherulite diameter of the blend was notably larger than the neat PLLA at the same isothermal crystallization temperatures. For example, the neat PLLA spherulite diameter at 100 °C was \sim 7 µm, while that of the PLLA/PBA blend was $\sim 13 \,\mu\text{m}$. Because PBA has a diluting effect on PLLA, the PLLA nuclei density decreased. Additionally, incorporation of low viscosity PBA also promoted the mobility of the PLLA molecular chain and





Figure 3 DSC melting curves after isothermal crystallization at different temperatures using a heating rate of 10°C per minute. (a) Neat PLLA and (b) PLLA/PBA 80/20. A full color version of this figure is available at Polymer Journal online.



Figure 4 Melting points for neat PLLA and PLLA/PBA 80/20 as a function of the crystallization temperature. A full color version of this figure is available at Polymer Journal online.



Figure 5 Polarized optical micrographs for neat PLLA and PLLA/PBA 80/20 morphologies formed at different temperatures. The left column shows images for neat PLLA: (a) 95 °C, (b) 100 °C, (c) 105 °C and (d) 110 °C. The right column shows images for PLLA/PBA 80/20: (e) 95 °C, (f) 100 °C, (g) 105 °C and (h) 110 °C. The scale bar is $10 \,\mu m$.

improved the spherulite growth rate. Therefore, the number of spherulites in the blend was lower than the PLLA under identical conditions, and the blend spherulite diameter was larger than the neat PLLA. The same phenomena have been observed for many crystalline/crystalline blends, such as poly(butylene succinate)/ PBA,¹⁴ polyhydroxybutyrate (PHB)/atactic R,S-PHB-OH²⁶ and poly(lactic acid)/poly(ε-caprolactone).²⁷ The difference in spherulite diameter between the neat and blended PLLA also suggests that the incorporation of PBA promotes PLLA molecular chain mobility.

Isothermal crystallization of neat PBA and PLLA/PBA blends in the low-temperature region

Neat PBA and the PLLA/PBA blend with a weight ratio of 20/80 were used to investigate the effects that the incorporation of PLLA has on PBA crystallization. The isothermal crystallization and melting behavior in the temperature region from 32 to 38 °C were investigated by DSC.

Figure 6 plots the relative crystallinity (Xt) versus the crystallization time for neat PBA and the PLLA/PBA blend. Using equation (1), the crystallization parameters, k, n and t_{0.5}, were determined and are listed in Table 2. The *n* value for the neat PBA is \sim 4, which suggests

а

Heat flow >

30

b

Heat flow >

30

327



38°C

36°C

34°C

32°C

70



Figure 6 Plots of the relative crystallinity degree versus crystallization temperature. (a) Neat PBA and (b) PLLA/PBA 20/80. A full color version of this figure is available at Polymer Journal online.

Table 2 Isothermal crystallization kinetics parameters of neat PBA and PLLA/PBA blend obtained by Avrami fit

Samples	T_c (°C)	n	K (min ⁻ⁿ)	t _{0.5} (min)
Neat PBA	32	4.2	1.03	0.91
	34	4.1	0.59	1.04
	36	4.1	0.24	1.28
	38	4.1	0.10	1.61
PLLA/PBA:20/80	32	4.3	5.3×10^{-2}	1.81
	34	4.1	$3.2 imes 10^{-2}$	2.10
	36	3.2	$1.8 imes 10^{-2}$	3.13
	38	3.4	4.8×10^{-3}	4.37

Abbreviations: PBA, poly(butylene adipate); PLLA, poly(L-lactide)

that the crystals grow three-dimensionally in a homogenous mode. For the PLLA/PBA blend, the *n* value varied from 3.2 to 4.3, which is close to 4. This result indicates that the blend had the same crystallization mechanism as the neat PBA.

On the basis of the $t_{0.5}$ value, the crystallization rate decreased with increasing crystallization temperature for both the neat PBA and PLLA/PBA blends. However, the blend crystallization rate was notably slower than that of neat PBA based on the $t_{0.5}$ values of the two samples. This trend was because PLLA is in a

Figure 7 DSC melting curves after isothermal crystallization at different temperatures using a rate of 10 °C per minute. (a) Neat PBA and (b) PLLA/ PBA 20/80. A full color version of this figure is available at Polymer Journal online.

Temperature/°C

1

Tm₆

∕Tm₇

Tm₄

60

Tm₅

40

Tmg

50

glass state in this temperature region, and solid PLLA limits the mobility of the PBA molecular chain, which decreases the crystallization rate.

PBA contains polymorphic structures, and its melting behavior is very complex. Figure 7 shows the melting curves for neat PBA and the PLLA/PBA blend after their isothermal crystallization in the temperature region from 32 to 38 °C. Figure 7a indicates that neat PBA has two melting peaks: T_{m1} and T_{m2} . As reported in Gan's work,¹⁸ neat PBA forms α crystals in this temperature range. T_{m1} is the original melting point, whereas T_{m2} is the melting point for the recrystallized α crystals.

The PLLA/PBA blend melting behavior significantly differed from the neat PBA (seen Figure 7b). For the temperature region from 36 to 38 °C, the blend had two melting peaks, T_{m1} and T_{m2} . However, at 32 °C, the blends had two different peaks (T_{m3} and T_{m4}) than those observed for 36 and 38 °C. Moreover, at 34 °C, the blend exhibited three melting peaks. This type of multiple melting behavior is similar to the melting of neat PBA polymorphic crystals.¹⁸ Therefore, the PLLA/PBA blend may have formed polymorphic structures in the temperature region from 32 to 38 °C.



Figure 8 WAXD patterns for PLLA/PBA 20/80 films after isothermal crystallization at different temperatures. (a) Neat PBA and (b) PLLA/PBA 20/80. A full color version of this figure is available at *Polymer Journal* online.

WAXD was used to investigate the structures of the neat PBA and PLLA/PBA blend crystals formed at 32, 34 and 36 °C (seen in Figure 8). The neat PBA sample exhibited three diffraction peaks ($\alpha_{(110)}, \alpha_{(020)}$ and $\alpha_{(021)}$). These results suggest that PBA only forms α crystals in this temperature region, which is in agreement with the literature.¹⁷ However, the diffractions peaks for the PLLA/PBA blend differ from those of neat PBA. At 36 °C, three diffraction peaks ($\alpha_{(110)}, \alpha_{(020)}$ and $\alpha_{(021)}$) ascribed to an α crystal were observed. At 32 °C, two apparently new peaks ($\beta_{(110)}$ and $\beta_{(020)}$) formed in positions different from those observed at 36 °C. Furthermore, at 34 °C, all of the α and β crystal peaks can be observed. Thus, the PLLA/PBA blend was determined to form polymorphic PBA crystals in the temperature range from 32 to 38 °C.

The PBA crystal structure is very sensitive to temperature changes.¹⁸ In neat PBA, the thermodynamically stable α crystal forms above 32 °C, and the metastable β crystal forms below 28 °C. For temperatures between 29 and 32 °C, PBA forms mixed α and β crystals. However, in the PLLA/PBA blend, the β and mixed crystals formed at 32 and 34 °C, respectively. Therefore, the addition of PLLA favored β crystal formation. This β crystal formation is related to dynamics and is kinetically preferred.¹⁸ The PBA crystallization rate in the PLLA/PBA blend decreased, as shown in Table 2. Accordingly, the very slow crystallization rate of the PLLA/PBA blend at 32 °C made the formation of the thermodynamic stable α crystal difficult. Thus, the metastable β crystal, which has a fast crystallization rate, formed. Increasing the temperature (34 °C or above) also increased the crystallization rate. The thermodynamically stable α crystal, which has a lower crystallization rate than the β crystal, could then form. These results are in agreement with our previous work, which showed that the PBA α crystal easily forms when the PBA crystallization rate increases owing to the addition of the nucleating agent multimethylbenzilidene sorbitol (TM6).28

Figure 9 shows the POM pictures of the neat PBA and PLLA/PBA blend formed in the temperature region from 32 to 38 °C. Both PBA and PLLA/PBA formed spherulites whose diameters increased with increasing crystallization temperature. However, the PLLA/PBA blend formed banded spherulites at temperatures from 32 to 34 °C, while neat PBA formed Maltese-cross spherulites across this range. The banded spherulite morphology of neat PBA is a consequence of



Figure 9 Polarized optical micrographs for neat PBA and PLLA/PBA blend morphologies after isothermal crystallization at different temperatures. The left column shows images for neat PBA: (a) $32 \,^{\circ}$ C, (b) $34 \,^{\circ}$ C, (c) $36 \,^{\circ}$ C and (d) $38 \,^{\circ}$ C. The right column shows images for PLLA/PBA 20/80: (e) $32 \,^{\circ}$ C, (f) $34 \,^{\circ}$ C, (g) $36 \,^{\circ}$ C and (h) $38 \,^{\circ}$ C.

lamellar twisting and is influenced by the crystallization temperature and molecular weight.²⁹ A neat PBA sample (Mw = 68 000 Da) formed banded spherulites at temperatures from 29 to 31 °C when in the mixed crystal form.³⁰ In contrast to neat PBA, the PLLA/PBA blend formed banded spherulites at 34 °C (mixed crystals) and 32 °C (only β crystals). Such a change may be caused by the presence of solid PLLA, which changes the PBA crystal growth environment. Therefore, blending can provide a valid method for adjusting the morphology of neat PBA.

CONCLUSIONS

The isothermal crystallization and melting behaviors of PLLA/PBA blends and the neat polymers differ significantly. In the hightemperature region, PBA is melted. The melted PBA chains with high mobility improve the isothermal crystallization rate and enlarge the PLLA spherulite diameter. Furthermore, the addition of PBA decreased the blend equilibrium melting temperature. However, the crystallization mechanism did not change with PBA loading. In the low-temperature region, PLLA is in a glass state. The PBA crystallization rate decreased for the blends because of the rigid PLLA chain. Additionally, PLLA greatly affected the polymorphism structures and crystallization morphology of PBA. According to the WAXD results, the PBA in the blends formed α and β crystal structures, while neat PBA only formed α crystals at the same temperature. POM demonstrated that banded PBA spherulites formed in blends. These results indicate that the crystallization rate, morphology and crystal structure of neat PLLA and PBA can be modulated via mutual blending.

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