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Using metamaterial nanoengineering to triple the superconducting critical temperature of bulk aluminum

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Recent experiments have shown the viability of the metamaterial approach to dielectric response engineering for enhancing the transition temperature, T_c , of a superconductor. In this report, we demonstrate the use of Al_2O_3 -coated aluminium nanoparticles to form the recently proposed epsilon near zero (ENZ) core-shell metamaterial superconductor with a T_c that is three times that of pure aluminium. IR reflectivity measurements confirm the predicted metamaterial modification of the dielectric function thus demonstrating the efficacy of the ENZ metamaterial approach to T_c engineering. The developed technology enables efficient nanofabrication of bulk aluminium-based metamaterial superconductors. These results open up numerous new possibilities of considerable T_c increase in other simple superconductors.

Recent theoretical^{1,2} and experimental³ work have conclusively demonstrated that the metamaterial approach to dielectric response engineering can be used to increase the critical temperature of a composite superconductor-dielectric metamaterial. Indeed, according to Kirzhnits *et al.*⁴ the superconducting properties of a material may be expressed via its effective dielectric response function, as long as the material may be considered as a homogeneous medium on the spatial scales below the superconducting coherence length. The electron-electron interaction in a superconductor may be expressed in the form of an effective Coulomb potential

$$V(\vec{q}, \omega) = \frac{4\pi e^2}{q^2 \varepsilon_{\text{eff}}(\vec{q}, \omega)}, \quad (1)$$

where $V = 4\pi e^2/q^2$ is the Fourier-transformed Coulomb potential in vacuum, and $\varepsilon_{\text{eff}}(q, \omega)$ is the linear dielectric response function of the superconductor treated as an effective medium. Based on this approach, Kirzhnits *et al.* have derived expressions for the superconducting gap, Δ , critical temperature, T_c , and other important parameters of the superconductor. Following this “macroscopic electrodynamics” formalism, it appears natural to use recently developed plasmonics⁵ and electromagnetic metamaterial⁶ tools to engineer and maximize the electron pairing interaction (Eq. 1) in an artificial “metamaterial superconductor”^{1,2} via deliberate engineering of its dielectric response function $\varepsilon_{\text{eff}}(q, \omega)$. For example, considerable enhancement of attractive electron-electron interaction may be expected in such actively studied metamaterial scenarios as epsilon near zero (ENZ)⁷ and hyperbolic metamaterials⁸. In both cases $\varepsilon_{\text{eff}}(q, \omega)$ may become small and negative in substantial portions of the relevant four-momentum (q, ω)

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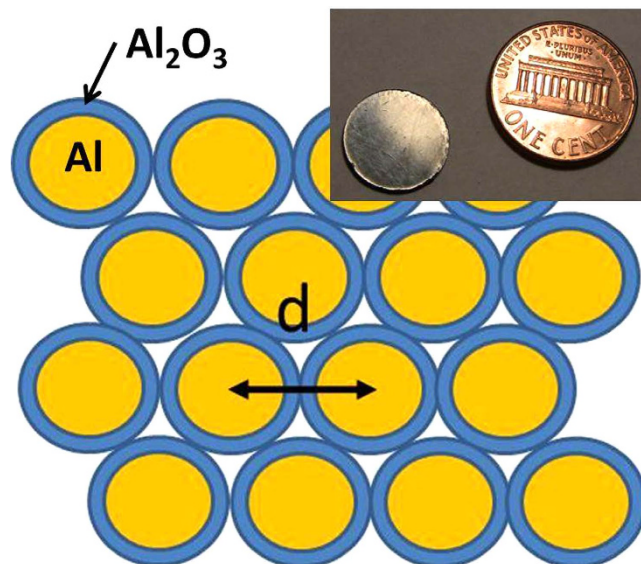


Figure 1. Schematic geometry of the ENZ metamaterial superconductor based on the core-shell nanoparticle geometry. The nanoparticle diameter is $d = 18$ nm. The inset shows typical dimensions of the fabricated bulk aluminium-based core-shell metamaterial.

space, leading to enhancement of the electron pairing interaction. This approach has been verified in experiments with compressed mixtures of tin and barium titanate nanoparticles of varying composition³. An increase of the critical temperature of the order of $\Delta T_c \sim 0.15$ K compared to bulk tin has been observed for 40% volume fraction of barium titanate nanoparticles, which corresponds to ENZ conditions. Similar results were also obtained with compressed mixtures of tin and strontium titanate nanoparticles. These results clearly demonstrated a deep connection between the fields of superconductivity and electromagnetic metamaterials. However, despite this initial success, the observed critical temperature increase was modest. It was argued in² that the random nanoparticle mixture geometry may not be ideal because simple mixing of superconductor and dielectric nanoparticles results in substantial spatial variations of $\varepsilon_{eff}(q, \omega)$ throughout a metamaterial sample. Such variations lead to considerable broadening and suppression of the superconducting transition.

To overcome this issue, it was suggested that an ENZ plasmonic core-shell metamaterial geometry, which has been developed to achieve partial cloaking of macroscopic objects⁹, should be implemented². The cloaking effect relies on mutual cancellation of scattering by the dielectric core (having $\varepsilon_d > 0$) and plasmonic shell (with $\varepsilon_m < 0$) of the nanoparticle, so that the effective dielectric constant of the nanoparticle becomes very small and close to that of vacuum (a plasmonic core with a dielectric shell may also be used). This approach may be naturally extended to the core-shell nanoparticles having negative ENZ behaviour, as required in the superconducting application. Synthesis of such individual ENZ core-shell nanostructures followed by nanoparticle self-assembly into a bulk ENZ metamaterial (as shown in Fig. 1) appears to be a viable way to fabricate an extremely homogeneous metamaterial superconductor.

The design of an individual core-shell nanoparticle is based on the fact that scattering of an electromagnetic field by a sub-wavelength object is dominated by its electric dipolar contribution, which is defined by the integral sum of its volume polarization⁹. A material with $\varepsilon > 1$ has a positive electric polarizability, while a material with $\varepsilon < 1$ has a negative electric polarizability (since the local electric polarization vector, $P = (\varepsilon - 1)E/4\pi$, is opposite to E). As a result, the presence of a plasmonic shell (core) cancels the scattering produced by the dielectric core (shell), thus providing a cloaking effect. Similar consideration for the negative ENZ case leads to the following condition for the core-shell geometry:

$$r_c^3 \varepsilon_c \approx -(r_s^3 - r_c^3) \varepsilon_s, \quad (2)$$

where r_c and r_s are the radii, and ε_c and ε_s are the dielectric permittivities of the core and shell, respectively. Eq. (2) corresponds to the average dielectric permittivity of the core-shell nanoparticle being approximately equal to zero. Working on the negative side of this equality will ensure negative ENZ character of each core-shell nanoparticle. A dense assembly of such core-shell nanoparticles will form a medium that will have small negative dielectric permittivity. Moreover, in addition to obvious advantage in homogeneity, a core-shell based metamaterial superconductor design enables tuning of the spatial dispersion of the effective dielectric permittivity $\varepsilon_{eff}(q, \omega)$ of the metamaterial, which would further enhance its T_c^2 . Spatial dispersion of a metamaterial is indeed well known to originate from plasmonic effects in its metallic constituents. In a periodic core-shell nanoparticle-based ENZ metamaterial spatial dispersion

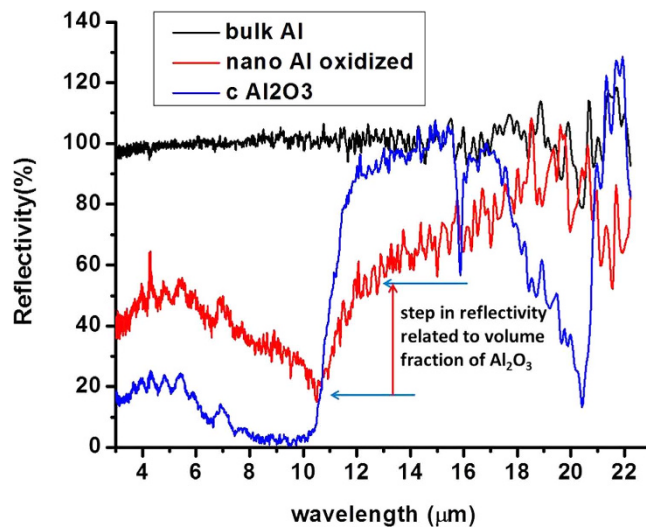


Figure 2. Comparison of the FTIR reflectivity spectrum of a typical core-shell Al_2O_3 -Al metamaterial sample with reflectivity spectra of bulk Al and Al_2O_3 samples. The step in reflectivity around $11\ \mu\text{m}$ may be used to characterize the volume fraction of Al_2O_3 in the core-shell metamaterial. The increased noise near $22\ \mu\text{m}$ is related to the IR source cutoff.

is defined by the coupling of plasmonic modes of its individual nanoparticles. This coupling enables propagating plasmonic Bloch modes and, hence, nonlocal effects.

Here, we report the first successful realization of such an ENZ core-shell metamaterial superconductor using compressed Al_2O_3 -coated aluminium nanoparticles, leading to tripling of the metamaterial critical temperature compared to the bulk aluminium. This material is ideal for the proof of principle experiments because the critical temperature of aluminium is quite low ($T_{c\text{Al}} = 1.2\ \text{K}^{10}$), leading to a very large superconducting coherence length $\xi = 1600\ \text{nm}^{10}$. Such a large value of ξ facilitates the metamaterial fabrication requirements while Al_2O_3 exhibits very large positive values of dielectric permittivity up to $\epsilon_{\text{Al}_2\text{O}_3} \sim 200$ in the THz frequency range¹¹. These results provide an explanation for the long known, but not understood, enhancement of the T_c of granular aluminium films^{12,13}.

The 18 nm diameter Al nanoparticles for these experiments were acquired from the US Research Nanomaterials, Inc. Upon exposure to the ambient conditions a $\sim 2\ \text{nm}$ thick Al_2O_3 shell is known to form on the aluminium nanoparticle surface¹⁴, which is comparable to the 9 nm radius of the original Al nanoparticle. Further aluminium oxidation may also be achieved by heating the nanoparticles in air. The resulting core-shell Al_2O_3 -Al nanoparticles were compressed into macroscopic, $\sim 1\ \text{cm}$ diameter, $\sim 0.5\ \text{mm}$ thick test pellets using a hydraulic press, as illustrated in the inset in Fig. 1.

The IR reflectivity of such core-shell metamaterial samples was measured in the long wavelength IR (LWIR) ($2.5\text{--}22.5\ \mu\text{m}$) range using an FTIR spectrometer, and compared with reflectivity spectra of Al and Al_2O_3 , as shown in Fig. 2. While the reflectivity spectrum of Al is almost flat, the spectrum of Al_2O_3 exhibits a very sharp step-like behaviour around $11\ \mu\text{m}$ that is related to the phonon-polariton resonance, which results from coupling of an infrared photon with an optic phonon of Al_2O_3 ¹⁵. The step in reflectivity is due to the negative sign of $\epsilon_{\text{Al}_2\text{O}_3}$ near resonance. This step-like behaviour may be used to characterize the volume fraction of Al_2O_3 in the core-shell metamaterial. In the particular case shown in Fig. 2, the volume fraction of Al_2O_3 in the core-shell metamaterial may be estimated, based on the Maxwell-Garnett approximation, as $\sim 39\%$, which corresponds to $(r_s - r_c) \sim 0.18r_c$. At $r_c \sim 9\ \text{nm}$ the corresponding thickness of Al_2O_3 appears to be $(r_s - r_c) \sim 1.6\ \text{nm}$, which matches expectations based on¹⁴.

The Kramers-Kronig analysis of the FTIR reflectivity spectra of the Al- Al_2O_3 sample also allows us to evaluate $\epsilon_{\text{eff}}(0, \omega)$ for the metamaterial in the LWIR spectral range. Plots of the real part of ϵ for pure Al and for the Al- Al_2O_3 core-shell metamaterial based on the Kramers-Kronig analysis of the data in Fig. 2 are plotted in Fig. 3. The plot in Fig. 3(a) clearly demonstrates that $\epsilon_{\text{Al-}\text{Al}_2\text{O}_3} \ll \epsilon_{\text{Al}}$ so that the ENZ condition was achieved in the sense that the initial dielectric constant of aluminium was reduced by a factor ~ 1000 . On the other hand, Fig. 3(b) demonstrates that the dielectric constant of the Al- Al_2O_3 core-shell metamaterials remains negative and relatively small above $11\ \mu\text{m}$. In particular, the large negative contribution to ϵ from the aluminium cores is compensated by the large positive contribution from the Al_2O_3 shells leading to the upturn of $\epsilon_{\text{Al-}\text{Al}_2\text{O}_3}$ that is observed near $20\ \mu\text{m}$ in Fig. 3(b) which is caused by the large positive value of $\epsilon_{\text{Al}_2\text{O}_3}$ in this spectral range. Note that while both metamaterials shown in Fig. 3(b) exhibit much smaller ϵ compared to the bulk aluminium, the metamaterial prepared using less oxidized aluminium nanoparticles exhibits considerably larger negative ϵ . The relatively large noise observed in the calculated plot of ϵ_{Al} in Fig. 3(a) is due to the fact that the aluminium reflectivity is close to 100% above $7\ \mu\text{m}$ so that the Kramers-Kronig-based numerical analysis of the reflectivity data does

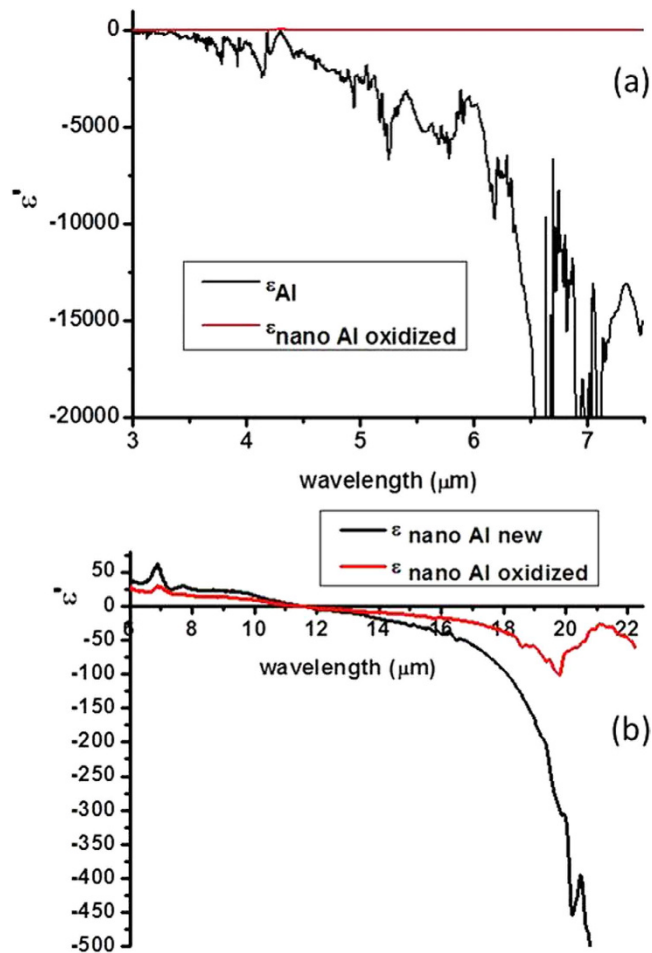


Figure 3. The plots of the real part of ϵ for pure Al and for the Al-Al₂O₃ core-shell metamaterial based on the Kramers-Kronig analysis of the FTIR reflectivity data from Fig. 2: **(a)** Comparison of ϵ for pure Al and for the Al-Al₂O₃ metamaterial clearly indicates that $\epsilon_{\text{Al-Al}_2\text{O}_3} \ll \epsilon_{\text{Al}}$. **(b)** Real part of ϵ for two different Al-Al₂O₃ core-shell metamaterials based on the Kramers-Kronig analysis. While both metamaterials shown in **(b)** exhibit much smaller ϵ compared to the bulk aluminium, the metamaterial prepared using less oxidized aluminium nanoparticles exhibits considerably larger negative ϵ .

not work reliably for pure aluminium samples in this spectral range. Another limitation on the accuracy of the analysis is the use of the finite spectral range (2.5–22.5 μm) of the FTIR spectrometer rather than the infinite one assumed by the rigorous Kramers-Kronig analysis. These limitations notwithstanding, we note that our result for pure aluminium is in good agreement with the tabulated data for ϵ_{Al} reported in¹⁶. Therefore, these results reliably confirm the ENZ character of the core-shell Al-Al₂O₃ metamaterial. It is also interesting to note that the same FTIR technique applied to the tin-BaTiO₃ nanocomposite metamaterials studied in³ also confirms their expected ENZ character as illustrated in Fig. 4. In both cases, the goal of metamaterial engineering was to create an effective superconducting medium with negative ENZ response. While both Figs 3 and 4 confirm that this goal has been achieved, the core-shell geometry of the developed Al-Al₂O₃ metamaterial has a clear advantage. The core-shell geometry guarantees a homogeneous spatial distribution of the effective dielectric response function, leading to tripling of T_c for the Al-Al₂O₃ core-shell metamaterial, compared to $\sim 5\%$ increase of T_c of the tin-BaTiO₃ random nanocomposite metamaterial developed in³.

The T_c of various Al-Al₂O₃ core-shell metamaterials was determined via the onset of diamagnetism for samples with different degrees of oxidation using a MPMS SQUID magnetometer. The zero field cooled (ZFC) magnetization per unit mass versus temperature for several samples with various volume fractions of Al₂O₃ is plotted in Fig. 5(a), while the corresponding reflectivity data are shown in Fig. 5(b). Even though the lowest achievable temperature with our MPMS SQUID magnetometer was 1.7 K, we were able to observe a gradual increase of T_c that correlated with an increase of the Al₂O₃ volume fraction as determined by the drop in reflectivity shown in Fig. 5(b). The reliability of the MPMS SQUID magnetometer at $T > 1.7$ K was checked by measurements of the T_c of bulk tin at $T_c = 3.7$ K, as described in ref. 3, in excellent agreement with the textbook data¹⁰.

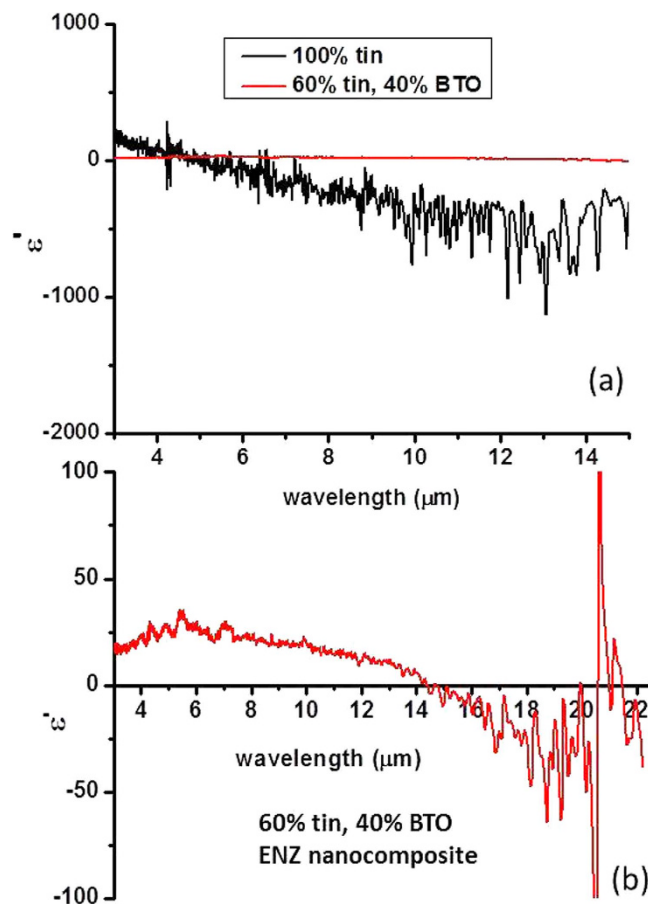


Figure 4. The plots of the real part of ϵ for pure tin and for the ENZ tin-BaTiO₃ nanocomposite metamaterial studied in ³: (a) Comparison of ϵ' for compressed tin nanoparticles and for the tin-BaTiO₃ nanocomposite metamaterial. (b) Real part of ϵ for the tin-BaTiO₃ nanocomposite metamaterial.

The observed increase in T_c also showed good correlation with the results of the Kramers-Kronig analysis shown in Fig. 3(b): samples exhibiting smaller negative ϵ demonstrated higher T_c increase. The highest onset temperature of the superconducting transition reached 3.9 K, which is more than three times as high as the critical temperature of bulk aluminium, $T_{cAl} = 1.2 \text{ K}^{10}$. All of the samples exhibited a small positive susceptibility that increased with decreasing temperature, consistent with the presence of small amounts of paramagnetic impurities. The discussed T_c values were determined by the beginning of the downturn of $M(T)$, where the diamagnetic superconducting contribution starts to overcome paramagnetic contribution, making this temperature the lower limit of the onset of superconductivity. Further oxidation of aluminium nanoparticles by annealing for 2 hours at 600 °C resulted in a T_c less than 1.7 K, our lowest achievable temperature. Based on the reflectivity step near 11 μm (see Fig. 5(b)), the volume fraction of Al₂O₃ in this sample may be estimated as ~50%, which corresponds to $(r_s - r_c) \sim 0.26r_c$. For $r_c \sim 9 \text{ nm}$, the corresponding thickness of Al₂O₃ was $(r_s - r_c) \sim 2.4 \text{ nm}$.

Thus, the theoretical prediction of a large increase of T_c in ENZ core-shell metamaterials² has been confirmed by direct measurements of $\epsilon_{eff}(0, \omega)$ of the fabricated metamaterials and the corresponding measurements of the increase of T_c . These results strongly suggest that increased aluminium T_c 's that were previously observed in very thin (<50 nm thickness) granular aluminium films^{12,13} and disappeared at larger film thicknesses were due to changes in the dielectric response function rather than quantum size effects and soft surface phonon modes¹³. As clearly demonstrated by our experimental data and the discussion above, the individual Al nanoparticle size is practically unaffected by oxidation, thus excluding the size effects as an explanation of giant T_c increase in our bulk core-shell metamaterial samples. The developed technology enables efficient nanofabrication of bulk aluminium-based metamaterial superconductors with a T_c that is three times that of pure aluminium and with virtually unlimited shapes and dimensions. These results open up numerous new possibilities of considerable T_c increase in other simple superconductors.

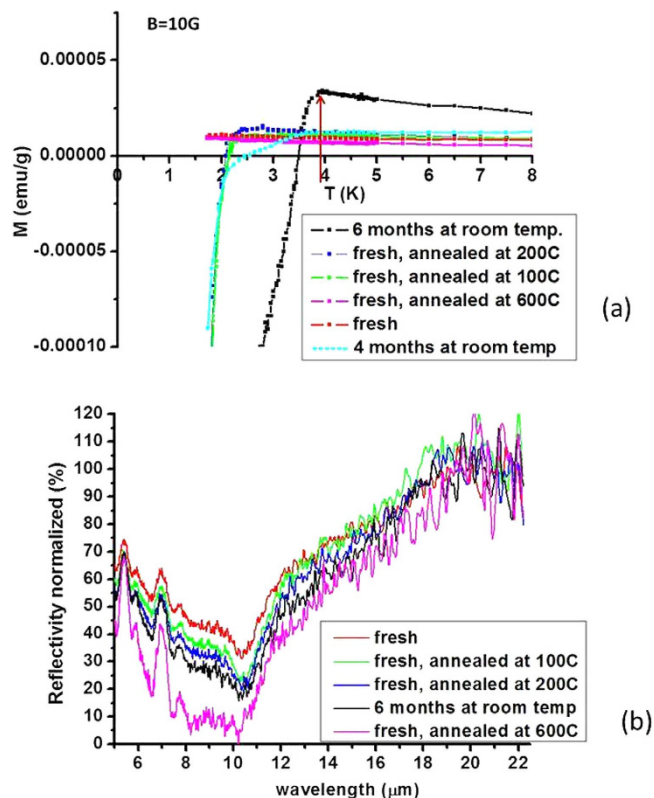


Figure 5. (a) Temperature dependence of zero field cooled magnetization per unit mass for several Al-Al₂O₃ core-shell metamaterial samples with increasing degree of oxidation measured in magnetic field of 10 G. The highest onset of superconductivity at ~3.9 K is marked by an arrow. This temperature is 3.25 times larger than $T_c = 1.2$ K of bulk aluminium. (b) Corresponding FTIR reflectivity spectra of the core-shell metamaterial samples. Decrease in reflectivity corresponds to decrease of the volume fraction of aluminium.

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Author Contributions

I.S., V.S. and M.O. wrote the main manuscript text, V.S., K.Z., T.G. and C.J. fabricated samples, V.S., J.P., M.O. and I.S. collected experimental data. All authors reviewed the manuscript.

Additional Information

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