Vacuum-Deposited Poly(*o*-phenylenediamine)/WO₃•nH₂O Nanocomposite Thin Film for NO₂ Gas Sensor

By Ashutosh TIWARI^{1,†,*} and Songjun LI^{2,†}

The vacuum-deposited thin film of hydrated tungsten oxide (WO₃•nH₂O) embedded poly(*o*-phenylenediamine) (PoPD/WO₃•nH₂O) nanocomposite was fabricated on an indium tin oxide (ITO) coated glass surface for potential NO₂ gas sensor application. The resulting PoPD/WO₃•nH₂O/ITO thin film was characterized using ultraviolet-visible spectroscopy (UV-vis), Fourier transform infrared spectroscopy (FT-IR), X-ray diffractometry (XRD), and electron microscopy (SEM). The composite thin film exhibited a crystalline surface morphology containing nanocrystalls of WO₃•nH₂O with a diameter ranging from 5 to 10 nm. The PoPD/WO₃•nH₂O/ITO film allowed for the low potential detection of NO₂ gas at concentration range from 0 to 1800 ppm. The NO₂ gas sensing characteristics were studied by measuring change in the current with respect to concentration and time. The current of the PoPD/WO₃•nH₂O/ITO film was linearly increased with an increase in concentration of NO₂ gas with a response of ~9 s.

KEY WORDS: Poly(*o*-phenylenediamine) / Hydrated Tungsten Oxide / Vacuum-Deposition / Thin Film / Nanocomposite / NO₂ Gas Sensor /

 NO_2 gas is a common air pollutant and its detection from the combustion environment is an obvious requirement. The gas sensors based on metal oxides are widely used for the detection of gases.^{1–3} Among them, tungsten oxide (WO₃) is one of the most attractive technological materials for gas sensors because of its distinctive electrochromic and catalytic properties.^{4–7} Recently, different types of potentiometric and amperometric gas sensors have been developed using WO₃ in the various forms such as sintered block, thick film and thin film as sensing probes.^{6–9} However, these sensors could provide precise data only in a limited concentration range. Also, they are expensive, large in size, and cannot operate at room temperature. With the advent of nanotechnology, nanostructured materials with novel characteristics provide new opportunities to address these challenges.

Gas sensors based on nanostructured materials have attracted much attention because of their increased sensitivity due to the high surface-to-volume ratios.¹⁰ Recent development of polymer-metal nanocomposites, with every imaginable combination of physical and chemical characteristics, has led to the fabrication of efficient gas sensors that can be used for a wide range of sensor applications.^{11,12} Polymer nanocomposites are easy to prepare and possess high sensing efficiency with a long shelf-life. Therefore in this work, we explore a novel material based on poly(*o*-phenylenediamine) (PoPD/WO₃•nH₂O) nanocomposite for NO₂ gas sensor application.

Poly(*o*-phenylenediamine), PoPD is a polyaniline derivative which can be realized *via* substitution of an amino group in an aniline nucleus. The oxidative polymer of *o*-phenylenediamine has apparently shown different characteristics of molecular structure and properties in comparison with polyaniline. It is widely used as electrochemical reduction catalyst, anticorrosion coating agent, sensors, etc.^{13,14} Poly(*o*-phenylenediamine) is a resinous polymer with amine (-NH₂) groups, which can be chosen as the orientation directing matrix for WO₃•nH₂O because of large quantities of amino groups on the PoPD units, which have a strong binding ability to inorganic materials.¹⁵ Moreover, in acidic condition, PoPD showed a cationic nature that may provide a template core environment to prepare quantum size composite materials. Nanosize WO₃ has gained considerable attention due to size-dependent electrical, chemical, and optical properties.¹⁶ Thus, it should be interesting to synthesize PoPD/hydrated tungsten oxide (WO₃•nH₂O) nanocomposite with employing cationic polymer PoPD.

In this study, the successful attempts have been made towards the fabrication of an efficient NO₂ gas sensor based on vacuum-deposited PoPD/WO₃•nH₂O nanocomposite thin film. The benefits of this technology contain self-assembly, nanoparticles, and high surface-to-volume ratio of sensing probe. The advantageous features of the present NO₂ gas sensor include low-cost, high selectivity, wide range of detection limit, and also sensor responses within ~9 s.

EXPERIMENTAL

Materials

o-Phenylenediamine (Alfa Aesar, 98%), sodium tungstate (Aldrich, 99.9%) and ammonium persulphate (Sigma, 99%) were used without further purification. All supplementary chemicals were of analytical grades. Indium-tin-oxide (ITO)

¹Division of Engineering Materials, National Physical Laboratory, Dr. K. S. Krishnan Marg, New Delhi 110012, India

²Key Laboratory of Pesticide & Chemical Biology of Ministry of Education, College of Chemistry, Central China Normal University, Wuhan 430079, China [†]Present Address: Department of Mechanical Engineering, University of Wisconsin-Milwaukee, Milwaukee-53211, Wisconsin, USA

^{*}To whom correspondence should be addressed (Tel: (+91) 11-32507819, Fax: (+91) 11-25726938, E-mail: tiwaria@mail.nplindia.org, and ashunpl@gmail.com (A. Tiwari) and Lsjchem@yahoo.com.cn (S. Li)).

coated Balzers glass sheets of resistance $15 \Omega/cm$ were used as substrates for the vacuum deposition of nanocomposite materials. The NO₂ gas standards were prepared by mixing two parts of NO gas (Zhuo Zheng gas limited company, 99.9%) with one part of oxygen (De-Luxe, 99%) in the gas bottles.¹⁷ The gas bottles were held for two weeks to allow any residual oxygen to react with the blended NO.

Synthesis of PoPD

PoPD was prepared by the polymerization reaction of *o*phenylenediamine using ammonium persulphate as initiator in the 5N HCl. *o*-Phenylenediamine (3 g, 0.278 moles) was mixed with 100 mL of 5N HCl under continuous stirring at 25 °C. Then, the *o*-phenylenediamine-HCl solution was taken into a three necked flask and the required amount of ammonium persulphate (0.5 g, 0.0219 moles) was continuously added drop wise to it over a period of 30 min. The resulting solution was further stirred for another 4 h. Then, the obtained orange yellow product was washed with 5N HCl. Finally, the product was dried under vacuum.

Synthesis of PoPD/WO3 • nH2O Nanocomposite

PoPD/WO₃•nH₂O nanocomposite was prepared *via in situ* polymerization of *o*-phenylenediamine in the presence of 5N HCl and sodium tungstate. *o*-Phenylenediamine (3 g, 0.278 moles) was mixed with 100 mL of 5N HCl under continuous stirring at 25 °C. Then, the ammonium persulphate (0.5 g, 0.0219 moles) was added drop wise to *o*-phenylenediamine-HCl solution over a period of 30 min. Next, 10 wt % of sodium tungstate was mixed in the above solution and was further stirred for 4 h at 25 °C. Then, the obtained dark yellow product was washed with 5N HCl. Finally, the product PoPD/WO₃•nH₂O nanocomposite was dried under vacuum.

Fabrication of Vacuum-Deposited PoPD/WO₃ • nH₂O Composite Thin Film

A rectangular pressed pallet with 12 mm diameter and 1.5 mm thickness of PoPD/WO₃•nH₂O nanocomposite was prepared using hydraulic pressure machine at 6 tons/cm² pressure. The pellet was used as a source of PoPD/WO₃•nH₂O for the preparation of thin film using vacuum thermal evaporation method at $\sim 10^{-5}$ torr pressure on the ITO coated glass substrates. Before making films, all the substrates were cleaned by successive washing with distilled water, trichloro-ethylene, and methanol under ultrasonic condition. The films were deposited under isothermal conditions and their thickness was measured using quartz thickness monitor.

Characterization

The adsorption spectra were recorded using an UV-visible spectrophotometer (Model 160 A, Shimadzu). FT-IR spectra were collected on a Perkin-Elmer (Spectrum BX II) spectrometer. X-Ray diffractogram was taken by Rigaku Rotaflex, RAD/Max-B, Rigaku Corporation, Japan instrument with a scanning speed of 1°/min. The surface morphology of the PoPD/ITO and PoPD/WO₃•nH₂O/ITO films was examined

using a LEO-440 scanning electron microscope (SEM) operated at 10 kV. The films were sputter-coated with a thin layer of gold ($\sim 20 \text{ nm}$) prior to the morphological examination.

Detection of NO₂ Gas

The sensors were placed in a sealed Plexiglas chamber of 60 cm³ volume. Argon was used as the carrier gas throughout the work. For humidity sensing, argon was bubbled through a bottle containing water and then mixed with dry argon. Electrical contacts were made to the sensors by connecting two short platinum wires to the electrodes using silver paste. The NO₂ gas was injected in the sensing chamber by an automatic gas-tight syringe. The total flow rate was constant at 1000 sccm, and the partial pressures of the test gases controlled with a mass flow controller. During the gas sensing study, NO₂ gas of desirable concentration was simultaneously supplied and unused NO₂ gas was taken out from chamber as exhaust gas. The changes in the current of PoPD/WO₃•nH₂O/ITO thin films on the exposure of NO₂ gas with respect to concentration and time were measured using a Keithley 2700 source meter having 181 nano-voltammeter and 195A digital multi-meter. The current passed through the PoPD/WO3 • nH2O thin film probes was quantitatively measured as electrical conductivity with respect to the varying concentration of NO2 gas ranging from 0 to 1800 ppm at 0.35 V. All experiments were conducted at room temperature (~ 25 °C), with the same sensor used for each set of measurements. Prior to each measurement the sensors were heated in vacuum at 100 °C for 1 h to remove the chemisorbed molecules.

RESULTS AND DISCUSSION

Synthesis of PoPD and PoPD/WO₃ • nH₂O Nanocomposite

The PoPD was synthesized by oxidative polymerization of o-phenylenediamine using ammonium persulphate as oxidizing agent.¹⁸ The ammonium persulphate was added drop-wise because the oxidation of o-phenylenediamine by ammonium persulphate is very sensitive to the reaction conditions. Oxidative polymerization of o-phenylenediamine occurs at the amine groups then a ladder PoPD polymer should be obtained. The oxidation reported here was done using $(NH_4)_2S_2O_8$ in the acidic condition, typically as the oxidation of aniline to polyaniline.¹⁹ For the polymer synthesis, a substoichiometric amount of oxidant was used. For the polymerization a sufficient amount of oxidizing agent was used to potentially allow a ladder PoPD polymer yielding. The reaction scheme for the synthesis of PoPD is shown in Figure 1A. During the polymerization reaction, the first drop of persulphate reacts with o-phenylenediamine-hydrochloride to give the intermediate products.

Thereafter, the formed intermediates react with excess *o*-phenylenediamine-hydrochloride in the reaction mixture to obtain PoPD. In the other hand, hydrolysis of Na₂WO₄ in HCl is known to be complex, giving different products or intermediates depending on the pH of the reaction medium.²⁰ In general, Na₂WO₄ hydrolyses to form WO₃•nH₂O. There-





Figure 1. Synthesis of the (A) PoPD and (B) PoPD/WO3 • nH2O nanocomposite.

fore, during the preparation of PoPD/WO₃ \cdot nH₂O nanocomposite, WO₃ \cdot nH₂O was formed, when blending the Na₂WO₄ with PoPD in presence of HCl. In the nanocomposite, WO₃ \cdot nH₂O will probably be embedded in the PoPD matrix through metal coordination and the hydrogen bonding as shown in Figure 1B.

CHARACTERIZATION

UV-visible Spectroscopy

UV-visible absorption spectra of PoPD and PoPD/WO₃•nH₂O nanocomposite film is shown in Figure 2. In the spectra of PoPD (Figure 2A), an absorption peaks at about 259 was observed due to the π - π * transition in benzeneoid ring which relates to the extent of conjugation between adjacent phenyl rings in the polymeric chain.

Moreover, a broad absorption band at 474 nm in PoPD was found due to the polarone state.²¹ In the spectra of PoPD/ WO₃•nH₂O nanocomposite (Figure 2B), a strong absorption peak at 278 nm was observed due to nanocrystalline WO₃. A broad absorption peak appeared in the region of 461 to 490 nm is due to the polarone state in the nanocomposite. Furthermore, the absorption peak at 258 nm becomes sharp in nanocomposite thin film. This observation may be due to the orientation and interaction of WO₃•nH₂O nanoparticles with PoPD.

FT-IR Spectroscopy

FT-IR spectrum of PoPD and PoPD/WO₃•nH₂O nanocomposite is shown in Figure 3. In the spectrum of PoPD (Figure 3A), the bands at 3336 and 3288 cm^{-1} are due to the characteristic N-H stretching vibrations of the -NH- group at



Figure 2. UV-visible absorption spectra of the (A) PoPD and (B) PoPD/ WO₃ • nH₂O nanocomposite thin films.

two different plan. The broad peak at around 3178 cm^{-1} can be assigned to the N-H stretching vibration of the -NH₂ group.

The two bands at around 1535 and 1659 cm⁻¹ are assigned to the stretching vibration of the C=C and C=N in the phenazine ring, respectively. The presence of the bands appearing at 752 and 608 cm⁻¹ is characteristic of the C-H out-of-plane bending vibrations of benzene nuclei in the phenazine skeleton. The peaks at 1230 and 1362 cm⁻¹ are associated with the C-N stretching in the benzenoid and quinoid imine units, respectively.²² Whereas, in the spectrum of PoPD/WO₃•nH₂O nanocomposite (Figure 3B), the absorption bands at 929, 852 and 806 cm⁻¹ was observed in addition



Wavenumber (cm¹)

Figure 3. FT-IR spectra of the (A) PoPD and (B) PoPD/WO_3 \cdot nH_2O nanocomposite thin films.



Polymer Journal

Figure 4. XRD spectra of the (A) PoPD and (B) PoPD/WO₃ • nH₂O nanocomposite thin films.



Figure 5. SEM image of the (A) PoPD, (B) drop casted PoPD/WO₃ • nH₂O nanocomposite, and (C) vacuum deposited PoPD/WO₃ • nH₂O nanocomposite thin films.

to the characteristic bands of PoPD. The bands at 929 cm^{-1} and 852 cm^{-1} are assigned to W-O stretching and at 806 cm^{-1} is due to W-O bending between tungsten and terminal oxygen. This observation clearly showed that formation of PoPD/WO₃•nH₂O nanocomposite.

X-Ray Diffraction

Figure 4 shows the XRD pattern of PoPD and PoPD/WO₃•nH₂O nanocomposite thin film. In the XRD pattern of PoPD (Figure 4A), the broad hump at $2\theta = 20^{\circ}$ suggests the amorphous nature of the PoPD polymer.

However, XRD pattern of PoPD/WO₃•nH₂O nanocomposite thin film (Figure 4B) shows the sharp reflection at $2\theta = 19$ (020), 23 (002), 28 (111), 32 (112), 34 (131), 39 (211), 48 (004), and 54° (222). These peaks correspond to hydrated WO₃•nH₂O nanoparticles that are present in the PoPD/ WO₃•nH₂O nanocomposite. The crystallite size of WO₃•nH₂O inside the PoPD/WO₃•nH₂O nanocomposite was calculated using the Scherrer equation²³ and was found to be in the range of ~8 nm.

SEM Study

The surface morphology of the PoPD, drop casted PoPD/WO₃•nH₂O, and vacuum deposited PoPD/WO₃•nH₂O nanocomposite thin films was examined by SEM. As shown in Figure 5A, the PoPD thin film was found to be porous weaved surface. The SEM image of drop casted PoPD/WO₃•nH₂O nanocomposite thin film (Figure 5B) shows the formation of rough microstructure having uniformly distributed cubic crystalline WO₃•nH₂O nanoparticles in the polymer.

It was observed that the cubic sphere of WO3·nH2O nanoparticles (crystallite size \sim 35–50 nm) consisted of a solid cubic shell that allowed polymer matrix out of the cubic structure. However, microstructure of vacuum deposited PoPD/WO₃•nH₂O nanocomposite thin film (Figure 5C) illustrates a homogeneous distribution of tiny crystalline WO₃•nH₂O nanoparticles in the porous PoPD polymer matrix. The estimated value of crystallite size was found to be about 5-10 nm, which is probably due to the formation of clusters of WO₃•nH₂O nanoparticles in the thin film. In this context, the vacuum deposited PoPD/WO3•nH2O nanocomposite film has a unique surface structure and can provide a high surface-tovolume ratio to NO₂ gas adsorption over the thin film, while drop casted PoPD/WO₃•nH₂O nanocomposite film has comparatively poor capability for the same. Moreover, during the vacuum deposition of the PoPD/WO3·nH2O nanocomposite film, the incorporation of WO₃•nH₂O nanoparticles into the PoPD polymeric network introduces porosity that may create the diffusion channels inside the polymeric networks. The uniformly distributed WO3 • nH2O nanoparticles along with the porosity in PoPD matrix are expected to be advantageous for gas sensor.

Gas Sensing Behavior

The NO₂ gas sensing behaviors of vacuum deposited PoPD/ ITO and PoPD/WO₃•nH₂O nanocomposite films were evaluated by the reported method.²⁴ The change in the current of the sensing probes (*c.f.*, PoPD/ITO and PoPD/WO₃•nH₂O/ ITO) as a function of time and NO₂ concentration was recorded at room temperature. The sensitivities of the sensors were estimated from the measured value of current in air and in presence of NO₂ gas. The graphs of changes in current of the PoPD/ITO and PoPD/WO₃•nH₂O/ITO probes as a function of time are shown in Figure 6.

With exposure of NO₂ gas (1200 ppm) to probes, the sudden increase in current was observed in the both case. The PoPD/ ITO film has response time of ~ 30 s but recovery time was almost double *i.e.*, about 22 s to PoPD/WO₃•nH₂O/ITO probe (Figure 6A). This observation may be due to the unique surface structure and electronic behavior of the vacuum deposited PoPD/WO₃•nH₂O nanocomposite film, can have ability to provide a high surface-to-volume ratio to NO₂ gas adsorption in a limited time period. Moreover, easy recovery of the vacuum deposition of the PoPD/WO3•nH2O nanocomposite sensor (Figure 6B) was responsible due the incorporation of WO₃•nH₂O nanoparticles into the PoPD polymeric network that introduced the porosity which probably created the good diffusion channels to NO2 gas inside the PoPD polymer/ WO₃•nH₂O nanocomposite networks. The uniformly distributed WO₃•nH₂O nanoparticles along with the porous PoPD matrix are expected to be a sharp and specific NO2 gas sensing response.

Moreover, the original current was achieved, when the NO_2 gas had been turned off and fresh air was introduced into the test chamber. The insert of fresh air in the test chamber remove the adsorb gas molecules from the probe surface and regain the



Figure 6. The current response of the (A) PoPD/ITO and (B) PoPD/ WO₃ • nH₂O/ITO nanocomposite probes, with respect to time ranging from 0 to 63 s on the exposure of 1200 ppm NO₂ gas at 0.35 V.



Figure 7. The current response of the (A) PoPD/ITO and (B) PoPD/ WO₃ • nH₂O/ITO nanocomposite probes on the exposure of 0 to 1800 ppm NO₂ gas at 0.35 V.

original current. The observed changes in the current of both the probes (*i.e.*, PoPD/ITO and PoPD/WO₃ \cdot nH₂O/ITO) are due to physical adsorption of NO₂ gas on the surface of respective films.

Figure 7 shows the variation of current versus NO₂ gas concentration profile of PoPD/ITO and PoPD/WO₃•nH₂O/ ITO probes. The current increases with an increase in NO₂ gas concentration ranging from 0 to 1800 ppm in case of PoPD/ WO₃•nH₂O/ITO probe but PoPD/ITO probe showed linearity with NO₂ gas concentration ranging from 0 to 1500 ppm. In general, the interaction of NO₂ gas with the π -electron network of PoPD capture the electron from the polymer and decrease the resistance of probe from 0 to 1500 ppm of NO₂ gas (Figure 7A). While, WO₃•nH₂O is an n-type semiconductor; they will create a more space charge region at the WO₃•nH₂O/ITO probe. During NO₂ gas adsorption, a high conductivity and sensing range of sensing probe was achieved in the PoPD/ WO₃•nH₂O/ITO probe due to reduction of space charge region in the PoPD/WO₃•nH₂O nanocomposite matrix (Figure 7B). The modulation of space charge region at the interface of WO₃•nH₂O-PoPD gives a high range of sensitivity for NO₂ gas and hence PoPD/WO₃•nH₂O/ITO probe can operate at room temperature. Also, in this study, a relatively fast response (~9 s) and recovery (~10 s) time was observed for NO₂ gas using PoPD/WO₃•nH₂O/ITO sensor. The observed fast response time of PoPD/WO₃•nH₂O thin film to NO₂ gas is well below, when compared to the other NO₂ gas sensors reported in the literature.²⁵⁻²⁸

Sensing Mechanism

In earlier studies, NO₂ was reported to have large effects on the conductivity of polymer films.^{29,30} It is interpreted that a charge transfer complex is formed between the PoPD/ WO₃•nH₂O film donor and NO₂ acceptor, and the charge carriers are the holes produced in the polymeric matrix. NO₂ is π -electron acceptor, and accepted electron would delocalize over the NO₂ planar structure. The high selectivity towards the NO₂ gas may be explained on the basis of charge transfer complex formed between the PoPD/WO₃•nH₂O film donor and NO₂ gas acceptor molecules to cause fluctuation in terms of current.^{31,32} These studies supported that the PoPD/ WO₃•nH₂O nanocomposite thin film has ability to sense NO₂ gas specifically.

Interference and Storage Stability

The effect of interferents (NH₃, NO and CO₂) was studied on the current responses of the PoPD/WO₃ \cdot nH₂O/ITO probe (Table I). These three substances were added into the gas sensing chamber at their normal physical concentration, *i.e.*, NH₃ (500 ppm); NO (500 ppm); and CO₂ (1000 ppm).

It was found that the presence of interferents had a negligible effects on the current response obtained; therefore, using this probe, NO₂ can be detected without any interference. Table II compares the characteristics of conducting polymer based NO₂ gas sensors as reported in the literature. From this table it is clearly observed that PoPD/WO₃•nH₂O/ITO probe exhibited a longer shelf life, higher selectivity, and shorter response time in a wider range.

The lifetime of the PoPD/WO₃ \cdot nH₂O/ITO probe was determined by measuring the current up to ~5 months at an interval of seven days. It was observed that the current decreased slowly with the increase in time period. After 5 months, the decrease in current was found to be about 32%.

CONCLUSIONS

 $PoPD/WO_3 \cdot nH_2O$ nanocomposite was synthesized by self assembled *in situ* polymerization reaction. $PoPD/WO_3 \cdot nH_2O$ nanocomposite thin film was successfully fabricated on ITO coated glass substrate using vacuum deposition method as efficient sensors for detection of NO_2 gas. The $PoPD/WO_3 \cdot nH_2O/ITO$ probe can be operated at room temperature for sensing NO_2 gas in the wide concentration rage. This is an added advantage of $PoPD/WO_3 \cdot nH_2O$ nanocomposite thin

Table I.	Effect of interference on the PoPD/WO3 • nH2O/ITO NO2				
gas sensor at 0.35 V					

SI. No.	Analyte/Interference	Current (mA)
1.	NO ₂ (1200 ppm)	278
2.	NO ₂ (1200 ppm) + NH ₃ (500 ppm)	286
3.	NO ₂ (1200 ppm) + NO (500 ppm)	282
4.	NO_2 (1200 ppm) + CO_2 (1000 ppm)	272

 Table II.
 Comparison between different materials tested for the NO2 gas sensors

Sensing probe	Linearity with [NO ₂], ppm	Response time, Seconds	Shelf- life, Months	Reference
PoPD/WO3•nH2O	0–1800	~9	~5	Present
nanocomposite thin film				work
PANI/polystyrenesulfonic acid	20–100	10	—	22
composite film				
PANI-SnO ₂ composite film	10-800	1000	—	12
PANI nanofibers	10–200	100	~4	23
Au-doped WO ₃	5–20	500	~3	24
Mesoporous WO3 thin films	3–100	60–120	—	25
Scandia-stabilised zirconia	10–500	8	—	26
(8 mol % Sc ₂ O ₃ -ZrO ₂) solid				
electrolyte and CuO + CuCr ₂ O ₄				
mixed-oxide				

film when compared to other types of sensors reported for sensing NO₂ gas. The response time and recovery time of PoPD/WO₃•nH₂O sensor was found to be ~9 s and ~10 s, respectively, which shows that the PoPD/WO₃•nH₂O sensor could be reused more frequently that extends the shelf life of the sensor.

Acknowledgment. Authors are thankful to the Department of Science and Technology, Govt. of India and to the National Science Foundation of China (Grant No. 20603010) for generous financial supports and to the Director, National Physical Laboratory, New Delhi, India for providing infrastructure facilities to carry out this work.

> Received: March 13, 2009 Accepted: May 25, 2009 Published: July 9, 2009

REFERENCES

- 1. E. Comini, Anal. Chim. Acta, 568, 28 (2006).
- 2. O. K. Tan, W. Cao, Y. Hu, and W. Zhu, Ceram. Int., 30, 1127 (2004).
- O. K. Tan, W. Caob, Y. Hua, and W. Zhu, Solid State Ionics, 172, 309 (2004).
- X. Wang, N. Miura, and N. Yamazoe, Sens. Actuators, B, 66, 74 (2000).
- C. J. Jin, T. Yamazaki, Y. Shirai, T. Yoshizawa, T. Kikuta, N. Nakatani, and H. Takeda, *Thin Solid Films*, 474, 255 (2005).
- J. L. Solis, S. Saukko, L. B. Kish, C. G. Graqvist, and V. Lantto, Sens. Actuators, B, 77, 316 (2001).
- I. Jimenez, J. Arbiol, G. Dezanneau, A. Cornet, and J. R. Morante, Sens. Actuators, B, 93, 475 (2003).

Polymer Journal

- M. Stankova, X. Vilanova, E. Llobet, J. Calderer, M. Vinaixa, I. Gracia, C. Canec, and X. Correig, *Thin Solid Films*, 500, 302 (2006).
- 9. K. Galatsis, X. X. Li, W. Wlodarski, and K. Kalantar-Zadel, Sens. Actuators, B, 87, 478 (2001).
- 10. A. Tiwari and S. Gong, *Electroanalysis*, 20, 1775 (2008).
- 11. M. K. Ram, O. Yavuz, V. Lahsangahand, and M. Aldissi, Sens. Actuators, B, 156, 750 (2005).
- 12. M. K. Ram, O. Yavuz, and M. Aldissi, Synth. Met., 151, 77 (2005).
- T. Ohsaka, T. Watanabe, F. Kitamura, N. Oyama, and K. Tokuda, *Chem. Commun.*, 16, 1072 (1991).
- C. Malitesta, F. Palmisano, L. Torsi, and P. G. Zambonin, *Anal. Chem.*, **62**, 2735 (1990).
- P. A. Kumar, M. Ray, and S. Chakraborty, J. Hazard. Mater., 143, 24 (2007).
- W. Morales, M. Cason, O. Aina, N. R. deTacconi, and K. Rajeshwar, J. Am. Chem. Soc., 130, 6318 (2008).
- 17. S. Vaughan, Gases & Instrumentation, 1, 26 (2007).
- 18. J. Han, G. Song, and R. Guo, Eur. Polym. J., 43, 4229 (2007).
- 19. A. Tiwari and V. Singh, Carbohydr. Polym., 74, 427 (2008).
- E. F. Sperenskaya, V. E. Mertsalova, and I. I. Kulev, *Russ. Chem. Rev.*, 35, 892 (1966).
- 21. A. Tiwari, V. Sen, S. R. Dhakate, A. P. Mishra, and V. Singh, Polym.

Adv. Technol., 19, 909 (2008).

- 22. A. Tiwari, J. Macromol. Sci., Part A: Pure Appl. Chem., 44, 735 (2007).
- H. P. Klug and L. E. Alexander, in "X-Ray Diffraction Procedures for Polycrystalline and Amorphous Materials," 2nd ed., Wiley Inter-Science, 1974, Chap. 9.
- 24. D. Xie, Y. Jiang, W. Pan, D. Li, Z. Wu, and Y. Li, *Sens. Actuators, B*, **81**, 158 (2002).
- X. B. Yan, Z. J. Hana, Y. Yanga, and B. K. Taya, *Sens. Actuators, B*, 123, 107 (2007).
- H. Xia, Y. Wang, F. Kong, S. Wang, B. Zhu, X. Guo, J. Zhang, Y. Wang, and S. Wu, *Sens. Actuators, B*, **134**, 133 (2008).
- L. G. Teoh, Y. M. Hon, J. Shieh, W. H. Lai, and M. H. Hon, Sens. Actuators, B, 96, 219 (2003).
- 28. W. Xiong and G. M. Kale, Sens. Actuators, B, 114, 101 (2006).
- 29. R. L. van Ewyk, A. V. Chadwick, and J. D. Wright, J. Chem. Soc., Faraday Trans 1, 76, 2194 (1980).
- S. Radhakrishnan and S. G. Joshi, J. Polym. Sci., Part C: Polym. Lett., 27, 127 (1989).
- 31. A. Tiwari, J. Polym. Res., 15, 337 (2008).
- T. G. J. V. Oirschot, D. V. Leeuwen, and J. Medema, *J. Electroanal. Chem.*, **37**, 373 (1972).