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ORIGINAL ARTICLE

Ring assembly of silica nanospheres mediated by amphiphilic block copolymers with oxyethylene moieties

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Silica nanospheres (SNSs) of ca. 15 nm in diameter assemble into nanoring structures with the aid of amphiphilic block copolymers containing the oxyethylene side chains poly[(2-ethoxyethyl vinyl ether)-block-(2-methoxyethyl vinyl ether)] [poly(EOVE-b-MOVE)]. Cryo-transmission electron microscope observation confirms that the ring assembly of SNSs occurs in the liquid phase. The SNSs favorably assemble into ring-like nanostructures in the presence of 1–2 wt% EOVE $_{100}$ -b-MOVE $_{310}$ at pH ~ 7.8 and 45 °C. Irregular aggregates of SNSs form at 60 °C, most likely because of the hydrophobic collapse of the thermoresponsive block copolymer. A series of poly(EOVE-b-MOVE) with varying block lengths successfully induce the ring assembly of SNSs, whereas a random copolymer fails, indicating that the polymer's molecular structure critically affects the assembly mode of SNSs. Interestingly, SNSs of a larger size (ca. 30 nm) assemble one-dimensionally into chain-like nanostructures in the presence of EOVE $_{100}$ -b-MOVE $_{310}$.

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INTRODUCTION

The self-assembly of colloidal nanoparticles (NPs) has attracted a great deal of attention for its fundamental scientific interest and technological significance in the fabrication of functional NP arrays. Among the assembled structures of NPs, NP rings are of interest because of their unique electronic,² magnetic³ and optical properties⁴, leading to applications in display,⁵ data storage,⁶ electronic resonators⁷ and spintronic devices,² among others. NP rings as two-dimensional nanostructures are often obtained by drying a suspension of NPs on solid substrates through the evaporation-driven hole formation⁸ and Marangoni flow,9 but the obtained ring structures are large, measuring up to several micrometers. Smaller ring structures can be prepared by using templates, such as lithographic patterns, 10 microspheres, 11 droplets^{4,5} and bubbles,^{12,13} during the solvent evaporation or liquid flow processes. The above methods are substrate-dependent, therefore limiting applications of the resulting ring structures. The liquid-phase self-assembly approach is highly attractive from a practical standpoint because it can provide free-standing ring nanostructures. Many studies have focused on the ring assembly of magnetic NPs such as Co,14 Ni-Co¹⁵ and magnetite, ^{16,17} which have anisotropic dipole interactions that drive ring assembly. Magnetic nanorings can be easily obtained

through the flux-closure of magnetic NPs. 16,18 Au nanorings have been frequently studied for their plasmonic property. The surface of Au NPs have been functionalized with amine and carboxyl groups at opposite ends 19 or with DNA molecules 20 to induce directional coupling between Au NPs. Ligand-stabilized Au NPs have been assembled to form ring structures through electrostatic interactions with a disk-shaped viral protein template. 21 However, the ring assembly of NPs in liquid phase without obvious anisotropic interactions or template is generally a challenging task.

Silica NPs with an amorphous nature and spherical shape are typical isotropic NPs. Silica nanospheres (SNSs), synthesized in the tetraethyl orthosilicate—water biphasic system with L-lysine as the base catalyst, are ideal model NPs for self-assembly in the liquid phase because they have monodisperse size distributions, colloidal stability and sizes that can be controlled to within 15–200 nm.²² We have previously reported the one-dimensional (1D) chain-like structures obtained from SNSs in the liquid phase mediated by the amphiphilic block copolymers PEO–PPO–PEO, which consist of poly(ethylene oxide) (PEO) and poly (propylene oxide) (PPO).^{23,24} The polymers affect the interactions among SNSs by adsorbing to the SNS surface via hydrogen bonding between the ether oxygen and silanol groups.^{23,24} The 1D chain-like

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structure is achieved when the electrostatic repulsion, steric repulsion and hydrophobic attraction between the SNSs with the surfaceadsorbed polymers are properly balanced.²⁴ In brief, the SNSs approach each other owing to weakened electrostatic repulsion at a specific pH and then attach to form a dimer; meanwhile, the surface-adsorbed amphiphilic polymers rearrange to cause steric crowding at the neck of the dimer. The steric crowding is believed to suppress lateral aggregation and favor the subsequent attachment of an SNS to the end of the dimer, leading to 1D chain-like structures.²⁴ Block copolymers in colloidal systems generally introduce a wealth of interactions, such as steric, depletion and hydrophobic interactions, 25 which can be exploited to manipulate NP assembly. 26-28 The polymermediated NP assembly strategy has considerable potential in the fabrication of novel assembled nanostructures through the specific control of the interactions among isotropic NPs.

The assembled structure of SNSs, a result of well-balanced interparticle interactions in the presence of block copolymers, should be critically influenced by the polymer structure, which remains largely unexplored. Here, we report that the 'fusion' of SNSs with a tailor-made block copolymer leads to a novel assembled nanostructure. SNSs assemble into ring nanostructures in the liquid phase with the aid of amphiphilic block copolymers consisting of 2-ethoxyethyl vinyl ether (EOVE) and 2-methoxyethyl vinyl ether (MOVE) units (poly(EOVE-b-MOVE), Figure 1). The poly(EOVE-b-MOVE)s are chosen for the following reasons. First, they interact with the SNS surface through hydrogen bonding between the oxyethylene side chains and silanol groups at the silica surface. Their molecular configuration on the SNS surface is likely to be different from that of the PEO-PPO-PEOs, which adsorb to the SNS through their main chains, thus, a different mode of SNS assembly can be expected. Second, poly(EOVE-b-MOVE)s are amphiphilic at mild temperatures (20-60 °C) because of the thermosensitive nature of the EOVE and MOVE units.^{29,30} The polymer amphiphile-derived interactions, such as steric repulsion and hydrophobic attraction, can be fully exploited to promote SNS assembly at a mild temperature. Finally, block lengths of the poly(EOVE-b-MOVE)s can be selectively and finely controlled through living cationic polymerization,²⁹⁻³¹ which enables systematic investigations of the SNS assembly mediated by these polymers.

EXPERIMENTAL PROCEDURE

Materials

Tetraethyl orthosilicate (TEOS) was purchased from Tokyo Chemical Industry Co., Ltd., Tokyo, Japan. I-lysine was purchased from Wako Pure Chemical Industries, Ltd., Tokyo, Japan. The purchased chemicals were used without further purification. Deionized water was used as the solvent in all experiments. EOVE and MOVE (Maruzen Petrochemical Co., Ltd., Tokyo, Japan) were washed with 10% aqueous sodium hydroxide and then with water, dried overnight over potassium hydroxide pellets and distilled over calcium hydride and then metallic sodium. 1,4-Dioxane (Wako Pure Chemical Industries; >99.5%) was distilled over calcium hydride and then lithium aluminum

Figure 1 Molecular structure of poly(EOVE -b - MOVE).

hydride. 1-(Isobutoxy)ethyl acetate (CH3CH(OiBu)OCOCH3) was prepared from isobutyl vinyl ether (Tokyo Chemical Industry) and acetic acid³¹ and was distilled twice over calcium hydride under reduced pressure. Toluene (Wako Pure Chemical Industries; 99.5%) was dried by passage through solvent purification columns (Glass Contour, Pure Process Technology, Nashua, NH, USA) and distilled over metallic sodium. Commercial Et, 5AlCl, 5 (Nippon Aluminum Alkyls, Ltd., Osaka, Japan; 1.0 m solution in toluene) was used without further purification. All reagents for polymerization, except for toluene, were stored in brown ampules under dry nitrogen.

Preparation of an SNS colloidal suspension

A colloidal suspension containing SNSs of 15 nm in diameter was prepared as previously reported.²² Typically, L-lysine (0.037 g) was dissolved in water (34.80 g) and then TEOS (2.60 g) was added. The mixture was allowed to react at 60 °C for 24 h, using a water bath with magnetic stirring at 500 rpm. Finally, a homogeneous and optically clear suspension of SNSs (2 wt% SiO₂, pH ~ 9.4) was obtained. Colloidal suspension of SNSs of ca. 30 nm in diameter was prepared via the seed-regrowth method. In brief, the as-prepared suspension of 15 nm-SNSs (6.86 g) was added into a lysine aqueous solution (lysine 29 mg, water 27.44 g), followed by the addition of TEOS (2.05 g). The mixture was allowed to react at 60 °C for 24 h using a water bath with magnetic stirring at 500 rpm, yielding a colloidal suspension of SNSs (2 wt% SiO2, pH \sim 9.3) of 30 nm in diameter.

Polymer synthesis and analysis

A series of block and random copolymers of EOVE and MOVE were synthesized by base-assisting living cationic polymerization according to previous reports.^{29–31} The following is the polymerization procedure for EOVE₁₀₀-b- $MOVE_{310}$ as an example ([EOVE]₀ = 1.0 M, [MOVE]_{add} = 3.0 M, [1-(isobutoxy)] ethyl acetate]₀ = 10 mm, [1,4-dioxane] = 1.2 m, $[Et_{1.5}AlCl_{1.5}]_0 = 20 \text{ mm}$, in toluene at 0 °C). A glass tube equipped with a three-way stopcock was dried using a heat gun under dry nitrogen for 10 min before use. Toluene (3.3 ml), 1,4-dioxane (0.60 ml), EOVE (0.80 ml) and 80 mm 1-(isobutoxy)ethyl acetate solution in toluene (0.75 ml) were added to the glass tube and cooled to 0 °C. Polymerization was started by the addition of 200 mm Et_{1.5}AlCl_{1.5} solution in toluene (0.60 ml) into the mixture. After 28 min, when the polymerization of EOVE is close to completion, the second monomer, MOVE (2.1 ml), was added to the reaction mixture. After 90 min, the polymerization was terminated with methanol containing a small amount of an aqueous ammonia solution (0.3 wt %). The reaction mixture was washed with aqueous HCl solution to remove any catalyst residue and was neutralized and concentrated under reduced pressure to give the desired polymer, which was then vacuum-dried over 6 h. The composition of EOVE and MOVE was calculated by ¹H nuclear magnetic resonance spectroscopy (NMR, JEOL JNM-ECA 500 spectrometer). The number-average molecular weight $(M_{\rm n})$ and the polydispersity index $(M_{\rm w}/M_{\rm n})$ $(M_{
m w}$: weight-average molecular weight) were calculated from gel permeation chromatography curves in chloroform at 40 °C with respect to 16 polystyrene standards (Tosoh; $M_{\rm n} = 291 - 1.09 \times 10^6$, $M_{\rm w}/M_{\rm n} < 1.1$). The characterization of the polymers obtained is summarized in Table 1.

Table 1 Characterization of poly(vinyl ether)s

		Degree of polymerization ^a			
Polymer		EOVE	MOVE	$M_{\rm n}^{\rm b}$	$M_{\rm w}/M_{\rm n}^{\rm b}$
1	EOVE ₁₀₀ -b-MOVE ₃₁₀	100 (100)	310 (300)	34 100	1.15
2 3	EOVE ₁₀₀ -b-MOVE ₁₉₄ EOVE ₅₀ -b-MOVE ₁₄₀	100 (100) 50 (50)	194 (200) 140 (150)	28 200 23 200	1.16 1.11
4	EOVE ₁₀₀ -r-MOVE ₂₁₀	100 (100)	210 (200)	33 100	1.13

Abbreviations: EOVE, (2-ethoxyethyl vinyl ether); MOVE, (2-methoxyethyl vinyl ether). ^aCalculated by ¹H NMR spectra. The theoretical degree of polymerization based on the feed monomer composition is presented in the parentheses ^bDetermined by gel permeation chromatography in CHCl₃, polystyrene calibration.



Polymer-mediated assembly of SNSs

In a typical procedure, poly(EOVE-*b*-MOVE) was dissolved at 2 wt% in the asprepared SNS suspension at room temperature, yielding a homogeneous polymer–SNS suspension. Subsequently, the pH value of the suspension was adjusted using 0.01–1 M aqueous HCl solutions. The suspension was then statically heated at 45 °C in an oven for several days.

Characterization

Morphology of the products was observed using a Hitachi S-900 scanning electron microscope (SEM) at an accelerating voltage of 6 kV. Silicon wafers were cleaned with Semico Clean 56 (Furuuchi Chemical, Tokyo, Japan) in an ultrasonic bath for 15 min and then with water for 15 min at room temperature, followed by rinsing with ethanol and drying in air. All of the samples were diluted 10 times with deionized water and then deposited onto clean silicon wafers by dip-coating at a withdrawal speed of 10 mm min⁻¹. Ultraviolet-ozone treatment of the samples was performed to remove any unreacted organic species. To minimize the charging-up of electrons during SEM observation, the samples were sputtered with Pt in an Ar atmosphere using an ion sputter system (Hitachi E-1030) prior to SEM analysis. For cryotransmission electron microscope (cryo-TEM) observation, a sample was diluted 10 times with deionized water and then dropped on a micro-grid. After making the liquid thin film of the sample on the micro-grid by blotting with filter paper, it was vitrified by rapidly immersing into liquid ethane using an EM CPC cryo-station (Leica Microsystems, Wetzlar, Germany). The vitrified sample was observed using a JEOL 2100F TEM operated at 200 kV. The temperature of the sample was maintained below -170 °C throughout the observation.

RESULTS

Block copolymerization

A series of block copolymers [poly(EOVE-b-MOVE)s] with varying compositions were prepared using base-assisting living cationic polymerization. $^{29-31}$ The resulting copolymers have narrow molecular weight distributions ($M_{\rm w}/M_{\rm n} \! \leqslant \! 1.16$); the target compositions are shown in Table 1. Block copolymer 1 has the highest molecular weight among all polymers. Polymer 2 has the same EOVE block length but a shorter MOVE block length as compared with 1. The sizes of both the EOVE and MOVE blocks of polymer 3 are approximately half of that of polymer 1. The EOVE and MOVE blocks turn from hydrophilic to hydrophobic with increasing temperature at ~20 °C and 60 °C, respectively. 29,30 Random copolymer 4 having the similar molecular weight and composition as that of block copolymer 2 was also synthesized.

Polymer-mediated ring assembly of SNSs in the liquid phase

The assembly experiments were conducted at 45 °C to ensure amphiphilicity of the block copolymers. The suspension containing block copolymer 1 (2 wt%) and SNSs (15 nm, 2 wt%) at pH 7.7 gains a bluish color after heating at 45 °C, suggesting the occurrence of Rayleigh scattering (Figure 2a). Figure 2b is an SEM image of the sample showing ring-like nanostructures composed of the SNSs. The number of SNSs comprising only one ring is mostly 3–5. The ring-like nanostructures have a size of ca. 30–50 nm, which are compatible sizes to cause Rayleigh scattering. Cryo-TEM observation is conducted to minimize the effects of the drying process and substrate on the sample structure. The cryo-TEM image (Figure 2c) shows ring-like nanostructures similar to those observed with SEM, indicating that the ring-like nanostructures should form in the liquid phase. In addition to closed-ring nanostructures, open rings are also observed.

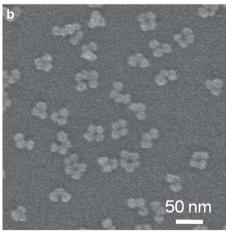
Effects of polymer concentration, pH and temperature on the ring assembly of SNSs

The effect of polymer concentration on ring assembly was examined using block copolymer 1 as an example. The polymer-free SNS suspension is colloidally stable at pH \sim 7.7 and 45 °C. 23,24 However, the SNS suspensions containing 0.1 and 0.5 wt% of 1 at pH 7.8 undergo gelation soon after heating at 45 °C. SEM observation of the gel reveals that the SNSs form dense irregular aggregates (Figures 3a and b). When the concentration of 1 is increased to 1 wt%, the suspension at pH 7.8 remains a homogenous fluid and becomes bluish after heating at 45 °C. SNS nanorings can be observed in the presence of 1 wt% of polymer 1 (Figure 3c), similar to our observations when using 2 wt% of 1 (Figure 2b).

The favorable pH for ring assembly is \sim 7.8 with 2 wt% of 1 at 45 °C (Figure 4a). The SNSs tend to form three-dimensional clusters rather than rings with decreasing pH value (Figures 4b and c). No gelation or phase separation of the suspension is observed even at a lower pH of \sim 6, whereas such phenomena are frequently observed when PEO–PPO–PEO block copolymers are used as the mediator.²⁴

Raising the temperature to 60 °C causes the formation of irregular particle aggregates at pH 7.8 and with 2 wt% of 1 (Figure 5), which are the optimal pH and polymer concentration for the ring assembly of SNSs at 45 °C. At this temperature, 1 should behave as a hydrophobic polymer because both the EOVE and MOVE blocks are hydrophobic at above 60 °C. 29,30





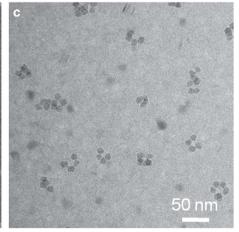


Figure 2 (a) A photograph of the suspension containing 1 (2 wt%) and SNSs at pH 7.7 after heating at 45 °C, (b) SEM and (c) cryo-TEM images of ring structures obtained from the above suspension.

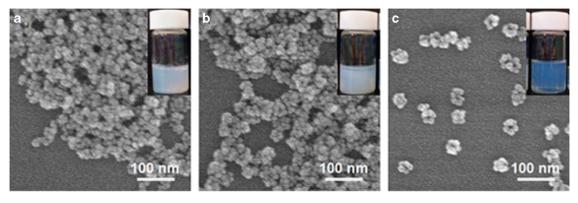


Figure 3 SEM images of the products obtained from the suspensions containing (a) 0.1 wt%, (b) 0.5 wt% and (c) 1 wt% of 1 and SNSs at pH 7.8 after heating at 45 °C. The insets in (a), (b) and (c) are digital photos of the corresponding samples.

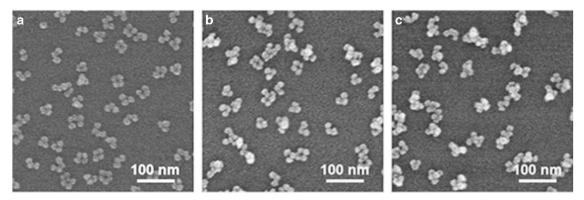


Figure 4 SEM images of the products obtained from the suspensions containing 2 wt% of 1 and SNSs at (a) pH 7.8, (b) pH 7.2 and (c) pH 6.8 after heating at 45 °C.

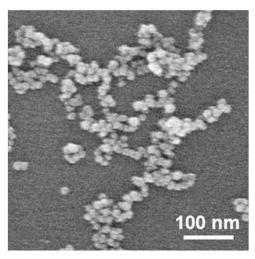


Figure 5 SEM image of the product obtained from the suspension containing 2 wt% of 1 and SNSs at pH 7.8 after heating at 60 °C.

Formation process of ring-like nanostructures

The formation process of ring-like nanostructures was examined using SEM observations of the block copolymer 1 (1 wt%)–SNS suspensions at pH 7.8 heated at 45 °C for different periods of time. SEM images of the products from suspensions heated for ≤1 h show many isolated SNSs and few ring-like nanostructures in addition to 1D chains

(Figures 6a–c). There are small clusters of SNSs on the silicon wafer, which might be caused by the drying process. The appearance of isolated SNSs obviously decreases and ring-like nanostructures emerge after 2–6 h of heating (Figures 6d–f). Ring-like nanostructures are dominant and the isolated SNSs are negligible after 24 h of heating (Figure 6g). The ring-like nanostructures remain unchanged after 3 days of heating (Figure 6h).

Effects of polymer structures on the ring assembly of SNSs

The effect of polymer structures on the SNS assembly was investigated at a fixed polymer concentration (2 wt%) at 45 °C. Block copolymers 2 and 3 with different block lengths also induce the ring assembly of SNSs. Polymer 2, having the same EOVE length but a shorter MOVE length compared with those of 1, gives nanorings at pH 7.8 (Figure 7a), which is similar to the optimal pH for 1. Reducing pH to 7.2 causes the accumulation of nanorings to form larger structures (Figure 7b), which is different from the case with 1 where the rings remain isolated. Nanorings are also observed with polymer 3 at pH 7.8 (Figure 7c), but the structure is less well-defined when compared with those obtained with 1 or 2. Similar to what is observed with polymer 2, the nanorings tend to accumulate with decreased pH (Figure 7d). The random copolymer 4, which has a similar molecular weight and composition as that of block copolymer 2, does not induce any ring formation. The 4-SNS suspension remains transparent at pH above 8.8, turns turbid at pH 8.3, and phase separates at pH below 7.8 (Figure 8a). SEM images reveal that many SNSs remain isolated at pH



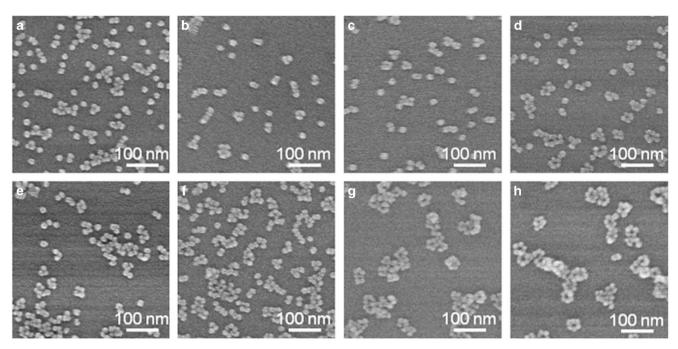


Figure 6 SEM images of the products obtained from the suspension containing 1 wt% of 1 and SNSs at pH 7.8 (a) before heating and (b-h) after heating at $45 \,^{\circ}$ C: (b) $0.5 \, h$, (c) $1 \, h$, (d) $2 \, h$, (e) $3 \, h$, (f) $6 \, h$, (g) $24 \, h$ and (h) $3 \, d$ ays.

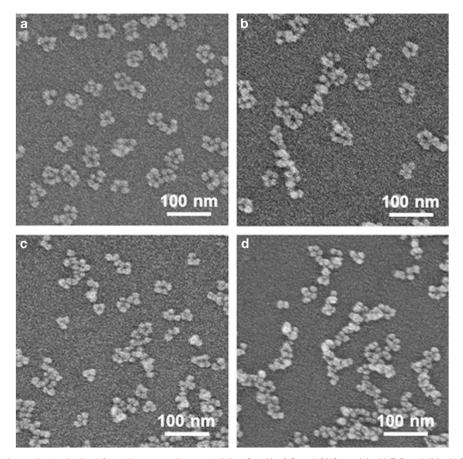


Figure 7 SEM images of the products obtained from the suspensions containing 2 wt% of 2 and SNSs at (a) pH 7.8 and (b) pH 7.2, and 2 wt% of 3 and SNSs at (c) pH 7.8 and (d) pH 7.2 after heating at 45 °C.

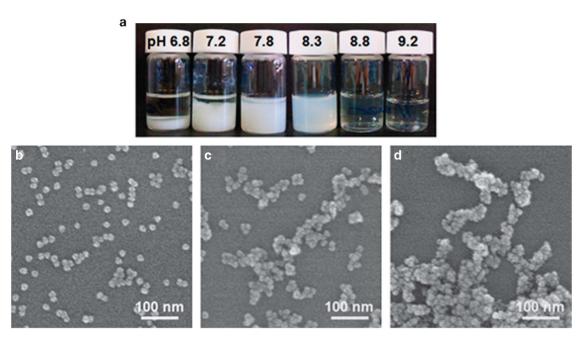


Figure 8 (a) Digital photo of the suspensions containing 2 wt% of 4 and SNSs within pH 6.8–9.2 after heating at 45 °C for 7 days. (b–d) SEM images of the products at (b) pH 8.8, (c) pH 8.3 and (d) pH 7.8.

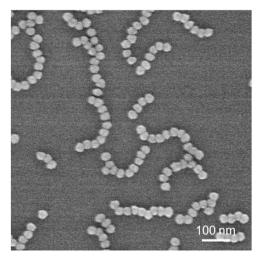


Figure 9 SEM image of the product from the suspension containing SNSs (30 nm) and 2 wt% of 1 after heating at $45\,^{\circ}$ C.

8.8 (Figure 8b) and they formed aggregates at lower pH values (Figures 8c and d).

1D assembly of 30 nm-SNSs

Interestingly, it is found that SNSs that are 30 nm in diameter one-dimensionally assemble into chain-like nanostructures in the presence of 2 wt% of 1 at pH 7.7 and 45 °C (Figure 9), similar to the phenomena observed when PEO–PPO–PEO block copolymers are used as the mediator.²⁴

DISCUSSION

A detailed study of the ring assembly system mediated by $EOVE_{100}$ -b- $MOVE_{310}$ (block copolymer 1) shows that specific experimental

conditions are required to achieve well-defined nanorings composed of SNSs. At least 1 wt % of 1 is required against 2 wt% of SNSs to obtain the nanorings (Figure 3). Lower concentrations of 1 (0.1 and 0.5 wt%) cause random particle aggregation and, consequently, gelation of the suspension. The particle aggregation at low polymer concentrations in polymer - NP colloidal systems is often caused by bridging flocculation, where a polymer molecule simultaneously adsorbs to more than two NPs and causes their aggregation.³² This tendency of random aggregation of SNSs at low polymer concentrations was also observed in our previous study using PEO-PPO-PEO block copolymers as the mediator.²⁴ An optimal pH for ring assembly also exists: nanorings composed of SNSs are obtained in high yields at pH 7.8 in the presence of 1 (2 wt%), whereas clusters of SNSs form at pH below 7.2 (Figure 4). It is known that colloidal silica has an isoelectric point at pH 2-3, above which the silica surface has negative Si-O groups.33 Decreasing the pH value reduces the number of Si-O⁻ groups at the silica surface and weakens the electrostatic repulsions among SNSs. In our previous study, the SNSs generally form random aggregates at pH below 7.0 in the presence of PEO-PPO-PEOs.²⁴ By contrast, the SNSs form isolated clusters instead of irregular aggregates at pH 6.8 in the presence of 1. We propose that the hydrophilic MOVE block of 1 provides considerable steric interactions to suppress the heavy aggregation of SNSs. This hypothesis is supported by the observation of aggregation of SNSs at pH ~ 7 with polymer 2 (Figure 7b), which has a shorter MOVE block than that of 1. The incubation temperature is also an important factor for the formation of nanorings. Only aggregates of SNSs are obtained with 1 at 60 °C (Figure 5), at which both the EOVE and MOVE blocks are hydrophobic. Polymer 1 that adsorbs to the SNS surface may hydrophobically interact, inducing aggregation of the SNSs.

A series of EOVE-b-MOVE block copolymers (1–3) successfully induce ring assembly of SNSs under similar optimal conditions (polymer concentration ~ 2 wt%, pH ~ 7.8) (Figure 4 and Figure 7), whereas the random copolymer 4 fails to give nanorings even within a



wide pH range 6.8–9.2 (Figure 8). Here, the inter-particle interactions in play may include electrostatic repulsion, caused by the surface charges of SNSs, and steric repulsion and hydrophobic attractions, induced by the surface-adsorbed amphiphilic polymers.²⁴ The suspension containing 4 and SNSs is turbid at pH 8.3, suggesting that the net attractive interaction is stronger in the 4 – SNS system when compared with the systems containing the block copolymers 1–3. These block copolymers adsorb to the SNS surface in such a manner that the hydrophilic MOVE blocks may populate the polymer-water interface, whereas the random copolymer adsorbs to the SNS surface homogeneously with both hydrophobic EOVE and hydrophilic MOVE units distributed at the polymer-water interface. The difference of polymer configurations at the polymer-water interface and the SNS surface may greatly affect the assembly mode of SNSs.

The mechanisms of ring assembly of the SNSs, mediated by poly(EOVE-b-MOVE), remain to be clarified. There are two possible routes for the isolated SNSs to assemble into ring structures. One possible route is that the SNSs consecutively attach to each other at a certain angle. However, this route does not explain why only a close ring structure instead of varied open structures (for example, a helix) is produced in this mode. The other possible route is that the SNSs initially assemble into a 1D chain-like structure, which subsequently bends into a ring structure. The formation of chain-like structures with SNSs of 30 nm in diameter confirms that the poly(EOVE-b-MOVE) is capable of mediating the 1D NP assembly, suggesting a relationship between the ring and 1D assembly. It is likely that the 1D chains observed at the early stages of ring assembly (Figures 6a-c) are precursors of the nanorings, and a driving force to achieve an energetically stable configuration forces their bending into the nanorings. The relationship between ring assembly and the percolation phenomena in NP-polymer systems is also worth considering. Here, SNSs form percolated large clusters or networks under specific experimental conditions. Ring-like small clusters might form near the percolation threshold, the value of which varies with the size, conformation and concentration of the polymers.³⁴ A more systematic study to understand the mechanisms of ring assembly will be conducted from both chemical and physical perspectives.

CONCLUSION

Liquid-phase ring assembly of SNSs with a diameter of ca. 15 nm is achieved using the thermosensitive block copolymers poly(EOVE-b-MOVE) as the mediator. A series of block copolymers EOVE50-b-MOVE₁₄₀, EOVE₁₀₀-b-MOVE₁₉₄ and EOVE₁₀₀-b-MOVE₃₁₀ promote the ring assembly of SNSs at pH ~7.8. However, the random copolymer EOVE₁₀₀-r-MOVE₂₁₀ fails to induce ring assembly, suggesting that the primary structure of polymer chains critically affects the assembly of SNSs. Using EOVE₁₀₀-b-MOVE₃₁₀ as an example, it is found that 1-2 wt% of polymer is necessary with 2 wt% of SNSs to induce ring assembly, and the optimal assembly temperature is 45 °C, at which the polymer is amphiphilic. Moreover, EOVE₁₀₀-b-MOVE₃₁₀ promotes the 1D assembly of larger SNSs (30 nm in diameter) under otherwise optimal pH, concentration and temperature conditions for the ring assembly of 15 nm-SNSs, implicating a relationship between ring and 1D assembly modes. Although the mechanisms of ring assembly remain to be explored, this study demonstrates that the 'fusion' of SNSs with block copolymers obtained with the state-of-theart polymer chemistry will open new avenues in liquid-phase particle assembly.

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