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ORIGINAL ARTICLE

Bragg-mirror-like circular dichroism in bio-inspired quadruple-gyroid 4srs nanostructures

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The smooth and tailorable spectral response of Bragg mirrors has driven their pervasive use in optical systems requiring customizable spectral control of beam propagation. However, the simple nature of Bragg mirror reflection prevents their application to the control of important polarization states such as circular polarization. While helical and gyroid-based nanostructures exhibiting circular dichroism have been developed extensively to address this limitation, they are often restricted by the spectral inconsistency of their optical response. Here we present the fabrication and characterization of quadruple-gyroid 4srs nanostructures exhibiting bio-inspired Bragg-mirror-like circular dichroism: a smooth and uniform band of circular dichroism reminiscent of the spectrum of a simple multilayer Bragg-mirror. Furthermore, we demonstrate that the circular dichroism produced by 4srs nanostructures are robust to changes in incident angle and beam collimation, providing a new platform to create and engineer circular dichroism for functional circular polarization manipulation.

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INTRODUCTION

Bragg mirrors are essential building blocks of many optical systems due to their ability to achieve near-total reflection of light across a precisely defined band of wavelengths. Formed by the constructive interference of successive Fresnel reflections from within a simple multilayer dielectric stack^{1,2}, the reflection spectra of Bragg mirrors at normal incidence are characterized by a smooth, uniform and strong band of unpolarized reflection surrounded by high transmission³. Through these tailorable reflection properties, Bragg mirrors have found use in numerous applications requiring precise spectral control of beam propagation such as dielectric mirrors, beamsplitters and filters⁴, fibre optic sensors and lasers⁵, and in many other distributed feedback laser systems⁶.

While the prevalence of devices facilitated by the reflection from Bragg mirrors is noteworthy, a growing demand to manipulate specific polarization states^{7,8} has driven a focus on nanostructures capable of normal incidence interaction with a particular polarization. Specifically, a demand for control of spin angular momentum carrying circular polarization^{8–10} has led to the development of chiral nanostructures such as helical spiral arrays^{11–19} and gyroids^{20–23} that exhibit circular dichroism: a differential reflection, transmission or absorption between left circularly polarized (LCP) and right circularly polarized (RCP) light. However, even though these structures can exhibit circular dichroism, their spectral behaviour is often dominated by sharp and narrow features^{13–21}, unpolarized reflection bands^{16–18,22},

and fluctuations in the sign of the circular dichroism^{17–22} when a Bragg-mirror-like response, defined here as a spectral response comparable to the simple, clean and smooth spectrum of a Bragg mirror, would be preferred.

In this article we demonstrate the presence of bio-inspired Bragg-mirror-like (Bragg-like for short) circular dichroism in quadruple-gyroid 4srs nanostructures that transmit only a single handedness of circular polarization. Taking inspiration from the strong but unpolarized Bragg-like reflections produced by the aligned single-gyroid 1srs-based wing-scales of the butterfly *Parides sesostris*, we utilize the laser direct write technique to create and align new 4srs nanostructures that exhibit smooth and strong circularly polarized transmission spectra reminiscent of the spectral response of a simple multilayer Bragg-mirror, and with similarities in its physical origin. We reveal a strong robustness of the circular dichroism band to variations in incident angle and beam collimation, and demonstrate the ability to incorporate defect modes within the circularly polarized stop-band that may enable the development of new chiral photonic devices.

MATERIALS AND METHODS

Crystallographic notation

Standard crystallographic notation with Miller indices is used to describe individual and families of directions and planes. The notations $\langle xyz \rangle$ and [xyz] are used to denote a family of directions equivalent by symmetry, and individual directions, respectively.

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The notation (xyz) is used to denote an individual plane that is orthogonal to the direction [xyz].

Numerical simulations

Calculation of photonic band-structure modes is performed using the open source software package MPB, developed by MIT. The **4srs** and **1srs** nanostructures are constructed within simple cubic (SC) and body centred cubic (BCC) primitive unit cells, respectively, from cylindrical or elliptical line segments terminated by spherical and elliptical caps. Each nano-structure is discretized across the lattice vectors with a grid resolution of 64 pixels per lattice unit, and the dielectric function is averaged over a mesh size of 8 at each grid point. After solution of each frequency eigenvalue (ω), the corresponding magnetic eigenfield ($H_{k,n}^*$) with wave vector k is analysed for its degree of circular dichroism and for its coupling amplitude to a wave of arbitrary polarization^{24,25}. To begin, the coupling amplitude between incident circularly polarized plane waves with wave vector $q \parallel k$ and the magnetic eigenfield in planes P orthogonal to k are computed via the overlap integrals

$$C_{k,n}^{\pm} = \left| \frac{1}{\iint_P dP} \iint_P \frac{1}{\sqrt{2}} (\hat{\mathbf{e}}_1 \pm i\hat{\mathbf{e}}_2) \cdot H_{k,n}^*(P) \right|^2 \tag{1}$$

where $\frac{1}{\sqrt{2}}(\hat{\mathbf{e}}_1 \pm i\hat{\mathbf{e}}_2)$ are the normalized circularly polarized plane waves and $H_{k,n}^*(P)$ is the normalized complex conjugate of the eigenfield in the plane P. The unit vectors \hat{e}_1 and \hat{e}_2 form an orthogonal set with k.

The degree of circular dichroism, *C*, which is termed the circular dichroism index, is given by the definition taken from the literature^{24–27}

$$C_{k,n} = \operatorname{sgn}(q \cdot \nabla_k \omega) \left(\frac{C_{k,n}^+ - C_{k,n}^-}{C_{k,n}^+ + C_{k,n}^-} \right)$$
 (2)

where sgn is the sign function. Similarly, the amplitude of coupling the mode to a wave of arbitrary polarization, β , which is termed the coupling index, is also given by the definition taken from the literature^{24–27}

$$\beta_{k,n} = C_{k,n}^+ + C_{k,n}^- \tag{3}$$

The circular dichroism and coupling indices are useful measures to quantify the nature of individual Bloch modes in terms of circular polarization, providing intuition for the full scattering problem that can only be solved when considering the complex band-structure including evanescent modes²⁸. We consider 64 different planes along k within the magnetic eigenfield, averaging C and β across all the planes to reveal final mean values. The coupling index for each mode in the band diagrams is represented as the size of the plot while the circular dichroism index of each mode in the band-diagram is represented by the colour of the plot. Modes with C < 0.5 are considered LCP and are plotted blue, while modes with C > 0.5 are considered RCP and are plotted red. Modes with $\beta < 0.1$ are considered low-coupling modes.

Numerical simulations of the transmission spectra are performed using the commercial finite element software CST Microwave Studio. The cubic non-primitive unit-cell is used to model the **1srs** nanostructure transmission while the SC primitive unit-cell is used to model the **4srs** nanostructure. The α =5° half apex angle illumination cone surrounding the [001] axis is modelled by a smoothed average of 95 plane wave simulations equally spaced over the cone of illumination.

Thick film deposition

Films of As_2S_3 of $\approx 16~\mu m$ thickness were prepared from stoichiometric As_2S_3 bulk glass using thermal evaporation²⁹. A temperature controlled Ta baffled boat housing the As_2S_3 was held at a temperature of 310 °C in a vacuum chamber pumped to a base pressure of 2×10^{-7} torr. A carousel under planetary rotation at the top of the chamber held 170 μ m thick glass substrates at room temperature and captured the evaporating As_2S_3 at a deposition rate of $\approx 0.6~\text{nm s}^{-1}$. In the as-deposited state, the As_2S_3 films were measured to have an approximate refractive index of 2.35 that increases to 2.43 after annealing³⁰. Raman spectroscopy was used to confirm that the deposited films contained the weakly interconnected As_4S_6 cage molecules that enable photo-polymerization^{29,31}

Laser nano-fabrication

The full details of the adaptive optics enhanced direct laser writing system have been described elsewhere²². An amplified Ti:Sapphire femto-second laser, operating at a central wavelength of 800 nm, a repetition rate of 10 kHz and with a pulse duration of 100 fs is expanded to fill the area of a liquid crystal-based SLM (HSPDM5120785-PCIe, Boulder Nonlinear Systems) that provided phase-only modulation of the laser beam for compensating systeminduced optical aberrations, as well as the strong spherical aberration that is imparted onto the beam when focusing through a glass coverslip and into the high refractive index As₂S₃. At this wavelength, the deposited film is transparent and no photo-polymerization occurs. Tracing the focus along the path of the 4srs nets enables photopolymerization back to As₂S₃ only of the 4srs geometry and at any size and orientation that is desired. The use of adaptive optics in the fabrication system is essential to achieve high-quality nano-fabrication in As₂S₃ films as strong spherical aberration that is imparted onto the focused laser beam by the mismatch in glass and As₂S₃ refractive indices can significantly distort fabricated structures such as gyroids²². The reflected beam from the SLM was imaged to the back aperture of a high numerical aperture (NA = 1.4) objective lens that was used to focus the beam into the As₂S₃. A three-dimensional piezo-electric translation stage and mechanical shutter were used to translate the sample and switch the beam such that the focus of the laser traced out the network segments of the 4srs within the As₂S₃ film. The 4srs nanostructures were typically written at a speed of 10 µm s⁻¹ and with an average laser power between 1.75 and 2.0 µW.

Wet chemical etching

After photo-polymerization, the As₂S₃ thick films were immersed for 3 min in a 2 mol-% ratio of diisopentylamine in dimethylsulfoxide at room temperature and with constant stirring. Immediately after etching, the films were immersed in acetone to cease the etching process.

Transmission spectroscopy

Transmission spectra were measured with a Fourier transform infrared spectrometer coupled to an infrared microscope. The microscope contained two Cassegrain reflection objectives for illumination and collection of transmitted light. Conversion of the broadband unpolarized light from the spectrometer to circular polarization was achieved via the addition of a broadband (2.5–10 μm) BaF2 wire grid linear polarizer and a broadband (2.5–7 μm) MgF2 circular polarizer. The incident beam was limited to a small cone of angles surrounding the [001] axis by placing a small aperture into the path of the objective lens, reducing the NA of the excitation from 0.68 NA to <0.1 NA such that the beam extended over a cone with a half

apex angle of $\alpha = 5^{\circ}$. The measured spectra were corrected for measurement-induced spectral shifts and scattering losses, and smoothed to remove noise.

RESULTS AND DISCUSSION

4srs intergrowth

The 1srs network (net) derives its name from the SrSi₂ crystal³², within which the Si atoms form a single chiral network with cubic symmetry I4₁32. Inflating the **1srs** net into cylindrical segments of finite radius forms a 1srs nanostructure, also known as a single-gyroid, that is well suited to experimental fabrication^{20–22}. A common unit-cell of the **1srs** nanostructure is the truncated octahedron with BCC basis^{24,25,33,34}, shown in Figure 1a. However, the only substantial circular dichroism within dielectric 1srs nanostructures^{24,25} is known to occur along the cubic <100> axes (see Materials and Methods for details of axes notation), often driving the use of a SC unit-cell²⁰⁻²² that aligns the four-screw single helices with the real-space lattice vectors, as shown in Figure 1b. It is the circular dichroism along these four-screw axes²⁴ that enables applications such as chiral beam-splitting²¹, but with performance limited by a far from Bragg-like spectral response that includes sharp and narrow features, unpolarized reflection bands, as well as changes in the sign of circular dichroism^{20–22}.

For optimal control of circular polarization, it is desirable for a nanostructure to exhibit Bragg-like circular dichroism, characterized by simple reflection of an LCP (RCP) incident wave with a spectral response equivalent to that of a Bragg mirror, and simultaneous complete transmission of a RCP (LCP) incident wave. In complex 3D nanostructures, smooth Bragg-like reflections often exist along the primitive reciprocal directions of the lattice. For the 1srs nanostructure, the primitive directions that are parallel to the <110> axes exhibit the broadest and cleanest Bragg-like band-gap (see Results and Discussion), a feature exploited by the butterfly *Parides sesostris*, which aligns the 1srs <110> axes within specific patches of its wing-scales to maximize structural colouration³⁵. The <110> axes of the 1srs nanostructure, however, do not exhibit sensitivity to circular polarization²⁵ and new structures must be investigated if Bragg-like circular dichroism is to be exploited.

The quadruple-gyroid **4srs** net^{24,32,36}, which is comprised of four individual yet nonintersecting **1srs** nets of the same handedness, is a geometry that provides access to circularly polarized Bragg-like reflection bands by combining a SC primitive lattice type with cubic circular dichroism²⁴. Building on the cubic symmetry of the **1srs** net, a **4srs** net is formed via simple half-lattice translations of three duplicate **1srs** nets—all of the same handedness—along each of the [100], [010] and [001] axes (Supplementary Fig. 1), ending in a **4srs** net such as the inflated nanostructure shown in Figure 1c. In this process, the four-screw single helices of the **1srs** are transformed into pairs of four-screw double helices²⁴, and a new SC primitive unit cell is formed with cubic symmetry $P4_232$, shown in Figure 1d. The length of the **4srs** primitive unit cell is half the length of the non-primitive cubic **1srs**

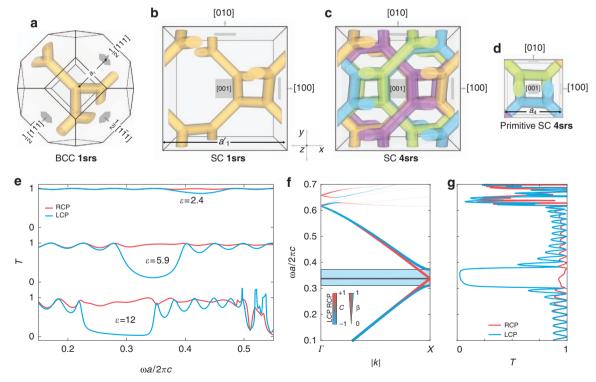


Figure 1 Geometry and numerical characterization of the quadruple-gyroid 4srs nanostructure. (a) A BCC unit cell of a RHD 1srs nanostructure (lattice parameter a_1). (b) A non-primitive cubic unit-cell of a RHD 1srs nanostructure (lattice parameter a_1') showing the four-screw helices along [001]. (c) A non-primitive cubic unit-cell of a RHD 4srs nanostructure made of four identical 1srs nanostructures. (d) A primitive SC unit-cell of a RHD 4srs nanostructure (lattice parameter a_4) centred on the four-screw double helix. The grey patches and arrows indicate the direction of the real-space lattice vectors for each of the unit cells. (e) Simulated transmission spectra along the <100> axes for 4srs nanostructures with a height of six unit cells, a volume fraction of Φ =36% and permittivities of ε =2.4, ε =5.9 and ε =12. (f) Band-diagram (Γ -X) and (g) transmission spectra along the <100> axes for a RHD 4srs nanostructure with ε =5.9, Φ =36% and a height of 12 unit cells. The colour of the modes in Γ represents the mode's circular dichroism index, Γ 0, while the linewidth of the plot indicates the mode's coupling index, Γ 10 (see Materials and Methods). The blue and grey shading indicates the location of the LCP stopband, and narrow band-gap, respectively.

unit cell, even though the structural size of the four individual 1srs nets remain the same, a result of the translation operation pair-wise exchanging the four network domains of the 4srs.

Numerical simulations

The translation operations used to form the 4srs nanostructure from a 1srs nanostructure induce two important transformations in its optical behaviour. First, the transformation of the primitive real-space lattice type from BCC to SC shifts the primitive reciprocal lattice vectors parallel to the <100> axes (from parallel to the <110> axes), ensuring that the Bragg-like reflection often associated with the primitive reciprocal lattice directions (see Results and Discussion) will be in a cubic basis. Second, the transformation of the four-screw single helices of the 1srs to the four-screw double helices of the 4srs enhances sensitivity to circular polarized light²⁴ along the cubically symmetric <100> axes. Figure 1e illustrates the Bragg-like circular dichroism properties that result from the combination of these transformations, with numerically simulated LCP and RCP transmission spectra (see Materials and Methods) along the [100] axis of a right handed (RHD) 4srs nanostructure plotted for the case of three relative **4srs** permittivities of $\varepsilon = 2.4$, $\varepsilon = 5.9$ and $\varepsilon = 12$ in a background of air $(\varepsilon = 1)$, corresponding to the dielectric contrasts produced by common polymer^{16,20,37}, chalcogenide glass^{22,29,31} and semiconductor^{38–40} fabrication techniques, respectively.

To characterize the circular dichroism of each transmission spectrum plotted in Figure 1e, we define the strength of circular dichroism as the differential transmission between LCP and RCP incident waves: $\Delta T = T_{\rm RCP} - T_{\rm LCP}$, and define the width of the circular dichroism band as the gap to mid-gap ratio: $\Delta \omega = 2(\omega_2 - \omega_1)/(\omega_2 + \omega_1)$, where ω_1 and ω_2 are the upper and lower frequency bounds of the dichroism band, respectively. Low-permittivity materials produce only weak ($\Delta T = 11\%$) circular dichroism, limiting their practicality in experimental realization of the **4srs** nanostructure. Materials with larger permittivities such as typical semiconductors allow significantly strong and broad ($\Delta T = 92.5\%$, $\Delta \omega = 40\%$) circular dichroism bands to be opened, but at the cost of significant complexity

in experimental realization^{38–40}. Between these two extremes, chalcogenide glasses such as arsenic trisulfide (As₂S₃), which can be nanostructured in a similar manner to many low-permittivity polymers^{22,29,31}, can also open broad bands of circular dichroism ($\Delta T = 85.9\%$, $\Delta \omega = 22.9\%$) suitable for experimental realization.

As with previous simulations of topologically equivalent constant mean curvature 4srs nanostructures²⁴, the origin of the dichroism band in the cylindrical 4srs nanostructure is a splitting of the LCP and RCP mode frequencies surrounding the narrow band-gap along the <100> axes²⁴. Figure 1f shows the numerically simulated bandstructure (see Materials and Methods) along the <100> axes for a cylindrical RHD As₂S₃ 4srs geometry with a dielectric volume fraction (Φ) of 36%, and is plotted alongside a simulation of the transmission spectra in Figure 1g for incident waves propagating through the same geometry. Each mode in the band-diagram in Figure 1f is analysed for its degree of circular dichroism and coupling amplitude (see Materials and Methods), plotted as the colour and linewidths of the curve, respectively. A clear separation of the first four LCP and RCP modes at the X point is evident, opening a LCP stop-band to create a band of circular dichroism 2.5 times wider than that produced by an equivalent RHD As₂S₃ 1srs nanostructure (Supplementary Fig. 2). A full band-structure analysis of the 4srs nanostructure can be found in Supplementary Fig. 3, highlighting the unique Bragg-like circular dichroism properties of the cubic <100> axes.

Laser nano-fabrication of 4srs nanostructures

Unlike the **1srs** geometry that is found ubiquitously in nature³² and which has been experimentally realized by means of self-assembly^{37,41,42}, bio-templating^{43,44}, laser direct write techniques^{20–22} and 3D printing⁴⁵, no **4srs** nanostructures have been either found or produced, although the occurrence of various triply intergrown geometries^{46–49} raises the hope that a self-assembly strategy for the **4srs** will eventually be found. To experimentally realize the **4srs** nanostructure for the first time, we utilized the laser direct write technique in the chalcogenide glass As_2S_3 to create cylindrical As_2S_3 **4srs** nanostructures in a three-step process as detailed in Figure 2a–2c

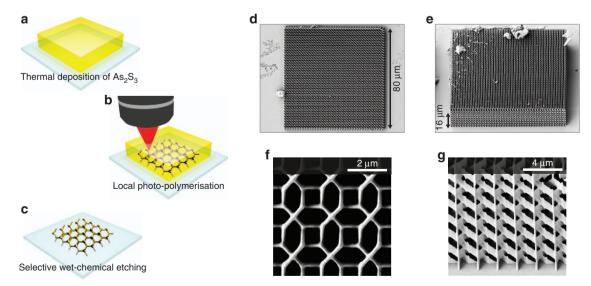


Figure 2 Three-step fabrication process of the As_2S_3 laser direct write technique and fabricated As_2S_3 4srs nanostructures. (a) Thermal deposition of As_2S_3 glass. (b) Local photo-polymerization with a focused femto-second laser beam, and (c) selective wet-chemical etching to reveal a free-standing nanostructure. (d) SEM image of a fabricated RHD 4srs nanostructure along the [001] axis. (e) SEM image of a fabricated RHD 4srs nanostructure along the [011] axis. (f, g) are close-up images of the views in (d) and (e) showing the geometry of the four-screw double helices.

and described in the Materials and Methods section. First, thick films of As_2S_3 were deposited onto glass substrates using a controlled deposition procedure that preferentially deposits the As_2S_3 as weakly interconnected As_4S_6 cage molecules^{29,31}. Second, the as-deposited films were locally and three-dimensionally photo-polymerized back to cross-linked As_2S_3 with an adaptive optics enhanced direct laser writing system²². Third, the films were wet-chemically etched to enable preferential dissolution of the unexposed As_4S_6 , leaving a free-standing As_2S_3 nano-structure with high dielectric contrast to the surrounding air.

Using this technique, **4srs** nanostructures with primitive lattice parameters between a_4 =1.75 and 2 µm were fabricated with linewidths between 150 and 240 nm along the [100] and [010] axes, and between 0.9 and 1.2 µm along the [001] axis, such that volume fractions covering the optimal value of Φ =19% (Supplementary Fig. 4h) were produced. In all cases, the nanostructures were oriented such that the [001] axis was perpendicular to the substrate surface and access to the four-screw double helices was possible.

Scanning electron microscope (SEM) images of fabricated **4srs** nanostructures along the [001] and [011] axes are shown in Figure 2d and 2e, with additional close-up images shown in Figure 2f and 2g showing the characteristic four-screw double helices that separate the band-edge mode frequencies²⁴ and lead to the opening of a circular dichroism band. The angled images along the [011] axis reveal a large degree of elongation of the network linewidth along the [001] axis. This elongation is the result of the elliptical shape of the diffraction limited focus in most laser writing systems, which for the system used here directly transforms the symmetric cylindrical segments of the **4srs**

to elliptic cylindrical segments with an ellipticity of e=5, where e is defined by the ratio of the [001] to [100] lines widths of the elliptical line segments. Such elongation breaks the cubic symmetry of the **4srs** nanostructure, and a variety of methods have been developed in order to restore it^{21,22,31}. However, as long as symmetry is maintained in the plane orthogonal to the [001] axis, the ellipticity will have negligible effect on the circular dichroism bands along the [001] axis (Supplementary Fig. 4) and hence preservation of cubic symmetry is not always necessary.

Transmission spectroscopy of 4srs nanostructures

To probe the circular dichroism of the 4srs nanostructures, Fourier transform infrared spectroscopy (see Materials and methods) was utilized to measure the circularly polarized transmission spectra across the mid-infrared wavelength range. The etched networks were mounted in the spectrometer such that the incident beam propagated parallel to the [001] axis, while the incident angle was limited to a narrow cone with a half apex angle (α) of 5° as shown in the measurement geometry in Figure 3a and 3b. Figure 3c and 3d shows the measured and simulated 4srs transmission spectra, respectively, for both LCP and RCP incident waves traversing a RHD 4srs nanostructure with $a_4 = 2 \mu m$ and lateral and axial linewidths of 235 and 1,175 nm, respectively, corresponding to a volume fraction of Φ = 17.7%. A strong reduction in transmission for the LCP incident beam is observed compared with the case of a RCP incident beam that propagates unaffected by the structure, resulting in the opening of a broad ($\Delta\omega = 18\%$) and strong ($\Delta T = 65\%$) band of Bragg-like circular dichroism. In comparison, the measured and simulated transmission

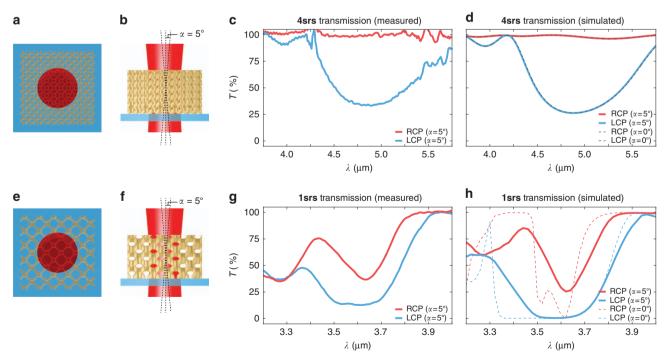


Figure 3 Experimental transmission characterization of the robust 4srs nanostructure circular dichroism and comparison with numerical simulations. (a, b) Transmission measurement geometry for the 4srs nanostructure. (c) Measured transmission spectra for a RHD 4srs nanostructure with a lattice parameter of $a_4 = 2 \, \mu$ m, a height of six unit cells, a lateral linewidth of 235 nm and e = 5. (d) Simulated transmission spectra at normal incidence ($\alpha = 0^{\circ}$) (dotted black lines) and at $\alpha = 5^{\circ}$ (solid coloured lines) for a 4srs geometry equivalent to the experimental measurement. (e, f) Transmission measurement geometry for the 1srs nanostructure. (g) Measured transmission spectra for a RHD 1srs nanostructure with a lattice parameter $a'_1 = 3 \, \mu$ m, a height of four unit cells, a lateral linewidth of 205 nm and e = 5. (h) Simulated transmission spectra at normal incidence ($\alpha = 0^{\circ}$) (dashed thin lines) and at $\alpha = 5^{\circ}$ (solid thick lines) for a 1srs geometry equivalent to the experimental measurement.

spectra for LCP and RCP incident waves traversing the [001] axis (Figure 3e and 3f) of a **1srs** nanostructure (see Supplementary Fig. 5 for SEM images) can be seen in Figure 3g and 3h, respectively, where two considerably narrower bands of circular dichroism with a maximum width of $\Delta\omega$ = 5.6% exist in a significantly more complex transmission spectra that includes dips in transmission for both left-and right-handed circularly polarized waves.

In addition to the presence of strong and broadband circular dichroism, comparison between the measured and simulated transmission spectra reveals a further advantage of the **4srs** nanostructure over similar **1srs** nanostructures. For the case of the **4srs** nanostructure, good agreement is found between the measured and transmission spectra when accounting for both the full $\alpha = 5^{\circ}$ half apex angle of the incident beam, as well as when the simulation is limited to normal incidence ($\alpha = 0^{\circ}$). The **1srs** nanostructure however, shows significant deviation between the numerical simulations at normal incidence and at $\alpha = 5^{\circ}$. This indicates excellent robustness of the **4srs** nanostructure to changes in both the incident angle, and to deviations from plane wave illumination surrounding the [001] axis.

The origin of the Bragg-like characteristics of the circular dichroism band lies in the coincidence of the four-screw double-helix axes with the direction of the primitive reciprocal lattice vectors. Along these primitive directions, the first Brillouin zone is met with the shortest possible wave-vectors such that other reciprocal lattice vectors that possess much larger wave-vector lengths cannot contribute to the band structure. In such a situation, the scattering problem reduces approximately to a single-mode problem equivalent to the situation of a Bragg mirror or simple multilayer stack²⁵ and accordingly, a simple band-gap with strong reflection is surrounded by a single pair of modes permitting high transmission is opened. In combination with the distinct topological continuities of opposite handed

screw motions along the four-screw double helices^{24,25} that drive a large separation of LCP and RCP mode frequencies, the band-gap is transformed into a broad band of circular dichroism with a spectral response reminiscent of the clean broadband reflection from a Bragg mirror.

The origin of the robustness to incident angle arises from the SC symmetry of the 4srs, which translates to a SC Brillouin zone in reciprocal space, shown in Figure 4a and 4b. Real-space propagation along the <100> axes corresponds to wave-vectors, k, terminating at the face of the Brillouin zone, such that small changes to the incident angle, θ , increase |k| in quadratic order. Accordingly, the band-edge mode frequencies undergo only a minor blue-shift as shown in Figure 4c, which plots the spectral position of the circular dichroism band for the experimental 4srs geometry as a function of θ . In contrast, the 1srs has BCC symmetry regardless of the choice of experimentally convenient non-primitive cubic unit cell, resulting in a face centred cubic (FCC) Brillouin zone in reciprocal space, shown in Figure 4d and 4e. In this case, real-space propagation along the <100> axes corresponds to wave-vectors that terminate at the Brillouin zone four-edge vertices, such that a small change in the incident angle result in a large decrease in lkl. Consequently, the band-edge mode frequencies undergo a strong red-shift as shown in Figure 4f, which plots the spectral position of the experimental 1srs circular dichroism bands (see Supplementary Fig. 6 for a band-diagram of this geometry) as a function of θ . For small angles of incidence, the disparate shift in mode frequencies between the two networks is present for all volume fractions and network ellipticities, as shown in Supplementary Fig. 7. This indicates that the switch from BCC to SC symmetry can provide robustness to angle of incidence in the presence of the most common geometric variations, so long as the topology of the network is preserved.

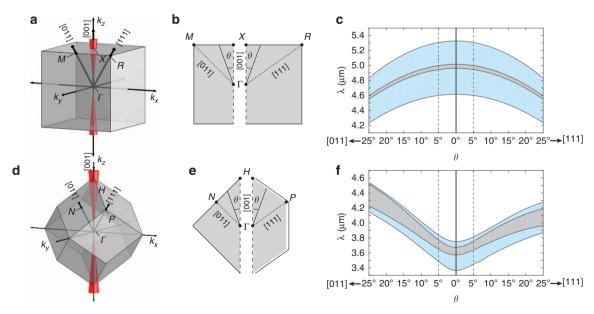


Figure 4 Origin of the robust circular dichroism in quadruple-gyroid 4srs nanostructures. (a) First Brillouin zone of a 4srs nanostructure and the measured wave-vectors (red cone) within the $\alpha=5^{\circ}$ half apex angle around [001]. (b) Partial cross-sections of the Γ -X-M and Γ -X-R Brillouin zone planes for the 4srs nanostructure. (c) Plot of the 4srs nanostructure circular dichroism bands (blue shading) and band-gap (grey shading) spectral positions as a function of incident angle (θ) from [001] to [011] and [111]. (d) First Brillouin zone of the 1srs nanostructure and the measured wave-vectors (red cone) within the $\alpha=5^{\circ}$ half apex angle around [001]. (e) Partial cross-sections of the Γ -H-N and Γ -H-P Brillouin zone planes for the 1srs nanostructure. (f) Plot of the 1srs nanostructure circular dichroism band spectral position as a function of incident angle (θ) from [001] to both [011] and [111].

One possible application of the **4srs** nanostructure is in the engineering of defect states^{50,51} within the circular dichroism band. In particular, quarter-twist planar defect layers^{52,53} can introduce a defect mode precisely at the mid-gap wavelength of a circular dichroism band^{52,53}. Supplementary Fig. 8 shows transmission spectra and SEM images of a fabricated RHD **4srs** nanostructure with a quarter-twist defect layer in the (001) plane, creating a single defect mode within the circular dichroism band and producing a peak in LCP transmission that is pinned to the mid-gap wavelength. Such defects may be useful for tailoring the position and strength of circular dichroism bands⁵², or increasing the photonic density of states for chiral lasing⁵¹.

CONCLUSION

We have shown that the quadruple-gyroid **4srs** nanostructure is a remarkable nanostructure that provides an exceptional ability to filter broad bands of circularly polarized light with strong resilience against deviations in the angle of incidence and beam collimation, even when the nanostructure possesses asymmetric cylindrical segments. Furthermore, we have shown that the **4srs** nanostructure allows for the creation of Bragg-like circular dichroism without spectral inconsistencies, unlike many other chiral structures that exhibit non-uniform circular dichroism and transmission ^{13–22}.

CONFLICT OF INTEREST

The authors declare no conflict of interest.

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